

**Supplementary Information for**  
**Ultra-Low Lattice Thermal Conductivity and Thermoelectric**  
**Performance in Monolayer  $\text{Mg}_3\text{As}_2$  and  $\text{Mg}_3\text{SbAs}$**

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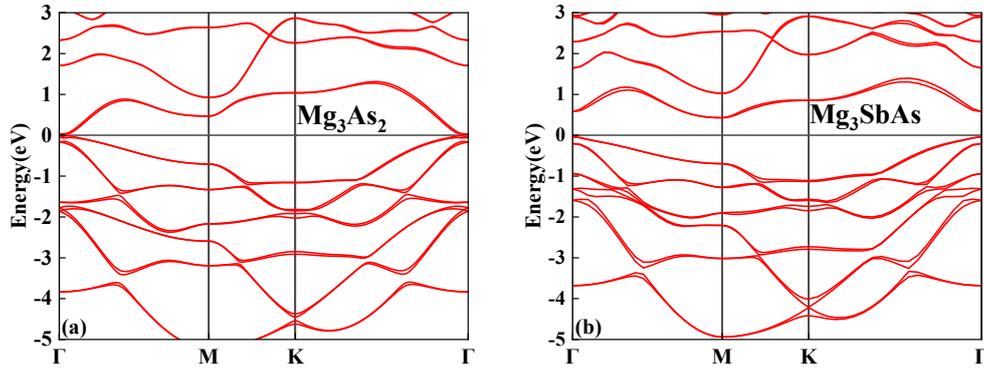
The monolayer  $\text{Mg}_3\text{SbAs}$  structure exists in two possible configurations. In  $\text{Mg}_3\text{SbAs}$  (1), Sb atoms occupy (0.333, 0.666, 0.375) and As atoms occupy (0.666, 0.333, 0.581); in  $\text{Mg}_3\text{SbAs}$  (2), Sb atoms occupy (0.666, 0.333, 0.588) and As atoms occupy (0.333, 0.666, 0.380). We calculated their formation energies and selected the one with lower formation energy,  $\text{Mg}_3\text{SbAs}$  (1), for subsequent calculations.

**Table S1**  $\text{Mg}_3\text{As}_2$ ,  $\text{Mg}_3\text{SbAs}$  (1), and  $\text{Mg}_3\text{SbAs}$  (2) formation energies

Matter	Formation Energies
$\text{Mg}_3\text{As}_2$	-1.6819532484 eV
$\text{Mg}_3\text{SbAs}$ (1)	-1.1946123617 eV
$\text{Mg}_3\text{SbAs}$ (2)	-0.8860788117 eV

To clarify the effect of SOC on the electronic structures of  $\text{Mg}_3\text{As}_2$  and  $\text{Mg}_3\text{SbAs}$ , we calculated the band structures with SOC. The results indicate that while the inclusion of SOC induces band splitting and a further reduction in the band gap, it does not significantly alter the overall band structure, particularly the conduction band minimum (CBM).

**Fig. S1** Band structures of (a)  $\text{Mg}_3\text{As}_2$  and (b)  $\text{Mg}_3\text{SbAs}$  calculated using SOC

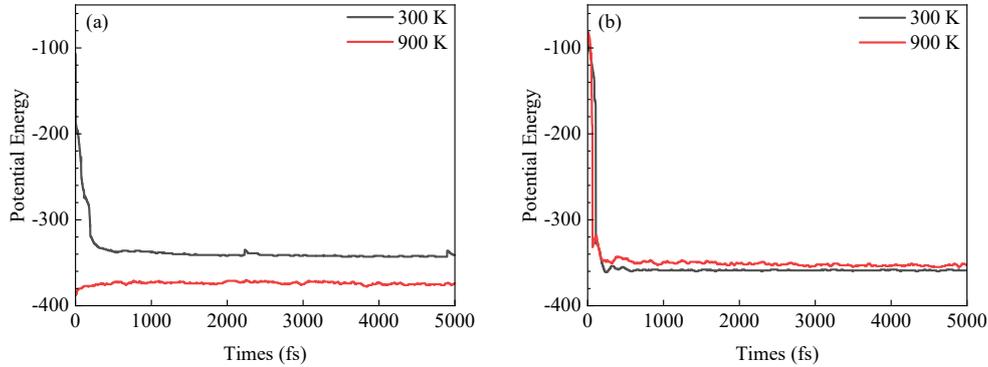


To evaluate the mechanical stability of monolayer  $\text{Mg}_3\text{As}_2$  and  $\text{Mg}_3\text{SbAs}$ , we investigated mechanical stability using the finite distortion method.

