[†] Supporting Information

Ultrathin dense LiF coverage coupled with a near-surface gradient fluorination lattice enables fast-charging long-life 4.6 V LiCoO₂

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Methods

Materials synthesis

The commercial 4.45 V pristine LCO (named as P-LCO, purchased from JOHN LONG in Beijing) was uniformly mixed with LiPF₆-containing absolute ethanol solution at a mass ratio of 100:1 and continuously stirred until it was completely dried. Next, the composite electrodes were further dried in a blast drying oven at 90 °C for 6 h (named as LCO@LPF-90), and then transferred to a muffle furnace, which was heated to different temperatures of 400, 600 and 700 °C at a rate of 5 °C/min and subjected to thermal fluorination for 2 h. Finally, the fluorinated LCO samples were obtained, and correspondingly named as F-LCO-400, F-LCO, and F-LCO-700, respectively.

Materials characterizations

The morphologies, phases and crystallographic structures were characterized by XRD using a parallel-beam XRD instrument (Smartlab, Rigaku, with Cu K α of wavelength 1.542 Å). The types and contents of doping elements of commercial LCO were determined by ICP-OES 7300DV (PerkinElmer). The surface chemistry was analyzed by XPS (Thermo Scientific K α spectrometer). TEM samples were prepared by dual beam focused ion beam electron microscopy (FIB, Helios 450HP, FEI) using a 2-30 kV Ga ion beam. STEM (ARM200, JEOL; JEM-3200FS, JEOL; ARM300, JEOL) coupled with EELS/EDS were performed at different voltages (60 kV, 200 kV) to collect scanning transmission electron microscopy images for atomic and structural analysis, elemental and spectral analysis. The soft-XAS (SXAS) was performed at beamline 02B and XMCD Experimental Station (SSRF). Total electron yield (TEY)

mode (10 nm for probe depth) was used to collect the O K-edge SXAS spectra. ESR measurements were performed at 9.45 GHz with a Bruker spectrometer at the temperature of 100 K using an Oxford Instruments ESR 9 liquid nitrogen cryostat. TOF-SIMS analysis was performed by a TOF. SIMS 5 spectrometer (ION-TOF GmbH) to analyze the surface chemical structure and for depth profiling. All detected secondary ions of interest had a mass resolution of >5000 and possessed negative polarity. A pulsed 30 keV Bi¹⁺ (20 ns) ion beam set at a high current mode was employed for depth profiling and a 500 eV Cs⁺ (negative) ion beam was utilized for the sputtering of the cycled electrodes with a typical sputtered area (300 µm × 300 µm). The typical analyzed area was 50 µm × 50 µm. The instruments used for in-situ differential electrochemical mass spectrometer (Hiden HPR-20) equipped with turbo molecular pump (Pfeiffer vacuum) and rotary pump (Edwards vacuum), self-made cold trap, an electrochemical cell (EL CELL GmbH) and a digital mass flowmeter.

Electrochemical measurements

For the coin-type half cells, the cathode composite slurry was prepared by first mixing LCO active materials (80%), Ketjen black (10%, ECP600JD) and poly (vinylidene fluoride) (10%, PVDF) in N-methyl-2-pyrrolidone (NMP). Then, the composite slurry obtained was coated onto aluminum foil and then dried at 100 °C for 12 h under vacuum. The cathode mass loading was controlled at 2.5-3 mg cm⁻². The assembly of Li||LCO battery was completed using CR-2016 coin-type cells in an argon glove box (both of O_2 and H_2O contents below 0.1 ppm), including the cathode (diameter 12 mm),

polypropylene separator (diameter 19 mm, Celgard 2500), lithium metal foil (diameter 15.6 mm), electrolyte (LB-372: 1M LiPF₆ in fluoroethylene carbonate (FEC): methyl (2,2,2-trifluoroethyl) carbonate (FEMC)) purchased from Suzhou DoDo Chem Network. All the coin cells were evaluated using a Land CT3002A battery test system in a constant current mode between 3.0 and 4.6 V (*vs.* Li⁺/Li) at 28 °C and 45 °C. The galvanostatic intermittent titration technique (GITT) was employed after 1 cycle by applying a pulse current of 0.1 C for 20 min with a time interval of 2 h in the voltage range of 3-4.6 V on a Land CT3002A battery test system. The cyclic voltammetry (CV) measurements, and electrochemical impedance spectroscopy (EIS) tests (100 kHz to 0.01 Hz, 5 mV) were measured on an electrochemical workstation (CHI760E, Shanghai Chenghua, China).

