

## Supplementary information

### **Intensive harmonized synapses with amorphous Cu<sub>2</sub>O-based memristors using ultrafine Cu nanoparticle sublayers formed *via* atomically controlled electrochemical pulse deposition**

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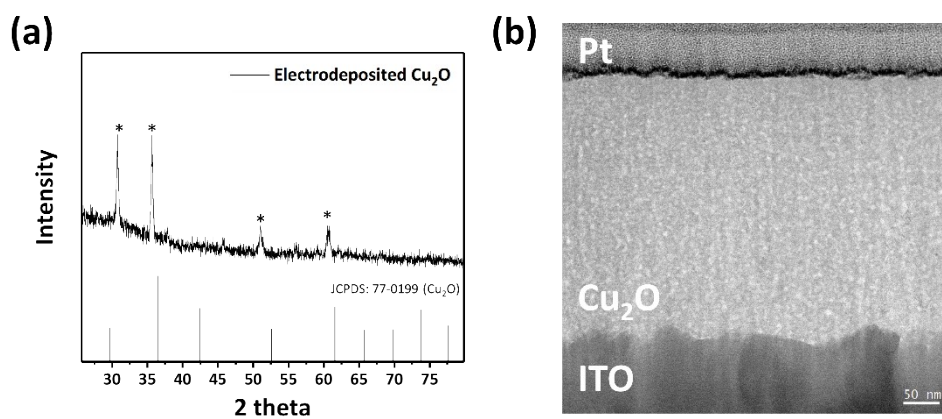
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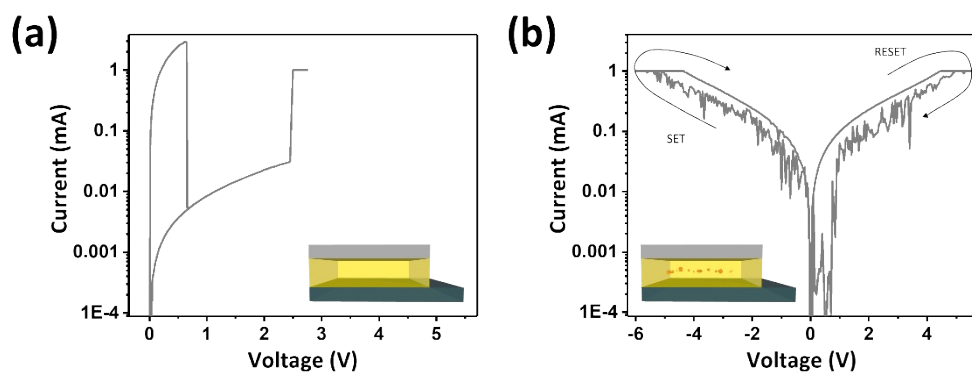
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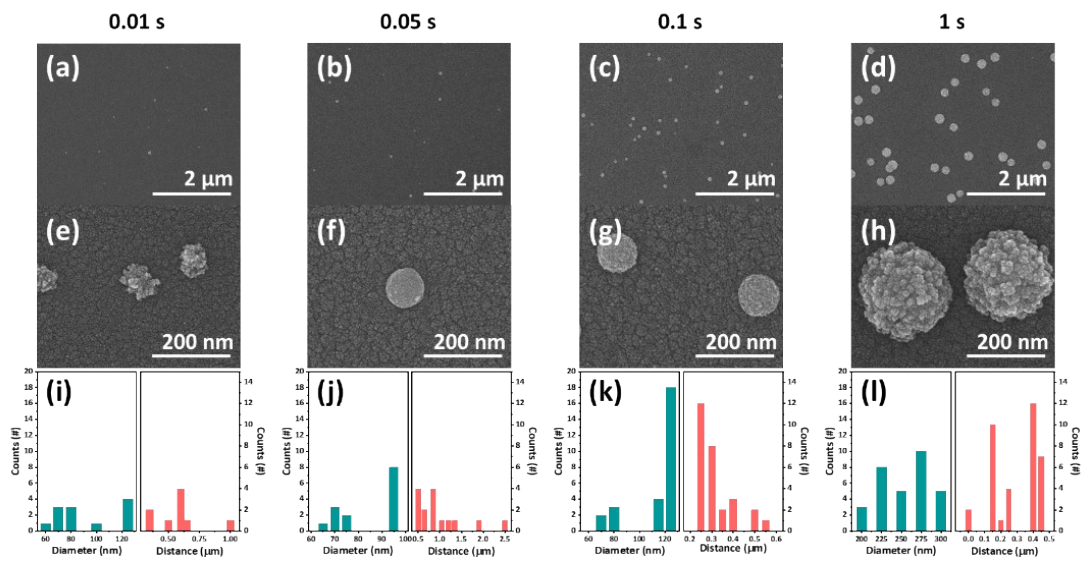
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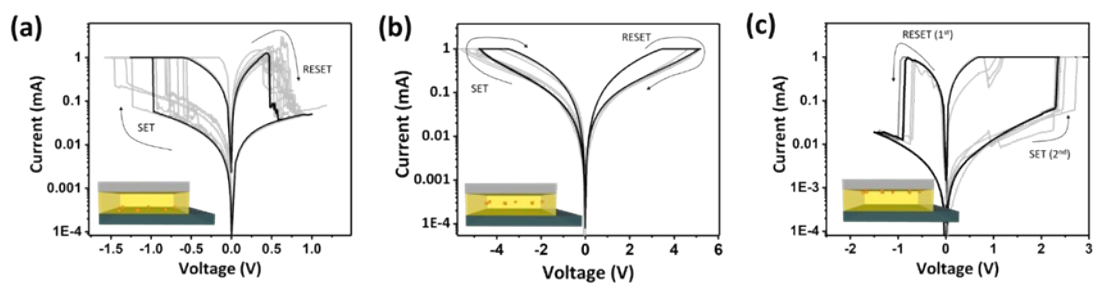
**Fig. S1** (a) X-ray diffraction patterns of electrodeposited  $\text{Cu}_2\text{O}$  film. The film is based on an amorphous matrix and has no peaks matching crystalline  $\text{Cu}_2\text{O}$ . The four-star marks are patterns of ITO substrate. (b) A transmission electron microscopy (TEM) image of electrodeposited  $\text{Cu}_2\text{O}$  film. No grain boundaries are observed.