**Table S2** Elastic Constants of  $\text{Mg}_3\text{As}_2$  and  $\text{Mg}_3\text{SbAs}$ 

Matter	$C_{11}$ (GPa)	$C_{12}$ (GPa)	$C_{66}$ (GPa)
$\text{Mg}_3\text{As}_2$	65.2649	34.8569	15.2040
$\text{Mg}_3\text{SbAs}$	60.8533	23.4988	18.6773

Calculations indicate that monolayer  $\text{Mg}_3\text{As}_2$  and  $\text{Mg}_3\text{SbAs}$  satisfy the mechanical stability criteria. To further evaluate the stability of these structures, we performed ab initio molecular dynamics (AIMD) simulations at temperatures ranging from 300 to 900 K using the Nosé-Hoover thermostat. A  $4 \times 4 \times 1$  supercell was employed to mitigate artificial interactions arising from periodic boundary conditions, with a time step of 1 fs and a total duration of 5 ps. As depicted in Fig. S2, the structure exhibits no significant distortion at the end of the simulation, and the total potential energy fluctuates around a constant value, confirming thermal stability at 900 K.

**Fig. S2** Thermal stability of (a)  $\text{Mg}_3\text{As}_2$  and (b)  $\text{Mg}_3\text{SbAs}$ 

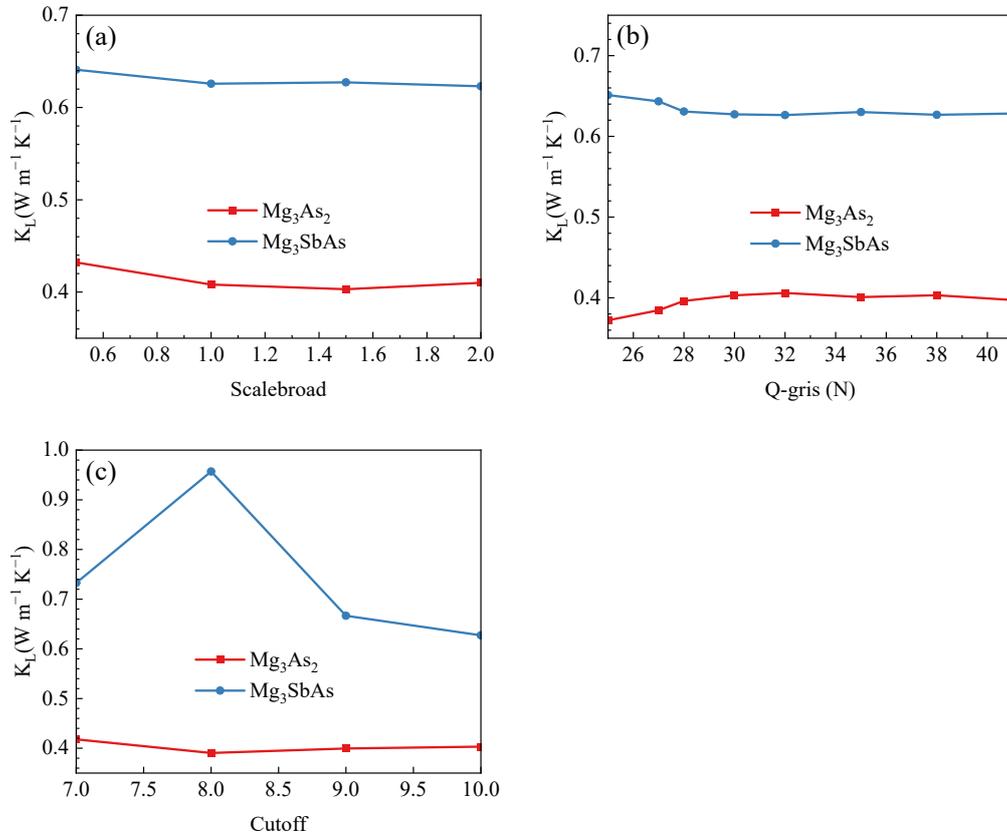
To evaluate the synthesizability of the materials, we calculated the exfoliation energies and compared them with those of the successfully prepared monolayer  $\text{Mg}_3\text{Bi}_2$ .

