Electronic Supporting Information:

Promoting Overall Sulfur Redox Kinetics for Li-S Batteries via Interfacial Synergy in a NiS-NiTe₂ Heterostructure-Modified Separator

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Electronic Supplementary Information contains Experimental Section and Fig. S1-S9.

Experimental Section:

Chemicals: All chemicals were analytical grade and used without further purification. Nickel nitrate hexahydrate (Ni(NO₃)₂·6H₂O) was purchased from Xiya Reagent Co., Ltd. Urea was obtained from Sinopharm Chemical Reagent Co., Ltd. Lithium sulfide (Li₂S), sulfur powder (S) and tellurium powder (Te) were procured from Shanghai Aladdin Bio-Chem Technology Co., Ltd. Multi-walled carbon nanotubes (CNTs, >99.9 %) were acquired from Nanjing Xianfeng Nano Co., Ltd.

Synthesis of $Ni(OH)_2$ nanoflowers: Ni(OH)_2 nanoflowers (Ni(OH)_2 NF) was synthesized using a hydrothermal method as per established literature procedures.^{S1} Specifically, 0.7 g of Ni(NO₃)₂·6H₂O and 0.3 g of urea were dissolved in 35 mL of distilled water. Following thorough stirring to ensure homogeneity, the resulting mixture was transferred to a 50 mL Teflon-lined autoclave and maintained at 120 °C for 12 h. The resulting sample was collected through centrifugation, subjected to multiple water rinses, and subsequently dried under vacuum conditions at 60 °C for 12 h.

Synthesis of NiS, NiTe₂, NiS-NiTe₂: A total of 50 mg of Ni(OH)₂ NF was positioned in a porcelain boat downstream, while mixed powders of S/Te, each weighing 100 mg, were placed in another porcelain boat upstream. The annealing process was conducted up to 600 °C for 2 h at a heating rate of 1 °C min⁻¹ under a N₂ atmosphere, resulting in the formation of NiS-NiTe₂. For comparison, NiS and NiTe₂ samples were obtained by incorporating only 100 mg of S or 100 mg of Te through the same aforementioned procedure, respectively.

Synthesis of modified separators and sulfur cathodes: The NiS-NiTe₂ modified separator was fabricated using a straightforward surface coating method. Specifically, 70 wt% of NiS-NiTe₂, 20 wt% of carbon nanotubes (CNTs), and 10 wt% of polyvinylidene fluoride (PVDF) were meticulously blended in 1-methyl-2-pyrrolidinone (NMP) solvent through vigorous stirring to form a homogeneous slurry. Subsequently, this slurry was coated onto a pristine Celgard 2500 polypropylene (PP) separator. The functionalized separator underwent overnight drying at 45 °C under vacuum conditions, followed by cutting into 19 mm diameter discs. The areal mass loading of the interlayer materials was approximately 0.3 mg cm⁻².

As for the sulfur-carbon nanotube (CNT-S) composite, it was prepared using a melt diffusion approach. Specifically, CNTs and sulfur powder were ground in a 1:4 weight ratio and subjected to heating at 155 °C for 10 h.

Synthesis of lithium polysulfide (Li_2S_6) solution and visual adsorption test: A Li_2S_6 solution was meticulously prepared by dissolving elemental sulfur and Li_2S (in a 5:1 molar ratio) in a mixed solvent comprising 1,2-dimethoxyethane (DME) and 1,3-dioxolane (DOL) (v/v = 1:1). The solution underwent vigorous stirring at 65 °C for 12 hours. Subsequently, 10 mg samples were introduced into 3 mL of a 5 mM Li_2S_6 solution. UV-Vis absorption spectra were acquired ex-situ after immersing the samples in the Li_2S_6 solution for 12 h. All these procedures were executed within an argon-filled glove box to maintain an inert atmosphere. Symmetric-cell assembly and evaluation of polysulfide conversion kinetics: The electrode was meticulously fabricated by blending active materials and polyvinylidene fluoride (PVDF) in a weight ratio of 9:1 within N-methyl-2-pyrrolidone (NMP) solvent. The resulting slurry was then coated onto an aluminum foil substrate. A 0.5 M Li₂S₆ electrolyte was prepared by stirring sulfur and lithium sulfide in a 1 M solution of bis(trifluoromethanesulfonyl)imide lithium salt (LiTFSI) in a solvent mixture of 1,3-dioxolane (DOL) and 1,2-dimethoxyethane (DME) (v/v = 1:1) at 65 °C within an argon-filled glove box overnight. For the electrochemical testing, two identical electrodes were employed as working and counter electrodes, both exhibiting a mass loading of 1.0 mg cm⁻². The electrolyte consisted of 25 μ L of 0.5 M Li₂S₆. Cyclic voltammetry measurements of the symmetric cells were conducted over a voltage window spanning from -1.0 to 1.0 V at various scan rates. Electrochemical impedance spectra were acquired across a frequency range of 0.01-10⁵ Hz.

