Supplementary file

Biological and physiochemical studies of electrospun polylactid/polyhydroxyoctanoate PLA/P(3HO)

scaffolds for tissue engineering applications

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1. Materials and methods

1.1. Specific surface area

The specific surface area (S_{BET}) of the samples was calculated using the BET formalism, analyzing the results of nitrogen adsorption/desorption at 77 K. Isotherms were measured using a Nova 2000 Quantachrome instrument, and the data was analyzed using the software provided with the instrument. Prior to the measurements, the samples were outgassed at 40 °C for 20 hours. It is worth mentioning that the use of 100-150 mg of materials in our experiments was necessary due to the limitations in the measuring cell size.

2. Results

2.1. S_{BET} results

Table S1. Measured surface areas of the fibrous samples.

Sample	SBET [m ² /g]
PLA	8.1
75PLA-25PHO	2.0
50PLA-50PHO	2.0
40PLA-60PHO	1.9

Table S2. Comparison of specific surface areas of fibers and composite materials prepared using biodegradable polymers.

Sample	S BET [m2/g]	Reference
PLA	8.1	This work
75PLA-25PHO	2.0	
50PLA-50PHO	2.0	
40PLA-60PHO	1.9	
PLA/Al ₂ O ₃ /Ag	n.d.	P. Kurtycz et al.[1]
Al ₂ O ₃ /Ag	181	
Non porous PLA fiber	1.45	Yoon et al.[2]
Porous PLA fiber	1.64	
Non porous fiber with glass beads	2.90	
Porous with glass beads	4.39	

PLA/sepiolite	n.d.	Sabzi et al[3]
Sepiolite	313.6	
PLA/NCC	n.d.	
NCC	17	
Scaffold 1/PLA fibers	0.5096	Chung et al.[4]
Scaffold 2/PLA fibers	2.3632	
PLA aerogel	10.12	Li et al.[5]
PLA	14.57	Spiridon et al[6]
PLA-4.5	26.24	Wang et al.[7]
PLA-5	31.00	
PLA-5.5	22.45	
PLA-6	20.88	
PLLA nanofiber membrane	4.64	Bai et al.[8]
PLLA nanofiber membrane – plasma treated	22.84	
P(3HB-co-4HB)	58.53	Zhijiang et al[9]
Photocatalyst CCZ/P(3HB-co-HHx) fiber	11.43	Selvin et al.[10]
P(3HB-co-4HB)/CA fiber mat 90:10	43.63	Zhijiang et al.[11]
P(3HB-co-4HB)/CA fiber mat 80:20	52.34	
P(3HB-co-4HB)/CA fiber mat 70:30	58.74	
P(3HB-co-4HB)/CA fiber mat 60:40	66.52	
PAN fiber mats	15.0	Chen et al.[12]

2.2. PLA/PHO fibers wettability



Figure S1. Water contact angle of the neat PLA and PLA/PHO fibers with increasing PHO content.



2.3. Mechanical properties

Figure S2. Stress/strain curves the neat PLA and PLA/PHO fibers with increasing PHO content. 2.4. Summary of PHA-based electrospun biomaterials for biomedical application

Presence of morphology defects is quite frequent phenomenon in PHA-based electrospun materials and is caused, mostly, by changes in electrospinning solution properties upon addition of other constituents (polymers or nanoparticles). Beads were observed in PHB/mcl-PHA[13], terpolymer P(3HB-4HB-3HV)[14], PHBV/PLLA[15], and neat PHBV[16] fibers, while fused regions were reported in P(3HB-*co*-3HH) and P(3HB-*co*-4HB)[17], PHBV/PHOHH[18], PHB/chitosan/bioglass composite[19], and PHBV/olive leaf extract[20] fibers, in the latter case they acted as mini reservoirs of polyphenols. PHBV electrospun fibers obtained using green