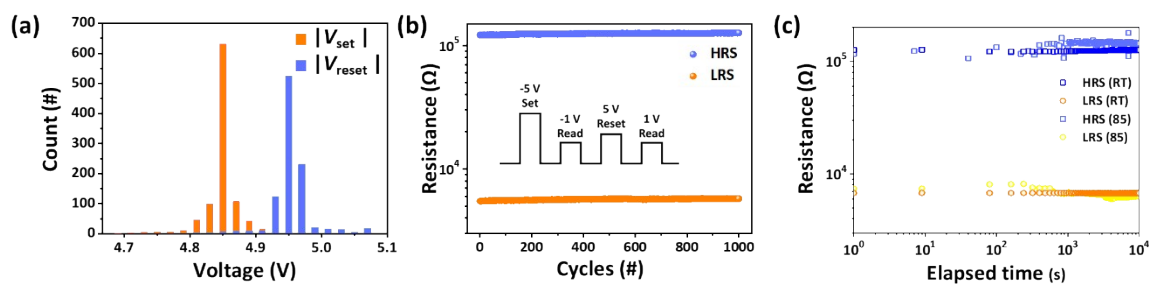
**Fig. S2** Representative resistive switching behaviors of (a) ITO/Cu<sub>2</sub>O/Pt device and (b) ITO/Cu<sub>2</sub>O/Sputtered Cu NPs/Cu<sub>2</sub>O/Pt device. (a) exhibits behaviors of unipolar resistive switching and electroforming-free. (b) exhibits a bipolar and noisy resistive switching.