 Li_2S nucleation measurements: For the liquid-solid conversion kinetics, a 0.2 M Li₂S₈ solution was prepared by blending elemental sulfur and Li₂S (in a 7:1 molar ratio) and dissolving the mixture in a tetraglyme solution with vigorous stirring. Investigations into the precipitation of Li₂S on various reactive surfaces were carried out using 2025 coin-type cells, employing a Celgard 2500 polypropylene (PP) membrane separator. In these experiments, 25 μ L of 0.5 M Li₂S₈ was applied to the cathode, and 25 μ L of LiTFSI electrolyte was added to the anode side. The cells underwent galvanostatic discharge at 0.1 mA down to 2.06 V, followed by maintaining the voltage at 2.05 V to allow Li₂S nucleation until the current dropped below 10⁻⁵ A. The Li₂S nucleation capacity was determined by calculating the integral area of the curve using Faraday's Law.

Li-S cell assembly and electrochemical performance measurements: The sulfur cathode materials (CNT/S) were mixed with Super P and PVDF in a weight ratio of 8:1:1 in NMP and stirred to obtain a uniform slurry, which was then coated on aluminum foil, vacuum-dried at 45 °C overnight, and cut into 12 mm-diameter disks to form the sulfur electrodes. The sulfur mass loading was 1 mg cm⁻². Li metal and Celgard 2500 PP were used as the anode and

separator, respectively. 1.0 M LiTFSI in DOL and DME (v/v =1:1) with 1 wt% LiNO₃ additive was used as the electrolyte. Electrochemical tests were conducted by assembling 2025 coin-type half-cells in an argon-filled glove box where the oxygen and water contents were below 0.1 ppm. The galvanostatic discharge/ charge measurements were conducted with a LAND CT2001A system (China) within a voltage window of 1.7-2.8 V under various C-rates (1 C=1675 mA g⁻¹). Cyclic voltammetry was performed with Chenhua electrochemical workstation (CHI 660E). Electrochemical impedance spectroscopy was conducted with Chenhua electrochemical workstation (CHI 660E) over the frequency range 0.01-10⁵ Hz. All electrochemical experiments were performed at 28 °C. The specific capacity of the Li-S battery was calculated according to the sulfur mass.

Materials characterization: Powder X-ray diffraction (PXRD) was executed using a X'Pert3Powder instrument (PANalytical, Netherlands) employing Cu Kα radiation at room temperature. Scanning electron microscopy (SEM) was carried out on a Nova NanoSEM 450 field-emission scanning electron microscope with acceleration voltages set at 15 kV. Transmission electron microscopy (TEM) and energy-dispersive X-ray spectroscopy (EDS) were performed utilizing a JEM-1400-plus instrument (Japan). Raman spectroscopy was conducted on an InVia-Reflex system by Renishaw, utilizing a laser source with a wavelength of 532 nm. X-ray photoelectron spectroscopy (XPS) data were acquired using an ESCALAB 250Xi instrument from Thermo Fisher Scientific. Thermo-gravimetric analysis (TGA) was conducted on an STA499f5 analyzer, applying a heating rate of 10 °C min⁻¹. Ultraviolet-visible (UV-Vis) spectra were obtained using a Hitachi U-3900H UV/Vis spectrophotometer.

Theoretical calculations: Theoretical calculations based on Density Functional Theory (DFT) were conducted in this study using the Vienna ab initio simulation package (VASP). Projector-augmented-wave (PAW) potentials were employed, and the generalized gradient approximations (GGA) of Perdew–Burke–Ernzerhof (PBE) pseudopotentials were applied for the exchange–correlation functional. The adhesion bonding energy (E_b) of Li₂S_n (n = 2, 4, 6, 8) was computed using the formula $E_b = E_{M/Li2Sn} - E_M - E_{Li2Sn}$, where $E_{M/Li2Sn}$ represents the total

energy of the adsorbed system, E_M is the total energy of the surface, and E_{Li2Sn} is the total energy of the isolated Li_2S_n polysulfide. In this context, more negative values indicate a greater likelihood of adsorbate-surface interactions due to the energy advantage.



Fig. S1 (a) XRD pattern of Ni(OH)₂; (b) SEM of Ni(OH)₂.



Fig. S2 XRD patterns of (a) NiS and (b) $NiTe_2$.



Fig. S3 SEM image of NiS-NiTe $_2$ at low magnification.



Fig. S4 (a) SEM of NiS, (b, c) corresponding elemental mapping of NiS.



Fig. S5 (a) SEM of NiTe₂, (b and c) corresponding elemental mapping of NiTe₂.



Fig. S6 SEM and cross-sectional SEM images of (a, b) PP separator and (c, d) NiS-NiTe₂ modified separator.



Fig. S7 Cyclic voltammetry curves of Li-S cells paired with (a) NiS and (b) NiTe₂.



Fig. S8 Cycling performances of the NiS-NiTe₂, NiS and NiTe₂-based Li-S batteries at the current density of 1 C.



Fig. S9 Geometrically stable configurations of $LiPS_n$ adsorption of NiS-NiTe₂, NiS and NiTe₂.

References:

S1. C. Gao, G. Wei, C. Wang, X. Zhou, X. Zhao, Q. Zhao, S. Wang and F. Kong, J. Alloys Compd, 2023, 959, 170545.