## Highly Stable LiTa<sub>2</sub>PO<sub>8</sub>-based Hybrid Solid Electrolytes Via In-situ

## Interfacial Formation Technique for Solid-State Lithium-Metal Batteries

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The Rietveld fitting analysis shows a good agreement between the observed and calculated diffraction patterns. The final R-values reached  $R_{wp} = 7.11\%$  and  $R_p = 5.31\%$  with a fit indicator of GOF = 3.14%.



Figure S1. Rietveld refinement result of LiTa<sub>2</sub>PO<sub>8</sub>.

Table S1.	Atomic	parameters	for	LiTa <sub>2</sub> PO <sub>8</sub> .
	1 10011110	parameters	101	

Site	<u>Np</u>	<u>X</u>	<u>y</u>	<u>Z</u>	Occ	Beq
Tal	8	0.24561	0.09570	0.25550	1	0.3395
Ta2	4	0.00000	0.34630	0.25000	1	0.3079
Ta3	4	0.00000	0.00000	0.00000	1	0.2921
P1	8	0.49930	0.20960	0.05960	1	0.5606
O1	8	0.05690	0.33430	0.06770	1	1.224
O2	8	0.38170	0.13710	0.40610	1	1.042
O3	8	0.35820	0.50640	0.12710	1	1.271
O4	8	0.04750	0.16450	0.54970	1	1.05
O5	8	0.14430	0.23420	0.28790	1	0.8133
O6	8	0.13790	0.05670	0.11610	1	1.469
07	8	0.37870	0.20230	0.15370	1	1.319
08	8	0.13680	0.46550	0.28240	1	1.358
Li1	8	0.24500	0.35700	0.14000	0.72	1.658
Li2	8	0.11200	0.38700	0.40000	0.21	1.658
Li3	4	0.50000	0.00000	0.00000	0.14	1.658

Phase name	Structure
Empirical formula	LiTa <sub>2</sub> PO <sub>8</sub>
Space group	C2/c
Scale	0.00004617(77)
Cell Volume (Å^3)	1199.845(63)
Wt% - Rietveld	100.000
Crystallite Size	
Cry size Lorentzian (nm)	98.43(79)
k: 1 LVol-IB (nm)	62.66(50)
k: 0.89 LVol-FWHM (nm)	87.60(70)
Crystal Linear Absorption Coeff.	692.099(36)
(1/cm)	
Crystal Density (g/cm^3)	5.84371(31)
Lattice parameters	
a (Å)	9.71332(31)
b (Å)	11.53696(33)
c (Å)	10.70698(33)
β (°)	89.9079(31)

Table S2. Summary of LiTa<sub>2</sub>PO<sub>8</sub> structural information analyzed by X-ray powder diffraction.

 Table S3. The goodness of fit parameter

GOF	3.14
R <sub>exp</sub>	2.27
R <sub>wp</sub>	7.11
R <sub>p</sub>	5.31
R <sub>exp</sub> -dash	3.83
R <sub>wp</sub> -dash	12.00
R <sub>p</sub> -dash	10.73
Weighted Durbin Watson	0.39



**Figure S2.** (a) –(e). Raman spectra and (f). Total ionic conductivity of PVDF-HFP+LiTFSI, PVDF-HFP+LiTFSI+SN (SPE), 0.5LTPO-HSE, 1LTPO-HSE, and 1.5LTPO-HSE



Figure S3. DTG curve of the LTPO powder and the membranes.

Particle size distribution of LTPO after 8 h ball-mill using 3D mixer. The d50 is approximately  $450 \ \mu m$ .



Figure S4. Particle size distribution of LTPO after ball-mill using 3D mixer.

Table S4. Total ionic conductivity ( $\sigma_t$  in mS cm<sup>-1</sup>) of LTPO pellet and 1LTPO-HSE membrane.

Temperature (°C)	LTPO pellet	1LTPO-HSE	LE-1LTPO-HSE (with
		membrane	10 μL) membrane
RT	0.17	0.59	1.17
30	0.21	0.84	2.25
40	0.30	1.03	4.03

50	0.40	1.27	4.33
60	0.55	1.56	4.58
70	0.63	1.91	4.76
80	0.82	2.35	5.13



**Figure S5.** Contact angle measurement at 0 s, 60 s and 180 s of *x*LTPO-HSE, where x = 0.5, 1 and 1.5.



**Figure S6.** (a). The EIS plot of the cells before and after cycling at 0.5C for three cycles. (g) (b). The first and third charge–discharge curves of the cells with different amounts of LE additives.

		Cell imp	pedance		Average	Initial Specific
		2)	2)		coulombic	Capacity
					efficiency (%,	(0.5C, mA h
Sample					CE <sub>avg</sub> )	<b>g</b> <sup>-1</sup> )
	R	Ъ	F	R <sub>ct</sub>		
	Before	After 3	Before	After 3		
	cycle	cycles	cycle	cycles		
LFP/LE(5 µL)-	6.3	8.5	265.9	202.3	97.9	133.1
1LTPO-HSE/Li			2000	_0	27.2	
LFP/LE(10 µL)-	4 1	47	190.2	120 7	09.4	120.6
1LTPO-HSE/Li	4.1	4./	109.2	120.7	70.4	139.0
LFP/LE(15 µL)-	4.1	4.8	183.2	114.9	97.4	142.5

<b>Fable S5.</b> Comparison of LFP/LE( $x \mu L$ )-1LTPO-HSE/Li cells with different amount of	of I	J	E.
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**Figure S7.** The voltage profile for the long-term cycling performance of (a) LFP/LE-1LTPO-HSE/Li cell with a rate of 2C/2C at 45 °C (b). LFP/LE-1LTPO-HSE/Li cell with a rate of 0.5C/2C at room temperature.

