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Supporting information

1 Negative Space Charge Modulated Ion Transport through

2 PEDOT: PSS Hydrogel Integrating Nanofluidic Channels for

3 High-Efficient Osmotic Energy Harvesting

4 Rui Zhu,^a Peng Sun,^d Guofeng Cui,^{a*} Yaoguang Yu,^{b*} Shaojun Ke,^c Jie Zhao,^c

10 ^d Guangdong Provincial Engineering Technology Research Center of Vacuum Coating Technologies and New Energy Materials, Siyuan

12 University, Guangzhou, Guangdong 510632, People's Republic of China.

^{5 &}lt;sup>a</sup>. Key Laboratory for Polymeric Composite & Functional Materials of Ministry of Education, Key Laboratory of Low-carbon Chemistry &

⁶ Energy Conservation of Guangdong Province, School of Chemistry, Sun Yat-sen University, Guangzhou 510275, P. R. China. E-mails:

^{7 &}lt;u>cuigf@mail.sysu.edu.cn</u>.

⁸ b. School of Materials, Sun Yat-sen University, Shenzhen 518107, P. R. China. E-mails: yuyg5@mail.sysu.edu.cn.

⁹ c School of Mechanical and Automotive Engineering, South China University of Technology, Guangzhou 510640, P. R. China.

¹¹ Laboratory, Guangzhou Key Laboratory of Vacuum Coating Technologies and New Energy Materials, Department of Physics, Jinan

14 Experimental Methods

To further verify the effect of the thickness and composition (PEDOT: PSS) of the hydrogel 15 membrane on the acquisition of osmotic energy, different single-factor experiments were executed 16 separately. Firstly, hydrogel membranes of 100 µm, 200 µm, 300 µm, 400 µm, 500 µm, and 600 17 18 µm sizes were prepared and assembled into the osmotic energy acquisition device to verify the effect of membrane thickness on osmotic energy performance with the concentration gradient of artificial 19 seawater and river water, respectively. Then, the thickness of the membrane was fixed at 236 µm, 20 21 and the effect of PEDOT: PSS addition on the output power was verified by controlling the amount of PEDOT: PSS added (1%, 3%, 5%, 7%, 10%, 15%, 20%, and 25%) with the concentration 22 gradient of artificial seawater and river water. 23



26 Fig. S1 Schematic of synthesis of conventional permeable membrane. The preparation of permeable

27 membranes by composite/ modified polymers with 2D materials was previously considered a viable

- 28 option, but the complicated process and lower stability posed significant challenges to its advancement.
- 29



31 Fig. S2 (a)-(c) C1s XPS spectra and (d)-(f) S2p XPS spectra of PEDOT: PSS hydrogel.

As depicted in Fig. 1(b), the shift of the $C\alpha = C\beta$ bond in PEDOT to lower 32 wavelengths, the slight shift of the C-S-C bond (579 cm⁻¹) to shorter wavelengths, 33 coupled with its significant enhancement upon H₂SO₄ doping, which these indicate the 34 increase of quinone structure in the main chain of PEDOT. As shown in Fig. S2(a)-(c), 35 the peak at 284 eV corresponds to the C-C/C=C bond in PSS, while the peak at 286 eV 36 corresponds to the C-O/C-S bond in PEDOT. Obviously, the increased ratio of C-37 C/C=C to C-O/C-S indicates that more sulfonates in the PSS chain segments form 38 hydrogen bonds with PVA or PA. Meanwhile, in the S(2p) XPS spectra (Fig. S2(d)-39 (f)), the peaks at 166-172 eV and 161-166 eV were attributed to the sulfur atoms in PSS 40 and PEDOT, respectively. The proportion of sulfur elements in PSS and PEDOT 41 gradually increases with further doping cross-linking, which was attributed to the 42 43 substitution of partial PSS by PA during the cross-linking process and the doping of H_2SO_4 . 44







50 Fig. S4 Conductivity test results of PEDOT: PSS hydrogels adding different amounts of (a) GA, (b) PA

51 and (c) H_2SO_4 , respectively.





