Supplementary Information for

Conductive RuO₂ Binders Enhance Mechanical Stability of Macroporous Nb-SnO₂ Particles as Cathode Catalyst Supports for High-Performance PEFCs

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Figure S1. SEM images of macroporous NTO/RuO₂–50 particles with different PMMA concentrations: (a) 0 wt.%, (b) 0.5 wt.%, (c) 1.0 wt.%, and (d) 2.0 wt.%.

We systematically investigated the effect of PMMA template concentration on the macroporous structure of NTO/RuO₂ particles by employing a constant RuO₂ binder concentration of 50 wt%. The morphological changes induced by different PMMA concentrations were characterized using SEM. In the absence of a PMMA template, the NTO/RuO₂ particles formed spherical structures of various sizes through aggregation. This phenomenon was attributed to the presence of only RuCl₃ and NTO nanoparticles in the precursor droplets, where NTO nanoparticles self-assembled into spherical shapes to minimize free energy, with the RuO₂ binder precipitating between the NTO nanoparticles. Upon the introduction of the 0.5 wt.% PMMA templates, a welldefined macroporous framework was observed, as illustrated in Figure S1b. When the PMMA template content was increased to 1.0 wt.%, the macroporous framework was preserved, and the number of open macropores on the particle surfaces increased, as shown in Figure S1c-d. The increase in the number of open macropores could enhance mass transport within the reaction environment, thereby improving the performance of PEFCs. However, a further increase in the PMMA concentration resulted in more particles with disrupted frameworks because the formation of additional open macropores led to a reduction in the thickness of the macroporous framework, causing structural instability and eventual collapse. Notably, the morphology of the final product is strongly influenced by the PMMA concentration within the precursor droplets. Although increasing the PMMA concentration in the precursor droplets increased the number of macropores in the final product, excessively high PMMA template concentrations diminished the thickness of the framework, making it too fragile to maintain the spherical morphology of the macroporous particles.



Figure S2. STEM images of macroporous NTO/ RuO_2 particles synthesized with

different concentrations of RuO_2 binder.

 Table S1. BET surface area measurements

Sample	S_{BET} (m ² /g)					
NTO nanoparticles	106.9					
NTO/RuO ₂ -20	71.1					
NTO/RuO ₂ -20	38.9					
NTO/RuO ₂ -30	38.1					
NTO/RuO ₂ –50	30.4					

* Calculated using the BET equation.



Figure S3. BET nitrogen adsorption isotherm plots of NTO nanoparticles and macroporous NTO/RuO₂ particles.

								R _{exp}	R _{wp}	R _p	Crytal
								(%)	(%)	(%)	size
			Site	X	У	Z	Occ				(nm)
Sn4 NTO/RuO ₂ - Ru 20 R meta	SnO ₂	Sn	2	0	0	0	1				17.0
		0	4	0.305	0.305	0.305	1				17.9
	BuO	Ru	2	0	0	0	1	- 200	6 72	5 17	0.21
	KuO ₂	0	4	0.305	0.305	0	1	2.88	0.73	5.47	9.31
	Ru metallic	Ru	2p	0.333	0.667	0.250	1				17.3
	SnO ₂	Sn	2	0	0	0	1	2.90	7.47	5.92	23.0
		0	4	0.305	0.305	0.305	1				
NTO/RuO ₂ -	RuO ₂	Ru	2	0	0	0	1				20.2
30		0	4	0.305	0.305	0	1				20.3
	Ru metallic	Ru	2p	0.333	0.667	0.250	1	•			48.4
	SnO	Sn	2	0	0	0	1				10.0
	51102	0	4	0.305	0.305	0.305	1				17.7
NTO/RuO ₂ -	RuO ₂	Ru	2	0	0	0	1	- 2 74	6.20	4 86	23.0
50		0	4	0.305	0.305	0	1	2.74	0.20	т.00	23.0
	Ru metallic	Ru	2p	0.333	0.667	0.250	1	-			46.7

Table S2. Fractional coordinates and occupancies for macroporous NTO/RuO2 particlessynthesized with different RuO2 binder contents.



Figure S4. SEM images of aggregated NTO/RuO₂ particles with different RuO₂ binder contents: (a) agg-NTO/RuO₂-0, (b) agg-NTO/RuO₂-20, (c) agg-NTO/RuO₂-30, and (d) agg-NTO/RuO₂-

50 particles.



Figure S5. XRD patterns of NTO nanoparticles.



Figure S6. (a)TEM image and (b) size distribution chart of NTO nanoparticles.

In our previous study, we successfully used NTO nanoparticles to prepare macroporous NTO/RuO₂ particles. NTO nanoparticles were synthesized using FSP, and their structural characteristics were analyzed using X–ray diffraction. The XRD patterns reveal that the NTO nanoparticles have a tetragonal SnO₂ structure, as illustrated in **Figure S5**. No discernible diffraction peaks corresponding to Nb are observed. The crystallite size of the NTO nanoparticles was determined to be approximately 9.95 nm using the Scherrer equation. The size distribution of these NTO nanoparticles was assessed based on TEM image analysis using the ImageJ software, and the resulting histogram of the size distribution is presented in **Figure S6**. The average nanoparticle size, as determined from TEM analysis, was 9.54 nm. A slight disparity in crystallite size was observed between the XRD and TEM analyses. This variance can be attributed to the unique necking structure inherent to the NTO nanoparticles synthesized via FSP. These NTO nanoparticles served as precursors for the synthesis of macroporous NTO/RuO₂ particles.



Figure S7. Cyclic voltammograms of carbon, NTO nanoparticles, and macroporous NTO/RuO₂ particles with varying RuO₂ binder contents before and after applying a potential of 1.3 V vs Ag/AgCl for 10 minutes.