Supporting Information

Towards 26% efficiency in inverted perovskite solar cells via

interfacial flipped band bending and suppressed deep-level traps

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Experimental Section

Materials

Formamidine iodide (FAI, >99.5%), methylammonium iodide (MAI, >99.5%) and methylammonium chloride (MACl, >99.5%) were gained from Greatcell Solar Materials Pty Ltd., Lead (II) iodide (PbI₂, >99.99%) was purchased form Xi'an E-Light New Material Co., Ltd. Cesium iodide (CsI, >99.99%), [4-(3,6-dimethyl-9H-carbazol-9-yl)butyl]phosphonic acid (Me-4PACz), piperazinium diiodide (PDI), C₆₀ and bathocuproine (BCP) were purchased from Xi'an Polymer Light Technology Crop. N, N Dimethylformamide (DMF, 99.8%), Dimethyl sulfoxide (DMSO, 99.8%) and chlorobenzene (CB) were obtained from Sigma-Aldrich. All materials used directly after purchase without further processing.

Device fabrication

The FTO substrates (2.0*2.0 cm²) were sequentially washed with deionized water, absolute ethanol, acetone and isopropanol in ultrasonic bath for 20 min. Then, the FTO substrates were further cleaned via UV-Ozone treatment for 15 min. After cooling to room temperature, the Me-4PACz (1 mmol in methanol) was spin-coated on the substrates at 5000 rpm 30 s and annealed at 100 °C 10 min.

The perovskite precursor solution was prepared by dissolving FAI, MAI, CsI, and PbI₂ in anhydrous DMF: DMSO=4: 1 (v: v) according to the stoichiometric formula of 1.5 $M Cs_{0.05}(FA_{0.98}MA_{0.02})_{0.95}PbI_3$, with 20 mol% MAC1. The solution was stirred at room temperature until dissolved and filtered through a polytetrafluoroethylene (PTFE) filter (0.22 µm) before use. Then, 150 µL perovskite precursor solution was dropped onto Me-4PACz substrates, spin-coated at 2000 rpm for 25 s and 5000 rpm for 30 s. The 200 μ L of chlorobenzene was poured on the substrates 10 s prior to the end of the program. Then, quickly transferred the substrates to annealing at 110 °C for 20 min. The supernatant of PDI saturated solution was spin-coated at 5000 rpm for 30 s on the upper surface of the annealed perovskite films, and then transferred to a hot stage of 100 °C annealing for 10 min. Then, The C₆₀ (27 nm) and BCP (7 nm) were sequentially deposited on the perovskite films by vacuum evaporation. Finally, Cu (120 nm) was deposited as back electrode through a mask (the effective area of the electrode is 0.075 cm²) by thermal evaporation under 9×10⁻⁵ Pa.

First-principles calculations

We performed the first-principles calculations in the frame of density functional theory (DFT) with the Vienna ab initio simulation package (VASP).^[1] The exchangecorrelation energy is described by the Perdew-Burke-Ernzerhof (PBE) form of generalized-gradient approximation (GGA) exchange-correlation energy functional.^[2] The structure optimizations of systems of PD⁺ ion before and after adsorption on PbI₂ terminal FAPbI₃ (100) surfaces with I vacancy and I interstitial defects have been carried out by allowing top layer atomic positions to vary and fixing lattice parameters and bottom layer atomic positions until the energy difference of successive atom configurations was less than 10⁻⁶ eV. The force on each atom in the relaxed structures was less than 0.015 eV/Å. The cutoff energy for the plane-wave basis set was set to 400 eV. The k-point spacing was set to be smaller than 0.03 Å⁻¹ over Brillouin zone (BZ).^[3] To calculate the deprotonation energy of PD⁺, FA⁺ and MA⁺ ions, they were put in 10 Å*10 Å*10 Å lattice. Their structure optimizations were performed by relaxing atomic positions and fixing lattice parameters.