57 Fig. S6 Nano-channel computational modeling.

59 MEASUREMENT OF OSMOTIC ENERGY CONVERSION

As shown in Fig. 4(f), the whole circuit diagram consists of three parts: the electrode, membrane, and external circuit. Among them, the membrane includes membrane resistance r_{int} and transmembrane diffusion potential. r_{int} denotes the ion transport resistance in the ion-selective membrane. E_{diff} is generated by the concentration gradient on both sides of the membrane, which can be calculated by the following equation:

$$E_{diff} = (2t_{+} - 1)\frac{RT}{zF}\ln\left(\frac{\gamma_{H}C_{H}}{\gamma_{L}C_{L}}\right)$$

66

where *R* is the universal gas constant, *T* is the Kelvin temperature, *z* is the ion valence state, and *F* is the Faraday constant. C_H and C_L are the ion concentrations at high concentrations and low concentrations, respectively. γ_H and γ_L are the ion activity coefficients of the high-concentration solution and low-concentration solution, respectively.

The measured
$$V_{oc}$$
 is the sum of E_{diff} and E_{redox} as in the following formula:
 $V_{oc} = E_{diff} + E_{redox}$

The electrodes can be abstracted as the redox potential E_{redox} , which is derived from the electrode potential drop when the two electrodes enter solutions of different concentrations. The E_{redox} can be calculated by the following equation:

$$E_{redox} = \frac{RT}{zF} \ln \left(\frac{\gamma_H C_H}{\gamma_L C_L} \right)$$

The external circuit consists of an external load r_L and an ammeter used to measure the current in the circuit.



- 82 Fig. S7 The device stability of power generation.
- 83



85 Fig. S8 The influence of the hydrogel membrane thickness in osmotic energy performance.



88 Fig. S9 The influence of PEDOT: PSS addition in osmotic energy performance.



91 Fig. S10 Schematic representation of the ICEG setup.

92 Table S1 Electrochemical parameters of a single unit of the ICEG.

	<i>V</i> _{0 (mV)}	V _{theo (m} V)	Permselectivity (%)	Conversion efficiency
				(%)
Configuration (I)	396	414	95.6	45.8

The parameters of a single unit of the PEDOT: PSS membrane pair were measured 93 in an ICEG device (Fig. S10). Ag/AgCl was used as the cathode and anode. The high 94 concentration (HC) and low concentration (LC) of the salt solutions were 3.0 M and 95 0.3 μ M, respectively. The membrane permselectivity (α) could be calculated as the 96 ratio between the measured membrane potential (V_{meas}) and the theoretical membrane 97 potential (V_{theo}): ^{21, 22} 98

$$\alpha(\%) = \frac{V_{meas}}{V_{theo}} \times 100$$

9

100

$$V_{theo} = N \frac{RT}{zF} \ln\left(\frac{a_H}{a_L}\right)$$

where N is the number of membranes, α is the average membrane permselectivity of 101 an anion and a cation exchange membrane, R is the universal gas constant (8.314 J·mol-102 103 ¹·K⁻¹), T is the absolute temperature (K), Z is the electrochemical valence, F is the

104 Faraday constant (96485 C·mol⁻¹), a_H is the activity of the concentrated solution 105 (mol·L⁻¹), and a_L is the activity of the diluted solution (mol·L⁻¹). For seawater (3.0 M 106 KCl) and river water (0.3 μ M KCl) the theoretical voltage difference over each 107 membrane for 100% selective membranes is 414 mV.