To evaluate room-temperature OER activity in aqueous solutions, electrocatalytic measurements were performed in a three-electrode cell using a glassy carbon rotating disk electrode (diameter: 3 mm) on an electrochemical workstation (CHI760E, Shanghai Chenghua, China). The Ag/AgCl was used as the reference electrode, and Pt wire was used as the counter electrode. 1 M KOH aqueous solution with continuous O_2 bubbling was used as the electrolyte. The remaining test steps were carried out as described in the previous literature.²³ Linear sweep voltammetry was conducted at 5 mV s⁻¹.

The pouch-type graphite||LCO full cells were completed in a dry room. The anode composite slurry was prepared by mixing graphite (94.3 wt%), acetylene black (1 wt%), styrene butadiene rubber (3.5 wt%, SBR) and carboxy methyl cellulose sodium (1.2

5 |

wt%, CMC) in deionized water. Then, the composite slurry obtained was coated onto the copper foil current collector and dried at 120 °C for 24 h under vacuum. The LCO cathode composite slurry was prepared by dispersing active materials (96%), ECP600JD (2%) and PVDF (2%) in NMP, followed by coating onto an Al foil current collector and drying at 120 °C for 24 h under vacuum. The graphite||LCO full cell consisted of LCO cathode, polypropylene separator, and graphite anode. LB-372 electrolyte consumption was about ~3.5g Ah⁻¹. The mass loading of the LCO electrode was about 12 mg cm⁻² (on both sides of the Al foil current collector). The N/P ratio (negative to positive capacity ratio) was controlled around 1.05. For the formation process, the assembled pouch-type full cells were first put through two complete electrochemical cycles before the cycling specified in the main text. During this process, the pouch cells were first charged at 0.02 C for 2 h. After keeping for 5 min, the full cells were charged to 4.5 V at a constant current mode at 0.2 C and then constant voltage at 4.5 V until the current dropped to 0.02 C. Then, the pouch cells were discharged to 3.0 V at 0.2 C during the 1st discharge process. For the 2nd cycle, the pouch cells were charged to 3.85 V at 0.2 C and rested at 45 °C for 24 h to complete the formation process. The subsequent rate performance and cyclability by GCD profiles were conducted at a voltage window of 3.0-4.5 V. All pouch cell electrochemical tests were performed on a Land CT3001k battery test system.

Computational Details

Spin-polarized density functional theory (DFT) calculations were conducted by Vienna ab initio Simulation Package (VASP) 5.4.¹ The generalized gradient

approximation (GGA) and the Perdew-Burke-Ernzerhof (PBE) exchange-correlation function describe the exchange-correlation interaction.² The ion-electron interactions are described by the frozen-core projector augmented wave (PAW) method.³ The GGA+U method was employed with Coulomb repulsion $U = 3.91 \text{ eV}.^4$ The LCO bulk structure was constructed with dimensions $5.66 \times 5.66 \times 14.13$. To investigate the effect of fluorination, one of the O atoms was substituted by F atom in bulk LCO. A $3 \times 3 \times$ 1 gamma-centered k-points grid was used for sampling the Brillouin zones in heterostructures, while $5 \times 5 \times 2$ in bulk LCO. $9 \times 9 \times 3$ gamma-centered k-points grids were used for self-consistent field (SCF) calculations in DFT. The plane wave basis energy cutoff was set to 550 eV. The convergence criteria are 0.04 eV/Å for maximum force, and 1×10^{-6} eV for energy changes. Electronic structure analysis was conducted with the help of Vaspkit package. The climbing image nudged elastic band (CI-NEB) method is conducted for simulating the migration of O atoms in pristine and F doped bulk LCO, with three climbing images between initial and final states generated through the transition state tools for VASP (VTST code).