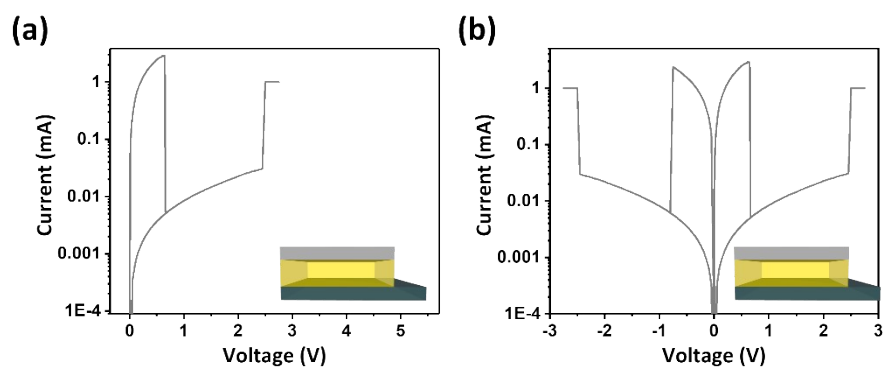
**Fig. S3** SEM images showing the time evolution of electrochemically generated Cu nanoparticles on ITO substrates.



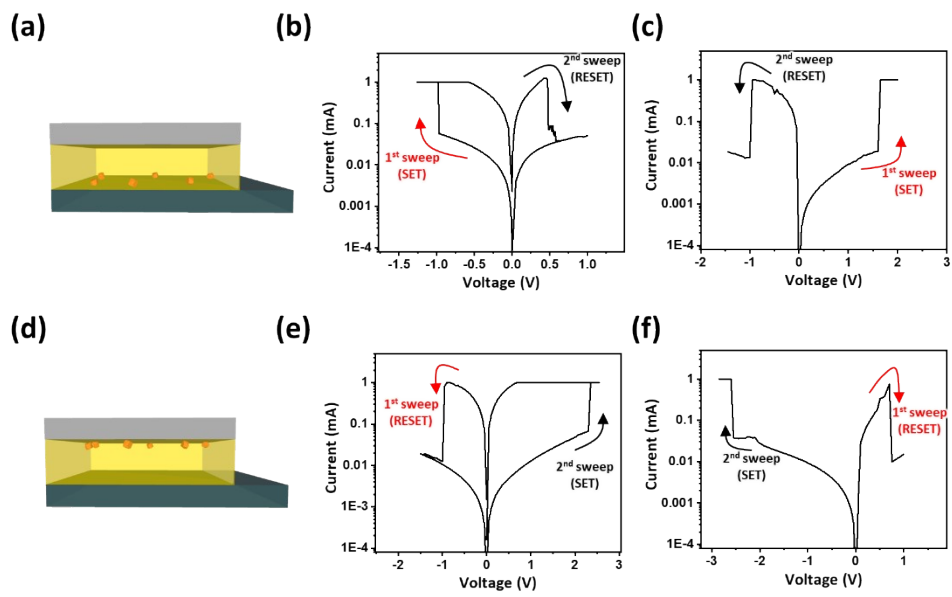
**Fig. S4** 1st, 10th, 20th ... 100th RS cycles under the IV-sweep mode for the device of (a) BU-Cu NPs, (b) CU-Cu NPs, and (c) TU-Cu NPs. The compliance current ( $I_{CC}$ ) was set 1 mA.



**Fig. S5** Electrical performances of the ITO/Cu<sub>2</sub>O/U-Cu NPs/Cu<sub>2</sub>O/Pt RRAM device: (a) The set/reset voltage distribution of 1000 repeated RS cycles. (b) Endurance test of HRS/LRS distribution of 1000 cycles at RT. (c) Retention test of HRS/LRS for 10<sup>4</sup> s at RT and 85 °C.



**Fig. S6** The unipolar set-reset characterization of pristine ITO/Cu<sub>2</sub>O/Pt device in (a) only positive sweep and (b) both positive and negative sweep.



**Fig. S7** (a) The schematic structure of BU-Cu NPs. The set-reset behavior of BU-Cu NPs with (b) negative set-positive reset and (c) positive set-negative reset. (d) The schematic structure of TU-Cu NPs. The set-reset behaviors of TU-Cu NPs with (e) negative reset-positive set and (f) positive reset-negative set.