108 The electrochemical conversion efficiency (η) can be calculated by the following 109 formula: ^{23, 24}

$$\eta_{max} = \frac{\alpha^2}{2}$$

Testing Energy conversion Material Solution Ref. area performance (mm^2) Artificial seawater PET membrane 0.03 $0.86 \ W/m^2$ [54] and river water PAM hydrogel Artificial seawater 28 $0.37 \ W/m^2$ [55] membrane and river water **SPEEK** Artificial seawater 28 $5.8 W/m^2$ [56] membranes and river water HEMAP hydrogel Artificial seawater 5.38 W/m² 0.03 [57] membrane and river water Loofah sponge Artificial seawater 0.0183 W/m² 60 [58] membranes and river water Artificial seawater MMT 0.03 2.1 W/m^2 [59] and river water membranes KANF Artificial seawater 4.8 W/m^2 0.03 [60] membranes and river water Artificial seawater 4.94 W/m^2 GOMs membrane 0.03 [61] and river water Artificial seawater 2.2 W/m^2 PPC membrane 0.075 [62] and river water

¹¹² Table S2 Comparison of osmotic energy conversion performance.

hydrogel	and river water			work
PEDOT: PSS	Artificial seawater	38.5	6.26 W/m ²	This
MMT membranes	and river water	0.03	4.13 W/III	[05]
	Artificial seawater		4.13 W/m^2	[65]
Glass membrane	and river water	0.05	4.10 W/III	[04]
	Artificial seawater	0.03	4.16 W/m^2	[64]
FOF membrane	solution	0.02	<i>5.7 w/m</i>	[05]
	0.5M/1mM NaCl	0.02	$5.7 W/m^2$	[63]

113 Note: Composition of simulated seawater (NaCl: 0.454M; MgSO4: 0.028M; MgCl2: 0.0257M; CaCl2:

114 0.0103M; KCl: 0.00972M; NaHCO₃: 0.00241M) and river water (NaCl: 0.01M).



- 116 In Concentration-induced Electricity G
- 117 Fig. S11 Electrochemical cell series device.



- 120 Fig. S12 River water pre-treatment unit. In practical application, flowing river water flows into the series-
- 121 connected ICEG devices through serpentine microchannels and settling tanks, which can effectively
- 122 buffer the current impact and reduce impurities.
- 123



125 Fig. S13 Schematic diagram of ICEG devices.



Fig. S14 (a) Schematic diagram of the electrical double layer mechanism (EDL) next to a negatively charged wall. (b) Diffusion-osmotic current induced by salt concentration difference. (c) Sketch of the nanofluidic osmotic energy conversion model. (d) Electrical distribution from charged wall to bulk solution.

132

133 We built a two-dimensional model to illustrate the osmotic energy conversion process, as shown in Fig. S14c. The model consists of a high concentration of seawater, a low concentration of river water 134 135 and a hydrogel nanochannel membrane. In the negatively charged nano-channels, the electrostatic force repels ions with the same charge (co-ions, negatively charged Cl⁻ ions in our work) near the channel 136 137 surface and attracts ions with the opposite charge (counter-ions, positively charge K⁺ ions in our work) to compensate the fixed charges (in our study negatively charges introduced by sulphuric or phosphoric 138 acid groups) on the surface, thus leading to a charge redistribution.²⁵ However, only a fraction of counter-139 140 ions ends up at the charged wall. A graded electric double-layer (abbreviated as EDL) is created between 141 the nano-channel wall and the electrolyte solution, with an increased concentration of counter-ions and 142 a decreased concentration of co-ions (Fig. S14a). The EDL refers to the Stern and diffusion layers. ²⁶⁻²⁸

The Stern layer is comprised entirely of counter-ions due to contact interactions. The diffusion layer is 143 the region from the Stern layer to the bulk solution, where the distribution of free ions is still governed 144 by the Coulomb force, electrically screened by the Stern layer. As depicted in Fig. S14b, the interaction 145 between counterions in the nano-channel and the charge of its inner wall is usually simplified by an 146 exponentially decaying potential.²⁹ Through this interaction, by which way the channel is filled with a 147 148 close to unipolar solution of counter-ions, and ion selectivity of transport can be observed. Based on the ion-selectivity mechanism, a streaming current is generated when the ionic solution is propelled by an 149 150 external mechanical driving force (a transmembrane concentration difference) through charged 151 nanofluidic channels (Fig. S14d). 30-34 In this way, the net charge (counter-ions, K+ in our work) within the diffused double layer moves at a certain velocity under the effect of a concentration gradient. At this 152 point, counter-ions (K⁺ ions in this work) spontaneously and preferentially diffuse along the diffused 153 double layer within the nano-channels. Ideally, the transport of counter-ions is proportional to the charge 154 155 density on the surface of the nano-channel. Conductance measurements reveal the charge-control ion 156 transportation within the PEDOT: PSS hydrogel membrane, as shown in Fig. 3(c), where the measured ionic conductance exhibited two distinct characteristic behaviors. ³⁵ Specifically, above the 0.5M 157 concentration point, the ionic conductance of the membrane follows the bulk rule of linearly increasing 158 159 with concentration. However, the conductance deviates from the bulk value and gradually approached 160 smoothing at concentrations less than 0.5M. In the low concentration region, the selective transport of ions tends to be controlled by the double electric layer within the nano-channel, as depicted in Fig. S14d. 161 162 Because of the negatively charged surface in the channel (the intrinsic negative charge of the nanochannel inner wall is introduced by sulfate or phosphate groups during the synthesis of the hydrogel), K⁺ 163