Supplementary Scheme 1| Fluorination interfacial reconstruction scheme of 4.6 V F-

LCO derived from 4.45 V commercial P-LCO.



Supplementary Figure 1| (a,b) STEM-ABF images of F-LCO, and (c) high resolution STEM-HAADF image of F-LCO, with EDS Line scan signals of O K-edge and F Kedges across from the surface to the bulk phase of F-LCO particle. (d) Co 2p XPS spectra from F-LCO and P-LCO, after surface fluorination, the position of Co²⁺ shifts to a high binding energy and transforms from Co-O to Co-F.



Supplementary Figure 2| (a) STEM-HAADF image of P-LCO (with EDS mapping area in the STEM-HAADF image, and line scan signal counts across from the surface to the bulk phase of LCO particle). (b-d) High resolution STEM-HAADF images of P-LCO by spherical aberration corrected transmission electron microscope at different nanoscale. (e) Atomic-resolution STEM HAADF image of subsurface for P-LCO at 2 nm scale and the ordered atomic structure model features of (003) planes corresponding to the image (d) (the zone axis is [-2 -1 0]). (f-h) EELS mapping of (f) Co and (g) O and (h) their overlay from image (e). (i-p) STEM-HAADF image and corresponding EDS line scan signal counts across from the surface to the bulk phase of LCO particles,



Supplementary Figure 3| (a) Long-term cycling performance of P-LCO, F-LCO and F-LCO-90, F-LCO-400, and F-LCO-700 obtained at a high rate of 3 C in the voltage of 3.0-4.6 V (*vs.* Li⁺/Li). (b) Long-term cycling performance of multiple different P-LCO and F-LCO cathodes at 3C.



Supplementary Figure 4| The corresponding discharge midpoint voltages of F-LCO

and P-LCO for 1000 cycles obtained at 3 C.



Supplementary Figure 5 Rate performance of F-LCO and P-LCO between 3.0 and 4.6 V in multiple half cells.



Supplementary Figure 6| The GCD profiles of (a) P-LCO and (b) F-LCO at different

rates between 3.0 and 4.6 V.



Supplementary Figure 7| The GCD profiles of F-LCO and P-LCO cathodes tested at

0.2 C within 3.0-4.6 V in half-cells.



Supplementary Figure 8 Cycling performance of F-LCO and P-LCO at 1 C under (a) 28 °C and (b) 45 °C.



Supplementary Figure 9| The GCD profiles of graphite||F-LCO pouch-type full cells measured at 0.1 C, 0.5 C, and 1 C within 3.0-4.5 V.



Supplementary Figure 10| The GITT curves of P-LCO and F-LCO at 0.2 C in the

voltage of 3.0-4.6 V.



Supplementary Figure 11 Nyquist plots of Li||LCO (*vs.* Li⁺/Li) batteries for P-LCO and F-LCO at open circuit voltage (OCV) and after three cycles CV curves at 0.1 m Vs⁻¹.



Supplementary Figure 12 | Nyquist plots of F-LCO and P-LCO after 100 cycles at 3

C.



Supplementary Figure 13| (a) Raman characterization of electrolyte composition. (b,c) TOF-SIMS depth profiles of interphase fragments acquired on the cycled (b) P-LCO and (c) F-LCO cathodes, illustrating the different composition of CEI layer, retrieved from coin-type half cells after the 1st cycle at 0.2 C.



Supplementary Figure 14 The F 1s XPS spectra for analyzing the chemical composition of cycled (a) P-LCO and (b) F-LCO cathodes after 1000 cycles. The significant LiF generated by LiPF_6 decomposition suggests that P-LCO interfacial instability triggers a sustained side reaction, while F-LCO exhibits a relatively stable low LiF signal.