solvents derived from levulinic acid were ribbon-like[21]. The addition of PHO has caused a decrease in blend PLA/PHO fiber diameter in our study and similar trend was observed for multiple PHA-based electrospun polymer blends[13], [22]–[24] and composites[19], [25]–[30]. The PLA/PHO blend fibers developed in this study exhibited increase in polymer degree of crystallinity which further reflected on corresponding mechanical and thermal properties. In contrast to this, most of the PHA-derived fibrillar biomaterials exhibited lower degree of crystallinity[13], [15], [22], [27], [29], [31], [32]. However, this was understandable considering that the starting polymer in these instances was either PHB or PHBV, both with much higher degree of crystallinity in comparison to the low crystalline PLA used as a carrier polymer for PHO in this study. In terms of biocompatibility, PLA/PHO fibers proved to be excellent biocompatible biomaterials, which is in agreement with the literature data (**Table S3**).

Polymer/blends/c omposites	Morphology	Thermal properties	Mechanical properties	Degree of crystallinity	WCA	Application	Ref.
PHB, PHBV, PHB/PHBV14 % w/v, CHCl ₃	Aligned fibers; decrease in diameter $(3.7\pm1.7 \text{ to} 3.2\pm1.3 \mu\text{m})$.	Decrease in $T_{\rm m}$ (173.1 to 167.0°C).	Improved upon blending.	Decrease in χ _c (55.6 to 28.2%).	Increase in WCA (115 to 121.6°).	No cytotoxic effect (mouse fibroblasts L929); biomedical application	Somba tmank hong et al.[22]
P(3HB-co-3HH), P(3HB-co-4HB), 1 wt%, HFIP ^a	Spindle and fused regions; decrease in fiber diameter in comparison to the neat PHB fibers (190 vs. 520 nm).	-	Mechanical properties comparable to human skin.	Copolymer fibers have lower χ_c (70 and 35% vs. 85% for PHBHH, PHB4HB and PHB fibers, respectively)	-	<i>In vivo</i> biodegradable (subcutaneous implantation in rats); skin tissue engineering.	Ying et al.[17]
PHBV/AgNP ^b , 5 wt%, 2,2,2- trifluoroethyl alcohol	Composite fibers have smaller diameter (630±20 vs. 770±40nm).	-	-	-	-	Antibacterial effect against <i>Klebsiella</i> <i>pneumoniae</i> and <i>Staphylococcus aureus</i> ; supports fibroblasts adhesion and proliferation and osteoblasts differentiation; joint surgery.	Xing et al.[25]
PHBV/PDLLA ^c , 3 or 6 wt%, CHCl ₃ /DMF ^d	Random and aligned fibers.	A slight increase in $T_{\rm m}$ (151 to 153°C).	Improved mechanical properties. In random fiber membranes: increased TS 6.0 ± 0.9 to 8.3 ± 0.5 MPa, E 0.11 ± 0.01 to 0.13 ± 0.02 Gpa and ϵ 270 ±60 to $540\pm30\%$	Decrease in the overall χ_c (43 to 10%).	-	<i>In vitro</i> biodegradable; tunable properties to specific application.	Cheng et al.[31]
PHBV/keratin, 6 wt%, HFIP	Decrease in fiber diameter (815±98 to	-	-	-	-	Supports NIH 3 TR cells viability; induces	Yuan et

 Table S3. Summary of different PHA-based electrospun biomaterials aimed for biomedical application.