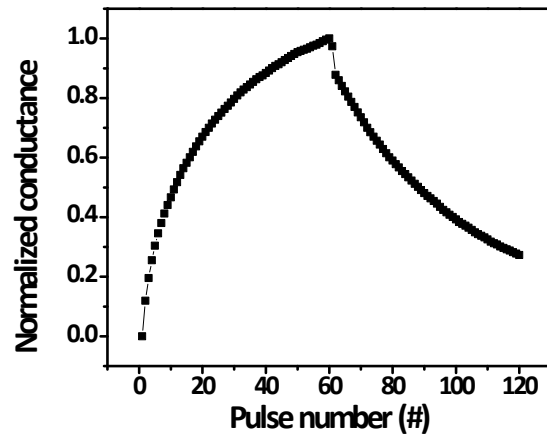
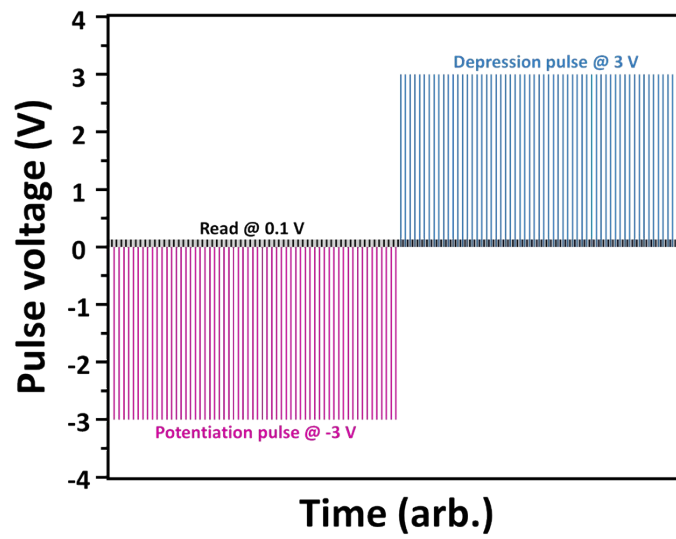
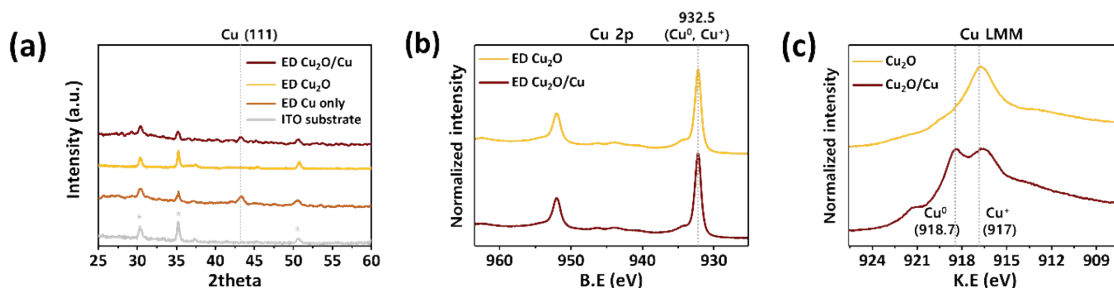


Fig. S8 LTP and LTD updates pristine  $\text{Cu}_2\text{O}$  RRAM device.



**Fig. S9** The employed pulse trains to test synaptic plasticity. The potentiation pulse and the depression pulse were applied after the read pulse, respectively, and a total of 60 potentiation pulses and 60 depression pulses were used. The pulse width and interval were set to 10  $\mu$ s and 50  $\mu$ s, respectively.