Supplementary Figure 15 The O 1s XPS spectra for chemical composition analysis of the cycled (a) P-LCO and (b) F-LCO cathodes after 1000 cycles. The relative strength of the lattice oxygen signal at 120 s qualitatively shows that the P-LCO surface produces the thicker CEI layer than that of F-LCO.



Supplementary Figure 16 Normalized TOF-SIMS depth profiles of representative species on cycled F-LCO cathodes after 1000 cycles. The maximum normalized intensity of CoOF⁻ lies between LiF_2^- and CoF_3^- , indicating the formation of a Co-O-F-heterogeneous bonding interface between LiF and LCO.



Supplementary Figure 17| The XRD patterns of P-LCO and F-LCO.



Supplementary Figure 18 (a) EELS line scan for F-LCO under high magnification STEM-HAADF mode and (b) 3D EELS spectra of Co *L*-edges after being normalized to L_3 -edge.



Supplementary Figure 19 (a) STEM-HAADF image of P-LCO particle after FIB fabrication. The lines in the figure represent the EELS line-scan. (b,c) EELS spectra of O K-edges in (b) and Co *L*-edges in (c) collected from P-LCO particle. (d) 3D EELS spectra of Co *L*-edges after being normalized to L_3 -edge of P-LCO particle.



Supplementary Figure 20| Normalized EELS spectra of Co *L*-edge (a) P-LCO and (b) F-LCO. (c) Perpendicular-mode EPR spectra of P-LCO and F-LCO electrodes at powder state. (d-e) OER electrocatalytic performances of P-LCO and F-LCO in 1 M KOH aqueous solution at room temperature. (d) Linear sweep voltammetry (LSV) curves at a scan rate of 5 mV s⁻¹. Potentials are *vs.* reversible hydrogen electrode (RHE). (e) Measured overpotential at 6 mA cm⁻² current density. (f-g) O K-edge SXAS spectra of (f) P-LCO and (g) F-LCO in different states of charge under TEY modes. (h-i) XPS analysis of (h) O 1s and (i) F 1s for P-LCO and F-LCO cathodes under 0.2 C charge/discharge state after 20 cycles.



Supplementary Figure 21 | XPS analysis of (a-b) F 1s, (c-d) P 2p, and (e-f) O 1s for

different LCO cathodes.



Supplementary Figure 22 |Spin density of LCO bulk with one substituted F atom.

Spin-polarized area is represented by yellow color.



Supplementary Figure 23 Diffusion pathway of O atom in bulk LCO with one substituted F atom. (a) Initial state, (b) transition state, and (c) final state are depicted with Li, O, Co, and F atoms represented by green, red, blue, and light blue spheres, respectively. The diffused O atom is represented by the yellow sphere.

Supplementary Table S1. Performance comparison of cycling performance of our F-LCO with

the reported 4.6 V LCO cathodes (vs. Li^+/Li).

Modification strategy	Rate (C/mA g ⁻¹)	Capacity	Retention	Refs.
Hydrothermal assisted Li/Al/F- modified LCO	27.4	171 mA h g ⁻¹ after 200 cycles	82.2% after 200 cycles	5
Ti/Al/Mg co-doped LCO	137	174 mA h g ⁻¹ after 100 cycles	86% after 100 cycles	6
Al/Ti bulk-doped and Mg surface-doped LCO	137	170 mA h g ⁻¹ after 200 cycles	80.2% after 200 cycles	7
LiMn _{1.5} Ni _{0.5} O ₄ coated LCO	50	182 mA h g ⁻¹ after 100 cycles	80.0% after 200 cycles	8
AlZnO coated LCO	185	121 mA h g ⁻¹ after 500 cycles	65.7% after 500 cycles	9
LATP coated LCO	137	180 mA h g ⁻¹ after 100 cycles	85.8% after 100 cycles	10
P/Ni co-doped LCO	137	188 mA h g ⁻¹ after 100 cycles	92.6% after 100 cycles	11
Se surface-doped LCO	70	189 mA h g ⁻¹ after 120 cycles	86.7% after 120 cycles	12
Trace SO ₂ in-situ modified LCO	280	176 mA h g ⁻¹ after 100 cycles	88% after 100 cycles	13
Al and F gradient-doped LCO	100	170.8 mA h g ⁻¹ after 200 cycles	86.9% after 200 cycles	14
Li-Al-PO ₄ coated LCO	137	180.4 mA h g ^{-1} after 200 cycles	88.6 after 200 cycles	15
LiAlH ₄ treated LCO	190	143.7 mA h g^{-1} after 500 cycles	71.6% after 500 cycles	16
Co _x B _y coated Mg-doped LCO	27 0	185.5 mA h g ^{-1} after 100 cycles	94.6% after 100 cycles	17
Mg ²⁺ and (PO ₄) ³⁻ co-doped LCO	270	137.5 mA h g^{-1} after 100 cycles	82.4% after 1000 cycles	18
LiCoPO4 coated LCO	2C/400	153 mA h g^{-1} after 1000 cycles	75% after 1000 cycles	19
MgF ₂ doped LCO	5C/1350	130 mA h g ⁻¹ after 1000 cycles	86.4% after 1000 cycles	20
V-doped LCO	5C/1350	110 mA h g ⁻¹ after 200 cycles	93.4% after 200 cycles	21
Se coated Mg-doped LCO	2C/400	128 mA h g ⁻¹ after 1000 cycles	68.5% after 1000 cycles	22

