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Supplementary Information

Heteroatomic interface engineering of octahedron VSe₂-ZrO₂/C/MXene composite derived from MXene-MOF hybrid as superior-performance anode for lithium-ion batteries

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Materials

Hydrofluoric acid (HF, 40%), zirconium tetrachloride (ZrCl₄, 98%), terephthalic acid (H₂BDC, 99%), N, N-Dimethylformamide (DMF, \geq 99.5%), glacial acetic acid (GR, \geq 99.8%) and selenium powder (Se, \geq 99.0%) were supplied by Sinopharm Chemical Reagent Co. Ltd. Vanadium aluminum carbide (V₂AlC, 98%, 400 mesh) was purchased from Jilin 11 Technology Co. Ltd.

Characterizations

The morphologies and interior structure were detected by field-emission scanning electron microscopy (FESEM, Tescan Mira 3 XH) and transmission electron microscopy (TEM, JEOL-2100F). X-ray diffraction spectroscopy (XRD, D/Max-2500/PC, RIGAKU, Japan) for measuring the phase structure was conducted with Cu K α radiation ($\lambda = 0.15406$ nm). The specific surface areas and pore size distributions were acquired by the nitrogen adsorption-desorption measurements on the AutosorbiQ analyzer (Quantachrome, USA). X-ray photoelectron spectroscopy (XPS) was collected in Escalab 250Xi spectrometer with Al K α radiation ($h\nu$ =1486.6eV). Electron paramagnetic resonance (EPR) spectra were obtained on Bruker ELEXSYS E580 spectrometer with a super high sensitivity cavity. Thermogravimetric analysis (TGA) was tested by thermal gravimetric analyzer (Mettler Toledo, Switzerland). The content of element was determined by inductively coupled plasma-atomic emission spectrometry (ICP-AES, Pe Avio200). Raman spectra were obtained using a laser of 532 nm by Raman microscope (XploRA PLUS).

Battery fabrication and electrochemical measurements

To study the electrochemical properties of half cells, the active materials, Super P and polyvinylidene fluoride (PVDF) were mixed in an agate mortar at a mass ratio of 8:1:1 into moderate N-methyl-2-pyrrolidinone (NMP) solvent to obtain the homogeneous slurry. Then, the slurry was bladed onto Cu foil and dried at 110 °C for 12 h under vacuum. The dried foil was cut into a circular electrode with a diameter of 14 mm on which the mass loading of the active material was 1.2-1.4 mg. The half coin cells (CR2032) were assembled in an argon-filled glove box with both the moisture and the oxygen content below 0.5 ppm. The electrolyte was formed by LiPF₆ (1.0 M) in ethylene carbonate/dimethyl carbonate /ethyl methyl carbonate (1:1:1 in volume), while Li disc was applied as a counter electrode and glass membrane as a separator. Galvanostatic charge/discharge measurements were conducted in the LAND CT2001A battery testing system setting the voltage range of 0.01 V-3.0 V. Cyclic voltammetry (CV) curves were investigated in the electrochemical workstation (AUTO-LAB PGSTAT302N) with the same voltage range at a scan rate of 0.2 mV s⁻¹ unless otherwise noted. Electrochemical impedance spectroscopy (EIS) data were recorded in the same electrochemical workstation with the frequency ranging from 0.01 to 100 kHz.

For full cells, the cathode was prepared in the same technique as the anode except for the active material of commercial LiFePO₄ and the Al foil current collector. For the testing of the full cell, the voltage window was set as 1.5-3.5 V with a designed anode/cathode reversible capacity ratio of ~1.1. Both the current density and specific capacity were calculated based on the mass of the anode.



Fig. S1. FESEM images of (a) V_2CT_x and (b) UiO-66.



Fig. S2. TEM images of (a) Se-ZrO₂/C and (b) VSe₂/MXene.



Fig. S3. XRD patterns of (a) UiO-66, (b) V_2CT_x and (c) UiO-66/ V_2CT_x .



Fig. S4. (a) Nitrogen adsorption-desorption isotherms and (b) pore size distributions of V_2CT_x .



Fig. S5. EPR spectra of Se-ZrO₂/C, VSe₂/MXene and VSe₂-ZrO₂/C/MXene.



Fig. S6. TGA curves of Se-ZrO₂/C, VSe₂/MXene and VSe₂-ZrO₂/C/MXene.



Fig. S7. CV curves at 0.2 mV s⁻¹ of (a) Se- ZrO_2/C and (b) VSe₂/MXene. GCD profiles of (c) Se- ZrO_2/C and (d) VSe₂/MXene.



Fig. S8. FESEM images of VSe₂-ZrO₂/C/MXene, VSe₂/MXene and Se-ZrO₂/C

electrodes after 100 cycles at the current density of 100 mA g⁻¹.



Fig. S9. (a) Cycling performance of meso-carbon microbeads. (b) and (c) Charge E_R of

VSe₂-ZrO₂/C/MXene and meso-carbon microbeads.



Fig. S10. Ex-situ Raman spectra of VSe₂-ZrO₂/C/MXene anode at different potential

states.



Fig. S11. TEM images of VSe₂-ZrO₂/C/MXene electrode after (a) discharging to 0.05

V and (b) charging to 3.0 V.



Fig. S12. (a) Discharge profiles for VSe₂-ZrO₂/C/MXene at the 10th and 100th cycles.
(b) Difference of the capacities at the 10th and 100th cycles and their differential vs. voltage.



Fig. S13. (a) Survey, (b) Zr 3d, (c) V 2p and (d) Se 3d XPS spectra for VSe₂-

ZrO₂/C/MXene after cycling.



Fig. S14. Nyquist plots of VSe₂-ZrO₂/C/MXene after different cycles.



Fig. S15. Schematic of VSe₂/MXene for Li-ion storage mechanism.

Table S1 Atomic concentration (at. %) of elements detected from XPS measurementfor Se-ZrO2/C, VSe2/MXene and VSe2-ZrO2/C/MXene.

Sample	С	Ν	0	Se	Zr	V
Se-ZrO ₂ /C	46.59	6.28	23.96	11.67	11.49	-
VSe ₂ /MXene	36.77	-	31.42	17.68	-	14.13
VSe ₂ -ZrO ₂ /C/MXene	52.88	3.55	17.11	12.04	6.25	8.18

Table S2 Comparison of the lithium-ion storage performances between VSe_2 - $ZrO_2/C/MX$ ene in this work and similar anode materials reported in the literatures.

Sample	Current density (mA g ⁻¹)	Cycle number	Capacity (mAh g ⁻¹)	Capacity retention	refs
UiO-66	38.8	100	118	65%	[S1]
SnO _x @UiO-66	50	100	994	85%	[S2]
VSe ₂ -NSs	2000	70	430	-	[S3]
	100	100	1020	119%	
VSe ₂ NSA/C	100	50	768	98%	[S4]
VSe ₂ /C	100	50	453	-	[S5]
VSe ₂ /graphene	100	60	630	~100%	[S6]
VSe ₂ -ZrO ₂ /C/MXene	100	210	1238.5	143%	This work

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