- 164 ions with opposite charges will be separated from identically charged Cl- ions in the channel, creating a
- 165 unipolar environment.

	C	Functional	Size for single	Price	
Membrane	Component	Group	sheet		
AMI-7001S	Gel polystyrene	Onetownow	1 2 x 1 0 x	\$350	
anion exchange	cross linked with		Quaternary 1.2 m × 1.0 m ×	per	
membrane ^a	divinylbenzene	Ammonium	0.43 mm	sheet	
CMI-7000S	Gel polystyrene	Carlanh and a	1 2 x 1 0 x	\$350	
cation exchange	cross linked with	Sulphonic	1.2 m × 1.0 m ×	per	
membrane ^a	divinylbenzene	Acid	0.43 mm	sheet	
PEDOT: PSS	PEDOT: PSS	Phytic Acid	1.2 m × 1.0 m ×	\$10 per	
membrane	hydrogel	PSS	1.0 mm	sheet	

167 Table S3 Price Comparison of Permeable Membranes.

168 ^a The commercial ion exchange membranes were provided by Membrane International Inc.

170 References

- 171 1. W. H. Shi, Z. W. Wang, H. Song, Y. Z. Chang, W. J. Hou, Y. P. Li, G. Y. Han, ACS
- 172 Appl. Mater. Interfaces, 2022, **14**, 35114-35125.
- 173 2. S. T. Wang, G. Q. Guo, X. X. Lu, S. M. Ji, G. X. Tan, L. Gao, *ACS Appl. Mater.*174 *Interfaces*, 2018, **10**, 19133-19142.
- 175 3. S. M. Zhang, Y. H. Chen, H. Liu, Z. T. Wang, H. N. Ling, A. Khademhosseini, *Adv. Mater.* 2020, **32**, 1904752.
- 177 4. Y. Xu, P. A. Patsis, S. Hauser, J. Pietzsch, Y. X. Zhang, Adv. Sci. 2019, 6, 1802077.
- 178 5. L. V. Kayser, M. D. Russell, D. Rodriquez, S. N. Abuhamdieh, D. J. Lipomi, Chem.
- 179 *Mater.* 2018, **30**, 4459-4468.
- 180 6. J. N. Arthur, S. Burns, C. M. Cole, Q. T. Barthelme, S. D. Yambem, *J. Mater.*181 *Chem. C* 2022, **10**, 13964.
- 182 7. N. M. Badawi, M. Bhatia, S. Ramesh, K. Ramesh, B. Shaik, S. F. Adil, *Polymers*2023, 15, 571.
- 8. Q. Wu, J. J. Wei, B. Xu, X. H. Liu, H. B. Wang, W. Wang, Q. G. Wang, W. G. Liu, *Scientific Reports* 2017, 7, 41566.
- 186 9. B. Y. Lu, H. Yuk, S. T. Lin, N. N. Jian, K. Qu, J. K. Xu, X. H. Zhao, *NATURE COMMUNICATIONS* 2019, **10**, 1043.
- 188 10. K. Ren, Y. Cheng, C. Huang, R. Chen, Z. Wang, J. Wei, *J. Mater. Chem. B* 2019,
 7, 5704.
- 11. J. S. Chen, J. F. Liu, T. Thundat, H. B. Zeng, ACS Appl. Mater. Interfaces 2019,
 11, 18720-18729.
- 12. J. Q. Han, Q. Q. Ding, C. T. Mei, Q. L. Wu, Y. Y. Yue, X. W. Xu, *Electrochimica Acta* 2019, **318**, 660-672.
- 13. T. Wang, Y. Zhang, Q. C. Liu, W. Cheng, X. R. Wang, L. J. Pan, H. X. Xu, *Adv. Funct. Mater.* 2018, 28, 1705551.
- 14. J. S. Chen, Q. Y. Peng, T. Thundat, H. B. Zeng, *Chem. Mater.* 2019, **31**, 45534563.
- 198 15. Y. Shi, M. Wang, C. B. Ma, Y. Q. Wang, X. P. Li, G. H. Yu, Nano Lett. 2015, 15,