	for charge			
	and			
	5C/1000			
	for			
	discharge			
Lathurized LCO	5C/1000	170 mA h g^{-1} after 600 cycles,	89.4% after 600 cycles,	23
		167 mA h g ⁻¹ after 100 cycles	99.0% after 100 cycles,	
		165 mA h g ⁻¹ after 300 cycles	96.0% after 300 cycles,	
F-LCO	3C/822	162 mA h g ⁻¹ after 500 cycles	94.3% after 500 cycles,	This work
		159 mA h g ⁻¹ after 700 cycles	94.3% after 700 cycles,	
		155 mA h g^{-1} after 1000 cycles	91.9% after 1000 cycles	

Doping elements	Proportions (mol%)	
Ti	4.59	
Mg	0.16	
Al	0.61	
В	0.57	

Supplementary Table S2. Doping element composition of commercial P-LCOs from ICP-OES.

Supplementary references

- 1. G. K. a. J. Furthmüller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, 54, 11169–11186.
- 2. K. B. a. M. E. J. P. Perdew, *Phys. Rev. Lett.*, 1996, 77, 3865-3868.
- 3. P. E. Blüochl, Phys. Rev. B: Condens. Matter Mater. Phys., 1994, 50, 17953-17979.
- C. L. Sun, X. B. Liao, F. J. Xia, Y. Zhao, L. Zhang, S. Mu, S. S. Shi, Y. X. Li, H. Y. Peng, G. Van Tendeloo, K. N. Zhao and J. S. Wu, *ACS Nano*, 2020, 14, 6181-6190.
- 5. J. Qian, L. Liu, J. Yang, S. Li, X. Wang, H. L. Zhuang and Y. Lu, *Nat. Commun.*, 2018, 9, 4918.
- J.-N. Zhang, Q. Li, C. Ouyang, X. Yu, M. Ge, X. Huang, E. Hu, C. Ma, S. Li, R. Xiao, W. Yang, Y. Chu, Y. Liu, H. Yu, X.-Q. Yang, X. Huang, L. Chen and H. Li, *Nat. Energy*, 2019, 4, 594-603.
- L. Wang, J. Ma, C. Wang, X. Yu, R. Liu, F. Jiang, X. Sun, A. Du, X. Zhou and G. Cui, *Adv. Sci.*, 2019, 6, 1900355.
- Z. Zhu, D. Yu, Z. Shi, R. Gao, X. Xiao, I. Waluyo, M. Ge, Y. Dong, W. Xue, G. Xu, W.-K. Lee, A. Hunt and J. Li, *Energy Environ. Sci.*, 2020, 13, 1865-1878.
- 9. T. Cheng, Z. T. Ma, R. C. Qian, Y. T. Wang, Q. Cheng, Y. C. Lyu, A. M. Nie and B. K. Guo, *Adv. Funct. Mater.*, 2021, **31**, 2001974.
- Y. Wang, Q. Zhang, Z. C. Xue, L. Yang, J. Wang, F. Meng, Q. Li, H. Pan, J. N. Zhang, Z. Jiang, W. Yang, X. Yu, L. Gu and H. Li, *Adv. Energy Mater.*, 2020, **10**, 2001413.
