

Supporting Information

Biomass-derived carbon sponges for use as sodium-ion capacitor electrodes

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Table S1. Electrochemical performance of state-of-the-art heteroatom-doped carbons as sodium stores.

Material	Capacity at 0.1 A g ⁻¹ (mAh g ⁻¹)	Capacity at 10 A g ⁻¹ (mAh g ⁻¹)	ICE (%)	Ref.
AGS-650	524	161	60	This work
3D N-doped carbon network	~ 240	123	38	[1]
N,S co-doped CNF	352	150	37	[2]
N-doped hierarchical porous hollow carbon spheres	300	107	44	[3]
N, O, P co-doped hierarchical porous carbon	312	72	40	[4]
N-doped carbon fibers	361	105	43	[5]
N/S-co-coped carbon nanoparticles	726	188	68	[6]
N, S co-doping hard carbon	250	102	<40	[7]
S-doped hard carbon	777	353	84	[8]
S-doped carbon nanosheets	548	133	58	[9]
S covalently linked TiO ₂ /C nanofiber	410	203	66	[10]
S/N doped 3D porous carbon nanosheets	392	138	33	[11]

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Table S2. Values of oxygen content and electrical conductivity determined for the as-obtained microporous carbon and the post-treated material.

	O content (wt%)	Electrical conductivity (S cm ⁻¹)
As-obtained	5.84	0.95
AGA-850	3.63	3.03

