Supporting Information

C₆N₂S Monolayer: An Auxetic Metal with Ultralow Diffusion Barrier and High Storage Capacity for Potassium Ion Batteries

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Computational Details

The particle swarm optimization (PSO) method within the evolutionary algorithm as implemented in the Crystal structure Analysis by Particle Swarm Optimization (CALYPSO) code¹ was employed to find the lowest energy structure of the C₆N₂S monolayer. Unit cells containing 1, 2, and 4 formula units (f.u.) were considered. In the first step, random structures with certain symmetries are built, in which atomic coordinates are generated by the crystallographic symmetry operations. Local optimizations using the VASP code were done with the conjugate gradients method and stopped when Gibbs free energy changes became smaller than 1×10^{-5} eV per cell. After processing the first-generation structures, 60% of them with lower Gibbs free energies are selected to construct the next generation structures with PSO. 40% of the structures in the new generation are randomly generated. A structure fingerprinting technique of bond characterization matrix is applied to the generated structures, so that identical structures are strictly forbidden. These procedures significantly enhance the diversity of the structures, which is crucial for structural global search efficiency. In most cases, structural search simulations for each calculation were stopped after generating 1000 ~ 1200 structures (e.g., about 20 ~ 30 generations).

The local structural relaxations and electronic properties calculations were performed in the framework of the density functional theory (DFT) within the generalized gradient approximation (GGA) as implemented in the VASP code.² The 2s²2p², 2s²2p³, and 3s²3p⁴ atomic orbitals were treated as valence states for C, N and S, respectively. The cut-off energy for the expansion of wave functions into plane waves was set to 600 eV in all the calculations.



Figure S1. The electronic band structure of the C_6N_2S monolayer calculated at the HSE06 level.



Figure S2. The top and side views of charge density difference (CDD) with an adsorbed Li (a) and Na (b) on the C_6N_2S monolayer.



Figure S3. The electron localization function (ELF) maps of (a) C_6N_2SLi , (b) C_6N_2SNa , and (c) C_6N_2SK .



Figure S4. The optimized structure of C_6N_2S after the removal of adsorbed K atoms.



Figure S5. Total density of states (TDOS) of C₆N₂S, C₆N₂SK, C₆N₂SK₂, C₆N₂SK₃, and C₆N₂SK₄.



Figure S6. The C₆N₂S bilayer structures of (a) aligned stacking, (b) *a*-direction translational stacking, (c) *b*-direction translational stacking, and (d) simultaneous translational stacking along the *a* and *b* directions.

Phase	Space Group	Lattice Parameters (Å, °)	Wyckoff Positions (fractional) Atoms	x	у	Z
C ₆ N ₂ S	Pmm2	<i>a</i> = 4.85602	C(2g)	0.00000	0.07137	0.51945
		b = 6.20827	C(2g)	0.00000	0.65564	0.47660
		c = 19.9284	C(2h)	0.50000	0.28295	0.49114
		$\alpha = 92.2986$	N(2h)	0.50000	0.15109	0.52118
		$\beta = 93.3793$	S(1b)	0.00000	0.50000	0.43326
		$\gamma = 66.9783$				

Table S2. Elastic Constants for the C₆N₂S Monolayer (in N/m).

Structure	C ₁₁	C ₁₂	C ₂₂	C ₆₆
C_6N_2S	322.309	7.339	38.890	85.595

Based on the mechanical balance conditions: $C_{11}C_{22} - C_{12}C_{12} > 0$ and $C_{66} > 0$, the C_3NS monolayer is mechanically stable.³

Table S3. The lattice parameters of $C_6N_2SK_n$ (n = 1-4).

lattice parameters	<i>n</i> = 1	<i>n</i> = 2	<i>n</i> = 3	<i>n</i> =4
<i>a</i> (Å)	7.398	7.419	7.412	7.420
<i>b</i> (Å)	18.747	18.003	18.285	18.220

References

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