## Electronic Supporting Information

## Observation of Magnetic Vortex Configuration in Non-stoichiometric $\mathrm{Fe}_{3} \mathrm{O}_{4}$ Nanospheres

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S1: SEM and XRD of $\alpha-\mathrm{Fe}_{2} \mathrm{O}_{3}$ (JCPDS: 89-0597)


S2: Size distribution of $\mathrm{Fe}_{3} \mathrm{O}_{4}$ NSs


S3: FT-IR of all the samples, i.e. LERs, SERs, and NSs. The detected peaks $549 \mathrm{~cm}^{-1}$ and $976 \mathrm{~cm}^{-1}$ represent iron oxide ( $\mathrm{Fe}_{3} \mathrm{O}_{4}$ here) and phosphate anion, respectively


S4: Simulated M-H hysteresis loop varying the size of NSs.



S5: (a) Simulated M-H hysteresis loop varying number of spheres at NSs diameter 700 nm . (b) Upper row contains full-sphere spin configuration rotating around vortex core axis (yellow arrow), i.e., Z-axis (perpendicular) and lower row with corresponding cut-sphere (hemisphere) to show rod-shaped vortex evolution towards Z-axis (parallel yellow dotted line).


S6: Focus-series of Fresnel contrast imaging technique for two isolated sample NS1 and NS2 and their corresponding reconstructed phase along different field axis.


S7: Hysteresis loss of LERs and SERs measured at 300 K applying field of $+/-50 \mathrm{kOe}$


S8: Energy profile of NSs obtained by means of micro-magnetic simulation where magnetostatic energy (demagnetizing) is dominated by exchange energy.

Table 1: Comparative study of lattice parameter, phase analysis, and magnetic measurement

| Sample | Lattice Parameter ( nm ) $\pm 0.0005$ | Phase Analysis |  |  |  | Magnetic Measurement |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  | XRD( $\pm 1$ ) |  | Mössbauer( $\pm 2$ ) |  | Verwey Transition Temp. (Tv) | $\begin{gathered} \mathrm{M}_{\mathrm{s}} \\ (\mathrm{emu} / \mathrm{g}) \end{gathered}$ |
|  |  | $\mathrm{Fe}_{3} \mathrm{O}_{4}(\%)$; | $\mathrm{Fe}^{0}$ (\%) | $\mathrm{Fe}_{3} \mathrm{O}_{4}(\%)$ | $\mathrm{Fe}^{0}$ (\%) |  |  |
| Long <br> Rods | $\begin{gathered} \mathrm{Fe}_{3} \mathrm{O}_{4} \\ =0.838 \\ \mathrm{Fe}^{0}=0.286 \end{gathered}$ | 85.5 | 14.5 | 84 | 16 |  | 132 |
| Short <br> rods | $\begin{gathered} \mathrm{Fe}_{3} \mathrm{O}_{4} \\ =0.838 \\ \mathrm{Fe}^{0}=0.286 \end{gathered}$ | 86 | 12.4 | 87 | 11 | 122 K | 108 |
| Sphere | $\begin{gathered} \mathrm{Fe}_{3} \mathrm{O}_{4} \\ =0.837 \end{gathered}$ | 100 | - | 100 | - |  | 76 |

Table 2: Magnetic properties

| Sample | Ms (emu/g) | $\mathbf{M r}(\mathbf{e m u} / \mathbf{g})$ | $\mathbf{H c}(\mathbf{O e})$ | $\mathbf{M r} / \mathbf{M s}$ |
| :--- | :--- | :--- | :--- | :--- |
| LERs | 132 | 31 | 232 | 0.23 |
| SERs | 108 | 17 | 229 | 0.16 |
| NSs | 70 | 10 | 21 | 0.14 |



S9: Electrical measurement device. (a) Four-probe contact and (b) Cryostat
Table 3. ${ }^{57} \mathrm{Fe}$ Hyperfine parameters, isomer shift (I.S.), quadrupole splitting (Q.S), hyperfine field ( $B_{h f}$ ) and spectral area (population distributions) for the diverse iron oxide preparations.