	720±124nm).					wound closure (<i>in vivo</i> test in mice); wound dressing/tissue engineering.	al.[33]
PHBV/PEO ^c , 20 wt%, CHCl ₃	Decrease in fiber diameter (2.8±0.1 to 1.3±0.2um); rougher surface due to phase separation.	$T_{\rm d}$ increase (279 to 285°C)	Decrease in TS, E and ϵ (1.8±0.2 to 0.4±0.2Mpa, 80±15 to 50±10Mpa, and 30±20 to 10±5%).	Decrease in PHBV χ _c (53 to 20%).	-	Biomedical application.	Bianco et al.[23]
PHB/mcl-PHA, CHCl ₃ /DMF	Decrease in fiber diameter $(1.27\pm0.37$ to 0.81 ± 0.37 µm); pronounced beads with mcl-PHA content increase.	-	-	Decrease in χ _c (47.2 to 28.7%).	-	Biomedical application	Azari et al.[13]
Terpolymer P(3HB-4HB- 3HV), 4 % w/v, CHCl ₃	Beaded morphology; 600- 1400 nm average fiber diameter.	-	Mechanical properties in following range: TS: 2.5±0.3 to 11.3±0.9Mpa, E: 73.5±5 to 288±62Mpa, ε: 29±9 to 376±188%.	_	WCA range: 73.1±0.5 to 92.6±0.4°).	No cytotoxic effect (L 929 fibroblast); supports hMSC differentiation into osteogenic and adipogenic phenotype.	Canad as et al.[14]
PHB/PLCL ^f , 6 wt%, CHCl ₃ /DMF	Decrease in fiber diameter (543.5±145 to 157±96.9 nm).	-	Increase in ε (10.6±1.4 to 55%).	-	Increase in WCA (88.6 to 120°).	Supports adhesion and proliferation of olfactory ensheating cells; neural tissue engineering.	Darana rong et al.[24]
PHBV/PLLA, 10 wt%, CHCl ₃ /DMF	Increase in fiber diameter (0.451±0.154 to 1.517±0.368um); beaded morphology with higher PHBV content; porous surface.	A slight decrease in $T_{\rm m}$ (153 to 148°C); a slight increase in $T_{\rm o}$ (251.6 to 265.9°C).	Increase in YM (19.9 to. 49.08MPa) and TS (0.35 to 0.78MPa).	Decrease in χ _c (43.5 to 1.1%).	-	Filtration and biomedical application.	Wagne r et al.[15]
PHBV/silk fibroin, 10 wt%, HFIP	Blend fibers have smaller diameter (380±65nm to 220- 350nm).	-	Increase in TS (1.31±0.2 to 4.84±0.52MPa), YM	-	Decrease in WCA (118.5±1.2 to	Supports mouse fibroblasts L929 adhesion and proliferation.	Lei et al. [26]

			(56.5±5.7 to 65.8±4.7MPa) and ε (5.1±0.7 to 44.3±4.0%).		70.9±1.9°).		
P(3HB- <i>co</i> - 4HB)/paclitaxel	Addition of the drug increased the diameter and caused bead free morphology (620±140 to 954±230 nm).	-	-	-	-	Biodegradable <i>in vitro</i> ; continuous drug release; drug eluting stents.	Sudesh et al.[34]
PHBV/PHOHH ^g , 15 wt%, CHCl₃/DMF	Addition of PHOHH has caused increase in blend fiber diameter (464±94 to 553±115 nm, for 75/25 and 65/35 ratios, respectively) and the appearance of fused regions; optimal ratio 75/25.	-	Decrease in TS (2.90 \pm 0.31 to 2.68 \pm 0.05 MPa for PHBV and PHBV/PHOHH 75/25, respectively) and YM (41.09 \pm 3.69 to 34.46 \pm 3.19MPa) and increase in ϵ (212.186 \pm 42.22 to 291.32 \pm 51.31 %).	Decrease in χ_c (20.5 to 16.1%, PHBV vs. PHBV/PHOHH).	-	Mechanical properties match skin, ideal for wound dressings.	Li et al.[18]
PHBV, 4 wt%, dichloromethane	Random (1.72±0.52 µm) and aligned (1.56±0.41µm,) fibers; beaded morphology.	-	-	-	-	Tissue engineering scaffolds.	Vega- Castro et al.[16]
PHB, PHBV, PHBV/PHOHD ^h , 4 wt%, CHCl ₃ , oxygen plasma treatment to improve hydrophilicity	PHB 2.6±0.4 μm; PHBV 2.7±0.4 μm; PHBV/PHOHD 4.0±0.2 μm (70/30); oxygen plasma treatment has decreased the fibers diameter due to physical etching.	Did not change significantly following processing and oxygen plasma treatment.	_	Decrease in χ_c due to addition of PHOHD (24.7 to 20.9%) and oxygen plasma treatment (38,8; 18,0 and 10.2% for PHB, PHBV and PHBV/mcl fibers,	Initial increase in WCA caused by addition of PHOHD eliminated by oxygen plasma treatment	Supports human dermal fibroblasts adhesion and proliferation; tissue engineering application.	Esmail et al.[32]