- 199 <u>6276-6281</u>.
- 200 16. M. Kotal, S. K. Srivastava, B. Paramanik, J. Phys. Chem. C 2011, 115, 1496-1505.
- 201 17. Z. X. Deng, Y. Guo, X. Zhao, P. X. Ma, B. L. Guo, *Chem. Mater.* 2018, **30**, 1729202 1742.
- 203 18. C. Qian, T. Higashigaki, T. Asoh, H. Uyama, ACS Appl. Mater. Interfaces 2020,
 204 12, 27518-27525.
- 205 19. J. S. Chen, Q. Y. Peng, T. Thundat, H. B. Zeng, *Chem. Mater.* 2019, **31**, 45534563.
- 207 20. F. F. Fu, J. L. Wang, H. B. Zeng, J. Yu, ACS Materials Lett. 2020, 2, 1287-1301.
- 208 21. E. Guler, R. Elizen, D. A. Vermaas, J. Membr. Sci. 2013, 446, 266-276.
- 209 22. P. Dlugolecki, A. Gambier, K. Nymeijer, *Environ. Sci. Technol.* 2009, 43, 6888210 6894.
- 211 23. J. Ji, Q. Kang, Y. Zhou, Adv. Funct. Mater. 2017, 27, 1603623.
- 212 24. D. K. Kim, C. H. Duan, Y. F. Chen, Microfluid. Nanofluid. 2010, 9, 1215-1224.
- 213 25. A. S. Dukhin, P. J. Goetz, Studies in Interface Science, 2010, 24, 21-89.
- 214 26. D.C. Grahame, Chem. Rev., 1947, 41, 441-501.
- 215 27. P. Delahay, John Wiley & Sons Inc, 1965.
- 216 28. M. A. Brown, A. Goel, Z. Abbas, Angew. Chem. Int. Ed., 2016, 55, 3790-3794.
- 217 29. W. Sparreboom, A. van den Berg, J. C. Eijkel, *Nat. Nanotechnol.*, 2009, 4, 713218 720.
- 219 30. F. H. Van der Heyden, D. Stein, C. Dekker, Phys. Rev. Lett., 2005, 95, 116104.
- 220 31. F. H. J. Van der Heyden, D. J. Bonthuis, D. Stein, C. Meyer, C. Dekker, Nano
- 221 *Lett.*, 2006, **6**, 2232-2237.
- 222 32. W. Guo, L. Cao, J. Xia, F. Q. Nie, W. Ma, J. Xue, Y. Song, D. Zhu, Y. Wang, L.
- 223 Jiang, Adv. Funct. Mater., 2010, 20, 1339-1344.
- 224 33. C. L. Rice, R. Whitehead, J. Phys. Chem., 1965, 69, 4017-4024.
- 225 34. H. Daiguji, P. Yang, A. J. Szeri, A. Majumdar, Nano Lett., 2004, 4, 2315-2321.
- 226 35. W. W. Xin, Z. Zhang, X. Y. Kong, L. Jiang, L. P. Wen, Nature Communications,
- 227 2019, **10**, 3876.