# **Supporting Information**

Revealing the Chemical Structure Dependent Carrier Trapping in One-Dimensional Antimony Selenide Photovoltaic Materials

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#### **Experimental Section**

#### Fabrication of Sb<sub>2</sub>Se<sub>3</sub> films

Sb<sub>2</sub>Se<sub>3</sub> films were fabricated on insulating substrate by thermal evaporation deposition method under a pressure of  $5 \times 10^{-4}$  Pa. Sb<sub>2</sub>Se<sub>3</sub> (99.99%, zhongnuoxincai) powder was utilized as thermal evaporation source, and suitable Sb (99.999%, Sinopharm) powder was co-evaporated to adjust the stoichiometry of Sb<sub>2</sub>Se<sub>3</sub> films. We selected a suitable evaporated rate of around 5 nm s<sup>-1</sup> and obtained Sb<sub>2</sub>Se<sub>3</sub> films around 300 nm. The substrate was preheated at 315 °C. Eventually, Sb<sub>2</sub>Se<sub>3</sub> films were annealed at 380 °C for 8 min in glove box filled with N<sub>2</sub>.

#### Fabrication of Sb<sub>2</sub>Se<sub>3</sub> solar cells

The FTO glass (TEC-A7) was cleaned in the order of DI water, isopropanol, acetone, ethanol, then it was cleaned by UV ozone for 15 min. Afterwards, CdS electron transporting layer was deposited on as-cleaned FTO glass by CBD method. Then, Sb<sub>2</sub>Se<sub>3</sub> film was deposited on CdS by thermal evaporation method. Afterwards, Spiro-OMeTAD was spin-coated on Sb<sub>2</sub>Se<sub>3</sub> film as hole transport layer. Eventually, Au electrode was evaporated on the Spiro-OMeTAD layer under a pressure of  $5 \times 10^{-4}$  Pa. The active area was defined as 0.04 cm<sup>-2</sup> by a metallic mask.

### Characterizations of films and devices

The crystallinity of Sb<sub>2</sub>Se<sub>3</sub> films was characterized by XRD (Bruker Advance D8 diffractometer) with Cu K $\alpha$  radiation ( $\lambda = 1.5406$  Å). The SEM images of Sb<sub>2</sub>Se<sub>3</sub> thin films were measured by FE-SEM (Hitachi SU8220) equipped with an EDS (Bruker) module. Raman spectroscopy (Horiba JobinYvon, LabRAM HR800) was measured to analyze chemical bonds of the films with 532-nm laser excitation. XPS results of the films were carried out with a monochromatic Al K $\alpha$  X-ray source to characterize the elemental composition of the surface by a Thermo Scientific K-Alpha+ instrument. UV-vis-NIR spectrophotometer (SOLID 3700) was used to measure the steady-state optical absorption of Sb<sub>2</sub>Se<sub>3</sub> films. The *J-V* performance of Sb<sub>2</sub>Se<sub>3</sub> films was measured by a Keithley 2400 equipment under solar-simulated AM 1.5G sunlight (a typical solar simulator based on xenon-lamp (Oriel Sol 3A)). Halogen lamp connected with monochromator (SPIEQ200) was as the single illuminated source to characterize the EQE of Sb<sub>2</sub>Se<sub>3</sub> solar cells.

Sb <sub>2</sub> Se <sub>3</sub> films	Se (at%)	Sb (at%)	Se/Sb
Sb-rich	58.40±0.92	41.60±0.92	1.41±0.05
Se-rich	60.95±0.35	39.05±0.35	1.56±0.02
Sb <sub>2</sub> Se <sub>3</sub>	59.96±0.36	40.04±0.36	1.50±0.02

**Table S1.** The statistical chemical composition of Sb<sub>2</sub>Se<sub>3</sub> films with 10 samples each for Sb<sub>2</sub>Se<sub>3</sub>, Se-rich and Sb-rich films probed by energy dispersive X-ray spectroscopy.

Table S2. Results of biexponential fitting of GSB decay.