				respectively).	(complete wetting).		
PHB/PHO, CHCl ₃	Randomly oriented fibers that mimic ECM; addition of PHO led to increase in blend fiber radius (from 336±98 to 744±462 nm for 1:0.25 and 1:1, respectively)	-	Decrease in stiffness upon addition of PHO (868±540kPa to 274±22kPa); matches articular cartilage.			Supports adhesion, migration, and synthetic activity of human articular cartilage chondrocytes; cartilage tissue engineering.	Ching et al.[35]
Combination of micro (PHB) and nanofibers (gelatin) crosslinked with genipin	Entangled network of micro (PHB, average diameter $1.21 \mu m$) and nanofibers (gelatin, $0.22 \mu m$); crosslinking with genipin increases the diameter of gel fibers to $0.338\pm0.193 \mu m$ and causes the change in surface roughness	-	TS 0.78±0.35MPa YM 44.8±11.2 MPa ε 1.85±0.37% .	Increase in χ _c (12 to 19.1%).	Increase in WCA (0 to 38.7±6.1°).	Improved biocompatibility; great attachment of Balb/3T3 fibroblasts; no hemolytic activity; <i>in</i> <i>vivo</i> wound healing assay in rats has shown that gelatin containing dressing has induced almost complete wound closure; diabetic wound treatment.	Sanhue za et al.[36]
PHB/HA ⁱ nanocomposite immersed in metacrylated gelatin	Fiber diaameter 2±0.2 μm.	Improved thermal stability due to the HA incorporation.	Decrease in TS and E upon soaking in gelatin due to partial hydrolysis of PHA fibers.	-	-	Supports <i>in vitro</i> mineralization; adhesion, proliferation, and osteogenic phenotype of immobilized mouse calvarial preosteoblasts; Bone tissue engineering.	Sadat- Shojai et al.[37]
PHB, PHB/Curcumin, PHB/Quercetin,	Incorporation of curcumin caused decrease in fiber	-	-	Decrease in χ _c (31.53 vs. 24.28, and 21.22% for	-	<i>In vitro</i> antioxidative activity; biomedical application	Vilche z et al.[27]

	diameter (1.370 \pm 0.282 to 0.761 \pm 0.123 µm); quercetin - increase in diameter (to 1.803 \pm 0.557 µm) due to better solubility and more favorable Cur-PHB interactions.			PHB, PHB/Cur and PHB/Que, respectively).			
P4HB, P4HB/peptide analogue of fibroblast growth factor (GF), 19 % w/v, CHCl ₃ /DMSO	Addition of GF has reduced fiber diameter (6,9±0.02, to 5.0±0.05 to 4.8+- 0.03 µm), for P4HB in CHCl ₃ , P4HB in CHCl ₃ /DMSO, and P4HB/GF/CHCl ₃ /D MSO, respectively; due to the increase in electrospinning solution conductivity.	A slight change in T_m (56.0 to 54.3 to 55.2°C, for P4HB in CHCl ₃ , P4HB in CHCl ₃ /DMSO, and P4HB/GF/CHCl ₃ /DMSO, respectively); Thermal stability independent of processing conditions and GF incorporation.	Addition of GF has reduced TS (28 to15MPa) increased ε (360 to 730% for P4HB in CHCl3 vs. P4HB/GF in CHCl ₃ /DMSO).	-	Increase in WCA (103 to 121°).	In vitro wound healing assay showed that the scaffold supports fibroblast migration, adhesion, and proliferation; wound dressing.	Kerido u et al.[28]
PHBV/OLE ^j , PHB/PHOHD/OL E	Incorporation of OLE caused bead formation and fused regions in PHB/PHOHD/OLE fibers; mini reservoirs of polyphenols.	-	-	-	-	Burst release in first 30 min, 90% release in 6h; Immunomodulatory effect on human dermal keratinocytes, downregulation of proinflammatory cytokines; wound healing.	De la Ossa et al.[20]
PHB/cellulose acetate (CA), 5 wt%,	Decrease in composite fibers diameter upon	Increase in $T_{\rm g}$ (12.6 to 36.5°C for PHB and	Decrease in TS (7.44±0.58 to	Decrease in the heat fusion of the crystalline phase	Decrease in WCA	Improved <i>in vitro</i> biodegradation; supports adhesion and	Zhijian g et al.[29]