Characterization

The ¹H NMR spectra were tested by 600 MHz nuclear magnetic resonance spectrometer (Avance III). X-ray photoelectron spectra (XPS) were acquired on an AXIS Ultra DLD X-ray photoelectron spectrometer and calibrated based on the C 1s peak (284.8 eV). Ultraviolet photoelectron spectroscopy (UPS) was measured by AXIS Ultra DLD machine. The SEM images were obtained by field emission scanning electron microscopy (JSM-7800F). Atom force microscopy (AFM) images were observed on the Bruke Bio-FastScan AFM using taping mode. X-ray diffraction (XRD) patterns were taken on a D8 Advance X-ray diffractometer with Cu Ka radiation as the X-ray source with a scan rate of 8° min⁻¹. The grazing incidence wide angle X-ray scattering (GIWAXS) spectra were obtained at beamline BL16B1 in the Shanghai Synchrotron Radiation Facility (SSRF) and the incidence angle of the X-ray beam was 0.4°. The PL quantum yield (PLQY), Steady state photoluminescence (PL) and time-resolved photoluminescence (TRPL) were carried out using FLS 1000 photoluminescence spectrometer. PL mapping images were obtained by Raman image-scanning electron microscope (RISE-MAGNA) with scanning area of 50*50 um². Time-of-flight secondary ion mass spectra (ToF-SIMS) were detected by ION TOF ToF SIMS 5-100 (Primary ion beam: Bi³⁺, 30 keV, incident angle: 45 deg, scanning area: 150*150 um², pixel: 128*128, beam current: 0.48 pA). The transient photovoltage (TPV) and transient

photocurrent (TPC) curves are detected by Fluxim Paios Spectrometer. The current density-voltage (*J-V*) curves of the device is measured under AM 1.5G illumination at 100 mW cm⁻² (calibrated with a standard Si solar cell) using an Abet Technologies Sun 2000 solar simulator and a Keithley 2400 source meter.

Femtosecond transient absorption (fs-TA)

The pump of fs-TA spectroscopy with broadband capabilities and 1 fs resolution is frequency-doubled to 400 nm. The probe pulses are generated by passing another fraction of the 800 nm pulses through the 2 mm thick sapphire crystal. Before white light generation, the 800 nm amplifed pulses are passed through a motorized delay stage. The delay interval is 1 ms. Depending on the movement of delay stage, the transient species are detected at different time scales following excitation. In order to reduce the error, the same detection light intensity is controlled, background processing is carried out and the number of integrations is adjusted to 2000 times before each sample test. The fluence of the pump pulse (~2.67 μ J/cm²) is calculated by the excitation beam spot size (diameter) ~0.08 cm. For hot carrier temperature calculation, the high energy tail of the TA spectrum can be approximately described by the Maxwell-Boltzmann distribution function:^[4,5]

$$\Delta A[E] \propto e^{-\frac{E-E_f}{k_B T_c}} \#(1)$$

Where, $E_{\rm f}$ is the quasi-Fermi energy and $k_{\rm B}$ is the Boltzmann constant. The carrier temperature ($T_{\rm c}$) can thus be obtained by fitting the high energy tail using the above equation. To ensure that the hot carriers have reached a quasi-equilibrium via the carrier-carrier scattering after the initial fs-pulse excitation, the hot carrier temperatures

are analyzed from TA spectra after 0.5 ps time delay.

Calculation of quasi-Fermi level splitting (QFLS) and $\Delta V_{oc-imp}^{nonrad}$ recombination loss

QFLS can be calculated from PLQY by the following formula:^[6]

$$QFLS = QFLS_{rad} + K_BTln(PLQY) = K_BTln\left(\frac{J_G}{J_{0,rad}} \times PLQY\right) \#(2)$$

Where, QFLS_{rad} is the radiation limit of semiconductor materials, which sets the maximum achievable splitting of the quasi-Fermi level without considering nonradiative radiation recombination. $K_{\rm B}$ is the Boltzmann constant, $J_{\rm G}$ is the photogenerated current density, $J_{0,\rm rad}$ is the dark state radiative recombination saturation current density. According to the detailed balance theory, the $J_{0,\rm rad}$ can be calculated by the following equations:^[6]

$$J_{0,rad} = q \int_{0}^{\infty} EQE_{PV}(E)\phi_{BB}(E)dE\#(3)$$
$$\phi_{BB}(E) = \frac{2\pi E^{2}}{h^{3}c^{2}} \frac{1}{\exp\left(\frac{E}{K_{B}T}\right) - 1}\#(4)$$

Where q is the elementary charge, EQE_{PV} is the photovoltaic external quantum efficiency, ϕ_{BB} is the black-body radiative spectrum, E is the photo energy, h is the Planck constant, and c is the light speed in vacuum. The calculated QFLS_{rad} is 1.286 eV. Finally, $\Delta V_{oc-imp}^{nonrad}$ recombination loss can be calculated from PLQY by the following formula:

$$\Delta V_{oc-imp}^{nonrad} = -K_B T \ln \frac{(PLQY)}{q} \#(5)$$

Finally, the calculated QFLS and $\Delta V_{oc-imp}^{nonrad}$ of different samples are shown in Table 1.



