

## Support information

**Leveraging all-fixed transfer framework to predict  
interpretable formation energy of MXene with hybrid  
terminals**

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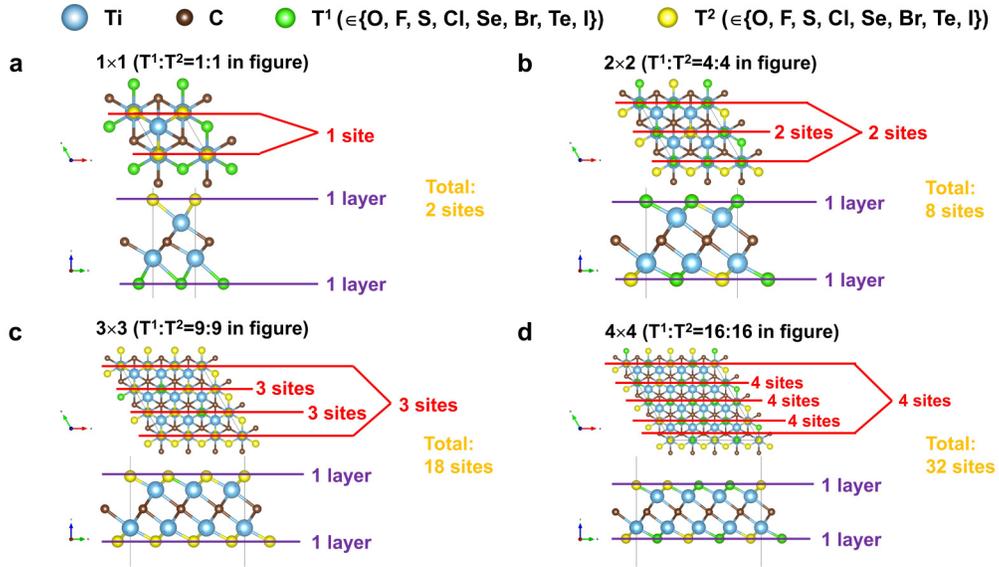


Fig. S1: The crystal structures of  $\text{Ti}_2\text{CT}_2$  in different lattice. (a) 1×1 lattice with 2 sites.  $T^1:T^2$  can only be 1:1. (b) 2×2 lattice with 8 sites.  $T^1:T^2$  can range from 1:7 to 7:1. (c) 3×3 lattice with 18 sites.  $T^1:T^2$  can range from 1:17 to 17:1. (d) 4×4 lattice with 32 sites.  $T^1:T^2$  can range from 1:31 to 31:1.

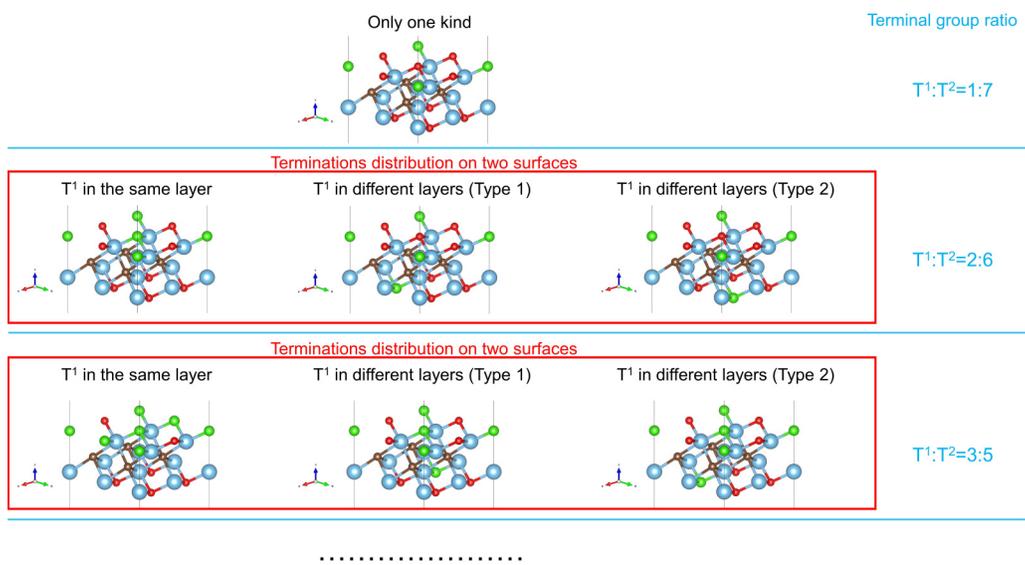


Fig. S2: The detailed displaying of terminal group ratio and terminations distribution on two surfaces.

