## **Supporting Information**

## A computational study on bifunctional 1T-MnS<sub>2</sub> with adsorption-

## catalysis effect for lithium-sulfur batteries

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Fig. S1 Density of states of (a)1T-MnS<sub>2</sub>, (b)1T-VS<sub>2</sub>, (c) 1T-TiS<sub>2</sub>, (d) 1T-ZrS<sub>2</sub> and (e) 1T-HfS<sub>2</sub>. The gray dashed line represents the Fermi level. The total density of states of 1T- $MS_2$ , p orbitals of sulfur atoms and d orbitals of metal atoms in 1T- $MS_2$  are represented by three different color curves. Our electronic structure calculations show 1T-TiS<sub>2</sub> and 1T-VS<sub>2</sub> are a semimetal and a metal respectively, and 1T-MnS<sub>2</sub>, 1T-ZrS<sub>2</sub> and 1T-HfS<sub>2</sub> are semiconductors.



Fig. S2 The adsorption configurations of all  $Li_2S_n$  (*n*=1,2,4,6,8) and  $S_8$  molecules on the surface of five 1T-*M*S<sub>2</sub>.



Fig. S3 The Bader charge transfer distribution of the adsorption configuration for  $Li_2S_2$  on five 1T-*M*S<sub>2</sub> surfaces. The black number represents the charge in the charge of the nearest atom, the negative number represents the charge gained by the atom, and the positive number represents the charge lost by the atom, and the red number is the bond length of S-S<sub>surface</sub> bond. (a) Isolated Li<sub>2</sub>S<sub>2</sub> molecule in vacuum, (b) Li<sub>2</sub>S<sub>2</sub>@1T-MnS<sub>2</sub>, (c) Li<sub>2</sub>S<sub>2</sub>@1T-VS<sub>2</sub>, (d) Li<sub>2</sub>S<sub>2</sub>@1T-TiS<sub>2</sub>, (e) Li<sub>2</sub>S<sub>2</sub>@1T-ZrS<sub>2</sub> and (f) Li<sub>2</sub>S<sub>2</sub>@1T-HfS<sub>2</sub>.The Li, S, Mn, V, Ti, Zr, and Hf atoms are marked as green, yellow, purple, red, cyan, thick green, and brown, respectively.



Fig. S4 The differential charge density for (a)  $Li_2S_4$  molecule and (b)  $Li_2S$  molecule on 1T-*M*S<sub>2</sub> surfaces. The bule and rose red surfaces correspond to the charge accumulation and depletion regions, respectively. The isovalue is 0.0055 e•Å<sup>-3</sup>. The green, yellow, purple, red, cyan, thick green and brown balls are Li, S, Mn, V, Ti, Zr and Hf atoms, respectively.



Fig. S5 The most stable adsorption configurations for Li atom on (a) 1T-MnS<sub>2</sub> and (b) 1T-VS<sub>2</sub> surfaces.



Fig. S6 The energy profile and atomic structure evolution for Li-ion diffusion on 1T-MnS<sub>2</sub> and 1T-VS<sub>2</sub> surfaces. The Li, S, Mn, and V atoms are marked as green, yellow, purple and red, respectively. The five images in the top row demonstrate the evolution of the atomic structure of Li ion diffusion on the surface of 1T-MnS<sub>2</sub> along C-B-C path, while the five images in the bottom row demonstrate the evolution of the atomic structure of Li ion diffusion on the surface of 1T-VS<sub>2</sub> along B-C-B path.

![](_page_5_Figure_2.jpeg)

Fig. S7 The schematic diagrams for the diffusion of Li ion on (a) 1T-MnS<sub>2</sub> surface and (b) 1T-VS<sub>2</sub> surface. The Li, S, Mn, and V atoms are marked as green, yellow, purple and red, respectively.

