Supporting Information

Fluorosulfide La_{2.7}Ba_{6.3}F_{8.7}S₆ with Double-Layer Honeycomb Structure Enabling Fluoride-Ion Conduction

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S.1: Refined structural parameters of La_{2.7}Ba_{6.3}F_{8.7}S₆ and La_{2.16}Ba_{6.3}Ca_{0.54}F_{8.16}S₆

Table S1. Refined structural parameters of La₃Ba₆F₉S₆ obtained by Rietveld refinement of X-ray diffraction data.

Atom	Wyckoff	g	x	У	Z	U / Å ³	BVS
Lal	3 <i>b</i>	1	0	0	1/2	0.011(13)	+2.03
Ba1	6 c	1	0	0	0.28286(9)	0.014(10)	+2.33
S1	6 c	1	0	0	0.1146(4)	0.028(4)	-1.78
F1	6 c	1	0	0	0.3673(9)	0.07(10)	-1.24
F2	3 <i>a</i>	1	0	0	0	0.14(2)	-0.64

Unit cell: trigonal *R*-3(148); *a* = 4.38603(6) Å, *c* = 32.4193(7) Å

 $R_{wp} = 11.01\%$, $R_p = 6.23\%$, goodness of fit S = 1.19

Table S2. Refined structural parameters of La_{2.7}Ba_{6.3}F_{8.7}S₆ obtained by Rietveld refinement of X-ray diffraction data. In order to avoid including errors for anions, the fluorine site occupancies were fixed.

Atom	Wyckoff	g	x	У	Z	U / Å ³
La1	3 <i>b</i>	0.387(5)	0	0	1/2	0.010(13)
Ba1	3 <i>b</i>	0.613(5)	0	0	=z(La1)	= <i>U</i> (La1)
La2	6 <i>c</i>	0.257(9)	0	0	0.28286(9)	0.015(10)
Ba2	6 <i>c</i>	0.743(9)	0	0	= z(La2)	= <i>U</i> (La2)
S 1	6 <i>c</i>	1	0	0	0.1147(4)	0.029(4)
F1	6 <i>c</i>	1	0	0	0.3672(9)	0.07(10)
F2	3 <i>a</i>	0.9	0	0	0	0.10(2)

Unit cell: trigonal *R*-3(148); *a* = 4.38602(6) Å, *c* = 32.4192(7) Å

 $R_{\rm wp} = 10.97\%$, $R_{\rm p} = 6.19\%$, goodness of fit S = 1.18

Atom	Wyckoff	g	x	У	Z		U / Å ²	
La1	3 <i>b</i>	0.457(2)	0	0	1/2	0.	0118(10)	
Ba1	3 <i>b</i>	0.543(2)	0	0	= z(La)	l) =	<i>U</i> (La1)	
La2	6 c	0.2216(10)	0	0	0.282748	8(5) 0.	0.01326(9)	
Ba2	6 c	0.7784(10)	0	0	= z(La2)	2) =	<i>U</i> (La2)	
S 1	6 c	1	0	0	0.114504	(14) 0.	03284(9)	
F1	6 c	0.9859(7)	0	0	0.366128	8(9) 0.	04877(9)	
F2	3 <i>a</i>	0.9281(14)	0	0	0	0	0.1157(2)	
Atom	U_{11} / Å ²	U_{22} / Å 2		U_{33} / Å 2	U_{12} / Å ²	U_{13} / Å ²	U_{23} / Å 2	
La1	0.007843(1	1) 0.007843((11)	0.01962(2)	0.00392(7)	0	0	
Ba1	$= U_{11}(\text{Lal})$	$= U_{22}(La)$	$= U_{22}(La1)$		$= U_{12}(La1)$	$= U_{13}(La1)$	$= U_{23}(La1)$	
La2	0.01489(8)) 0.01489((8)	0.0100(12)	0.00745(4)	0	0	
Ba2	$= U_{11}(La2)$	$= U_{22}(La)$	12)	$= U_{33}(La2)$	$= U_{12}(La2)$	$= U_{13}(La2)$	$= U_{23}(La2)$	
S1	0.0310(12)) 0.0310(1	2)	0.0365(3)	0.01549(8)	0	0	
F1	0.0505(12)) 0.0505(1	2)	0.0453(18)	0.02525(7)	0	0	

0.1211(9)

0

0.0565(17)

0

Table S3. Refined structural parameters of $La_{2.7}Ba_{6.3}F_{8.7}S_6$ obtained by Rietveld refinement of neutron diffraction data.

