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## **Electronic Supplementary Information**

### Unraveling interfacial compatibility of ultrahigh nickel cathode and chloride

### solid electrolyte for stable all-solid-state lithium battery

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#### **Experimental Section**

**Synthesis of Li<sub>2</sub>TaCl<sub>7</sub> SEs.** The Li<sub>2</sub>TaCl<sub>7</sub> were obtained by ball milling LiCl (anhydrous, Aladdin) and TaCl<sub>5</sub> (anhydrous, Aladdin). The ratio of LiCl and TaCl<sub>5</sub> powders was set as 2:1 and mixed and then loaded into tungsten carbide pots and ball-milled at 450 rpm for 180 h. The mass ratio of tungsten carbide ball mill beads to the precursors is 55:1. The obtained fine solid electrolyte powders were collected and stored in the glovebox for further application and characterization. All of the preparation processes were conducted in an argon-filled glove box with H<sub>2</sub>O and O<sub>2</sub> concentrations less than 0.1 ppm.

**Characterizations.** The obtained solid electrolyte powders were cold-pressed into 10 mm diameter pellets by a hydraulic press (YLJ-15T-LD, Hefei Kejing Materials Technology Co., Ltd.) at 370 MPa for 3 minutes for alternating current (AC) impedance measurement. The ionic conductivity and activation energy were obtained by impedance in the temperature range from 25 to 60 °C with an applied frequency of 1 Hz to 7 MHz and a constant voltage of 20 mV using a Bio-Logic VMP3. Powder X-ray diffraction patterns were obtained by the Philips X'Pert PRO SUPER X-ray diffractometer using Cu K $\alpha$  radiation ( $\lambda$  =1.54178 Å). The exposed morphologies of ASSLB composite cathodes with different cycled states in depth were captured with the application of Ga<sup>+</sup> with a 30 kV focused ion beam (FIB, ORION Nanofab, Zeiss). Prior to the measurement, the cathodes were dismantled from the ASSLBs and stuck on the sample tables carefully in an argon-filled glovebox with H<sub>2</sub>O and O<sub>2</sub> concentrations less than 0.1 ppm.

**Electrochemical measurements of ASSLBs.** Single-crystal LiNi<sub>0.92</sub>Co<sub>0.05</sub>Mn<sub>0.03</sub>O<sub>2</sub> (S-NCM92, provided by Hefei Gotion High-tech Power Energy Co., Ltd.) was firstly mixed with Li<sub>2</sub>TaCl<sub>7</sub> and polytetrafluoroethylene (PTFE) (Guangdong Canrd New Energy Technology Co., Ltd.) with a mass ratio of 75:22:3 and then mixed by a miniature vibration mixer (MSK-SFM-12 M, Hefei Kejing Materials Technology Co., Ltd) for 60 min as cathode composite powders. 100 mg of Li<sub>2</sub>TaCl<sub>7</sub> powder was first placed into a polyetheretherketone (PEEK) model (10 mm diameter) and pressed at 1.5 tons for 1 min to form a SE layer. The composite cathode powder with a range of areal capacity of 1-5 mAh cm<sup>-2</sup> was then spread over the surface of the Li<sub>2</sub>TaCl<sub>7</sub> and pressed at 2.5 tons for another 3 min. 40 mg of Li<sub>6</sub>PS<sub>5</sub>Cl powder was dispersed evenly on the other surface of the Li<sub>2</sub>TaCl<sub>7</sub> layer and then pressed at 3 tons for 3 min. Then, a piece of In foil (0.1 mm thickness, 10 mm diameter, 3A Materials) was attached to the surface of Li<sub>6</sub>PS<sub>5</sub>Cl. A Li foil (6 mm diameter, China Energy Lithium Co., Ltd.) with a weight ratio of Li: In=1:50 was subsequently attached to the In foil. Subsequently, the ASSLB was pressed at 1 ton for

another 1 min and then placed into a custom-made stainless-steel casing (Ningbo Zhengli New Energy Technology Co., Ltd.) with a constantly applied pressure of 100 MPa for 24 hours to form the Li-In alloy. The ASSLB was placed into a custom-made stainless-steel casing for galvanostatic cycling. Galvanostatic cycling (current density: 0.191, 1, and 3 mA cm<sup>-2</sup>) of the ASSLBs was conducted at 30 °C with a constantly applied pressure of 100 MPa. The voltage range of the cycling was 2.8-4.1 V, 2.8-4.3 V, 2.8-4.6 V, and 2.8-4.8 V versus Li<sup>+</sup>/Li using a LAND-CT2001A and Neware-CT-4008T battery cycler. For the aging of ASSLBs, galvanostatic charging was conducted at a current density of 0.191 mA cm<sup>-2</sup>, followed by constant voltage charging and aging at the designated cut-off voltage for a specified duration. All of the preparation processes were conducted in an argon-filled glove box with H<sub>2</sub>O and O<sub>2</sub> concentrations less than 0.1 ppm.

