Supplementary Information (SI) for Journal of Materials Chemistry C. This journal is © The Royal Society of Chemistry 2025

# Assembly of van der Waals Structure from CVD-Grown 2-dimensional Materials Using Plasma-Treated Polyvinyl Chloride

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# Methods

## 1. Fabrication of vdW stacking

Preparation of P-PVC stamp: To fabricate a P-PVC stamp for picking up CVD-grown 2D materials, a PVC stamp composed of a PVC film, PDMS dome, gel sheet, and slide glass was first fabricated. More details on the fabrication process are provided in previous studies<sup>1,2</sup>. A triple or single micro PDMS dome was fabricated on a glass slide to control the contact area between the stamp and 2D materials. A gel sheet (Gel-Pak, PF-20-X4) was attached to the sliding glass, and a PDMS droplet was deposited on the gel sheet and heated to form a PDMS dome. A commercially available PVC food wrap (Riken Technologies) was placed over the PDMS dome. The surface of the PVC/PDMS stamp was cleaned with isopropyl alcohol prior to use. The PVC stamp was then placed inside an ICP apparatus (Harrick Plasma, PDC-32G) and plasma-treated for 40 s. The plasma was generated using a radio-frequency power supply of 9 W after back pumping the chamber to a vacuum of approximately 280 Pa without adding any other gases. This pressure is a pivotal factor in reducing the amount of residual PVC, which was optimized as shown in Figure S9. The plasma pressure is critical as it determines the ion bombardment energy of the plasma, which can modify the molecular weight of the PVC during plasma treatment, thus influencing the glass transition, mechanical properties, solubility, and thermal stability. These properties are important for a successful transfer.

**CVD synthesized WS2 and MoS2:** In the CVD growth of WS<sub>2</sub>, a three-zone tube furnace is utilized as the growth equipment. The precursor materials used are high-purity WO<sub>3</sub> (tungsten trioxide) and high-purity sulfur (S). The carrier gases employed are argon (Ar) at a flow rate of 200 sccm and hydrogen (H<sub>2</sub>) at a flow rate of 20 sccm. During the process, the WO<sub>3</sub> is heated to 1000 °C, while the S is heated to 200 °C. A sapphire substrate was positioned 7 cm below the WO<sub>3</sub>. The growth process is carried out for 10 minutes, resulting in the formation of WS<sub>2</sub> on the substrate.

In the growth of MoS<sub>2</sub>, a dual-zone tube furnace with an 80mm diameter is used. Molybdenum trioxide (MoO<sub>3</sub>) with a purity of 99.999% serves as the molybdenum source, while solid S of 99.999% purity is used as the sulfur source. Ar gas is employed as the carrier gas. During the growth process, the MoO<sub>3</sub> is heated to 650 °C, and the sulfur is heated to 180 °C. The growth is conducted under a pressure of 4000 Pa and lasts for 5 minutes. This process results in the formation of triangular MoS<sub>2</sub> crystals on a SiO<sub>2</sub>/Si substrate.

Assembling vdW stacking from CVD-grown 2D materials using P-PVC stamp: Prior to transfer, the as-synthesized 2D materials were annealed at 300 °C for 1 h in an Argon-gas environment with 4% hydrogen. This stabilizes the adhesion between the substrate and 2D materials, prevents wrap-up, and cleans the top surface of the CVD-synthesized 2D materials by removing the adsorbed precursors. As a results, annealing can enhance the interlayer interactions for vdW stacking. Although annealing can enhance the adhesion between the substrate and 2D materials and cause the pick-up of the samples more difficult, this issue can be easily resolved using a P-PVC stamp and water-assisted pick-up method. The primary process of assembling a vdW heterostructure began with the selection of desired 2D crystals using an optical microscope. Subsequently, the desired 2D crystals were picked up by gently touching them with the stamp at approximately 50-70 °C and the chip surface was simultaneously injected with DI water. This

process was repeated to sequentially select the 2D materials and assemble the desired vdW stack on the P-PVC stamp surface. Finally, the resulting stack structure was transferred onto a targeted substrate at approximately 160 °C. During this step, the vdW stack and sacrificial layers were detached from the PVC and dropped onto the target substrate.

The cleaning process for removing the residual sacrifice layer: After the vdW stack structure was transferred onto the substrate, two steps were required to remove the sacrificial layer. The first step was to immerse the sample in a THF solution for 2 h, which dissolved most of the PVC sacrificial layer. The second step was to anneal the sample at 300 °C for 12 h in an environment of 4% hydrogen and 96% argon. This process removed the residual polymers from the sample top and enhanced interlayer vdW interactions<sup>3</sup>.

### 2. Structure characterization of 2D materials

**PL and Raman spectroscopy:** We used Renishaw inVia with a wavelength of 532 nm to perform PL and Raman spectroscopic measurement. A grating of 600/mm and 3000/mm were used for the PL and Raman spectroscopic measurements, respectively. The peak positions were calibrated using a standard Si wafer by aligning the Si Raman peaks at 520.5 cm<sup>-1</sup>.

