

Electronic Supplementary Information

**Insight into the photoelectric characteristic and photocatalytic
water splitting in van der Waals heterostructures Cs₂PbI₄/MX₂
(M=Mo, W; X=Se, S)**

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Abstract

The photoelectric characteristic and photocatalytic water splitting in van der Waals heterostructures Cs₂PbI₄/MX₂ (M=Mo, W; X=Se, S) were carried out based on first principle calculations. Four stable heterostructures were obtained with indirect narrow gaps. Among them, Cs₂PbI₄/MoSe₂, Cs₂PbI₄/MoS₂ and Cs₂PbI₄/WSe₂ belong to type-II heterostructures, and Cs₂PbI₄/WS₂ is type-I heterostructure. High optical absorption efficiency and high carrier mobility imply that type-I heterostructure Cs₂PbI₄/WS₂ has a broad potential application prospect in light-emitting devices. Moreover, the optical absorption efficiency up to 10⁻⁵cm⁻¹, the spatial separation interval of photogenerated electron-hole (~3.30), and type-II band edge alignment

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mean that heterostructures are easy to realize electron-hole separation and transfer, reduce the recombination probability of electron and holes, and thus enabling solar energy conversion and highly efficient photocatalytic water splitting. Especially, Cs₂PbI₄/MoSe₂ heterostructure can trigger HER and OER reaction spontaneously at $pH=0$ under equilibrium potential 1.23 V with a limiting reaction barrier of 1.16 eV for OER, which is close to 1.128 eV in WSe₂/MoSe₂ heterojunction reported recently, implying that its may be an excellent photocatalyst. In addition, the limiting reaction barrier of Cs₂PbI₄/WSe₂ heterostructure is 1.27 V, low overpotential for OER is 0.31 V, and high power conversion efficiency indicating that it may have a potential applications in both solar cells and photocatalytic water splitting.

Computational details The Heyd-Scuseria-Ernzerhof (HSE06) hybrid functional included the SOC effect is employed in simulating the electric properties of heterostructures.¹ The cutoff energy was set to 350 eV, and the Convergence criteria for energy and force are set at 10⁻⁵ eV and 0.002 eV/Å, respectively. The Brillouin zone grid is set to 6×4×1 for $1 \times \sqrt{2} \text{Cs}_2\text{PbI}_4 / 2 \times \sqrt{7} \text{WSe}_2$, $1 \times \sqrt{2} \text{Cs}_2\text{PbI}_4 / 2 \times \sqrt{7} \text{WSe}_2$, and $1 \times \sqrt{2} \text{Cs}_2\text{PbI}_4 / 2 \times \sqrt{7} \text{WSe}_2$ heterostructures, respectively.

The Gibbs free energies for the hydrogen evolution reaction (HER) and oxygen evolution reaction (OER) and are estimated by means of the model proposed by Nørskov et al.²⁻³

The HER can be driven along the following steps:



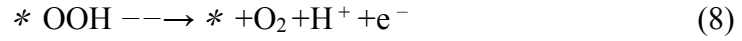
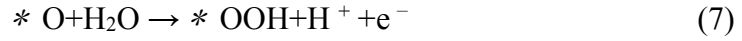
The corresponding free energy ΔG_{*H} is obtained by the following formula:

$$\Delta G_{*H} = \Delta E_{*H} + \Delta E_{ZPE} - T\Delta S \quad (3)$$

$$\Delta E_{*H} = E_{*H} - E^* - 1/2 E_{H2} \quad (4)$$

where ΔE_{*H} is the hydrogen chemisorption energy, where E_{*H} , E^* and E_{H2} are the energies of absorbed slab, pure slab and H_2 gas, respectively.

Moreover, the OER follows four elementary steps:



wherein, $*$ represents the absorption site on the slab for intermediates.

The Gibbs reaction free energy (ΔG) of each step is estimated by the following formula:

$$\Delta G = \Delta E + \Delta E_{ZPE} - T\Delta S + \Delta G_U + \Delta G_{pH} \quad (9)$$

where, ΔE is the total energy difference between reactants and products of reactions, ΔE_{ZPE} is the zero point energy correction, ΔS is the vibrational entropy change at finite temperature T , $\Delta G_U = -eU$ incorporates the effect of the electrode potential, where e is elementary charge, U is the electrode potential ($U=1.23$), ΔG_{pH} is the correction of the H^+ free energy. $pH = 0$ are employed in our calculations.