Time-of-flight secondary ion mass spectrometry (ToF-SIMS). For the time-of-flight secondary-ion mass spectrometry (ToF-SIMS) studies, ION-ToF-SIMS 5-100 was used with the pressure of the analysis chamber below  $1.1 \times 10^{-9}$  mbar. The imaging with delay extraction mode with pulsed 30 keV Bi<sup>3+</sup> (0.48 pA pulsed current) ion beam and a cycle time of 100 µs was applied for analysis. To minimize the effect of mass interference, the spectrometry (bunched) mode was used for surface analysis to enable a high signal intensity and a high mass resolution. The analysis area was set to 100 × 100 µm<sup>2</sup> and rasterized with 128 × 128 pixels, and every patch was analyzed with 1 frame and 1 shot per pixel and frame. The primary ion current was ca. 0.48 pA, and the stop condition was set to a primary ion dose of  $10^{12}$  ions cm<sup>2</sup>. To quantitatively compare the specific signal intensity, we collected 10 mass spectra at different areas on each cathode.

### **Supplementary Figures**



**Fig. S1 Characterization of the Li<sub>2</sub>TaCl<sub>7</sub>. a**, PXRD patterns of Li<sub>2</sub>TaCl<sub>7</sub>. The representative diffraction peaks patterns can be indexed with LiCl (ICSD no. 26909). **b**, SEM images for the surface of cold-pressed Li<sub>2</sub>TaCl<sub>7</sub> pellet. **c**, Nyquist plots of the EIS measurement results of Li<sub>2</sub>TaCl<sub>7</sub> with nonreversible electrodes in the temperature range from 25 to 60 °C. **d**, Arrhenius plot of ionic conductivities of the Li<sub>2</sub>TaCl<sub>7</sub>. The ionic conductivity of Li<sub>2</sub>TaCl<sub>7</sub> at room temperature is around 7.07 mS/cm.



**Fig. S2 Characterizations of the S-NMC92. a**, PXRD pattern of S-NMC92 powders. **b**, SEM images of S-NMC92 powders.



**Fig. S3 a-c**, Representative corresponding normalized dQ/dV curves of ASSLBs cycled with cut-off charging voltages of 4.3, 4.6, and 4.8 V, respectively.



**Fig. S4 Evolution of internal resistance of ASSLBs during the cycling process. a-c**, Nyquist plots of the EIS measurement results of ASSLBs under pristine, 200<sup>th</sup>, and 400<sup>th</sup> cycle state.



**Fig. S5 a-l**, Nyquist plots of the EIS measurement results of ASSLBs during the aging process with a constant voltage of 4.3 V.



**Fig. S6 a-l**, Nyquist plots of the EIS measurement results of ASSLBs during the aging process with a constant voltage of 4.6 V.



**Fig. S7 a-l**, Nyquist plots of the EIS measurement results of ASSLBs during the aging process with a constant voltage of 4.8 V.



**Fig. S8 a**, The equivalent circuit employed to fit the Nyquist plots of the EIS results of ASSLBs. **b-d**, Representative fitted results of Nyquist plots of the EIS results of ASSLBs with the cut-off voltage of 4.3, 4.6, and 4.8 V.



**Fig. S9 a-c**, The fitted resistance values of ASSLBs with constant current charged to 4.3, 4.6, and 4.8 V. **d-f**, The fitted resistance values of ASSLBs with constant voltage charged to 4.3, 4.6, and 4.8 V.



Fig. S10 a-b, Schematic illustration of the fabricated ASSLBs and the Focused ion beam technique used to expose the inside cathode. c, SEM image of the exposed surface of the pristine cathode. d-f, SEM images of the exposed surface of the cathode cycled with the charging voltage of 4.3 V at the 50<sup>th</sup>, 200<sup>th</sup>, and 400<sup>th</sup> cycle. g-i, SEM images of the exposed surface of the cathode cycled with the charging voltage of 4.6 V at the 50th, 200th, and 400th cycle. j-l, SEM images of the exposed surface of the cathode cycled with the charging voltage of 4.8 V at the 50<sup>th</sup>, 200<sup>th</sup>, and 400<sup>th</sup> cycle. The physical contact loss was marked by yellow arrows. Scale bar: 1 µm.



**Fig. S11** ToF-SIMS exemplary secondary ion images of negatively charged fragments of the composite cathodes with the OCV and aged state at 4.3, 4.6, and 4.8 V.



