

Supplementary information for: High oxygen-vacancy diffusivity predicted for perovskite oxide $\text{Ca}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{2.5}^{\dagger}$

Alexander Bonkowski,^a Caitlin Perkampus,^a and Roger A. De Souza^{*a}

The interactions in our molecular dynamics (MD) calculations are described with a short-range potential of the Buckingham form and a long-range coulombic term:

$$V_{ij}(r) = A_{ij} \exp\left(-\frac{r}{\rho_{ij}}\right) - \frac{C_{ij}}{r^6} + \frac{q_i q_j}{4\pi \epsilon_0 r}$$

The empirical parameters A_{ij} and ρ_{ij} represent Pauli repulsion; C_{ij} describes dispersion (van der Waals interactions); q_i and q_j are the formal ion charges, and ϵ_0 is the vacuum permittivity. We employ a set of parameters that has been used previously to examine BSCF materials.^{1,2} We expand the model to include Ca^{2+} by using parameters derived for the same oxygen–oxygen potential. The potential parameters are given in Table S1. All MD calculations were carried out in the LAMMPS code^{3,4}, with periodic boundary conditions, in an isothermal-isobaric (NpT) ensemble ($p_{\text{damp}} = 1 \text{ ps}$, $T_{\text{damp}} = 0.1 \text{ ps}$) at $p = 1013.25 \text{ mbar}$, and a timestep of $\delta t = 1 \text{ fs}$. Metropolis Monte Carlo calculations⁵ were carried out as implemented in the LAMMPS code.

Table S1 Buckingham type potential parameters.

Interaction	A_{ij} / eV	ρ_{ij} / Å	C_{ij} / $\text{eV} \text{\AA}^{-6}$	source
$\text{Ca}^{2+}-\text{O}^{2-}$ (I)	1340.18	0.3214	0.0	6
$\text{Ca}^{2+}-\text{O}^{2-}$ (II)	1090.4	0.3437	0.0	7
$\text{Ba}^{2+}-\text{O}^{2-}$	1214.4	0.3522	0.0	1,8
$\text{Sr}^{2+}-\text{O}^{2-}$	774.2	0.3538	0.0	9
$\text{Co}^{3+}-\text{O}^{2-}$	1329.82	0.3087	0.0	9,10
$\text{Fe}^{3+}-\text{O}^{2-}$	1156.36	0.3299	0.0	10
$\text{O}^{2-}-\text{O}^{2-}$	22764.3	0.1490	43.0	9,10

Simulations at 500 K show a cubic cell parameter of $a(\text{BSCF}) = 3.799 \text{\AA}$, exactly the same as Shiiba *et al.*², and $a(\text{CSCF}) = 3.744 \text{\AA}$, slightly lower due to $r_{\text{Ca}^{2+}} < r_{\text{Ba}^{2+}}$.

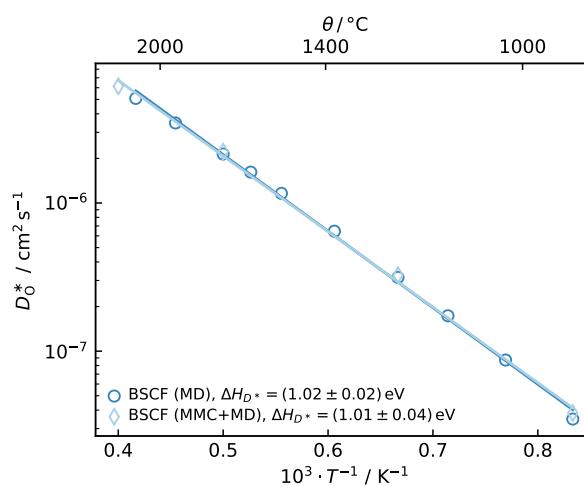


Fig. S1 Oxygen tracer diffusion coefficients versus inverse temperature for cubic $\text{Ba}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{2.5}$, obtained from MD simulations, with and without preceding MMC simulations.

^a Institute for Physical Chemistry, RWTH Aachen University, 52074 Aachen, Germany. Fax: +49 241 80 92128; Tel: +49 241 80 94739; E-mail: desouza@pc.rwth-aachen.de

Notes and references

- 1 C. Fisher, M. Yoshiya, Y. Iwamoto, J. Ishii, M. Asanuma and K. Yabuta, *Solid State Ionics*, 2007, **177**, 3425–3431.
- 2 H. Shiiba, C. L. Bishop, M. J. D. Rushton, M. Nakayama, M. Nogami, J. A. Kilner and R. W. Grimes, *J. Mater. Chem. A*, 2013, **1**, 10345.
- 3 S. Plimpton, *J. Comput. Phys.*, 1995, **117**, 1–19.
- 4 A. P. Thompson, H. M. Aktulga, R. Berger, D. S. Bolintineanu, W. M. Brown, P. S. Crozier, P. J. in 't Veld, A. Kohlmeyer, S. G. Moore, T. D. Nguyen, R. Shan, M. J. Stevens, J. Tranchida, C. Trott and S. J. Plimpton, *Comput. Phys. Commun.*, 2022, **271**, 108171.
- 5 N. Metropolis, A. W. Rosenbluth, M. N. Rosenbluth, A. H. Teller and E. Teller, *J. Chem. Phys.*, 1953, **21**, 1087–1092.
- 6 G. Mather, M. Islam and F. Figueiredo, *Adv. Funct. Mater.*, 2007, **17**, 905–912.
- 7 G. V. Lewis and C. Catlow, *J. Phys. C: Solid State Phys.*, 1985, **18**, 1149.
- 8 M. T. Buscaglia, V. Buscaglia, M. Viviani and P. Nanni, *J. Am. Ceram. Soc.*, 2004, **84**, 376–84.
- 9 M. Islam, M. Cherry and C. Catlow, *J. Solid State Chem.*, 1996, **124**, 230–237.
- 10 M. Cherry, M. Islam and C. Catlow, *J. Solid State Chem.*, 1995, **118**, 125–132.