Electronic Supporting Information

Observation of Magnetic Vortex Configuration in Non-stoichiometric Fe₃O₄ Nanospheres

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S1: SEM and XRD of α -Fe₂O₃ (JCPDS: 89-0597)



S2: Size distribution of Fe₃O₄ NSs



S3: FT-IR of all the samples, i.e. LERs, SERs, and NSs. The detected peaks 549 cm⁻¹ and 976 cm⁻¹ represent iron oxide (Fe₃O₄ here) and phosphate anion, respectively



S4: Simulated M-H hysteresis loop varying the size of NSs.



Cylindrical rod-shaped spin configuration in vortex core along Z-axis (perpendicular to the XY plane)

S5: (a) Simulated M-H hysteresis loop varying number of spheres at NSs diameter 700 nm. (b) Upper row contains full-sphere spin configuration rotating around vortex core axis (yellow arrow), i.e., Z-axis (perpendicular) and lower row with corresponding cut-sphere (hemisphere) to show rod-shaped vortex evolution towards Z-axis (parallel yellow dotted line).



S6: Focus-series of Fresnel contrast imaging technique for two isolated sample NS1 and NS2 and their corresponding reconstructed phase along different field axis.



S7: Hysteresis loss of LERs and SERs measured at 300 K applying field of +/- 50 kOe



S8: Energy profile of NSs obtained by means of micro-magnetic simulation where magnetostatic energy (demagnetizing) is dominated by exchange energy.

Sample	Lattice Parameter	Phase Analysis				Magnetic Measurement		
	(nm)	XRD(±1)		Mössbauer(±2)		Verwey	Ms	
	±0.0005	Fe ₃ O ₄ (%);	Fe ⁰ (%)	Fe ₃ O ₄ (%)	Fe ⁰ (%)	Transition	(emu/g)	
						Temp.		
						(Tv)		
Long	Fe ₃ O ₄	85.5	14.5	84	16		132	
Rods	=0.838							
	Fe ⁰ = 0.286							
Short	Fe ₃ O ₄	86	12.4	87	11	122 K	108	
rods	=0.838							
	Fe ⁰ = 0.286							
Sphere	Fe ₃ O ₄	100	-	100	-		76	
	=0.837							

Table 1: Comparative study of lattice parameter, phase analysis, and magnetic measurement

Table 2: Magnetic properties

Sample	Ms (emu/g)	Mr (emu/g)	Hc (Oe)	Mr/Ms
LERs	132	31	232	0.23
SERs	108	17	229	0.16
NSs	70	10	21	0.14





89: Electrical measurement device.	(a) Four-probe	contact and (b) Cryostat
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Table 3. ⁵⁷ Fe Hyperfine parameters, isomer shift (I.S.), quadrupole splitting (Q.S), hyperf	ine
field (B _{hf}) and spectral area (population distributions) for the diverse iron oxide preparation	ns.

Sample	Phase	site	Hyperfine parameters				
			I.S.(mm/s)	Q.S.(mm/s)	B _{hf} (T)	Population (%)	
			±0.01	±0.01	±0.1	±2	
LERs	Fe ₃ O ₄	A(Fe ³⁺)	0.29	-0.07	49.48	18.86	
		B (Fe ³⁺ , Fe ²⁺)	0.69	-0.15	45.70	35.74	
		A(Fe ³⁺)	0.32	-0.002	49.70	26.54	
	Fe		-0.02	0	33.00	18.85	
SERs	Fe ₃ O ₄	A(Fe ³⁺)	0.32	-0.04	49.64	16.43	
		B (Fe ³⁺ , Fe ²⁺)	0.63	0.07	45.33	33.64	
		A(Fe ³⁺)	0.32	-0.002	49.70	41.17	
	Fe		-0.02	0	32.96	8.8	
NSs	Fe ₃ O ₄	A(Fe ³⁺)	0.31	-0.064	48.83	55.58	
		B(Fe ²⁺)	0.79	-0.37	46.52	21.20	
		B(Fe ³⁺)	0.38	0	47.64	23.22	

The obtained hyperfine fitting results for LERs and SERs can be discussed considering possible model of core/shell structure assuming that some sample region in the core obey the bulk Fe_3O_4

composition (i.e., stoichiometric) though the remaining portion are non-stoichiometric at vacancies/defects probably within the shell/surface region, in agreement with previous results reported elsewhere ¹. The presence of stoichiometric Fe₃O₄ in these sample is the signature of the possible Verwey phase transition, which occurs near 120 K ². Below the Verwey temperature T_V , the valence states of iron ions in both A- and B-sites are mostly stable, though above T_V an electron exchange among Fe²⁺ and Fe³⁺ ions in octahedral B-sites appears and hence these valence states get unstable, giving rise to changes in many physical properties along with structural change from monoclinic to cubic, in agreement with the clear observation of Verwey phase transition in ZFC curve, shown in Figure 4a.

Moreover, it is well known that magnetic moment (M) per formula unit for the stoichiometric Fe_3O_4 is 'M' =4 μ_B . In the present work, the part of non-stoichiometry in LERs, SERs, and NSs led to the magnetic moment 'M' = 3.76 μ_B with δ = 0.12 and hence can be represented by non-stoichiometric formula $Fe_{2.88}O_4$, 'M' = 3.68 μ_B with δ = 0.14 and represented by $Fe_{2.86}O_4$, and 'M' = 3.52 μ_B with δ = 0.16 and represented by $Fe_{2.88}O_4$ for LERs, SERs, and NSs respectively. The increased number of vacancies and surface effects from LERs to NSs is believed to be due to a rise in concentration of phosphate anions during synthesis.

References

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