Scheme S1. Deprotonation of (1) PDI, (2) MAI and (3) FAI. After DFT calculation, the deprotonation energy of PDI is -3.326 eV, while that of FAI and MAI are -1.612 eV and -1.854 eV, indicating that PDI is more prone to deprotonation thermodynamically. (4) The amine-cation reactions between MA and FA⁺ to generate MFA⁺. (5) Piperazine forms a complex with FAI through hydrogen bonding.



Fig. S1. XRD spectra of Control, PDI-treated and PI-treated perovskite films. * denoted PbI₂.



Fig. S2. Top-view SEM image of perovskite film after PDI treatment for 200 times.



Fig. S3. Top-view SEM images of Control, PDI-treated and PI-treated perovskite films.



Fig. S4. AFM images of Control and PDI-treated perovskite films.



Fig. S5. 2D GIWAXS patterns of (a) Control and (b) PDI-treated perovskite films. (c) Azimuthally integrating scattering intensity along the ring at q=10 nm⁻¹ of Control and PDI-treated perovskite films.



Fig. S6. (a) Steady state PL and (b) TRPL decays of the Control and PDI perovskite films.



Fig. S7. The full ¹H NMR spectra of FAI in DMSO-d6. ¹H signals of δ ~2.5 ppm and 3.3 ppm are attributed to H₂O and DMSO-d6.



Fig. S8. The full ¹H NMR spectra of MAI in DMSO-d6.



Fig. S9. The full ¹H NMR spectra of FAI+MAI+PbI₂ in DMSO-d6 aging for 1 h at 60 °C.



Fig. S10. The full ¹H NMR spectra of FAI+MAI+PbI₂ in DMSO-d6 aging for 48 h at 60 °C. A new peak at δ =2.81 ppm appeared in the perovskite solution, which was attributed to MFAI.



Fig. S11. The full ¹H NMR spectra of PDI+FAI+MAI+PbI₂ in DMSO-d6 aging for 1 h at 60 °C.



Fig. S12. The full ¹H NMR spectra of PDI+FAI+MAI+PbI₂ in DMSO-d6 aging for 48 h at 60 °C.



Fig. S13. XPS spectra of (a) C 1s and (b) Pb 4f of Control and PDI-treated perovskite films.



Fig. S14. Density of states of perovskites with iodine vacancies (a) before and (b) after PDI treatment, and iodine interstitials (c) before and (d) after PDI treatment.

	CK	1 µm	NK	1 µm	IL	1 µm	Si K	1 µm
C ₆₀	a de la compañía	and latter						Martina
PDI		and the second	atter the				and the second	
Glass							a Maria da	

Fig. S15. Cross-section SEM and EDS images of PDI/C_{60} stack.



Fig. S16. (a) Steady state PL and (b) TRPL decays of the Control/C60 and PDI/C60 samples.



Fig. S17. (a) The pseudo color plot and (b) normalized pump-probe fs-TA spectra of Control perovskite films.



Fig. S18. Extracted hot carrier temperatures with delay time for Control, $Control/C_{60}$ and PDI/C_{60} samples.



Fig. S19. Absolute bleach amplitude at early delay time at 3.1 eV excitation of Control, C_{60} and PDI/C_{60} samples.



Fig. S20. Nanosecond transient absorption dynamics for Control/C₆₀, SAM/Control and PDI/C₆₀ samples.



Fig. S21. Relevant integration of the temperature-dependent PL intensity of control and PDI-treated perovskite film and fitting curves for E_b .



Fig. S22. Depth profiling XPS spectra of C 1s and N 1s for perovskite/ C_{60} stack with etching time.