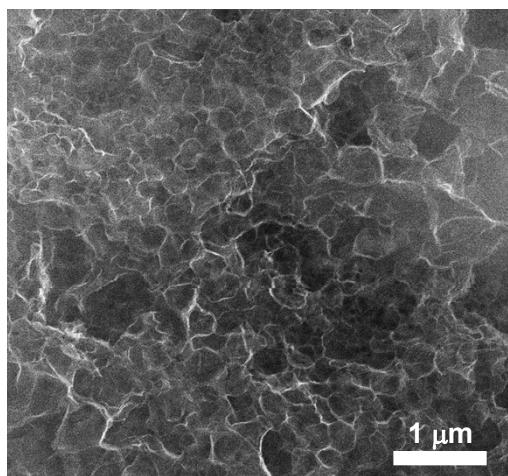


Figure S1. SEM micrograph of the non-doped material AG-600

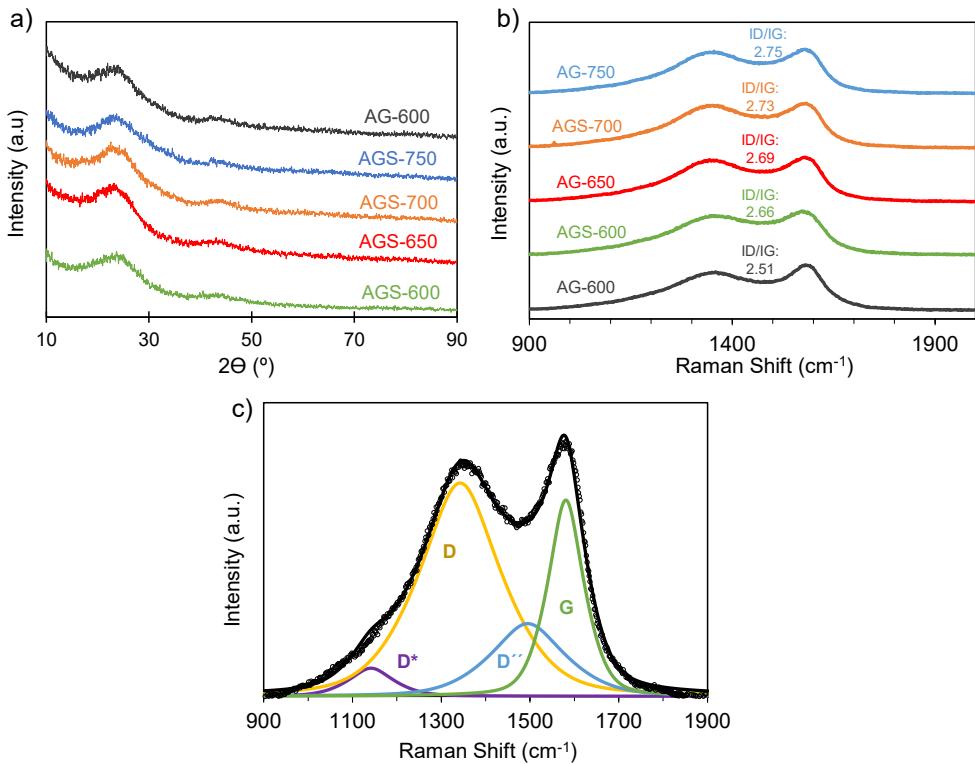


Figure S2. a) X-ray diffractograms and b) first-order Raman spectra of the S-doped carbon sponges and the non-doped material; c) example of deconvolution of the Raman spectrum of AGS-650 into four bands ($D^* \sim 1150 \text{ cm}^{-1}$, $D \sim 1300 \text{ cm}^{-1}$, $D'' \sim 1500 \text{ cm}^{-1}$ and $G \sim 1600 \text{ cm}^{-1}$).

With increased doping temperature, there is a decrease of the heteroatom content and an increase of the electrical conductivity of the samples (Table 1), which points to a progressive ordering of the carbon structures. Accordingly, the increase of the I_D/I_G ratio observed with the rise of the doping temperature points to a decrease of defects according to the stage 2 of the three-stage Raman model proposed by Ferrari et al. (*Physical review B*, 61(20), 14095).

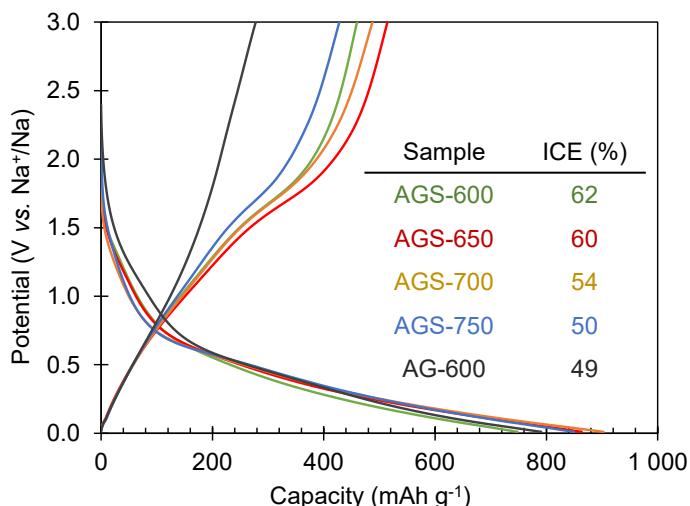


Figure S3. GCD profiles of the negative electrode materials corresponding to the first cycle recorded at 0.1 A g⁻¹ (inset: ICE values).

