Supporting Information

Zero-dimensional Cadmium-based Metal Halide with Near-Unity Photoluminescence Quantum Efficiency

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Experimental Procedures

Materials: Cadmium chloride (AR) and m-xylylenediamine were purchased from Aladdin. Antimony trichloride (99.9%) and manganese chloride (99.0%) were purchased from Macklin. Hydrochloric acid was purchased from Sinopharm Chemical Reagent Co., Ltd., and all reagents and solvents were used without further purification.

Synthesis of $(C_8H_{14}N_2)_2CdCl_6$: C₈H₁₂N₂ (1 mmol) and CdCl₂ (0.5 mmol) were dissolved in HCl (10 mL) at 90 °C and stirred for 1 hour to form a clear solution. The solution was then transferred to a programmed cooling oven, and the solution was then cooled to room temperature at a rate of 2 °C h⁻¹. The crystals were filtered and cleaned with isopropyl alcohol and dried in a furnace overnight at 30 °C. For ion-doped (C₈H₁₄N₂)₂CdCl₆ crystals, a certain content of SbCl₃ or MnCl₂ is added to the hydrochloric acid system of the raw material, or SbCl₃ and MnCl₂ are added at the same time. The next operation is the same as above.

Characterization Techniques: Powder X-ray diffraction (PXRD) data were collected on a Rigaku D/max-IIIA diffractometer with Cu K α (1.54 Å) radiation at 293 K. The photoluminescence excitation (PLE) and photoluminescence emission (PL) spectra were obtained on Edinburgh Instruments FLS 980. The temperaturedependent PL spectrum was measured by its matched temperature control system. The PLQY measurement and time-resolved PL (TPRL) spectra were recorded on an Edinburgh FLS 1000. The diffuse reflectance spectra of the microcrystalline powders were recorded on a UV/Vis spectrophotometer (V670-Jasco spectrophotometer) calibrated by simultaneous measurement of the substance (BaSO4 powder) and then converted to absorbance using the Kubelka-Munk theory. The actual doping concentration was measured by inductively coupled plasma atomic emission spectroscopy (ICP-AES, optimal 8000). XPS measurements were performed on Thermal Scientific Escalab 250 Xi-UPS. An Al K α (1486.6 eV) X-ray was used as the excitation source for XPS.

Computational Methods of the charge balance model after doping with Sb^{3+} : First-principles density functional theory (DFT) is used to study the charge compensation mechanism of Sb³⁺ substituted Cd²⁺ in (C₈H₁₄N₂)₂CdCl₆. The first-principles calculations were performed using the projected augmented-wave method as implemented in the Vienna Ab initio Simulation Package (VASP). We used the VASPKIT code for post-processing of the VASP calculated data. The structural optimizations, convergence tests, and estimation of wavefunctions were adopted with the Perdew-Burke-Ernzerhof (PBE) functional. Plane-wave cutoff energy, the width of smearing and kmesh-resolved value were set to 400 eV, 0.1 eV, 0.06 Π /Å, and 20 Å, respectively. The chemical potential of Cl and H atom are calculated from the gaseous molecules within the huge lattice box. A 2*2*1 supercell containing 220 atoms was constructed to build model 1 and model 2. When estimating the formation energy, cut-off energy, and k-point mesh were set as 400 eV and the 1 × 1 × 1 Monkhorst-Pack grid, respectively. The convergence criterion for the electronic energy was 10⁻⁵ eV, whereas the structures were relaxed until the Hellmann-Feynman forces were smaller than 0.02 eV/Å.

Computational Methods of Band structure and Density of States: The first-principles calculations were performed using the projected augmented-wave method as implemented in the Vienna Ab initio Simulation Package (VASP). We used the VASPKIT code for postprocessing of the VASP calculated data. The structural optimizations and convergence tests were adopted with the Perdew-Burke-Ernzerhof (PBE) functional. The doped structures originate from a 2*2*2 supercell of the prototype ($C_8H_{14}N_2$)₂CdCl₆ (440 atoms) with one Cd atom replaced by the Sb atom, which corresponds to 12.5% doping content. The plane-wave cut off energy, the width of smearing, and kmesh-resolved value were set to 500 eV, 0.05 eV, and 0.03 $2\pi/Å$, respectively. The convergence criterion of the total energy was 10⁻⁵ eV/atom, and the residual Hellmann-Feynman force was 0.02 eV/Å. The GGA+U methods were applied to calculate the band structure and density of states. Within the GGA+U scheme, a single effective parameter, Ueff = U - J, is used for Mn and Sb, respectively.

	SCs		
(C ₈ H ₁₄ N ₂) ₂ CdCl ₆ : X% Sb	Sb(mg/L)	Cd(mg/L)	Ratio(Sb/Cd)
X=10	0.027	8.692	0.29%
X=20	0.043	6.162	0.64%
X=30	0.077	8.341	0.85%
X=40	0.155	10.740	1.33%

Table S1. ICP-AES data of X% Sb3+-doping (C₈H₁₄N₂)₂CdCl₆

Table S2. DFT simulation of two different charge balance forms after doping with Sb³⁺.

