Electronic Supplementary Information

Morphology-controlled green synthesis of Tellurium Nanostructures

and application of Te/MXene hybrid structure

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Figure S1. Experimental method and operation process of synthesizing Te NWs



Figure S2. Wide-range XPS spectra of Te nanotube prepared at 90 °C. Obvious Te 3d peaks are demonstrated by the XPS spectra of Te nanotubes, similar to those of Te nanowires.



Figure S3. High-resolution XPS spectrum of the C 1s peak for Te/C nanocables.



Figure S4. (a) OM image of the synthesized tellurene on SiO_2 substrate. Scale bar, 30 µm. (b) SEM image of synthesized tellurene. Scale bar, 5 µm. (c) Raman spectra of the synthesized tellurene.

The images in Figure S4 exhibit the characterizations of the synthesized 2D tellurene. F igure S4a is the OM image of tellurene on a SiO₂ substrate, in which various uniform 2D nanosheets were observed. The SEM image in Figure S4b confirms the lateral dimension of tellurene $\sim 10 \mu m$. Raman spectroscopy was used to explore the crystal structure of the o btained tellurene, and the spectra are shown in Figure S4c. Three active modes located at 92, 121, and 141 cm⁻¹ are donated as the E1, A1, and E2 vibration modes, respectively, c onsistent with the reported works.¹⁻³



Figure S5. Transfer curve of the synthesized Te NWs film. $V_d=0.1$ V.

For the transfer curve in Figure S5, the drain current slightly changed with the increase of gate voltage from -20 V to 20 V, and high drain current could be obtained at low drain voltage, indicating the high metallic conductivity of Te NWs. As reported, high junction resistance exists between individuals in nanotube films⁴ and the poorly conductive amorphous carbon shells of nanocables decrease the conductivity of Te nanocable film⁵, so Te NWs were employed for the application of transparent conductive electrodes in this work.



Figure S6. (a) OM image of the synthesized $Ti_3C_2T_x$ flakes on SiO₂ substrate. Scale bar, 2 um. (b) SEM images of synthesized tellurene. Scale bar, 2 μ m. (c) XRD pattern of the synthesized $Ti_3C_2T_x$.

The investigations of the synthesized $Ti_3C_2T_x$ flakes are shown in Figure S6. The OM image in Figure S6a and SEM image in Figure S6b indicate the exfoliated large-area and thin $Ti_3C_2T_x$ flakes. The XRD pattern in Figure S6c further confirms the high quality of the achieved $Ti_3C_2T_x$ flakes, in which a weak peak at 39° is observed.^{6, 7}



Figure S7. Schematic of the fabrication process of Te NWs/MXene hybrid structures.

Figure S7 is the schematic of fabricating Te NWs/MXene hybrid structure. Firstly, the substrate was cleaned with ethanol (C_2H_5OH) and deionized (DI) water. Then, in sequence, Te NWs and MXene solutions were spray-coated on the prepared substrate.



Figure S8. Sheet resistances and transmittances of Te and Te NWs/MXene hybrid structures with different solution concentrations of Te NWs.

The sheet resistances and transmittances at 550 nm of pure Te and Te NWs/MXene hybrid structures were measured and summarized in Fig. S8. With the spray-coating of the MXene layer, significant performance enhancements were demonstrated by Te/MXene hybrid structures with lower sheet resistances and similar transmittances. When the solution concentration of Te NWs is 10 mg/mL, the sheet resistance of Te film is difficult to measure, so 20 mg/mL is the minimum concentration investigated in this work.



Figure S9. Sheet resistances and transmittances of Te NWs/MXene hybrid structure with different solution concentrations of MXene. Te NWs/MXene1: Te NWs/MXene hybrid structure with one layer of MXene. Te NWs/MXene2: Te NWs/MXene hybrid structure with two layers of MXene.

Material	Method	Sheet resistance (Ω/sq)	Transmittance (%)	Refs.
$Ti_3C_2T_x$ in water	Spray- coating	1300	68	8
$Ti_3C_2T_x$ in ethanol	Spray- coating	3500	58	8
rGO	Filtration	565	59	9
graphene	CVD growth	280	80	10
Te/MXene	Spray- coating	156	70	This work

Table S1 A comparison of various transparent conductive electrodes.

rGO: reduced graphene oxide.

Various transparent conductive electrodes based on nanomaterials are summarized in Table 1. Compared with pure $Ti_3C_2T_x$ TCEs, the Te/MXene hybrid structure shows a lower sheet resistance with a higher transmittance, which is supposed to be attributed to (1) the Te NWs distributed across the MXene flakes act as conductive bridges and reduce the contact resistance.¹¹ (2) the Te/MXene hybrid structure could be a resistivity in parallel problems, the total resistance of Te/MXene TCE is lower than every individual component in the hybrid conductive structures.¹² Therefore, compared to pure MXene film, Te/MXene hybrid TCE demonstrates a better performance.

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