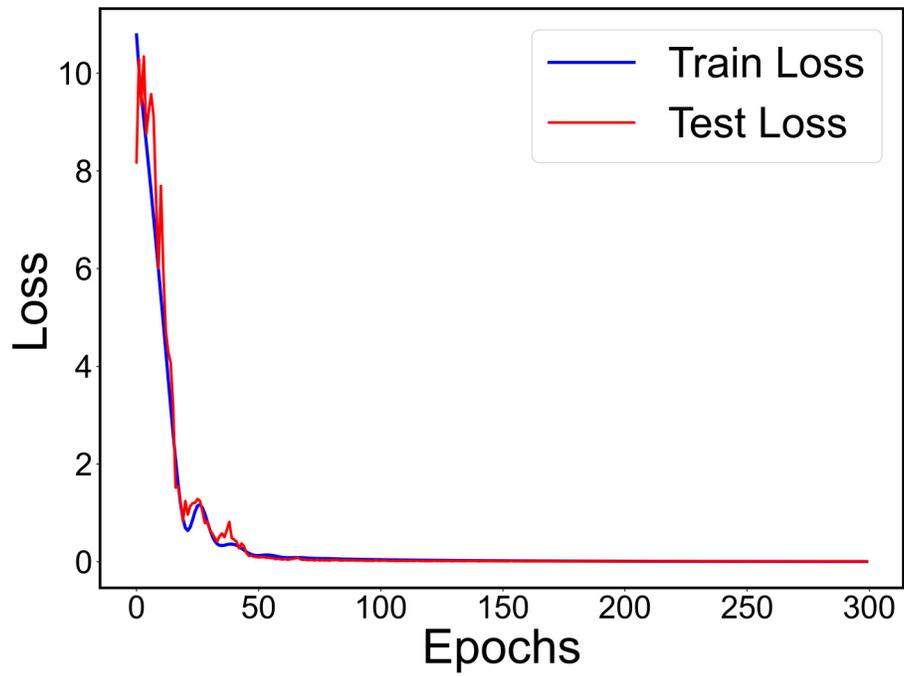


Fig. S3: The learning rate of the transferred model in the first transfer. The target training set is  $1 \times 1$   $Ti_2CT_2$  lattice that has 36 individuals.

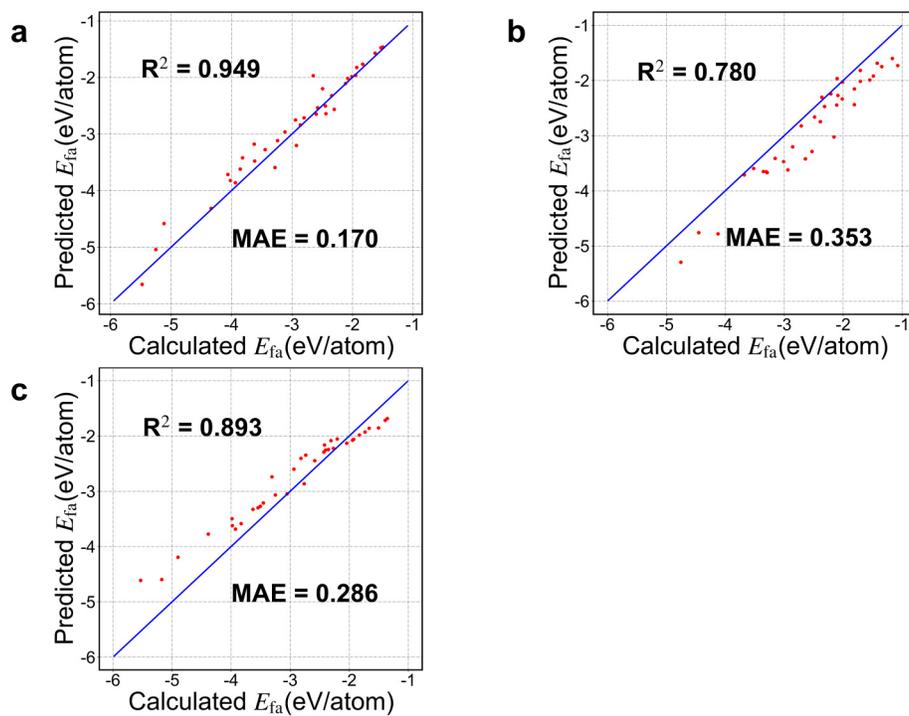


Fig. S4:  $R^2$  of the model trained on  $1 \times 1$   $Ti_2CT_2$  lattice dataset and predicts optimized  $1 \times 1$  lattice of (a)  $Ti_2NT_2$ , (b)  $V_2CT_2$ , (c)  $Ti_3C_2T_2$ .



