Electronic Supplementary Information

Two-dimensional ferromagnetic semiconductors of monolayer BiXO₃ (X = Ru, Os) with direct band gaps, large magnetic anisotropy and high Curie temperatures

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Figure S1. Electron localization function (ELF) of BiRuO₃ and BiOsO₃ monolayers. (a) The side view of the crystal structure. (b-c) The ELF of BiRuO₃ and BiOsO₃ on (001) plane. The position of the cut plane for ELF is indicated by the blue dashed line in (a).



Figure S2. Schematically illustration of the X…X (X = Ru, Os) distances for the exchange interactions $J_1(d_{J1})$, $J_2(d_{J2})$, and $J_3(d_{J3})$ in 2D BiXO₃.

Normally, the next-nearest-neighbor interaction J_2 should be larger than that of the second-next-nearest-neighbor interaction J_3 . But some exceptions show J_2 can be smaller than J_3 . One typical example is the $Cr_2Ge_2Te_6$ monolayer, whose J_2 is 0.08 meV, while its J_3 displays a larger absolute value of 0.16 meV¹. Importantly, our predicted BiRuO₃ monolayer shares the same structure with that of $Cr_2Ge_2Te_6$. In fact, the exchange interaction is sensitive to the atom distances². In our predicted systems, we believe the non-local magnetic exchange interactions, i.e. the superexchange interaction can be an important factor to affect the energy of magnetic exchange interactions between ions at different distances (see Figure S2). We notice that for BiOsO₃, the Os…Os distance for J₂ is $d_{J_2}^{Os} = 5.39$ Å and that for J₃ is $d_{J_3}^{Os} = 6.23$ Å (Table S1). For BiRuO₃, the Ru…Ru distance for J₂ is $d_{J_2}^{Ru} = 5.29$ Å and that for J₃ is $d_{J_3}^{Ru} = 6.11$ Å. The difference D_{J_2} of J₂ between the two systems is $d_{J_2}^{Os} - d_{J_2}^{Ru} = 0.10$ Å. Similarly, we can get the difference for J₃ is $D_{J_3} = 0.12$ Å. As D_{J_3} is larger than D_{J_2} , it means the J₃ interaction in BiRuO₃ monolayer could be significantly enhanced due to the relatively short Ru…Ru distance (compared with Os…Os distance). This can be one reason why the J₃ is greater than J₂ in BiRuO₃.



Figure S3. The band structures of monolayers (a-e) BiRuO₃ and (f-g) BiOsO₃ under tensile strain at PBE+U level. The Fermi level has been shifted to zero.



Figure S4. The calculated *d* orbital projected partial density of states (PDOS) near the Fermi level of Ru and Os atoms for monolayer (a) BiRuO₃ and (b) BiOsO₃. The Fermi level has been shifted to zero in each panel.



Figure S5. The *d* orbital-resolved MAE of Ru and Os atoms for (a-c) BiRuO₃ and (d-f) BiOsO₃ monolayers under strains of (a)/(d) -3%, (b)/(e) 0%, and (c)/(f) +2%.

The origin of large MAE

According to the second-order perturbation theory proposed by Wang *et al.*³, the MAE consists of two items, where the contributions of the same spin channel with upward orientation (uu) and downward orientation (dd) can be expressed as

$$MAE = E_z - E_x = \xi^2 \sum_{o, u} \frac{|\langle o \mid L_z \mid u \rangle|^2 - |\langle o \mid L_x \mid u \rangle|^2}{E_o - E_u}$$

where ξ is the SOC constant, o (u) stand for the occupied (unoccupied) states, and the energy of occupied (unoccupied) states is represented by E_o (E_u). L_z and L_x are the angular momentum operators. Likewise, the contributions of the coupling between opposite spin channel (ud) can be expressed as

$$MAE = E_z - E_x = \xi^2 \sum_{o, u} \frac{|\langle o \mid L_x \mid u \rangle|^2 - |\langle o \mid L_z \mid u \rangle|^2}{E_o - E_u}$$
(2)

Based on the equations above, the MAE is determined by matrix elements of the spin-orbital interaction between occupied and unoccupied states. The projected DOS of d orbitals of Ru/Os atoms in Fig. S4 shows that the the occupied and unoccupied spin-up d states of Ru/Os atoms are near the Fermi level. The d orbital-resolved MAE of Ru and Os atoms for strain-free and strained BiXO₃ monolayers are presented in Fig. S5. For the strain-free structures (Fig. S5b and e), the large MAE in BiXO₃ are mainly contributed by the matrix element differences between d_{xy} and d_{x2-y2} as well as d_{yz} and d_{z2} orbitals, and both contributions are positive. Under compressive strain, the contributions to the MAE from the matrix element differences d_{xy} and d_{x2-y2} as well as d_{yz} and d_{z2} orbitals are significantly increased. Consequently, the values of MAE for both structures increase as well. Under tensile strain, we can see the matrix element differences d_{xy} and d_{x2-y2} as well as d_{yz} and d_{z2} decrease, leading to the reduced values of MAE under tensile strain. Therefore, the evolution of the spin-orbital matrix element differences between d orbitals of Ru and Os atoms is responsible for the change of MAE in monolayers BiRuO₃ and BiOsO₃ under strain effect.

Table S1. The X…X (X = Ru, Os) distances for J₂ and J₃ in the BiXO₃ monolayer, as well as the difference between atom distance $D_{J_x} (D_{J_x} = d_{J_x}^{Os} - d_{J_x}^{Ru})$.

Exchange interaction	Os…Os (Å)	Ru…Ru (Å)	$D_{J_{x}(\text{\AA})}$
J_2	$d_{J_2}^{Os} = 5.39$	$d_{J_2}^{Ru} = 5.29$	$D_{J_{2=}0.10}$

J ₃	$d_{J_3}^{Os} = \frac{1}{6.23}$	$d_{J_3}^{Ru} = 6.11$	$D_{J_{3=}} 0.12$

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