![](_page_6_Figure_0.jpeg)

Fig. S8 The schematic diagram for the decomposition of  $Li_2S$  on (a) 1T-MnS<sub>2</sub> surface and (b) 1T-VS<sub>2</sub> surface. The direction of the arrow represents the migration path of one of the Li ions in Li<sub>2</sub>S molecule as it breaks away from the molecule. The Li, S, Mn, and V atoms are marked as green, yellow, purple and red, respectively.

![](_page_6_Figure_2.jpeg)

Fig. S9 The energy profile and atomic structure evolution for the decomposition of Li<sub>2</sub>S

on 1T-MnS<sub>2</sub> and 1T-VS<sub>2</sub> surfaces. The Li, S, Mn, and V atoms are marked as green, yellow, purple and red, respectively. The three images in the top row demonstrate the evolution of the atomic structure of Li<sub>2</sub>S decomposition on the surface of 1T-MnS<sub>2</sub>, while the three images in the bottom row demonstrate the evolution of the atomic structure of Li<sub>2</sub>S decomposition on the surface of 1T-MnS<sub>2</sub>.

![](_page_7_Figure_1.jpeg)

Fig. S10 The stability test for 1T-MnS<sub>2</sub>. (a) The phonon dispersion curve, (b) AIMD simulation curve and (c) biaxial stress-strain response of 1T-MnS<sub>2</sub>. Inset images are top and side views of 1T-MnS<sub>2</sub> after AIMD simulation of 30 ps at 400 K.

![](_page_7_Figure_3.jpeg)

Fig. S11 The adsorption energy of  $Li_2S_n$  and  $S_8$  on (a) 1T-VS<sub>2</sub>, (b) 1T-ZrS<sub>2</sub> and (c) 1T-HfS<sub>2</sub> surfaces in vacuum and implicit solvent environment.

![](_page_8_Figure_0.jpeg)

Fig. S12 The top views of adsorption configurations of Li<sub>2</sub>S<sub>n</sub> and S<sub>8</sub> on 1T-MnS<sub>2</sub> surface.

![](_page_8_Figure_2.jpeg)

Fig. S13 Density of states for the adsorption configuration of  $Li_2S_n$  and  $S_8$  on 1T-MnS<sub>2</sub> surface. Note that due to large energy gap in spin-down channel, only spin-up density of states is shown for the convenience of viewing energy gap influenced by adsorbed molecules. The Fermi level is set to 0 eV.

Table S1 The calculated bond length (*d*) and ICOHP of S-S<sub>surface</sub> bond on  $1T-MS_2$  surfaces when Li<sub>2</sub>S molecule is adsorbed on  $1T-MS_2$  surfaces.

	$MnS_2$	VS <sub>2</sub>	TiS <sub>2</sub>	ZrS <sub>2</sub>	HfS <sub>2</sub>
<i>d</i> (Å)	2.03	2.07	2.05	2.07	2.15
ICOHP (eV)	-6.18	-5.58	-5.90	-5.61	-4.42

Table S2 Possible adsorption sites and the adsorption energy of Li ion on  $1T-MS_2$  surfaces. The bold part indicates the most likely site and the strongest adsorption among A, B, C sites for Li ion. NA represents A is local maximum site, and Li ion cannot be anchored to A site.

	Adsorption	Adsorption energy of Li ion, $E_{ads}$ (eV)					
11-1/152	А	В	С				
MnS <sub>2</sub>	-2.93	-3.57	-3.78				
$VS_2$	NA	-3.39	-3.33				
TiS <sub>2</sub>	NA	-3.65	-3.68				
$ZrS_2$	NA	-3.17	-3.27				
HfS <sub>2</sub>	NA	-2.92	-2.97				

Table S3 Possible diffusion modes and barriers of Li ion on 1T-MS<sub>2</sub> surfaces.