Fig. S23. Depth profiling XPS spectra of C 1s and N 1s for perovskite/PDI/C₆₀ stack with etching time.



Fig. S24. Surface contact potential of Control and PDI-treated PSCs.



Fig. S25. Analysis of perovskite bandgap from the EQE spectrum by taking its derivative spectrum. The bandgap was estimated to be about 1.55 eV.



Fig. S26. *J-V* curves of Pip- and PI-treated devices with 0.075 cm² in reverse scan and forward scan.



Fig. S27. Independent PCE certification for Inverted PDI-treated PSCs of a small area (0.075 cm²) at FuJian Metrology Institute (National PV Industry Measurement and Testing Center). The certified efficiency is 25.87% under reverse scan (I_{sc} of 1.903 mA, V_{oc} of 1.181 V and FF of 86.36%). The quasi-steady-state efficiency is 25.52% under

maximum power point tracking measured by National PV Industry Measurement and

Testing Center.



Fig. S28. Nyquist plots of Control and PDI-treated PSCs.



Fig. S29. (a) TPV and (b) TPC of Control and PDI-treated PSCs.



Fig. S30. The current density and steady power output measured at the maximum power point for 300 s of Control and PDI-treated PSCs.



Fig. S31. Statistical distribution of PCE, V_{oc} , J_{sc} , and FF for 27 Control and PDI-treated PSCs.



Fig. S32. EQE spectra and integrated J_{sc} of PDI-treated device with 1.028 cm².



Fig. S33. Long-term storage stability of the unencapsulated Control and PDI-treated devices stored in a dark nitrogen environment.

Sample	Scan	$V_{\rm oc}\left({ m V} ight)$	$J_{\rm sc}~({ m mA~cm^{-2}})$	FF (%)	PCE (%)
Pip	Reverse	1.164	24.69	81.87	23.53
	Forward	1.156	24.58	81.84	23.25
PI	Reverse	1.163	25.16	83.98	24.28
	Forward	1.163	24.93	82.43	23.88

Table S1. Photovoltaic parameters of Pip- and PI-treated devices.

Reference	$E_{\rm g}/{ m eV}$	$V_{\rm oc}/{ m V}$	$J_{sc}/mA \ cm^{-2}$	FF/%	PCE/%
J. Am. Chem. Soc., 2020, 142, 20134-20142 ^[7]	1.54	1.17	24.92	80.17	23.37
Sci. Adv., 2021, 7, eabe8130 ^[8]	1.55	1.17	23.9	83.6	23.4
Joule, 2022, 6, 2849-2868 ^[9]	1.55	1.17	24.95	83.31	24.32
Science, 2022, 376, 416-420 ^[10]	1.55	1.18	25.68	82.32	25.0
Science, 2022, 376, 73-77 ^[11]	1.55	1.20	24.7	82	24.3
Nat. Photon., 2022, 16, 352-358 ^[12]	1.56	1.15	24.9	83.46	23.91
Nature, 2022, 611, 278-283 ^[13]	1.53	1.15	26.13	84.6	25.49
Science, 2022, 375, 434-437 ^[14]	1.56	1.19	24.8	82.9	24.3
Science, 2023, 380, 404-409 ^[15]	1.56	1.21	24.78	84.65	25.39
Science, 2023, 379, 683-690 ^[16]	1.55	1.21	25.08	84.37	25.56
Nature Energy, 2023, 8, 462-472 ^[17]	1.55	1.19	24.78	83.07	24.50
This work (0.075 cm ²)	1.55	1.18	25.67	86.47	26.15 (Certified: 25.87)
This work (1.028 cm ²)	1.55	1.18	25.00	82.12	24.18

Table S2. Summary of the highest reported parameters of inverted PSCs.

References:

- 1. J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, 77, 3865-3868.
- 2. G. Kresse and D. Joubert, *Phys. Rev. B*, 1999, **59**, 1758-1775.
- 3. H. J. Monkhorst and J. D. Pack, *Phys. Rev. B*, 1976, **13**, 5188-5192.
- M. B. Price, J. Butkus, T. C. Jellicoe, A. Sadhanala, A. Briane, J. E. Halpert, K. Broch, J. M. Hodgkiss, R. H. Friend and F. Deschler, *Nat. Commun.*, 2015, 6, 1-8.
- 5. Y. Yang, D. P. Ostrowski, R. M. France, K. Zhu, J. van de Lagemaat, J. M. Luther and M. C. Beard, *Nat. Photonics*, 2016, **10**, 53-59.
- 6. P. Caprioglio, M. Stolterfoht, C. M. Wolff, T. Unold, B. Rech, S. Albrecht and D. Neher, *Adv. Energy Mater.*, 2019, **9**, 1901631.