Sb <sub>2</sub> Se <sub>3</sub> films	$A_1$	$ au_1$ (ps)	$A_2$	τ <sub>2</sub> (ps)	τ <sub>ave</sub> (ps)
Sb-rich	-0.135	0.08	-0.865	0.31	0.30
Se-rich	-0.031	0.07	-0.969	0.20	0.20

 Table S3. Results of biexponential fitting that rely on Levenberg-Marquardt algorithm.

Sample	$A_1$	τ <sub>1</sub> (ps)	<i>A</i> <sub>2</sub>	τ <sub>2</sub> (ps)	τ <sub>ave</sub> (ps)
Sb-rich Sb <sub>2</sub> Se <sub>3</sub>	0.115	14.6	0.649	3891.1	3888.5
Se-rich Sb <sub>2</sub> Se <sub>3</sub>	0.128	20.2	0.703	4256.3	4252.7
Sb-rich Sb <sub>2</sub> Se <sub>3</sub> on CdS	0.024	8.3	0.447	1972.7	1972.2
Se-rich Sb <sub>2</sub> Se <sub>3</sub> on CdS	0.075	4.7	0.553	1410.7	1410.1



Figure S1. UV-vis spectroscopy of Sb-rich and Se-rich Sb<sub>2</sub>Se<sub>3</sub> films.



**Figure S2.** Transient kinetic traces of (a) Sb-rich Sb<sub>2</sub>Se<sub>3</sub> films and (b) Se-rich Sb<sub>2</sub>Se<sub>3</sub> films and their fits (solid lines).



Figure S3. (a-d) Statistical  $V_{\text{OC}}$ ,  $J_{\text{SC}}$ , FF and PCE of Se-rich and Sb-rich Sb<sub>2</sub>Se<sub>3</sub> devices.

Note S1. TA characterization.

The carrier dynamics was obtained from femtosecond transient absorption spectroscopy (fs-TAS). The fs-TAS instrument we used comprises ultrafast laser system, light conversion, sample transport, delay generation and probe detection. Broad UV/vis probe band is achieved by generating continuous white light in different media (e.g., CaF<sub>2</sub>, sapphire); the choice of media and seed energy determine the range of spectrum and shape of the generated continuous light.<sup>1</sup> Transient absorption (TA) measurements are performed on a pump-probe system (Helios, Ultrafast System) with the maximum delay time of ~8000 ps using a motorized optical delay line under ambient conditions. The pump pulses at 400 nm wavelength (~800  $\mu$ W average power at the sample) are delivered by an ultrafast optical parametric amplifier (OPera Solo) excited by a regenerative amplifier (Coherent Astrella, 800 nm, 1 kHz), seeded with a mode-locked Ti:sapphire oscillator (Coherent Vitara, 800 nm, 80 MHz) and pumped with a LBO laser (Coherent Evolution-50C, 1 kHz system). A small amount of 800 nm femtosecond pulses from the regenerative amplifier are used to pump a sapphire crystal to create a 320-780 nm white light continuum as probe pulses. Interpulse fluctuations of continuous white light probe pulses are corrected using a self-splitting reference beam (Reference). The mechanical chopper operates at a frequency of 500 Hz and is used for modulating the pump pulses. The TA decay is fitted with biexponential function:

$$\Delta A(t) = \sum_{i=1}^{N} A_i exp\left(-\frac{t}{\tau_i}\right) \tag{1}$$

where t is the probe delay time,  $A_i$  and  $\tau_i$  are amplitudes of TA and decay lifetimes, respectively. Here, the number of components (N) to satisfactorily fit the experimental data is 2. The average lifetime ( $\tau_{ave}$ ) was estimated from the fitting parameters according to the following equation.

$$\tau_{ave} = \sum_{i=1}^{2} A_i \tau_i^2 / A_i \tau_i \tag{2}$$

Note S2. Calculation of trap-assisted recombination rate coefficient.