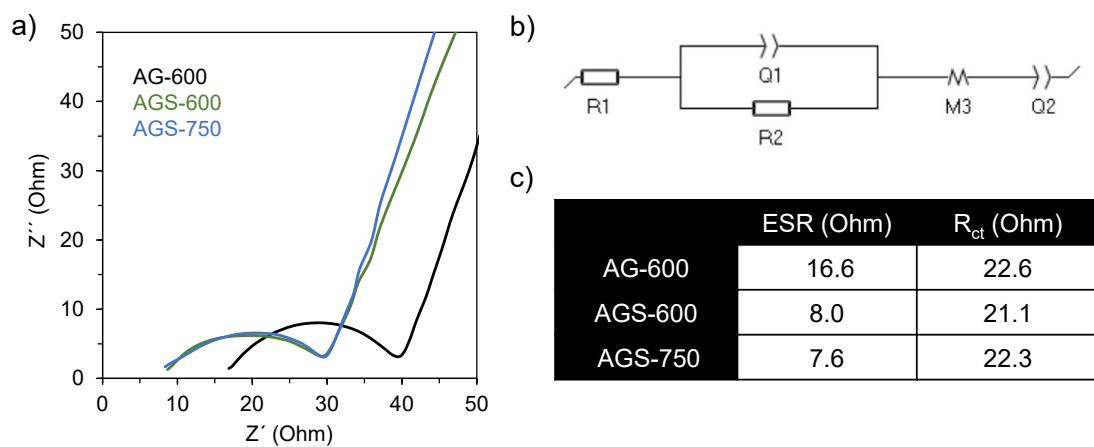


Figure S4. A) Nyquist plots of the cells recorded after the first sodiation-desodiation cycle; b) equivalent electric circuit model used for fitting the Nyquist plots and c) values of equivalent series resistance (ESR) and charge transfer resistance (R_{ct}) calculated based on this model.