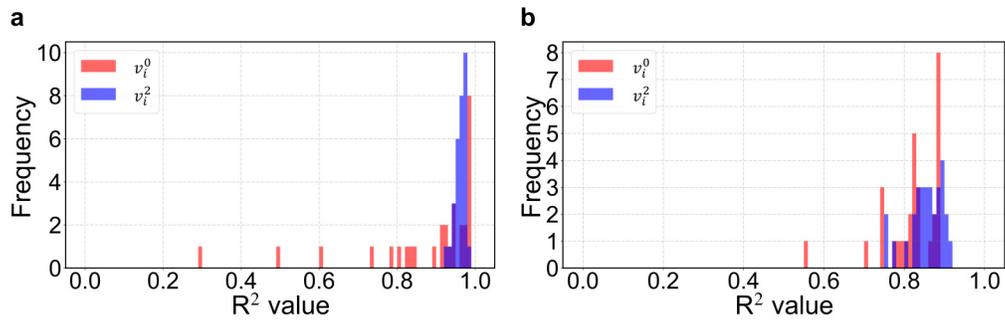


Fig. S6: The frequency distribution of the performance of feature  $v_i^0$  and feature  $v_i^2$ . Both of them were used to train on the  $1 \times 1$   $\text{Ti}_2\text{CT}_2$  lattice and to predict the  $1 \times 1$  (a)  $\text{Ti}_2\text{NT}_2$  and (b)  $\text{Ti}_3\text{C}_2\text{T}_2$  lattice. This process is conducted totally 30 times in both (a) and (b).

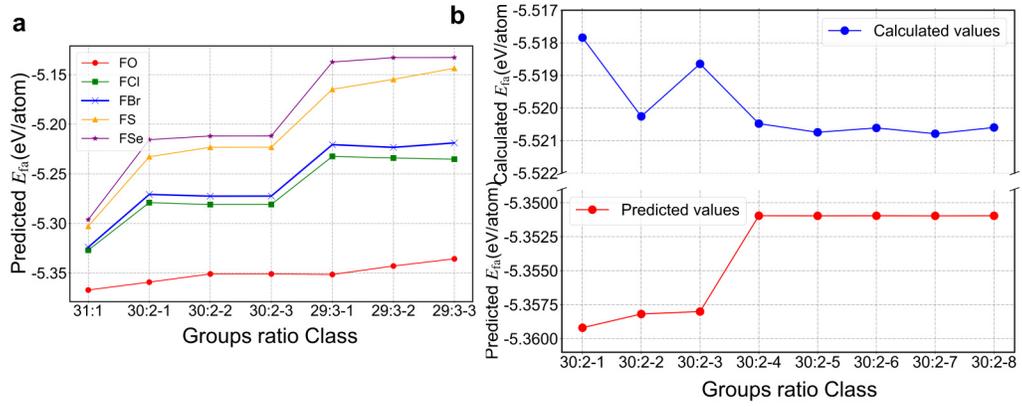


Fig. S7: The  $E_{fa}$  of screening results and their comparing with calculated results. (a) Screening results of the  $4 \times 4$   $Ti_3C_2T_2$  lattice with the lowest  $E_{fa}$ . “30:2-1, 30:2-2, 30:2-3” are sampled from all kinds of structures under the ratio of 30:2. The same with “29:3-1, 29:3-2, 29:3-3”. (b) Results comparing the prediction of all kinds structures of  $4 \times 4$   $Ti_3C_2T_2$  FO lattice to calculation of those under the ratio of 30:2.

**Table. S1: The training and testing sets of different kinds of MXene.**

Training set			Testing set			Corresponding figures
Lattice size	MXene type	Number of data	Lattice size	MXene type	Number of data	\
1×1	Ti <sub>2</sub> C	36	1×1	Ti <sub>2</sub> N	36	Fig. S3a
				V <sub>2</sub> C	36	Fig. S3b
				Ti <sub>3</sub> C <sub>2</sub>	36	Fig. S3c
			2×2	Ti <sub>2</sub> C	560	Fig. 4a
			3×3	Ti <sub>2</sub> C	72	Fig. 4b
			4×4	Ti <sub>2</sub> C	144	Fig. 4c
				Ti <sub>2</sub> C (without optimization)	144	Fig. 4d
2×2	Ti <sub>2</sub> C	560	3×3	Ti <sub>2</sub> C	72	Fig. 5a
			4×4	Ti <sub>2</sub> C	144	Fig. 5b
				Ti <sub>2</sub> C (without optimization)	144	Fig. 5c