	E _p (eV)	E _d (eV)	$\mu_{Sb}(eV)$	$\mu_{Cd}(eV)$	$\mu_{H}(eV)$	$\mu_{Cl}(eV)$	E _f (eV)
Case 1	-1214.86	-1219.59	-3.86	-0.73		-1.78	9.63
Case 2	-1210.81	-1219.59	-3.86	-0.73	-0.03		11.94

Where E_f is the formation energy; E_d and E_p are the total energies of the doped and undoped crystals, respectively; μ_{Sb} and μ_{Cl} are the chemical potentials of the added Sb and Cl atoms, respectively; μ_{Cd} is the chemical potential of the removed Cd atom.

Table S3. ICP-AES data of X% Mn2+-doped (C8H14N2)2CdCl6 SCs

(C ₈ H ₁₄ N ₂) ₂ CdCl ₆ : X% Mn	Mn(mg/L)	Cd(mg/L)	Ratio(Mn/Cd)
X=5	0.012	11.300	0.22%
X=20	0.040	10.990	0.74%
X=30	0.147	21.040	1.43%
X=40	0.171	14.630	2.39%

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20 30 2 Theta (degree)

40

$(C_8H_{14}N_2)_2CdCl_6$: X ₁ % Sb and X ₂ % Mn	Sb(mg/L) or Mn(mg/L)	Cd(mg/L)	Ratio(Sb/Cd or Mn/Cd)
X1=30	0.092	21.375	0.40%
X ₂ =70	0.249	21.375	2.38%
a.	- Simulated XRD - (C ₄ H ₁₄ N ₃) ₂ CdCl, (Tre) Mulue M	(C _s H ₁ N ₂) after 600	sCdCle lays

Table S4. ICP-AES data of X₁% Sb³⁺-doped and X₂% Mn²⁺-doped (C₈H₁₄N₂)₂CdCl₆ SCs



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20 30 2 Theta (degree)

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Figure S2. (a) XPS spectrum of $(C_8H_{14}N_2)_2CdCl_6$: Sb³⁺. (b) XPS analysis of Sb 3d in $(C_8H_{14}N_2)_2CdCl_6$: Sb³⁺.



Figure S3. (a) The optimized structure of $(C_8H_{14}N_2)_2CdCl_6$. (b) Model A, one Sb³⁺ occupies the one lattice site of Cd²⁺, generating one interstitial Cl⁻ ion in the lattice. (c) Model B, two Sb³⁺ substituting two Cd²⁺, making one C₈H₁₂N₂ replace one C₈H₁₄N₂²⁺ ion.



Figure S4. (a) Optical absorption spectra of $(C_8H_{14}N_2)_2CdCl_6$ and $(C_8H_{14}N_2)_2CdCl_6$: Sb³⁺. (b) Absorption band edge fitting of 0.64% Sb³⁺-doped $(C_8H_{14}N_2)_2CdCl_6$.



Figure S5. CIE chromaticity coordinate of $(C_8H_{14}N_2)_2CdCl_6: 0.64\% Sb^{3+}$.



Figure S6. The PLQY of Sb³⁺-doped (C₈H₁₄N₂)₂CdCl₆ with different Sb³⁺ contents.



Figure S7. (a) The emission spectrum under different excitation wavelengths (from 250 nm to 370 nm with an interval of 20 nm) and (b) excitation spectrum under different excitation wavelengths (from 500 nm to 650 nm with an interval of 30 nm) of $(C_8H_{14}N_2)_2CdCl_6$: 0.64% Sb³⁺.



Figure S8. (a) Temperature-dependent PL spectrum of Sb³⁺-doped ($C_8H_{14}N_2$)₂CdCl₆ under 327 nm excitation. (b) FWHM of the PL spectrum as a function of temperature for ($C_8H_{14}N_2$)₂CdCl₆: 0.64% Sb³⁺ and the red line is the fitted curve.



Figure S9. (a) Density of states of $(C_8H_{14}N_2)_2CdCl_6$ and (b) Sb^{3+} -doped $(C_8H_{14}N_2)_2CdCl_6$.



Figure S10. EDS mapping of the 1.43% Mn²⁺-doped (C₈H₁₄N₂)₂CdCl₆ sample.



Figure S11. (a) Optical absorption spectra of $(C_8H_{14}N_2)_2CdCl_6$ and $(C_8H_{14}N_2)_2CdCl_6$: Mn^{2+} . (b) Absorption band edge fitting of 1.43% Mn^{2+} -doped $(C_8H_{14}N_2)_2CdCl_6$.



Figure S12. CIE chromaticity coordinates of (C₈H₁₄N₂)₂CdCl₆: 1.43%Mn²⁺.



Figure S13. (a) The density of states of Mn^{2+} -doped $(C_8H_{14}N_2)_2CdCl_6$.



Figure S14. The PLQY of Mn^{2+} -doped ($C_8H_{14}N_2$)₂CdCl₆ with different Mn^{2+} contents.



Figure S15. Contribution of PL intensity for Sb and Mn luminescence centers.