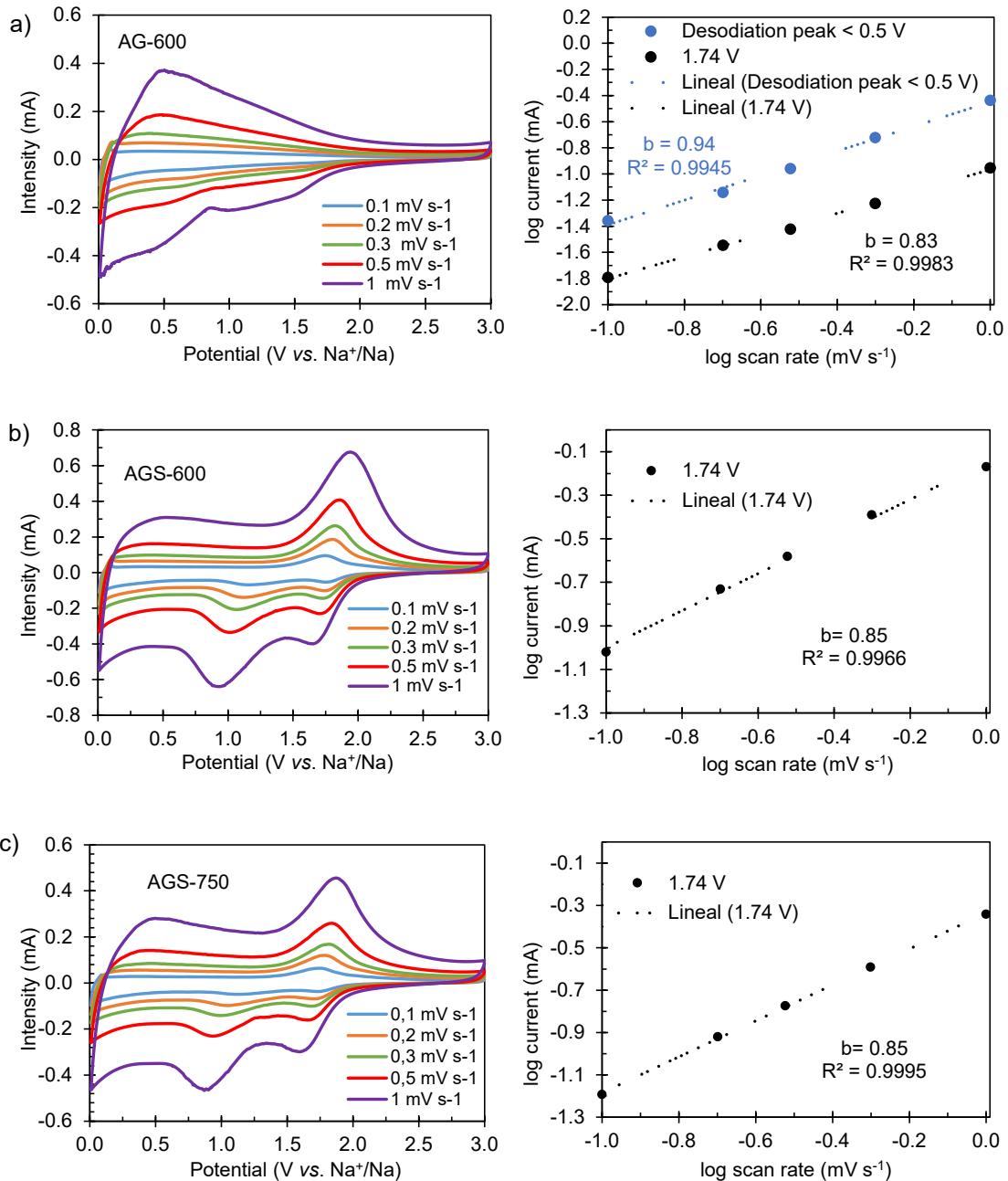


Figure S5. CVs recorded at different potential sweep rates (left) and determination of the b parameter at 1.74 V (right) for (a) AG-600, (b) AGS-600 and (c) AGS-750.

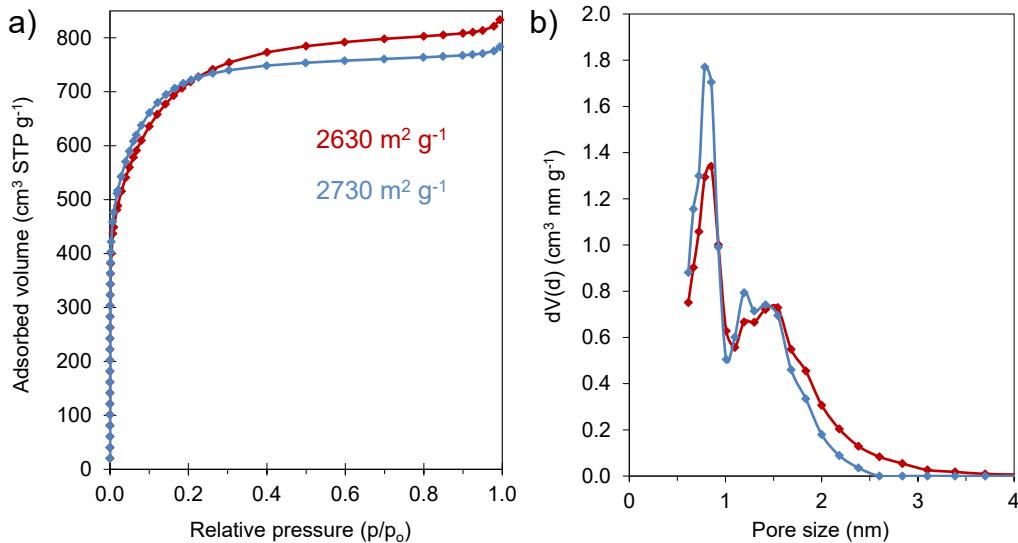


Figure S6. Nitrogen adsorption isotherm and pore size distribution corresponding to the microporous carbon before (blue) and after (red) the post-activation thermal treatment.

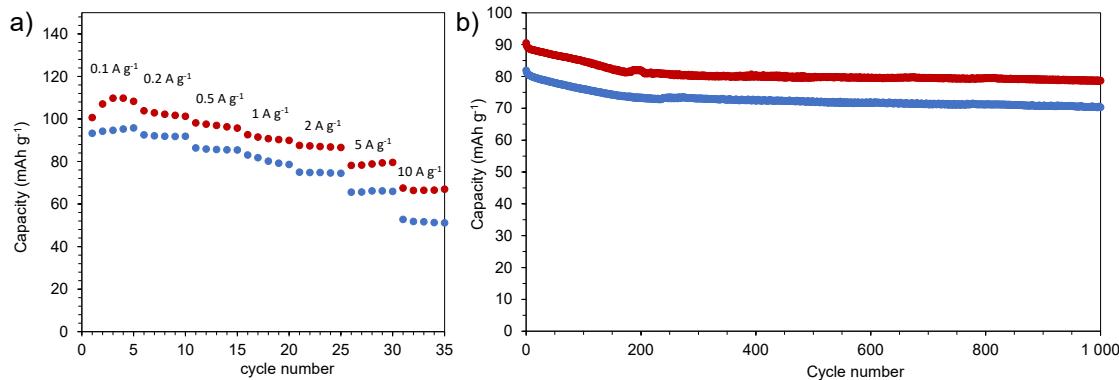


Figure S7. (a) Rate capability and (b) long term cycling stability test for the microporous carbon before (blue) and after (red) the post-activation thermal treatment.