#### 3. Structure characterization of P-PVC stamp

**FTIR:** FTIR (Shimadzu, IRSpirit) was used to analyze the variation in the chemical components of the PVC film before and after the plasma and after annealing treatments. Specifically, we utilized the absorption mode and focused on the plasma-treated surface of P-PVC using an ATR unit. The ATR unit with a diamond prism and an incident angle of 30 °resulted in a penetration depth ( $d_p$ ) of 814 nm for the PVC film, as calculated using the following equation.

$$d_p = \frac{\lambda}{2\pi n_1 \sqrt{\sin\theta^2 - (n_1/n_2)^2}}$$

Where  $\lambda$  is the wavelength of the incident light,  $n_1$  and  $n_2$  are the refractive index of the ATR crystal and sample, respectively, and  $\theta$  is the incidence angle.

**Liquid Chromatography:** Liquid chromatography (Toso, HLC-8320GPC) was used to determine the molecular weight of the PVC films under diverse conditions. To prepare the sample, the PVC film was dissolved in THF (2 mg/mL). The measurement was conducted at 40 °C, with a flow rate of 0.6 mL/min, using a total of 20 µL of the solution. The measurements lasted for 15 min. The differential mobility of the PVC molecules inside the PTFE column depends on their molecular weights, allowing the individual components of PVC to be quantified based on the sequentially detected absorption signals at the column outlet. By analyzing these signals, the molecular weight of the PVC film was determined by fitting it onto a cubic curve. In addition to the four peaks corresponding to the PVC component, several downward peaks originating from THF were observed after 10 min.

**DSC:** A DSC (Netzsch, DSC 3500) was used to determine the glass transition temperature and melting point of the PVC films. In particular, we focused on the glass transition temperature, which is the temperature at which the PVC nanostructure becomes disordered, resulting in a sudden decrease in hardness. This transition is crucial because it affects the suitability of the soft stamp for picking up 2D materials. Additionally, we measured the melting point, which is vital for detaining 2D materials with a sacrificial layer because it causes the layer to melt and separate from the PVC, adhering to the substrate. Notably, we analyzed the entire PVC film, including both

the plasma-treated surface and the pristine sections that were not in contact with the reactive plasma.

**DI-contact angle measurement:** To examine the surface energy of the PVC surface, we used a home-made contact-angle measurement apparatus to measure the contact angle of DI water on the PVC film at room temperature (20 °C). Before measuring the contact angle, the PVC film was flattened, attached to a glass slide, and subjected to various treatments such as plasma treatment and annealing.

Figure



**Fig. S1.** The (a)(b)optical image, the (c)(d) PL intensity mapping and the (e) corresponding PL spectrum at the marked location of monolayer  $MoS_2$  before (on SiO<sub>2</sub>) and after transfer by P-PVC (on SiO<sub>2</sub>), respectively. Scale bar is 20  $\mu$ m.



**Fig. S2**. (a) Pick up WS<sub>2</sub> in sequence and assemble the twist bilayer structure on the P-PVC stamp. (b) drop off the twist bilayer WS<sub>2</sub> on a sapphire substrate.



Fig. S3. assembly of vdWs using CVD synthesized MoS2 and mechanically exfoliated graphene. The structure of the vdWs stacking is shown in the inset. Scale bar is  $30 \mu m$ .



**Fig. S4**.(a) Optical and (b) AFM image of twsit bilayer  $WS_2$  sample transferred on SiO<sub>2</sub> substrate. The dashed square in (a)(c) indicate the location of the AFM measurement. The monolayer region is clean, indicates little residual of PVC after transfer and cleaning, while some bubbles and wrikles appear at the bilayer region, which is strongly influenced by the the transfering circumstance and quality of 2D material.



Fig. S5. PL spectrum of bilayer WS<sub>2</sub> with twist angel from 2 to 57 degrees.



Fig. S6. PL intensity mapping of bilayer WS<sub>2</sub> with a twist angel of 57 degrees.



Fig. S7. Raman spectrum of bilayer  $WS_2$  with twist angel from 2 to 57 degrees.

(a)

Temperature (°C)	50	60	70	80	90	100	110	120	13 0	140	150	160	170	180
PVC-exfolilated 2D materials														
PVC-CVD 2D materials_water														
P-PVC-CVD 2D materials_water														



**Fig. S8.** (a) Temperature parameters of the transfer, including pick-up (red region) and drop-off (purple region), and the (b) corresponding expected adhesion variation between 2D materials /PVC stamp ( $E_{2d-PVC}$ ), and between 2D materials/substrate ( $E_{2d-sub}$ ) under various temperatures with two different experimental setups.



**Fig. S9.** the optical image of PVC residual (a) before and (b) after the cleaning with P-PVC treated under different plasma condition. (Low, Me, and Hi represent the plasma power, which is 6.8 W, 10.5 W, and 18 W, respectively)



Figure S10. Schematic of the assembly of multi-layer vdW stacking using P-PVC.