| Sample | Phase | site | Hyperfine parameters |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $\begin{aligned} & \text { I.S. }(\mathrm{mm} / \mathrm{s}) \\ & \pm 0.01 \end{aligned}$ | $\begin{aligned} & \text { Q.S.(mm/s) } \\ & \pm 0.01 \end{aligned}$ | $\begin{aligned} & \mathrm{B}_{\mathrm{hf}}(\mathrm{~T}) \\ & \pm 0.1 \end{aligned}$ | $\begin{aligned} & \text { Population (\%) } \\ & \pm 2 \end{aligned}$ |
| LERs | $\mathrm{Fe}_{3} \mathrm{O}_{4}$ <br> Fe | $\begin{aligned} & \mathrm{A}\left(\mathrm{Fe}^{3+}\right) \\ & \mathrm{B}\left(\mathrm{Fe}^{3+}, \mathrm{Fe}^{2+}\right) \\ & \mathrm{A}\left(\mathrm{Fe}^{3+}\right) \end{aligned}$ | $\begin{aligned} & \hline 0.29 \\ & 0.69 \\ & 0.32 \\ & -0.02 \end{aligned}$ | $\begin{aligned} & \hline-0.07 \\ & -0.15 \\ & -0.002 \\ & 0 \end{aligned}$ | $\begin{aligned} & 49.48 \\ & 45.70 \\ & 49.70 \\ & 33.00 \end{aligned}$ | $\begin{aligned} & 18.86 \\ & 35.74 \\ & 26.54 \\ & 18.85 \end{aligned}$ |
| SERS | $\mathrm{Fe}_{3} \mathrm{O}_{4}$ <br> Fe | $\begin{aligned} & \mathrm{A}\left(\mathrm{Fe}^{3+}\right) \\ & \mathrm{B}\left(\mathrm{Fe}^{3+}, \mathrm{Fe}^{2+}\right) \\ & \mathrm{A}\left(\mathrm{Fe}^{3+}\right) \end{aligned}$ | $\begin{aligned} & \hline 0.32 \\ & 0.63 \\ & 0.32 \\ & -0.02 \end{aligned}$ | $\begin{aligned} & \hline-0.04 \\ & 0.07 \\ & -0.002 \\ & 0 \end{aligned}$ | $\begin{aligned} & 49.64 \\ & 45.33 \\ & 49.70 \\ & 32.96 \end{aligned}$ | $\begin{array}{\|l\|} \hline 16.43 \\ 33.64 \\ 41.17 \\ 8.8 \end{array}$ |
| NSs | $\mathrm{Fe}_{3} \mathrm{O}_{4}$ | A(Fe $\left.{ }^{3+}\right)$ <br> $\mathrm{B}\left(\mathrm{Fe}^{2+}\right)$ <br> $\mathrm{B}\left(\mathrm{Fe}^{3+}\right)$ | $\begin{array}{\|l\|} \hline 0.31 \\ 0.79 \\ 0.38 \\ \hline \end{array}$ | $\begin{aligned} & \hline-0.064 \\ & -0.37 \\ & 0 \end{aligned}$ | $\begin{aligned} & 48.83 \\ & 46.52 \\ & 47.64 \end{aligned}$ | 55.58 <br> 21.20 <br> 23.22 |

The obtained hyperfine fitting results for LERs and SERs can be discussed considering possible model of core/shell structure assuming that some sample region in the core obey the bulk $\mathrm{Fe}_{3} \mathrm{O}_{4}$
composition (i.e., stoichiometric) though the remaining portion are non-stoichiometric at vacancies/defects probably within the shell/surface region, in agreement with previous results reported elsewhere ${ }^{1}$. The presence of stoichiometric $\mathrm{Fe}_{3} \mathrm{O}_{4}$ in these sample is the signature of the possible Verwey phase transition, which occurs near $120 \mathrm{~K}^{2}$. Below the Verwey temperature $T_{V}$, the valence states of iron ions in both $A$ - and $B$-sites are mostly stable, though above $T_{V}$ an electron exchange among $\mathrm{Fe}^{2+}$ and $\mathrm{Fe}^{3+}$ ions in octahedral B -sites appears and hence these valence states get unstable, giving rise to changes in many physical properties along with structural change from monoclinic to cubic, in agreement with the clear observation of Verwey phase transition in ZFC curve, shown in Figure 4a.

Moreover, it is well known that magnetic moment (M) per formula unit for the stoichiometric $\mathrm{Fe}_{3} \mathrm{O}_{4}$ is ' M ' $=4 \mu_{\text {B. }}$. In the present work, the part of non-stoichiometry in LERs, SERs, and NSs led to the magnetic moment ' M ' $=3.76 \mu_{\mathrm{B}}$ with $\delta=0.12$ and hence can be represented by nonstoichiometric formula $\mathrm{Fe}_{2.88} \mathrm{O}_{4}$, ' $\mathrm{M}^{\prime}=3.68 \mu_{\mathrm{B}}$ with $\delta=0.14$ and represented by $\mathrm{Fe}_{2.86} \mathrm{O}_{4}$, and ' M ' $=3.52 \mu_{\mathrm{B}}$ with $\delta=0.16$ and represented by $\mathrm{Fe}_{2.88} \mathrm{O}_{4}$ for LERs, SERs, and NSs respectively. The increased number of vacancies and surface effects from LERs to NSs is believed to be due to a rise in concentration of phosphate anions during synthesis.

## References

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