Table S1. The lattice parameters a and b of the optimized monolayers Cs_2PbI_4 and MX_2 with different size. And the lattice parameters a , and b of the constructed initial heterostructures $\text{Cs}_2\text{PbI}_4/\text{MX}_2$ ($\text{M}=\text{Mo}, \text{W}; \text{X}=\text{Se}, \text{S}$).

| Heterostructures | Configurations | Cs_2PbI_4 | | MX_2 | | initial heterostructures | |
|---|--|---------------------------|-----------------|-----------------|-----------------|--------------------------|-----------------|
| | | $a(\text{\AA})$ | $b(\text{\AA})$ | $a(\text{\AA})$ | $b(\text{\AA})$ | $a(\text{\AA})$ | $b(\text{\AA})$ |
| $\text{Cs}_2\text{PbI}_4/\text{MoSe}_2$ | $1 \times \sqrt{2} / 2 \times \sqrt{7}$ | 6.299 | 8.908 | 6.588 | 8.715 | 6.443 | 8.811 |
| | $1 \times \sqrt{5} / 2 \times 4$ | 6.299 | 14.084 | 6.588 | 13.176 | 6.443 | 13.630 |
| | $1 \times 1 / 2 \times \sqrt{3}$ | 6.299 | 6.299 | 6.588 | 5.705 | 6.443 | 6.002 |
| $\text{Cs}_2\text{PbI}_4/\text{WSe}_2$ | $1 \times \sqrt{2} / 2 \times \sqrt{7}$ | 6.299 | 8.908 | 6.580 | 8.704 | 6.439 | 8.806 |
| | $1 \times \sqrt{5} / 2 \times 4$ | 6.299 | 14.084 | 6.580 | 13.160 | 6.439 | 13.622 |
| $\text{Cs}_2\text{PbI}_4/\text{MoS}_2$ | $1 \times \sqrt{2} / 2 \times \sqrt{7}$ | 6.299 | 8.908 | 6.328 | 8.372 | 6.314 | 8.640 |
| | $\sqrt{2} \times 2 / 3 \times \sqrt{13}$ | 8.907 | 12.598 | 9.492 | 11.408 | 9.200 | 12.002 |
| $\text{Cs}_2\text{PbI}_4/\text{WS}_2$ | $1 \times \sqrt{2} / 2 \times \sqrt{7}$ | 6.299 | 8.908 | 6.365 | 8.419 | 6.332 | 8.664 |

Table S2. The band gap $E(\text{eV})$, electron affinity $A(\text{eV})$, and ionization energy $I(\text{eV})$ of monolayers Cs_2PbI_4 and MX_2 based on HSE+SOC and PBE functional, respectively.

| | Cs_2PbI_4 | MoSe_2 | WSe_2 | MoS_2 | WS_2 |
|--|---------------------------|----------------------|----------------------|----------------------|----------------------|
| $E(\text{PBE})$ | 1.83 | 1.51 | 1.64 | 1.74 | 1.87 |
| $-A(\text{PBE})$ | -3.130 | -3.758 | -3.428 | -4.169 | -3.804 |
| $-I(\text{PBE})$ | -4.961 | -5.267 | -5.074 | -5.905 | -5.679 |
| $E(\text{HSE}^{\alpha=0.25}+\text{SOC})$ | 1.62 | 2.44 | 2.11 | 2.42 | 2.47 |
| $E(\text{HSE}^{\alpha}+\text{SOC})$ | $1.89^{\alpha=0.35}$ | $1.56^{\alpha=0.07}$ | $1.67^{\alpha=0.15}$ | $1.87^{\alpha=0.10}$ | $2.01^{\alpha=0.15}$ |
| $-A(\text{HSE}^{\alpha}+\text{SOC})$ | -3.521 | -3.995 | -3.591 | -4.369 | -3.904 |
| $-I(\text{HSE}^{\alpha}+\text{SOC})$ | -5.406 | -5.555 | -5.262 | -6.237 | -5.920 |
| Experiment ⁴ | 1.86 | 1.57 | 1.65 | 1.90 | 1.94-1.99 |

Table S3. The band gap (eV),VBM edge (eV) and CBM edge (eV) of monolayers Cs₂PbI₄ and MX₂ in heterostructures based on HSE+SOC and PBE functional, respectively.

| | Functional | Cs ₂ PbI ₄ | MoSe ₂ | Cs ₂ PbI ₄ | WSe ₂ | Cs ₂ PbI ₄ | MoS ₂ | Cs ₂ PbI ₄ | WS ₂ |
|---------|------------|----------------------------------|-------------------|----------------------------------|------------------|----------------------------------|------------------|----------------------------------|-----------------|
| Bandgap | | 2.065 | 1.488 | 1.892 | 1.733 | 2.031 | 1.231 | 1.431 | 2.054 |
| CBM | PBE | 1.857 | 1.197 | 1.686 | 1.244 | 2.019 | 0.940 | 1.266 | 1.283 |
| VBM | | -0.208 | -0.290 | -0.206 | -0.489 | -0.011 | -0.291 | -0.206 | -0.269 |
| Bandgap | | | | 1.773 | 1.867 | 1.631 | 1.519 | 1.622 | 1.824 |
| CBM | HSE+SOC | | | 1.552 | 1.464 | 1.412 | 1.191 | 1.397 | 1.535 |
| VBM | | | | -0.220 | -0.403 | -0.218 | -0.329 | -0.225 | -0.289 |