Figure S11. Distance in between  $(\Delta \omega)$  A1g and E2g modes in monolayer and bilayer WS<sub>2</sub> with different twist angles.



Figure S12. The pick-up of WS2 and that dropped off on to substrate.



Figure S13. screenshot of the process stacking of  $MoS_2$  in SI movie.

									Ref.	<b>Targeted 2D materials</b>	Other advantages & disadvantages	Spatial resolved micro-scale transfer	large scale transfer	Sequential pick-up	Drop-off	Pick-up	Methology	Stamp / Polymer
MoS2, WS2 ACS Nano 9 6178 (2015)	√ damage to the sapphire substrate √ PS is brittle(not suitable for large aera)	×	×	×	wet	Wet + NaOH	PS direct spin coating on substrate	f. PS	Nanoscale 4 6637(2021) Appl. Phys. Lett. 104 203506 (2014)	MoS2 (on etchable substrate)	<sup>4</sup> The adhesion between substrate and 2D flass do not influence pick-up × Substrate on twushle × 2D flake on StO_Si substrate only × Additional annealing in Ar/H2 to remove PMMA	×	0	×	PMM	Etching (NaOH/KOH)	PMMA direc	
TMD on SiO2 ACS Appl. Nano Mater. 2 532	✓ Less residual, and wrikles	×	0	×	Wet (CA removed by acet	Wet + NaOH or NH4F and	CA direct spin coating on s	g. CA (Cellulose ace	ACS Nano 9 5510 (2015) ACS Nano 5 9927 (2011)	WS2	✓ The adhesion between substrate and 2D flaks do not influence pick-up X For metal substrate only	×	o	×	A removed by chemical method	Bubbling transfer (NaCl solution)	t spin coating on targeted 2D:	a. PMMA
MoS2 (2019) Nat. Commun. 9 97:	√ Large scale, all dry proc ✓ Enables Roll to roll tran	ightarrow	0	×	n) (no drop off	Dry + watei IF	lbstrate Transfer with solid ) stamp	i. PET/EV (Ethylen-Vinyl A	Nanoscale 9 19124 (2017)	MoS2	The adhesion between substrate     ✓)       and 2D flaks do not influence     ✓       pick-up     pickate not reusable       Substrate not reusable     2D flakes on water soluable       2D flakes on water soluable     substrate only (NaS, and NaCl)	×	o	×		Desolve the water-soluble sacrificial layer	laks	
N 9 (2018) ACS Na	ress ✓ Good adh sfer × Need to tu PVP and chemical				) (PVP is	7	EVA/PET Transfer w	A j. l .cetate) (Polyvir	Carbon 116 167 (2017)	MoS2	Less polymer residual All dry process (fast)	o	0	×	Dry release	Heating at 70 ∘C	Transfer with solid PDMS/DMSO stamp	b. PDMS/DMSO Dimethyl sulfoxide)
MoS2, WS2 mo 10 5237 (2016)	vesion ✓ Cle une the contact of ✓ Lau 2d Taks with extra X spin (N-vinylpyrrolidone)		0	×	Wet PMI s water soluble) hea	Dry	vith solid PVA/PVP Use stamp	PVA/PVP k nylpyrrolidone)	Nanoscale 8 10677 (201	MoS2 on SiO2/Si substr	✓ All dry process (fast) × For MoS2 on SiO2 substr only	o	0	×	Dry release	Dry+water@70 °C	Transfer with solid PD stamp	c. PDMS (Dimethylpolysilox
MoS2, WS2, MoSe2 Nature 550 229 (2017)	an interface rge aera a coating process (1 <sup>st</sup> layer)	×	0	0	MA should be removed by ting or chemical solution	Dry	STRT/PMMA/2D flake as stamp	. TRT tape/PMMA	(6) IEEE Access 8 70488 (	ate WSe2	<ul> <li>PVA is water-solubili</li> <li>rate (less residual)</li> <li>good adhesion to 2D</li> <li>(high servensitie)</li> <li>X Hard for uniform</li> <li>transfer</li> </ul>	o	0	×	Wet release (PVA remove by wat	Dry pickup	DMS Transfer with sol PDMS/DMSO star	d. PDMS/PV/ ane) (polyvinyl alcoh
MoS2, WS2,	<ul> <li>✓ Fast assembling of vdWs hetrostructure</li> <li>✓ For both micro and macro scale fabrication</li> </ul>	0	0	0	Wet (PVC removed by THF+annealing)	Dry + water @70°C	Use P-PVC/PDMS as stamp	This study: P-PVC	2020) ACS Nano 8 11522 (2014)	MoS2 on sapphire	ity / Large young' modulus -> less deformation of 2D 'Inaks flaks X Weak adhesion between PS and 2D flaks	×	0	×	Wet (PS removed by THF)	Wet+ water	lid PS direct spin coating on mp substrate	A e. Polystyrene (PS) hol)

## Table S1. comparison of transfer method for CVD-grown 2D flakes<sup>4,5,14,6–13</sup>

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