**Table. S2: The reference values of  $E_{fa}$  of non-optimized  $4 \times 4$   $Ti_{n+1}C_nT_x^1T_{2-x}^2$  ( $n=1,2$ ) lattice(values in () is the calculated values).**

Base and Pair		Elements ratio				
		31:1	30:2	29:3	27:5	24:8
$Ti_2C$	FO	-5.89(-5.58)	-5.84(-5.57)	-5.78(-5.56)	-5.68(-5.53)	-5.54(-5.48)
	FCl	-5.84(-5.53)	-5.75(-5.46)	-5.65(-5.4)	-5.46(-5.27)	-5.17(-5.06)
	FBr	-5.8(-5.51)	-5.66(-5.43)	-5.52(-5.35)	-5.23(-5.19)	-4.83(-4.92)
	FS	-5.8(-5.51)	-5.66(-5.43)	-5.52(-5.34)	-5.24(-5.17)	-4.83(-4.9)
	FSe	-5.79(-5.5)	-5.63(-5.41)	-5.48(-5.31)	-5.17(-5.12)	-4.7(-4.82)
$Ti_3C_2$	FO	-5.37(-5.53)	-5.35(-5.52)	-5.34(-5.51)	-5.32(-5.48)	-5.28(-5.43)
	FCl	-5.33(-5.47)	-5.28(-5.41)	-5.23(-5.34)	-5.14(-5.21)	-5.02(-5)
	FBr	-5.32(-5.46)	-5.27(-5.38)	-5.22(-5.3)	-5.13(-5.13)	-4.99(-4.85)
	FS	-5.3(-5.46)	-5.22(-5.38)	-5.15(-5.3)	-5.01(-5.12)	-4.8(-4.85)
	FSe	-5.3(-5.45)	-5.21(-5.36)	-5.13(-5.27)	-4.98(-5.07)	-4.74(-4.78)

The Supporting information of the calculation method of lithium storage capacity:

Table. S3: The formation energy of a single Li atom adsorbed on  $T_C$ ,  $T_M$ ,  $T_T$  and  $T_t$  sites.

Element pair	Adsorbed sites	Formation energy
FO	$T_M$	-0.76484
	$T_C$	-0.94663
	$T_T$	unfavorable
	$T_t$	-0.93221
FCl	$T_M$	-0.73536
	$T_C$	-0.91498
	$T_T$	-0.22675
	$T_t$	0.33698
FBr	$T_M$	-0.72826
	$T_C$	-0.91475
	$T_T$	-0.21825
	$T_t$	0.48174
FS	$T_M$	-0.78245
	$T_C$	-0.99712
	$T_T$	unfavorable
	$T_t$	-0.43601
FSe	$T_M$	-0.77543
	$T_C$	-0.99603
	$T_T$	unfavorable
	$T_t$	-0.27374

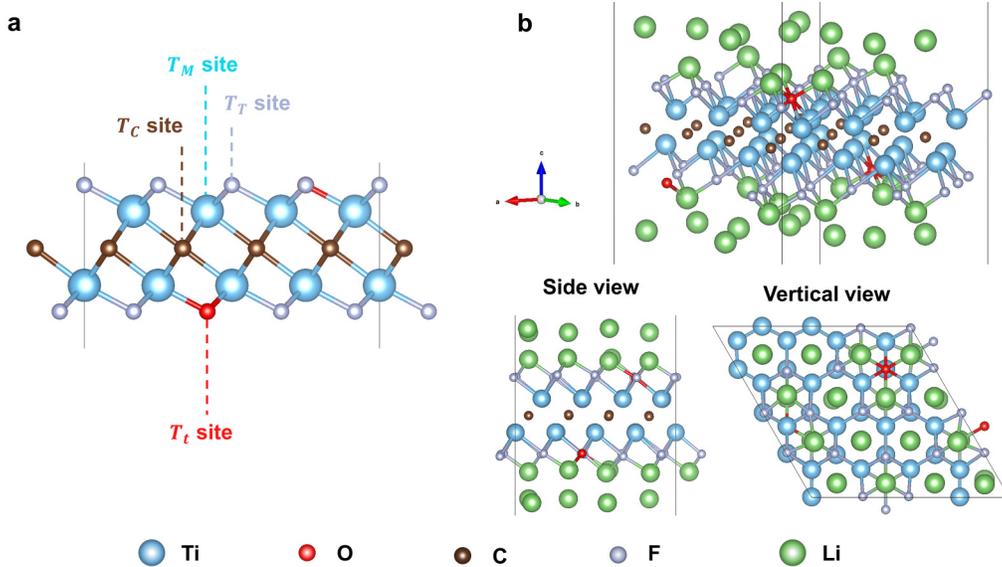


Fig. S8: The verification of the possible adsorbed sites on the surface of MXene. (a) The chosen of the possible adsorbed sites. (b) The calculated structure of FO MXene that fully covers Li atoms on the  $T_C$  sites.