Unit cell: trigonal *R*-3(148); *a* = 4.3870457 (13) Å, *c* = 32.43296(3) Å

0.1130(3)

 $R_{wp} = 8.84\%$, $R_e = 1.41\%$, goodness of fit S = 6.25

0.1130(3)

F2

Atom	Wyckoff	g	x	У	Ζ	U / Å ²
Lal	3 <i>b</i>	0.4612(17)	0	0	1/2	0.01426(9)
Ba1	3 <i>b</i>	0.5032(9)	0	0	= <i>z</i> (La1)	= <i>U</i> (La1)
Cal	3 <i>b</i>	0.0356(8)	0	0	= <i>z</i> (La1)	= <i>U</i> (La1)
La2	6 c	0.1293(8)	0	0	0.282977(5)	0.01464(8)
Ba2	6 c	0.7984(4)	0	0	= z(La2)	= <i>U</i> (La2)
Ca2	6 c	0.0722(4)	0	0	= z(La2)	= <i>U</i> (La2)
S1	6 c	1	0	0	0.114960(11)	0.02684(9)
F1	6 c	0.9987(7)	0	0	0.365078(7)	0.05043(9)
F2	3 a	0.7225(15)	0	0	0	0.07977(3)
Atom	U_{11} / Å 2	U_{22} / Å 2	U_{33} / Å 2	U_{12} / Å 2	U_{13} / Å 2	U_{23} / Å 2
Lal	0.0118(11)	0.0118(11)	0.0192(2)	0.00592(7	r) 0	0
Ba1	$= U_{11}(La1)$	$= U_{22}(La1)$	$= U_{33}(La1)$	$= U_{12}(\text{Lal})$) = $U_{13}(La1)$	$= U_{23}(La1)$
Cal	$= U_{11}(La1)$	$= U_{22}(La1)$	$= U_{33}(La1)$	$= U_{12}(\text{Lal})$) = $U_{13}(La1)$	$= U_{23}(La1)$
La2	0.01711(8)	0.01711(8)	0.0097(11)	0.00856(4) 0	0
Ba2	$= U_{11}(La2)$	$= U_{22}(La2)$	$= U_{33}(La2) = U_{12}(La2)$		$= U_{13}(\text{La2})$	$= U_{23}(La2)$
Ca2	$= U_{11}(La2)$	$= U_{22}(La2)$	$= U_{33}(\text{La2}) = U_{12}(\text{I})$		$= U_{13}(\text{La2})$	$= U_{23}(La2)$
S1	0.0289(11)	0.0289(11)	0.0227(2) 0.01444(6) 0	0
F1	0.0566(11)	0.0566(11)	0.0381(18)	0.02830(6) 0	0
F2	0.0748(3)	0.0748(3)	0.0897(5) 0.0374(18		3) 0	0

Table S4. Refined structural parameters of $La_{2.16}Ba_{6.3}Ca_{0.54}F_{8.16}S_6$ obtained by Rietveld refinement of neutron diffraction data.

Unit cell: trigonal *R*-3(148); *a* = 4.3640735(13) Å, *c* = 32.34269(3) Å

 $R_{\rm wp}\!=$ 5.55%, $R_{\rm e}\!=$ 1.32%, goodness of fit $S\!=$ 4.20





Figure S1. Flow-chart of structural determination for powder diffraction. The square includes the structural method with underline and program name written italic. The dotted circle indicates the structural parameters, whereas the circle with grey background indicates the experimental data.



Figure S2. XRD patterns ($\lambda = 1.5418$ Å) of the synthesized compounds and reference data for Ba₂LaF₇(00-048-0099)¹ from International Centre for Diffraction Data(ICDD). The sharps(#) and asterisks(*) indicate LaFS and Ba₂LaF₇ phases, respectively.



Figure S3. (a) XRD patterns ($\lambda = 1.5418$ Å) of the synthesized compound weighted the molar ratio BaS:LaF₃ = 70:30 and reference data for LaF₃(01-084-9414)², BaS(03-065-7901)³, LaFS(01-070-7567)⁴, Ce₂SrF₄S₂(01-070-7574)⁴ from International Centre for Diffraction Data(ICDD). (b) Refined crystal structure of La₃Ba₆F₉S₆. (c) Rietveld patterns of La_{2.7}Ba_{6.3}F_{8.7}S₆. Red plus, black lines, and blue lines denote observed, calculated, and difference, respectively. Green ticks represent the Bragg position of La_{2.7}Ba_{6.3}F_{8.7}S₆. (d) Refined crystal structure of La₃F_{8.7}S₆ (space group: *R*-3). The unit cell is shown by the solid lines.



Figure S4. Enlarged view near double honeycomb fluoride ion layers and the view along *ab*-plane. of (a) $La_{2.7}Ba_{6.3}F_{8.7}S_6$, (b) LaF_2I^5 and (c) $La_{0.9}Ba_{0.1}F_{2.9}^6$. All the compounds form the fluoride-ion conduction layers with a honeycomb structure along *ab*-plane.