 Table S6. Heat flux values for LFP/LE-1LTPO-HSE/Li and LFP/1LTPO-HSE/Li coin cells.

		Heat flux ( $Q_t$ , J g <sup>-1</sup> ) values of cells					
Operation	C rate	LFP/LE-1L	TPO-HSE/Li	LFP/1LTPO-HSE/Li			
Temperature		Charge	Discharge	Charge	Discharge		
45 °C	2C/2C	-130.0	-135.9	-126.4	-134.2		
	3C/3C	-172.2	-181.6	-111.2	-123.1		
	2C/2C	-103.8	-117.6	-71.9	-76.5		
55 °C	3C/3C	-151.1	-163.8	-67.9	-78.8		

LiFePO<sub>4</sub> composite cathode slurry was prepared to assemble a pouch cell. The same procedure was followed for the preparation step as mentioned in preparation of LiFePO<sub>4</sub> composite cathode

material section. The dried composite cathode was cut to a dimension of 3.3 cm  $\times$  4.8 cm for LiFePO<sub>4</sub> cathode, 3.5 cm  $\times$  5.0 cm for Li metal anode and 3.8 cm  $\times$  5.3 cm for the LTPO-HSE in a glovebox. The active material loadings of LiFePO<sub>4</sub> in the composite cathodes for the coin cell and pouch cell were 3.5 and 4.5 mg cm<sup>-2</sup>, respectively. Charge and discharge experiments were conducted on the batteries within the range of 2.5 to 4.0 V at room temperature.



Figure S8. The voltage profile for the  $3 \text{cm} \times 5 \text{cm} \text{LFP/LE-1LTPO-HSE/Li}$  pouch cell with a rate of 0.1C/0.1C at room temperature,

We investigated the applicability of the LTPO-HSE membrane in the presence of LE (113  $\mu$ L, based on the area of pouch cell) for pouch cell test with a dimension of 3 cm × 5 cm. One side of the carbon-coated Al foil was coated with a composite cathode material consisting of LiFePO<sub>4</sub> and assembled with Li foil (100  $\mu$ m thickness) using the LE-1LTPO-HSE membrane. The pouch cell underwent cycling with a current of approximately 11 mA at a rate of 1C. The voltage curves depicted in Fig. S8, measured at 0.1C and between 2.5–4.0 V, demonstrate that the LFP/LE-

1LTPO-HSE/Li cell exhibited a capacity of 7.95 mA h (123.93 mA h  $g^{-1}$ ), and even after 10 cycles, it maintained over 100% of its initial capacity at room temperature. This suggests that the use of LTPO-HSE in the presence of LE additives holds promise for applications in SSLMBs.

In the P 2p region, the peak of PO<sub>4</sub> in LFP cathode surface in the LFP/LE-1LTPO-HSE/Li cell is located at 133.1–133.7 eV, which is the P–O/P = O of the phosphate group in LFP and LTPO. The peak at 134.5 is also observed for the LiDFP additives. However, only two peaks for the PO<sub>4</sub> groups (in LFP and LTPO) are observed in LFP/1LTPO-HSE/Li cell. The presence of LiDFP additive helped for the formation of stable CEI thin film on the surface of the composite cathode material.

According to the F1s, it can be seen that there are Li-F, C-F, B-F and P-F peaks observed on the CEI layer LFP composite cathode material in the LFP/LE-1LTPO-HSE/Li cell. On the other hand, only two peaks of Li-F and C-F are observed on the LFP composite cathode surface in LFP/1LTPO-HSE/Li cell. The LiF intensity is much higher in LFP/1LTPO-HSE/Li cell than the cell with LE additive. The presence of thick LiF CEI layer on the cathode surface may diminish the diffusion of Li<sup>+</sup> ion which can decrease its capacity through long-term cycling. The content of LiF in LFP/LE-1LTPO-HSE/Li cell is very low, indicating the little decomposition product of the electrolyte as well as electrode components. The presence of LTPO as a catholyte (3% added during the composite cathode preparation) is also detected from the Ta 4f spectra of the CEI layer in the composite cathode material. It is shown in Fig. S9(d) that the intensity of Ta<sup>5+</sup> is higher in LFP/LE-1LTPO-HSE/Li cell than without LE. This is due to the fact that, the presence of LE additives prevents the decomposition of LTPO catholyte by forming stable CEI layer on its surface.



**Figure S9.** XPS spectra of (a). B 1s, (b). P 2p, (c). F 1s and (d). Ta 4f of the chemical composition of the surface films on the cycled LiFePO<sub>4</sub> composite cathode; (I). LFP/LE-1LTPO-HSE/Li and (II). LFP/1LTPO-HSE/Li.



Figure S10. SEM images of (a). LiFePO<sub>4</sub> pristine powder, (b). LiFePO<sub>4</sub> composite cathode material. EDX mapping of (c). LiFePO<sub>4</sub> pristine powder, (i) Fe; (ii) P; and (iii) O. and (d). LiFePO<sub>4</sub> composite cathode material, (i). Fe, (ii). P, (iii). Ta, (iv). N, (v). O, (vi). S, (vii). F, and (viii). C.