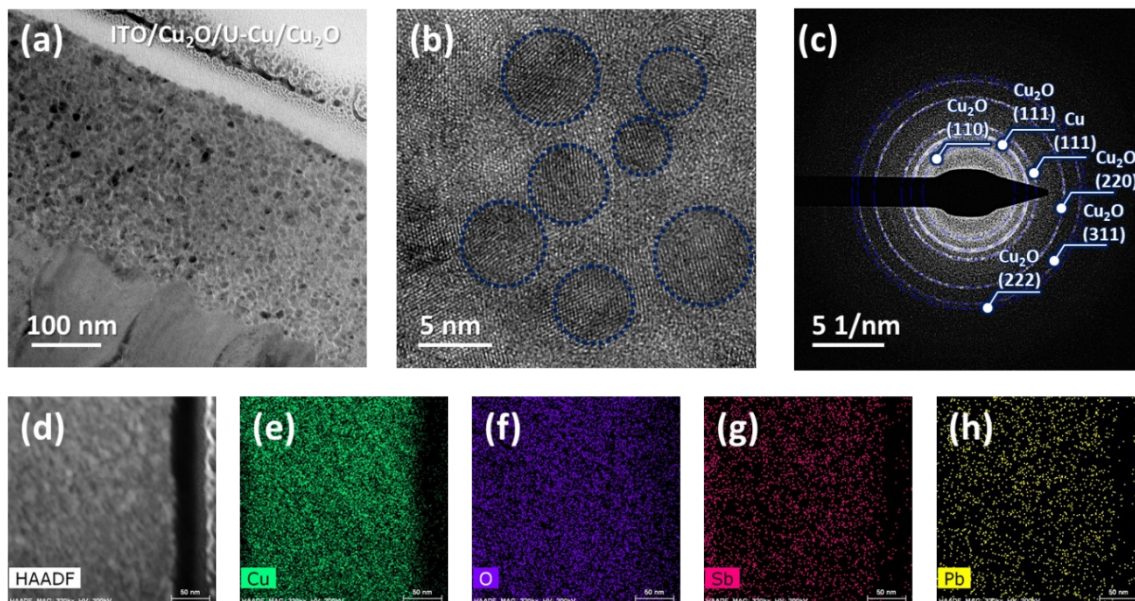


**Fig. S10** (a) XRD analysis of the electrodeposited  $\text{Cu}_2\text{O}/\text{Cu}$ ,  $\text{Cu}_2\text{O}$ , and  $\text{Cu}$  films together with ITO reference:  $\text{Cu}_2\text{O}$  (yellow line, ED @ -0.4 V vs vs.  $\text{Ag}/\text{AgCl}$ ) and  $\text{Cu}$  film (orange line, ED @ -1 V vs vs.  $\text{Ag}/\text{AgCl}$ ). (b) and (c) XPS and AES analysis of electrodeposited  $\text{Cu}_2\text{O}$  film (yellow line) and  $\text{Cu}_2\text{O}$  film with  $\text{Cu}$  nanoparticles (brown line).

First, to analyze the presence and phase of the formed  $\text{Cu}$ -based phase at  $-0.45$  V or  $-1$  V vs.  $\text{Ag}/\text{AgCl}$ , XRD analysis of the thin film deposited with an intentionally high thickness (300 nm) was additionally performed (Fig. S10a). In the XRD data, specific ED  $\text{Cu}_2\text{O}$  peaks were not observed for the embedded nanoparticles in the amorphous matrix (yellow line). However, the ED  $\text{Cu}$  metal with strong crystallinity was confirmed to have a (111) peak at  $43.3^\circ$ , which was also produced on  $\text{Cu}_2\text{O}$  by the additional pulse deposition method (red line).

Second, the XPS is one of the best methods for investigating the chemical composition and electronic structure of surface transformations. Through XPS analysis, as shown in **Figs. S10b and 1c**, the surfaces of  $\text{Cu}_2\text{O}$  deposited at  $-0.4$  V and pulse electrodeposited  $\text{Cu}_2\text{O}/\text{Cu}$  were compared. The main peaks of  $\text{Cu } 2p_{3/2}$  consisted of  $\text{Cu}^0$  at 932.61 eV,  $\text{Cu}^+$  at 932.28 eV and  $\text{Cu}^{2+}$  at  $934.7 \pm 0.1$  eV. For both the ED  $\text{Cu}_2\text{O}$  and ED  $\text{Cu}_2\text{O}/\text{Cu}$ , only a single peak at 932.5 eV was observed, and no additional satellite peaks related to  $\text{Cu}^{2+}$  ( $\text{CuO}$  phase) were observed. However, the peak at 932.5 eV can be assigned to either  $\text{Cu}^+$  or  $\text{Cu}^0$  because their binding energies almost overlap in the spectrum of  $\text{Cu } 2p_{3/2}$ .