CHCl ₃ /DMF	addition of CA (670±220 to 160±52 nm for PHB and PHB/CA 60/40, respectively).	PHB/CA 60/40), respectively) due to inhibited mobility of polymer chains (interactions with CA); decrease in T_m (152.4 to 147.6°C for PHB and PHB/CA 60/40, respectively).	4.52±0.34MPa), YM (867.5±201.2 to 806,9±168.2MPa) and ε (22.70±1.77 to 6.53±0.24%) for PHB and PHB/CA 60/40, respectively).	(72.4 to 25.8 J/g for PHB and PHB/CA 60/40, respectively).	(123.0±3.5 to 60.00±1.25 °) for PHB and PHB/CA 60/40, respectivel y).	proliferation of fibroblasts; wound dressing/tissue engineering.	
PHB/chitosan/bio glass, trifluoroacetic acid	Decrease in diameter due to increase in electrical conductivity of the spinning solution; presence of fused fiber regions.	-	Increase in TS (0.95±1.5 to 3,42±2.98 MPa) and E (109.9±0.05 to 210.96±0.09 MPa).	-	-	Supports differentiation of human exfoliated deciduous teeth stem cells to odontoblast- like cells and odontogenic phenotype; dentin tissue engineering.	Khoro ushi et al.[19]
PHBV/chitosan/H A, 8% w/v, HFP	Decrease in nanocomposite fiber diameter (405±74 to 256±110 nm, for PHBV and PHBV/Ch/HA).	-	Decrease in TS (5.72+- 0.42 to $3.55+-0.22$ MPa), decrease in YM (151.26±1.61 to 101.93±4.52 MPa) and decrease in ε (31.81±2.45 to 4.4±0.08%).	-	Decrease in WCA (126.10±1. 08 to 78.80±2.28 °).	Slow <i>in vitro</i> biodegradation (up to 60days); supports proliferation and osteogenic phenotype of human fetal osteoblasts; <i>in vitro</i> biomineralization; bone tissue engineering.	Zhang et al.[30]
PHBV, green solvents derived from levulinic acid (obtained from hemi-or cellulose fraction of biomass).	Ribbon-like fibers obtained using a mixture of formic acid and MEK (methyl ethyl ketone, a derivative of levulinic acid).	-	In agreement with the literature.	-	In agreement with the literature.	Supports adhesion and proliferation of adenocarcinoma human alveolar basal epithelial cells (A549)- <i>in vitro</i> model of lung endothelium; <i>in vitro</i> model of biological	Lapom arda et al.[21]

barriers.

^a1,1,1,3,3,3-hexafluoro-2-propanol
^b silver nanoparticles
^c poly(D,L-lactic acid)
^d dimethylformamide
^e polyethylene oxide
^f poly(L-lactide-*co-&*-caprolactone)
^g poly(3-hydroxyoctanoate-*co*-3-hydroyhexanoate)
^h poly(3-hydroxyoctanoate-*co*-3-hydroydecanoate)
ⁱ hydroxyapatite
^j olive leaf extract rich in polyphenols

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