- F. Li, X. Deng, F. Qi, Z. Li, D. Liu, D. Shen, M. Qin, S. Wu, F. Lin, S.-H. Jang, J. Zhang, X. Lu, D. Lei, C.-S. Lee, Z. Zhu and A. K. Y. Jen, *J. Am. Chem. Soc.*, 2020, 142, 20134-20142.
- 8. S. Chen, X. Xiao, H. Gu and J. Huang, *Sci. Adv.*, 2021, 7, eabe8130.
- Z. Zhu, K. Mao, K. Zhang, W. Peng, J. Zhang, H. Meng, S. Cheng, T. Li, H. Lin, Q. Chen, X. Wu and J. Xu, *Joule*, 2022, 6, 2849-2868.
- Z. Li, B. Li, X. Wu, S. A. Sheppard, S. Zhang, D. Gao, N. J. Long and Z. Zhu, *Science*, 2022, **376**, 416-420.
- R. Azmi, E. Ugur, A. Seitkhan, F. Aljamaan, A. S. Subbiah, J. Liu, G. T. Harrison, M. I. Nugraha, M. K. Eswaran, M. Babics, Y. Chen, F. Xu, T. G. Allen, A. U. Rehman, C.-L. Wang, T. D. Anthopoulos, U. Schwingenschlogl, M. De Bastiani, E. Aydin and S. De Wolf, *Science*, 2022, **376**, 73-77.
- H. Chen, S. Teale, B. Chen, Y. Hou, L. Grater, T. Zhu, K. Bertens, S. M. Park, H. R. Atapattu, Y. Gao, M. Wei, A. K. Johnston, Q. Zhou, K. Xu, D. Yu, C. Han, T. Cui, E. H. Jung, C. Zhou, W. Zhou, A. H. Proppe, S. Hoogland, F. Laquai, T. Filleter, K. R. Graham, Z. Ning and E. H. Sargent, *Nat. Photonics*, 2022, 16, 352-358.
- Q. Jiang, J. Tong, Y. Xian, R. A. Kerner, S. P. Dunfield, C. Xiao, R. A. Scheidt, D. Kuciauskas, X. Wang, M. P. Hautzinger, R. Tirawat, M. C. Beard, D. P. Fenning, J. J. Berry, B. W. Larson, Y. Yan and K. Zhu, *Nature*, 2022, 611, 278-283.
- 14. X. Li, W. Zhang, X. Guo, C. Lu, J. Wei and J. Fang, Science, 2022, 375, 434-437.
- S. Zhang, F. Ye, X. Wang, R. Chen, H. Zhang, L. Zhan, X. Jiang, Y. Li, X. Ji, S. Liu, M. Yu, F. Yu, Y. Zhang, R. Wu, Z. Liu, Z. Ning, D. Neher, L. Han, Y. Lin, H. Tian, W. Chen, M. Stolterfoht, L. Zhang, W.-H. Zhu and Y. Wu, *Science*, 2023, **380**, 404-409.
- W. Peng, K. Mao, F. Cai, H. Meng, Z. Zhu, T. Li, S. Yuan, Z. Xu, X. Feng, J. Xu, M. D. McGehee and J. Xu, *Science*, 2023, **379**, 683-690.
- X. Zheng, Z. Li, Y. Zhang, M. Chen, T. Liu, C. Xiao, D. Gao, J. B. Patel, D. Kuciauskas,
 A. Magomedov, R. A. Scheidt, X. Wang, S. P. Harvey, Z. Dai, C. Zhang, D. Morales, H.
 Pruett, B. M. Wieliczka, A. R. Kirmani, N. P. Padture, K. R. Graham, Y. Yan, M. K.
 Nazeeruddin, M. D. McGehee, Z. Zhu and J. M. Luther, *Nat. Energy*, 2023, 8, 462-472.