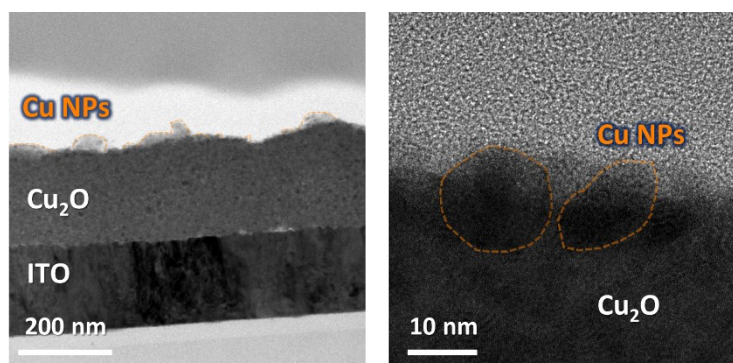
To further confirm this contribution, we examined the  $\text{Cu}L_{3M_{4.5}M_{4.5}}$  Auger peaks, from which the modified Auger parameter was calculated (**Fig. S10c**). This parameter also allowed us to distinguish between  $\text{Cu}^+$  and  $\text{Cu}^0$ , which show almost the same chemical shift as  $\text{Cu}2p$ . Based on the pristine  $\text{Cu}_2\text{O}$  peak,  $\text{Cu}_2\text{O}/\text{Cu}$  had an additional shoulder at 918.7 eV, originating from  $\text{Cu}$ . The distinct separation of the two shoulders indicates that  $\text{Cu}_2\text{O}$  and  $\text{Cu}$  were simultaneously exposed on the surface of the active layer.



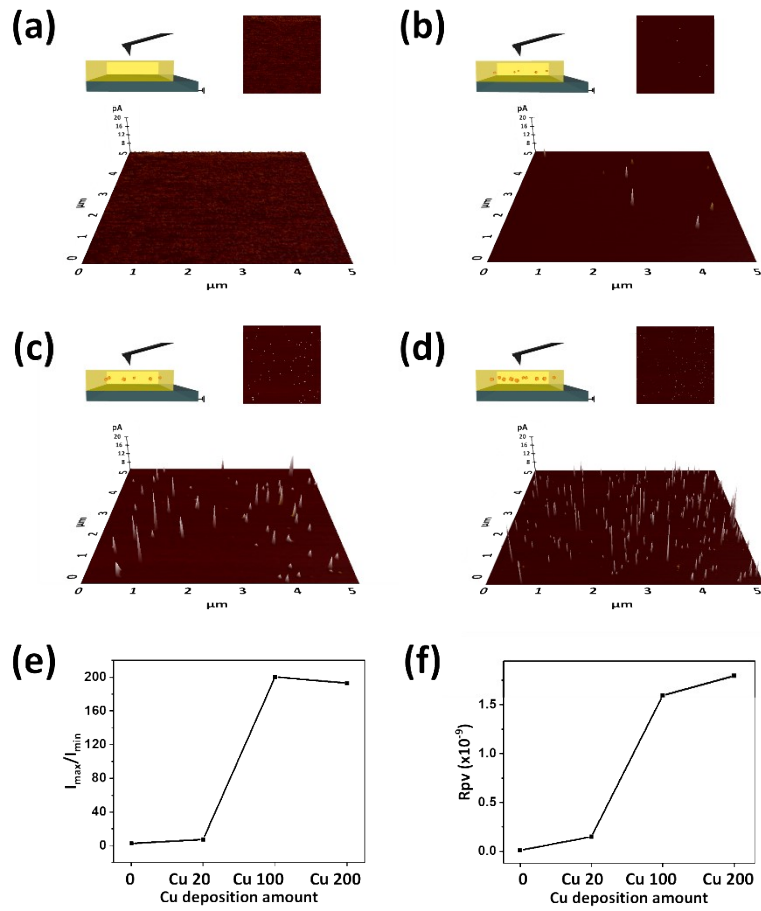
**Fig. S11** (a) The low magnified and (b) high magnified HRTEM images. (c) SAED pattern of  $\text{Cu}_2\text{O}$  active layer. (d) The HAADF-STEM image of  $\text{Cu}_2\text{O}$  active layer and elemental mapping for (e) Cu, (f) O, (g) Sb, and (h) Pb elements.