Investigating the formation energy of a single Li atom adsorbed on different sites of  $4 \times 4$   $Ti_2C$  FX MXenes (X represents O, Cl, Br, S Se), we found that Li prefers to occupy the  $T_C$  site compared with  $T_M$  site or  $T_T/T_t$  site, as shown in Table. S3 and Fig. S7a. Thus, we step further to investigate the structures of the same MXenes that fully occupied by Li atoms in all  $T_C$  sites. Fig. S7b shows the structure of the lowest total energy of FO MXene adsorbing Li atoms (FCl, FBr FS, FSe are shown in Fig. S8), but such structure can't form since its formation energy is positive. Apart from FSe and FBr MXenes, most of the Li atoms which are on the surface of MXene can still settle on  $T_C$  sites with the increasing of Li atoms' number. It inspired us to take a new way to add the adsorbed atoms that adheres to the four principles below:

1. The formation energy produced by adding a new Li into the structure must be negative. Known as the  $E_{fi}$  in  $E_{fi} = E_0^1 - E_0^2 - n\mu_{Li}$  must be negative. ( $E_0^1$  and  $E_0^2$  represent the structures that have or haven't added the new atoms.  $n$  is the number of the newly added atoms one time while  $\mu_{Li}$  is the chemical potential of a single Li atom.)

2. After adding the new atoms, the structures that have been optimized by VASP once again should not decompose.
3. The newly added atoms should occupy the  $T_C$  sites as much as possible.
4. The newly added atoms should be adsorbed on the first layer as much as possible.

If any of these four principles don't satisfied in one adding cycle, the newly added atoms will change the original sites to another random sites and the structure will be recalculated, until all the sites have been tried but the newly added Li can't occupy the  $T_C$  or be adsorbed on the first layer while the  $E_{fi}$  is still negative. When the  $E_{fi}$  becomes positive or the structure can't afford to avoid decomposing after an adding cycle, the total number of the adsorbed Li atoms before this cycle will be treated as the theoretical capacity.

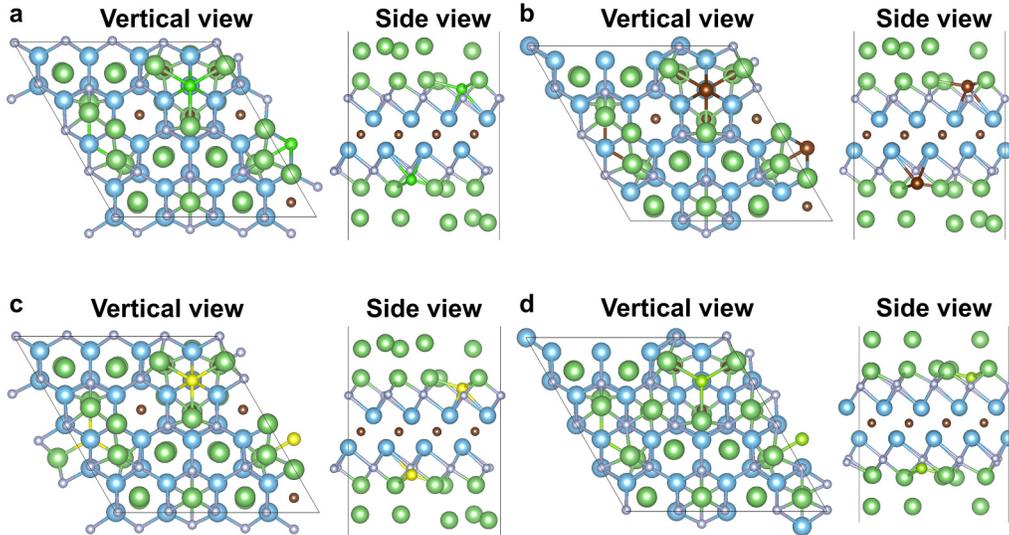


Fig. S8: The optimized structure of (a) FCl, (b) FBr, (c) FS, and (d) FSe MXene fully adsorbing Li atoms on  $T_C$  sites.

Meanwhile, in order to increase efficiency, we chose the element ratio of F:X equals to 30:2 and 24:8, where the "X" atoms can be evenly divided into two sides of the  $Ti_2C$ . So, we can add two Li atoms one time, with the first Li in one side and the second in the other side, utilizing the symmetrical equivalence to reduce errors.