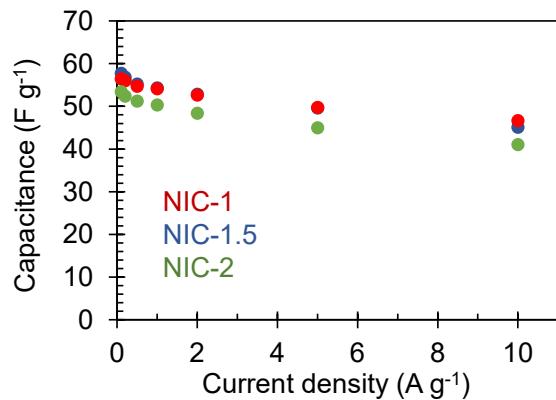


Figure S8. Rate capability of the full cells expressed in terms of capacitance.

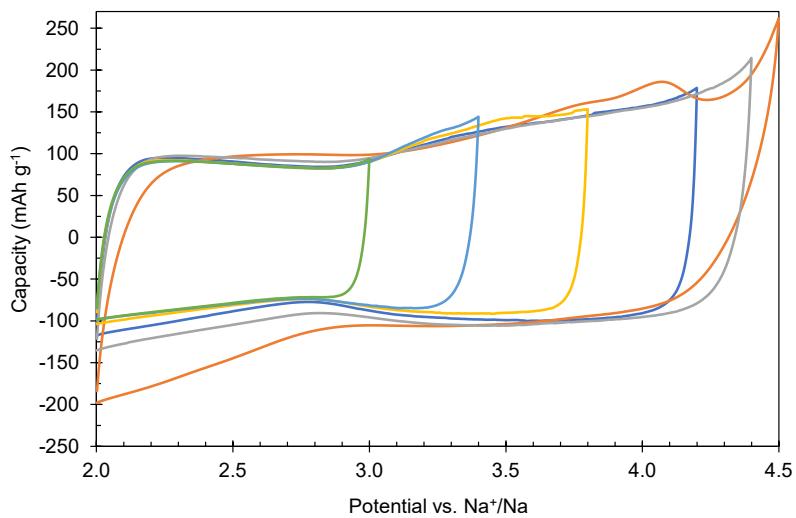


Figure S9. Cyclic voltammograms recorded for AGA-850 at increasing voltage window (potential sweep rate of 2 mV s^{-1}).

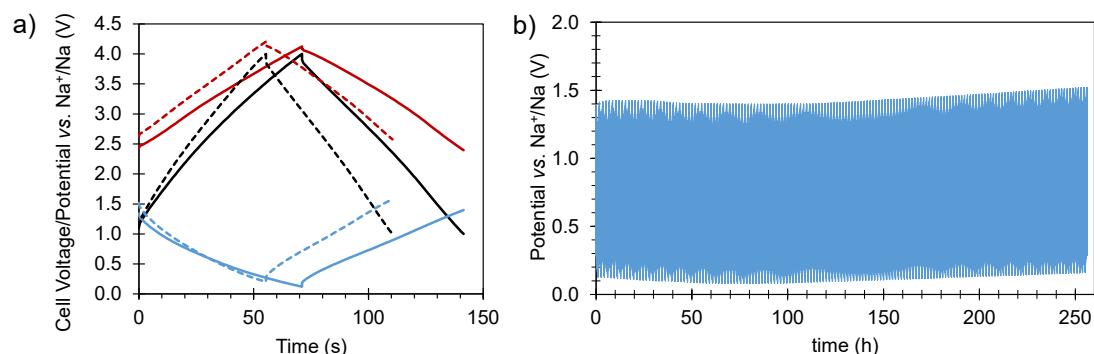


Figure S10. a) Cell voltage and electrode potential plots corresponding to the first (solid lines) and 10000th (dashed) cycles at 2 A g^{-1} ; b) evolution of the negative electrode potential window during the whole cycling stability test.