Third, the pulse-deposited  $\text{Cu}_2\text{O}$  active layer with nanoparticles embedded in an amorphous matrix was confirmed through the TEM image and energy-dispersive X-ray (EDX) analysis, as shown in **Fig. S11**. A SAED pattern of pulse deposited  $\text{Cu}_2\text{O}$  active layer (**Fig. S11c**) revealed six sets of rings, indicating the coexistence of Cu and  $\text{Cu}_2\text{O}$  phases. In addition, the Sb and Pb dopants were evenly dispersed, leading to an amorphous growth (**Figs. 11d–h**).

To clearly observe the Cu NPs, HR-TEM analysis was performed with ITO/ $\text{Cu}_2\text{O}$ /U-Cu. The Cu NPs obtained from the electrochemical deposition were uniformly generated on the surface (**Fig. S12**).



**Fig. S12** HRTEM image of ITO/ $\text{Cu}_2\text{O}$ /U-Cu NPs.



**Fig. S13.** Conductive atomic force microscope (C-AFM) analyses of ITO/Cu<sub>2</sub>O/U-Cu NPs/Cu<sub>2</sub>O/Pt devices in HRS state. The local current images from (a) pristine Cu<sub>2</sub>O device, (b) U-Cu 20 device, (c) U-Cu 100 device, and (d) U-Cu 200 device. (e) The ratio of maximum current to minimum current. The values are 2.55, 7.66, 200.25, and 192.85. (d) The peak-to-valley of the selected region ( $R_{pv}$ ), which represents the difference in  $I_{max}$  and  $I_{min}$ , had values of 9.203E-12A, 1.49E-10A, 1.595E-9A, and 1.795E-9A. The Cu 100 device and the Cu 200 device have candidate sites that can concentrate the injected electric field.

**Table S1** Performance comparison.

Structure	Active layer	Switching type	Set voltage (V)	Reset voltage (V)	Endurance (Cycle)	Retention (Time)	Nonlinearity (%) (LTP/LTD)	Accuracy (%)	Ref.
ITO/Cu <sub>2</sub> O /U-Cu/Cu <sub>2</sub> O Pt	Cu <sub>2</sub> O	Bi-polar	-3	+3	2000	10 <sup>4</sup> s	2.43/0.13	<b>85.17</b>	This work
Ta /ZnSnO/TiN	ZnSnO	Bi-polar	-1	1.5	2000	10 <sup>4</sup> s	35.51/22.21	45	1
Pt/a-Ta <sub>2</sub> O <sub>5</sub> /TiN	Ta <sub>2</sub> O <sub>5</sub>	Bi-polar	-2.4	1.3	100	10 <sup>4</sup> s		85	2
Pt/HfAlO <sub>x</sub> /TiN	HfAlO <sub>x</sub>	Bi-polar	-0.8	1	200	10 <sup>4</sup> s	211.9/89.12	44.9	3
Al/HfO <sub>2</sub> /Ti/TiN	HfO <sub>2</sub>	Bi-polar	2	-1.5	240	10 <sup>2</sup> s	16.53/0.99	10	4
Pt/Li <sub>2</sub> TiO <sub>3</sub> /Pt	LTO	Bi-polar	-2	2	2000	10 <sup>5</sup> s	5.25/1.45	83	5
Pt/AlN/WN/Cu	AlN	Bi-polar	0.7 V	-0.4 V	500	10 <sup>4</sup> s	2.3 / 3.5		6
ITO/Al:HfO <sub>x</sub> /HfO <sub>x</sub> /Al	HfO <sub>x</sub>	Bi-polar	1.4 V	-2.4 V		10 <sup>4</sup> s	8.9 / 5.6		7

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