Possible diffusion modes						
$1T-MS_2$	B-B	C-C	B-C	A-A	A-B	A-C
$MnS_2$	0.24	0.24	0.24	≥0.24	≥0.65	≥0.85
$VS_2$	0.12	0.12	0.12	NA	NA	NA
$TiS_2$	0.17	0.17	0.17	NA	NA	NA
$ZrS_2$	0.18	0.18	0.18	NA	NA	NA
$HfS_2$	0.21	0.21	0.21	NA	NA	NA

Table S4 The calculated adsorption free energy  $(G_{ads})$  of Li<sub>2</sub>S<sub>n</sub> and S<sub>8</sub> molecules, which was defined as  $G_{ads} = G_{total} - G_{MS_2} - G_{molecule}$ .  $G_{MS_2}$ ,  $G_{molecule}$  and  $G_{total}$  are the Gibb free energies of the 1T- $MS_2$  surface, the isolated molecule and the whole adsorbed system, respectively.

	$MnS_2$	$VS_2$	TiS <sub>2</sub>	$ZrS_2$	$HfS_2$
$G_{ads}(Li_2S)$	-3.92	-3.02	-3.14	-2.11	-1.56
$G_{ads}(Li_2S_2)$	-2.91	-2.06	-2.07	-1.39	-1.05
$G_{ads}(Li_2S_4)$	-1.94	-1.28	-1.49	-0.88	-0.56
$G_{ads}(Li_2S_6)$	-0.67	-0.72	-0.78	-0.52	-0.41
$G_{ads}(Li_2S_8)$	-0.61	-0.70	-0.82	-0.48	-0.38
$G_{ads}(S_8)$	-0.07	-0.16	-0.16	-0.11	-0.07

Table S5 The calculated all  $\Delta G$  when SRR occurs on the five 1T-*M*S<sub>2</sub> surfaces and in vacuum without catalyst.

	$MnS_2$	$VS_2$	TiS <sub>2</sub>	$ZrS_2$	$HfS_2$	Vacuum
$\Delta G(\mathrm{S_8}^* \to \mathrm{Li_2S_8}^*)$	-3.18	-3.17	-3.29	-3.01	-2.95	-2.63
$\Delta G(\text{Li}_2\text{S}_8^* \rightarrow \text{Li}_2\text{S}_6^*)$	-0.12	-0.07	-0.02	-0.09	-0.08	-0.05
$\Delta G(\text{Li}_2\text{S}_6^* \rightarrow \text{Li}_2\text{S}_4^*)$	-0.98	-0.27	-0.42	-0.07	0.14	0.29
$\Delta G(\text{Li}_2\text{S}_4^* \rightarrow \text{Li}_2\text{S}_2^*)$	-0.06	0.13	0.33	0.41	0.42	0.91
$\Delta G(\mathrm{Li}_2 \mathrm{S_2}^* \to \mathrm{Li}_2 \mathrm{S}^*)$	-0.10	-0.05	-0.16	0.18	0.39	0.90

Table S6 The calculated energy gap (eV) of pristine  $1T-MnS_2$  and the total adsorbed system where sulfur species is on  $1T-MnS_2$  surface. The tag "\*" represents the molecule absorbed on  $1T-MnS_2$  surface.

Spin channel	1T-MnS <sub>2</sub>	Li <sub>2</sub> S*	$Li_2S_2*$	$Li_2S_4*$	Li <sub>2</sub> S <sub>6</sub> *	$Li_2S_8*$	$S_8$ *
Spin-up	0.10	0.00	0.00	0.00	0.10	0.09	0.06
Spin-down	1.97	1.94	1.91	1.97	1.97	1.94	1.97

	-4%-MnS <sub>2</sub>	-2%-MnS <sub>2</sub>	2%-MnS <sub>2</sub>	4%-MnS <sub>2</sub>
$\Delta G(\mathrm{S_8}^* \to \mathrm{Li}_2 \mathrm{S_8}^*)$	-3.15	-3.23	-3.27	-3.22
$\Delta G(\text{Li}_2\text{S}_8^* \rightarrow \text{Li}_2\text{S}_6^*)$	-0.16	-0.08	-0.03	0.00
$\Delta G(\text{Li}_2\text{S}_6^* \rightarrow \text{Li}_2\text{S}_4^*)$	-0.69	-0.83	-0.90	-1.14
$\Delta G(\mathrm{Li}_2 \mathrm{S_4}^* \to \mathrm{Li}_2 \mathrm{S_2}^*)$	0.21	0.17	0.15	0.05
$\Delta G(\mathrm{Li}_2 \mathrm{S_2}^* \to \mathrm{Li}_2 \mathrm{S}^*)$	0.03	0.03	-0.12	-0.09

Table S7 The calculated all  $\Delta G$  when SRR occurs on four deformable 1T-MnS<sub>2</sub> surfaces.