S. 3: Fluoride ion conductivity and electrochemical window of La_{2.7}Ba_{6.3}F_{8.7}S₆

Figure S5. Schematic illustration of the cell and equivalent circuit for the evaluation of the ionic conductivity of the synthesized compounds and the confirmation of conduction carrier species of the synthesized sample. (a) Schematic illustration of the symmetric cell for the evaluation of the ionic conductivity of the sample by AC impedance and (b) the equivalent circuit model of the symmetric cell. The model is composed by the electrolyte component (R_b : bulk resistance, C_b : bulk capacitance, R_{gb} : grain boundary resistance, C_{gb} : grain boundary capacitance) based on the ion conduction in the solid and the electrode interface resistance (C_{ei}) due to the polarization effects at the electrode/electrolyte interfaces. (c) Schematic illustration of the blocking cell for the confirmation of the dominant carrier species of the sample by AC impedance and DC conductivity measurements. (d) The equivalent circuit model of the blocking cell. The model consists of electrolyte (sample and two PbSnF4 on both sides) components, the resistance component (R_{ei} : electrode interface resistance, C_{ei} : electrode interface resistance, C_{di} : double layer capacitance) due to the decomposition of PbSnF4 by fluoride ion migration between Pb electrode and PbSnF4.



Figure S6. Confirmation of conduction carrier of La_{2.7}Ba_{6.3}F_{8.7}S₆. (a)Nyquist plots from AC impedance measurement and (b) Voltage-time profile of SS|Pb|PbSnF4|La_{2.7}Ba_{6.3}F_{8.7}S₆|PbSnF4|Pb|SS blocking cell at 423 K. (c) *I-V* plots for data obtained 2 s after applying current. Since PbSnF4 is an almost pure fluoride ion conductor, the cell conducts only fluoride ions under DC method, while all mobile ions in the cell conduct under AC impedance measurements⁷. The resistance value of SS|Pb|PbSnF4|La_{2.7}Ba_{6.3}F_{8.7}S₆|PbSnF4|Pb|SS symmetrical blocking cell observed in the AC impedance measurements is 7552 ohm, whereas the resistance value of the same cell in observed in the DC method indicates fluoride ion as conduction carrier of La_{2.7}Ba_{6.3}F_{8.7}S₆. However, the absolute resistance value in this measurement is smaller than that observed in Fig. 3. This is probably due to PbSnF4 has infiltrated the gaps around the pellet, reducing the effective thickness. Further investigation of this method is required.



Figure S7. (a) X-ray diffraction patterns of the synthesized La_{2.7-x}Ba_{6.3}Ca_xF_{8.7-x}S₆ (x = 0, 0.27, 0.54) and diffraction peaks of the refined La_{2.7}Ba_{6.3}F_{8.7}S₆ ($\lambda = 0.7504$ Å). The asterisks (*) represent the peaks of impurities. (b) Composition dependence of lattice parameters for La_{2.7-x}Ba_{6.3}Ca_xF_{8.7-x}S₆ (x = 0, 0.27, 0.54). From La_{2.7}Ba_{6.3}F_{8.7}S₆ to La_{2.16}Ba_{6.3}Ca_{0.54}F_{8.16}S₆, the lattice parameters of *a* and *c* gradually decrease in line with Vegard's law. The decreased lattice volume is attributed to the varying ionic radii of lanthanum (103 pm) and calcium (100 pm) in the La_{2.7-x}Ba_{6.3}Ca_xF_{8.7-x}S₆ phase. (c) Arrhenius plot of the sintered La_{2.7-x}Ba_{6.3}Ca_xF_{8.7-x}S₆. (d) Neutron Rietveld refinement of La_{2.16}Ba_{6.3}Ca_{0.54}F_{8.16}S₆. Red circle, blue line and pink line stands for the observed, calculated, and difference, respectively. Green ticks represent the Bragg position of La_{2.16}Ba_{6.3}Ca_{0.54}F_{8.16}S₆.



Figure S8. Cyclic voltammogram of the $SS|Pt|La_{2.7}Ba_{6.3}F_{8.7}S_6|Pb/PbF_2|SS$ cell at 423 K. The electrochemical potential window of $La_{2.7}Ba_{6.3}F_{8.7}S_6$ exceeds 4.0 V.

S. 4: Mechanism of fluoride ion conduction



Figure S9. (a) Three-dimensional nuclear density distributions calculated by maximum-entropy method using neutron diffraction data for $La_{2.7}Ba_{6.3}F_{8.7}S_6$ at 1.0 fm Å⁻³. (b) Enlarged view of the nuclear density distributions along *ab*-plane.



Figure S10. Fluoride ion conduction pathway calculated by BV-based energy calculation of $La_{2.7}Ba_{6.3}F_{8.7}S_{6.}$

References

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