**Fig. S12 ToF-SIMS mass spectra of negatively charged fragments. a-d,** Evolution of the mass spectra of  $TaCl_x^-(a)$ ,  $TaClO^-(b)$ ,  $TaO_x^-(c)$ , and  $ClO^-(d)$  fragments of the composite cathodes with the OCV and aged states at 4.3, 4.6, and 4.8 V, respectively.



**Fig. S13 a-b,** Boxplots of the normalized intensity of  $TaO_x^-$  (a) and ClO<sup>-</sup> (b) fragments of the composite cathodes with the OCV and aged states at 4.3, 4.6, and 4.8 V, respectively.



**Fig. S14** ToF-SIMS exemplary secondary ion images of negatively charged fragments of the composite cathodes with the pristine and cycled state at 4.3, 4.6, and 4.8 V.



**Fig. S15 ToF-SIMS mass spectra of negatively charged fragments. a-d,** Evolution of the mass spectra of  $TaCl_x^-$  (a),  $TaClO^-$  (b),  $TaO_x^-$  (c), and  $ClO^-$  (d) fragments of the composite cathodes with the pristine and cycled states at 4.3, 4.6, and 4.8 V, respectively.



**Fig. S16 a-b,** Boxplots of the normalized intensity of  $TaO_x^-$  (a) and ClO<sup>-</sup> (b) fragments of the composite cathodes with the OCV and cycled states at 4.3, 4.6, and 4.8 V, respectively.



Fig. S17 Scheme diagram of the degradation pathways of Li<sub>2</sub>TaCl<sub>7</sub>-LiNi<sub>0.92</sub>Co<sub>0.05</sub>Mn<sub>0.03</sub>O<sub>2</sub>.

**Note S1**. When operated at low cut-off voltages, stable cycling can be achieved due to the good oxidation stability of Li<sub>2</sub>TaCl<sub>7</sub> and reversible lithiation and de-lithiation processes of S-NCM92, particularly at cut-off voltages of 4.3 V and notably at 4.1 V. However, when operated at higher cut-off voltages of 4.6 and 4.8 V, oxidation reactions in Li<sub>2</sub>TaCl<sub>7</sub> occur, leading to the decomposition of the amorphous matrix. This decomposition disrupts the Li-ions conduction network, significantly increasing the resistance of migration of Li-ions and polarization of ASSLBs. Meanwhile, the lattice oxygen liberated from the charged S-NCM92 predominantly interacts with the TaCl<sub>6</sub><sup>-</sup> sublattice within Li<sub>2</sub>TaCl<sub>7</sub>, leading to the formation of Ta-Cl-O and Ta-O species. These reactions generate side-products that obstruct Li-ion migration, significantly increasing interface resistance and adversely affecting the ASSLBs' capacity and lifespan.



Fig. S18 a, Electrochemical curves of ASSLBs first charged to 4.1 V and then aged and held for 240 h.b-n, Nyquist plots of the EIS results of ASSLBs during the aging process with a constant voltage of 4.1 V.



**Fig. S19 Interface stability of ASSLBs as a function of aging time with a charging voltage of 4.1 V revealed by in-situ EIS**. Evolution of fitted electrolyte resistance, cathode-electrolyte interface resistance, and anode-electrolyte interface resistance plotted as a function of aging time.



Fig. S20 a-b, Comparison of electrochemical performance of top-notch chloride SE-based ASSLBs, including areal capacity (a) and lifespan (b).

# **Supplementary Tables**

**Table S1**. Comparison of electrochemical performance, including discharge capacity and lifespan, of chlorides-based ASSLB with designed areal capacity over 3 mAh cm<sup>-2</sup> operated with different cut-off voltages, as exhibited in **Figure S20**.

Chlorides	Cathodes	Cut-off voltage (V)	Loading (g cm <sup>-2</sup> )	Current (mA cm <sup>-2</sup> )	Capacity (mAh g <sup>-1</sup> )	Cycles (no.)	Retention (%)	References
Li <sub>2</sub> In <sub>0.33</sub> Sc <sub>0.67</sub> Cl <sub>4</sub>	LCO	4.5	52.46	1.2	80	500	100 (50 °C)	Nat. Energy 2022, 7(1): 83-93
	NCM85	4.3	21.59	0.49	172	80	108	
Li <sub>3</sub> InCl <sub>6</sub>	NCM622	4.2	11.9	0.17	125	70	60	ACS Energy Lett. 2022, 7, 2979–2987
		4.3			130		35	
		4.4			160		46	
		4.5			175		34	
Li2TaCl7	NCM92	4.3	24.3	1	187.1	110	70	Our work
			25.2	3	144.1	220	72	
		4.1	23.6	1	150.2	200/300	91/80	
			22.2	3	112.1	600	82	