- 11. N. Qin, Q. Gan, Z. Zhuang, Y. Wang, Y. Li, Z. Li, H. Iftikhar, C. Zeng, G. Liu, Y. Bai, K. Zhang and Z. Lu, *Adv. Energy Mater.*, 2022, **12**, 2201549.
- 12. Z. Zhu, H. Wang, Y. Li, R. Gao, X. Xiao, Q. Yu, C. Wang, I. Waluyo, J. Ding, A. Hunt and J. Li, *Adv. Mater.*, 2020, **32**, e2005182.
- X. Tan, T. Zhao, L. Song, D. Mao, Y. Zhang, Z. Fan, H. Wang and W. Chu, *Adv. Energy Mater.*, 2022, **12**, 2200008.
- W. Huang, Q. Zhao, M. Zhang, S. Xu, H. Xue, C. Zhu, J. Fang, W. Zhao, G. Ren, R. Qin, Q. Zhao, H. Chen and F. Pan, *Adv. Energy Mater.*, 2022, 12, 2200813.
- X. Wang, Q. Wu, S. Li, Z. Tong, D. Wang, H. L. Zhuang, X. Wang and Y. Lu, *Energy Storage Mater.*, 2021, **37**, 67-76.
- P. Wang, Y. Meng, Y. Wang, L. Chen, Z. Zhang, W. Pu, J. Li, C. Yang and D. Xiao, *Energy Storage Mater.*, 2022, 44, 487-496.
- J. Chen, H. Chen, S. Zhang, A. Dai, T. Li, Y. Mei, L. Ni, X. Gao, W. Deng, L. Yu, G. Zou, H. Hou, M. Dahbi, W. Xu, J. Wen, J. Alami, T. Liu, K. Amine and X. Ji, *Adv. Mater.*, 2022, 34, 2204845.
- W. Kong, D. Zhou, Q. Zhang, D. Wong, K. An, C. Schulz, N. Zhang, J. Zhang and X. Liu, *Adv. Funct. Mater.*, 2022, **33**, 2211033.
- X. R. Yang, C. W. Wang, P. F. Yan, T. P. Jiao, J. L. Hao, Y. Y. Jiang, F. C. Ren, W. G. Zhang, J. M. Zheng, Y. Cheng, X. S. Wang, W. Yang, J. P. Zhu, S. Y. Pan, M. Lin, L. Y. Zeng, Z. L. Gong, J. T. Li and Y. Yang, *Adv. Energy Mater.*, 2022, **12**, 2200197.
- W. Kong, J. Zhang, D. Wong, W. Yang, J. Yang, C. Schulz and X. Liu, *Angew. Chem., Int. Ed.*, 2021, 60, 27102-27112.
- W. J. Kong, D. Wong, K. An, J. C. Zhang, Z. H. Chen, C. Schulz, Z. J. Xu and X. F. Liu, *Adv. Funct. Mater.*, 2022, **32**, 2202679.
- 22. A. Fu, Z. Zhang, J. Lin, Y. Zou, C. Qin, C. Xu, P. Yan, K. Zhou, J. Hao, X. Yang, Y. Cheng, D.-

Y. Wu, Y. Yang, M.-S. Wang and J. Zheng, Energy Storage Mater., 2022, 46, 406-416.

M. Z. Cai, Y. H. Dong, M. Xie, W. J. Dong, C. L. Dong, P. Dai, H. Zhang, X. Wang, X. Z. Sun,
S. N. Zhang, M. Yoon, H. W. Xu, Y. S. Ge, J. Li and F. Q. Huang, *Nat. Energy*, 2023, 8, 159-168.