**Table S3** Exfoliation energies of  $\text{Mg}_3\text{SbAs}$  and  $\text{Mg}_3\text{As}_2$ 

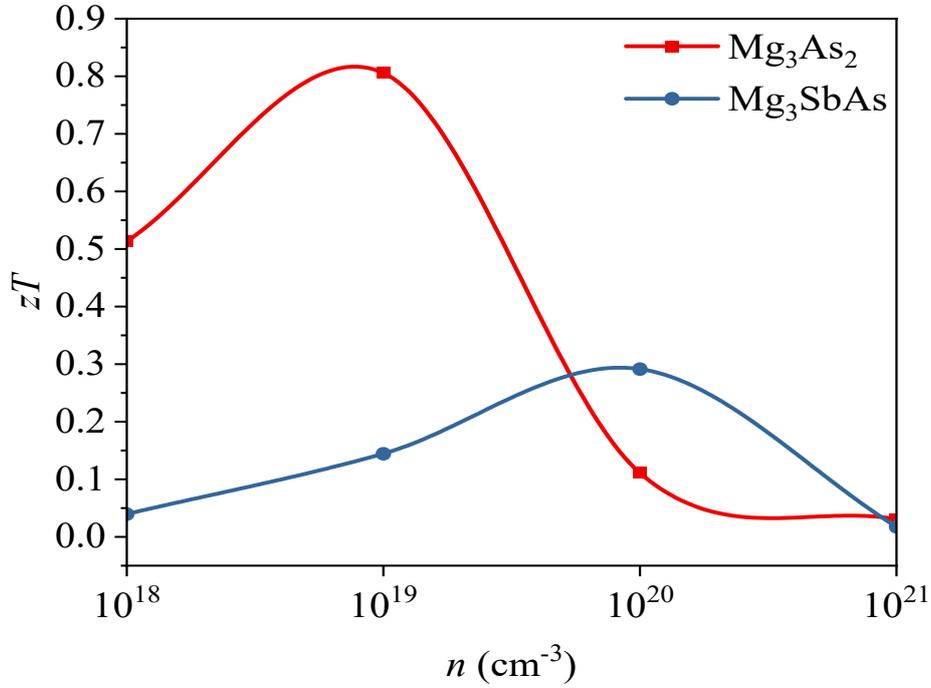
Matter	Exfoliation Energy
$\text{Mg}_3\text{As}_2$	83.2906 meV/ $\text{\AA}^2$
$\text{Mg}_3\text{Bi}_2$	52.1932 meV/ $\text{\AA}^2$
$\text{Mg}_3\text{SbAs}$	64.7775 meV/ $\text{\AA}^2$

To accurately describe the lattice thermal conductivity, we performed convergence tests with respect to the interaction range, the q-point sampling density  $n_{\text{grid}}$ , and the Gaussian broadening scalebroad, as shown in Fig. S3 of the Supplementary Information. After testing, we decided to use the configuration with scalebroad = 1.5, q-point = 30, and cutoff = 10.

**Fig. S3** Convergence tests of (a) Gaussian broadening (broad) and interaction range, (b) q-point grid ( $n_{\text{grid}}$ ), and (c) interaction range for  $\text{Mg}_3\text{As}_2$  and  $\text{Mg}_3\text{SbAs}$



**Fig. S4**  $zT$  values of  $\text{Mg}_3\text{As}_2$  and  $\text{Mg}_3\text{SbAs}$  at 300 K as a function of carrier concentration



To accurately analyze the bipolar effect of the material, we calculated the intrinsic carrier concentrations at different temperatures. When the intrinsic carrier concentration approaches the doping concentration, the bipolar effect becomes significant; when the intrinsic carrier concentration exceeds the doping concentration, the bipolar effect suppresses the thermoelectric performance of the material.

**Table S4** Intrinsic carrier concentration

Temperature(K)	$\text{Mg}_3\text{As}_2$	$\text{Mg}_3\text{SbAs}$
300 K	$8.83 \times 10^{16} \text{ cm}^{-3}$	$3.32 \times 10^{12} \text{ cm}^{-3}$
400 K	$5.65 \times 10^{17} \text{ cm}^{-3}$	$2.27 \times 10^{14} \text{ cm}^{-3}$
500 K	$1.85 \times 10^{18} \text{ cm}^{-3}$	$3.17 \times 10^{15} \text{ cm}^{-3}$
600 K	$4.31 \times 10^{18} \text{ cm}^{-3}$	$1.88 \times 10^{16} \text{ cm}^{-3}$
700 K	$8.16 \times 10^{18} \text{ cm}^{-3}$	$7.15 \times 10^{16} \text{ cm}^{-3}$
800 K	$1.35 \times 10^{19} \text{ cm}^{-3}$	$1.95 \times 10^{17} \text{ cm}^{-3}$
900 K	$2.05 \times 10^{19} \text{ cm}^{-3}$	$4.42 \times 10^{17} \text{ cm}^{-3}$

**Fig. S5** Variation of electrical conductivity with respect to the CBM, VBM, and band gap  $E_g$  for (a)  $Mg_3As_2$  and (b)  $Mg_3SbAs$ .