## Some discussion of Li-ion diffusion

According to the number of stable adsorption sites shown in Fig. 4(a) in the text and considering the reversibility of diffusion, there are 6 possible diffusion modes (A-A, B-B, C-C, A-B, A-C, B-C) for Li ion on MnS<sub>2</sub> surface, and there are 3 possible diffusion modes (B-B, C-C, B-C) for Li ion on VS<sub>2</sub>, TiS<sub>2</sub>, ZrS<sub>2</sub>, HfS<sub>2</sub> surfaces, where the mode with the same initial and final sites is defined as the same mode. As shown in Fig. 4 in the text, a single Li ion diffusing on MnS<sub>2</sub>, ZrS<sub>2</sub>, HfS<sub>2</sub>, and TiS<sub>2</sub> surfaces along C-B-C path (C-C mode) faces a barrier of 0.24 eV, 0.21 eV, 0.18 eV and 0.17 eV, while Li ion diffusing on VS<sub>2</sub> surface along B-C-B path (B-B mode) faces a barrier of 0.12 eV.

In the following, we will use the adsorption energy of Li on different sites and the energy barrier of our calculated diffusion path to argue that the diffusion barrier of other possible paths is not less in energy than the path in our calculation.

Take VS<sub>2</sub> surface as an example. We use CI-NEB method to calculate the diffusion barrier from a B site to its nearest B site and discovered that the lowest energy path is B-C-B with a barrier of 0.12 eV. Other possible paths, i.e. C-C and B-C, is equivalent to or is a subset of the B-C-B path, and therefore they should have the same energy barrier of 0.12 eV. Similar argument can be applied to  $TiS_2$ ,  $ZrS_2$  and  $HfS_2$  surfaces.

For MnS<sub>2</sub> surface, we use CI-NEB method to calculate the diffusion barrier from a C site to its nearest C site and discovered that the lowest energy path is C-B-C with a barrier of 0.24 eV. Therefore, the diffusion paths B-C and B-C-B share the same energy barrier of 0.24 eV. **It is worth emphasizing that the energy barrier will be no less than the absolute value of the energy difference (that is, the adsorption energy difference) at the two nearest local stable sites of the diffusion path**. When Li-ion diffusion is along A-C or A-B mode, the absolute value of the adsorption energy difference between the initial and final sites is 0.85 eV or 0.65 eV (Table S2), which is higher than the energy barrier (0.24 eV) along C-B-C path. Similarly, when Li-ion diffusion is along A-A or B-B mode, Li ion may pass through C site due to strong adsorption of Li ion at C site, making the diffusion A-C-A or B-C-B path due to the strong adsorption at C site, hence the energy barrier cannot be no less than 0.24 eV. Therefore, a single Li ion diffusing on MnS<sub>2</sub> surfaces faces a minimum barrier of 0.24 eV. eV.

In conclusion, a single Li ion diffusing on  $MnS_2$ ,  $ZrS_2$ ,  $HfS_2$ ,  $TiS_2$ , and  $VS_2$  surfaces faces a minimum barrier of 0.24 eV, 0.21 eV, 0.18 eV, 0.17 eV, and 0.12 eV, respectively. In addition, Li-ion diffusion can be along B-C-B-C-B-C-... or C-B-C-B-C-B-C-B-... path. Possible diffusion modes and barriers of Li ion on  $1T-MS_2$  surfaces are summarized in Table S3.