Equation S3 shows the evolution of carrier density after pump excitation.

$$-\frac{dN(t)}{dt} = k_1 N + k_2 N^2 \tag{3}$$

where N(t) is the carrier density;  $k_1$  is the monomolecular recombination rate coefficient at low carrier density;  $k_2$  is the bimolecular recombination rate coefficient at high carrier density. Herein, we only focus on  $k_1$  because trap-assisted recombination is one of monomolecular recombination. Thus, we can simplify the Equation S3 as follows:

$$-\frac{dN(t)}{dt} = k_1 N \tag{4}$$

For further adjustment,

$$\frac{dN(t)}{N} = -k_1 dt \tag{5}$$

Integrate both sides of the Equation S5 at the same time.

$$\int_{0}^{N} \frac{dN}{N} = -k_1 \int_{0}^{t} dt \tag{6}$$

Then,

$$nN = -k_1 t + C \tag{7}$$

Where C is a constant. When t = 0,  $N = N_0$  ( $N_0$  is the initial carrier density). Thus,  $C = \ln N_0$ .

Half-lifetime  $(t_{1/2})$  is generally used for first-order decay process. Suppose that when  $N_0$  decays to half,  $t = t_{1/2}$ ,

$$ln\frac{1}{2}N_0 = -k_1 t_{1/2} + lnN_0 \tag{8}$$

Therefore,

$$t_{1/2} = \frac{ln2}{k_1}$$
(9)

Changing the excitation intensity will vary the initial carrier density. Since  $k_1$  is the intrinsic decay rate constant, the time for the initial carrier density decaying to half  $(t_{1/2})$  is fixed, thus the decay curve of free electron-hole recombination remains unchanged and independent of excitation intensity.

If trap-assisted recombination of electrons and holes occurs, we assume that holes are trapped firstly and then recombine with electrons. In this case, we denote  $N_{\text{TH}}$  and  $N_{\text{FH}}$  as density of trapped and free holes, respectively;  $N_{\text{TE}}$  and  $N_{\text{FE}}$  as density of electrons bound to trapped holes and free electrons, respectively;  $k_{\text{TH}}$  as intrinsic trapping rate constant of holes and  $k_{\text{TE}}$  as trapping rate constant of electrons captured by pre-trapped holes;  $N_0$  as the initial carrier density;  $t_{\text{TH}}$  as hole capture time and  $t_{\text{TE}}$  as electron capture time. Since the holes are trapped at first, we have following equations.

$$\frac{dN_{TH}}{dt} = N_{FH}k_{TH} = (N_0 - N_{TH})k_{TH}$$
(10)

Integrate both sides of Equation S10 at the same time.

$$\int_{0}^{N} \frac{dN_{TH}}{N_{0} - N_{TH}} = k_{TH} \int_{0}^{t_{TH}} dt$$
 (11)

According to the integration result, when  $N = N_{\text{TH}}$ ,

$$ln\left(1 - \frac{N_{TH}}{N_0}\right) = -k_{TH}t_{TH} \tag{12}$$

The trapped holes then continue to attract free electrons,  $dN_{rr}$ 

$$\frac{dN_{TE}}{dt} = N_{FE} \left( N_{TH} - N_{TE} \right) k_{TE} \approx \left( N_{TH} - N_{TE} \right) k_{TE}$$
(13)

Integrate both sides of Equation S13 at the same time.

$$\int_{0}^{N} \frac{dN_{TE}}{N_{TH} - N_{TE}} = k_{TE} \int_{0}^{t_{TE}} dt$$
 (14)

According to the integration result, when  $N = N_{\text{TE}}$ ,

$$ln\left(1 - \frac{N_{TE}}{N_{TH}}\right) = -k_{TE}t_{TE}$$
(15)

It can be seen that the  $t_{\text{TE}}$  is related to  $N_{\text{TH}}$ . When the  $N_{\text{TH}}$  increases, the electron capture rate increases, and the recombination rate of electrons and holes increases, thus the dynamic decay accelerates. The variability of excitation intensity will change the  $N_0$ , which in turn influences the  $N_{\text{TH}}$  and carrier capture time  $t_{\text{TE}}$ , leading to different decay curves.

## Reference

1 L. A. Baker, S. E. Greenough and V. G. Stavros, J. Phys. Chem. Lett., 2016, 7, 4655-4665.