Table S4. The open-circuit voltage (V_{OC}), conduction band offset (CBO) and power conversion efficiency (PCE) of Cs₂PbI₄/MSe₂ (M=Mo,W) heterostructures. E_d^g is donor bandgap based on PBE and HSE+SOC, respectively.

| Functional | Heterostructure | Size | E_d^g | CBO | V _{oc} | PEC (%) |
|------------|---|---|---------|-------|-----------------|---------|
| PBE | Cs ₂ PbI ₄ /MoSe ₂ | $1 \times \sqrt{5} / 2 \times 4$ | 2.065 | 0.660 | 1.405 | 17.1% |
| | Cs ₂ PbI ₄ /WSe ₂ | $1 \times \sqrt{2} / 2 \times \sqrt{7}$ | 1.892 | 0.442 | 1.450 | 21.9% |
| HSE+SOC | Cs ₂ PbI ₄ /MoSe ₂ * | $1 \times \sqrt{2} / 2 \times \sqrt{7}$ | 1.885 | 0.474 | 1.111 | 16.8% |
| | Cs ₂ PbI ₄ /WSe ₂ | $1 \times \sqrt{2} / 2 \times \sqrt{7}$ | 1.773 | 0.088 | 1.385 | 23.9% |

* represents the data based isolated monolayers.

Table S5. Carrier effective masses m_e and m_h , reduced mass μ , static dielectric constant ϵ and exciton binding energies of Cs₂PbI₄/MSe₂ (M=Mo,W) heterostructures. Edg is donor bandgap based on PBE functional.

| Heterostructure | Direction | m_e | m_h | μ | ϵ | E_e |
|---|-----------|-------|-------|-------|------------|-------|
| Cs ₂ PbI ₄ /MoSe ₂ | <i>a</i> | 0.266 | 2.26 | 0.238 | 5.233 | 0.473 |
| | <i>b</i> | 0.657 | 0.739 | 0.348 | 5.209 | 0.698 |
| Cs ₂ PbI ₄ /WSe ₂ | <i>a</i> | 0.598 | 6.419 | 0.547 | 4.905 | 1.237 |
| | <i>b</i> | 1.645 | 0.196 | 0.175 | 5.014 | 0.379 |
| Cs ₂ PbI ₄ /MoS ₂ | <i>a</i> | 0.476 | 0.591 | 0.264 | 5.035 | 0.567 |
| | <i>b</i> | 0.471 | 0.958 | 0.316 | 5.107 | 0.659 |
| Cs ₂ PbI ₄ /WS ₂ | <i>a</i> | 0.854 | 0.694 | 0.383 | 4.758 | 0.920 |
| | <i>b</i> | 0.089 | 0.733 | 0.794 | 4.787 | 1.885 |

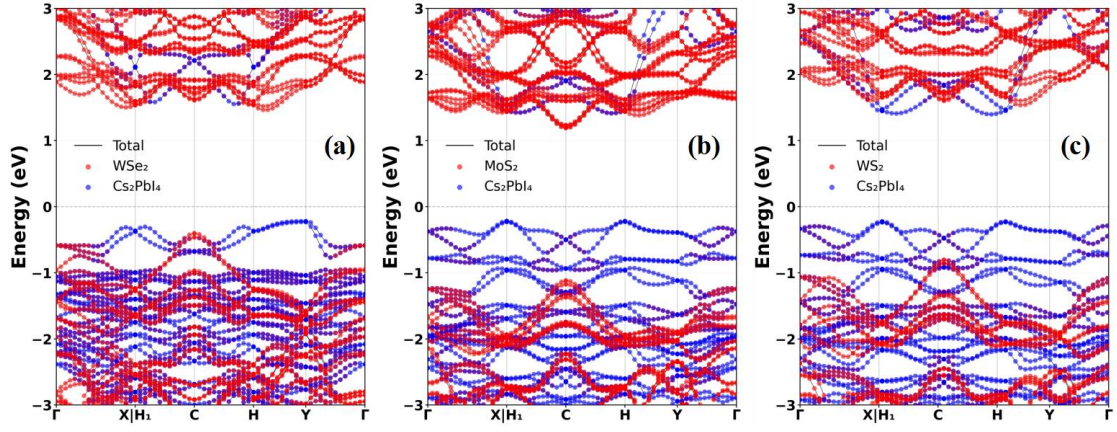


Fig.S1. Projected band structures of heterostructures (a) $\text{Cs}_2\text{PbI}_4/\text{WSe}_2$, (b) $\text{Cs}_2\text{PbI}_4/\text{MoS}_2$ and (c) $\text{Cs}_2\text{PbI}_4/\text{WS}_2$ based on HSE+SOC.

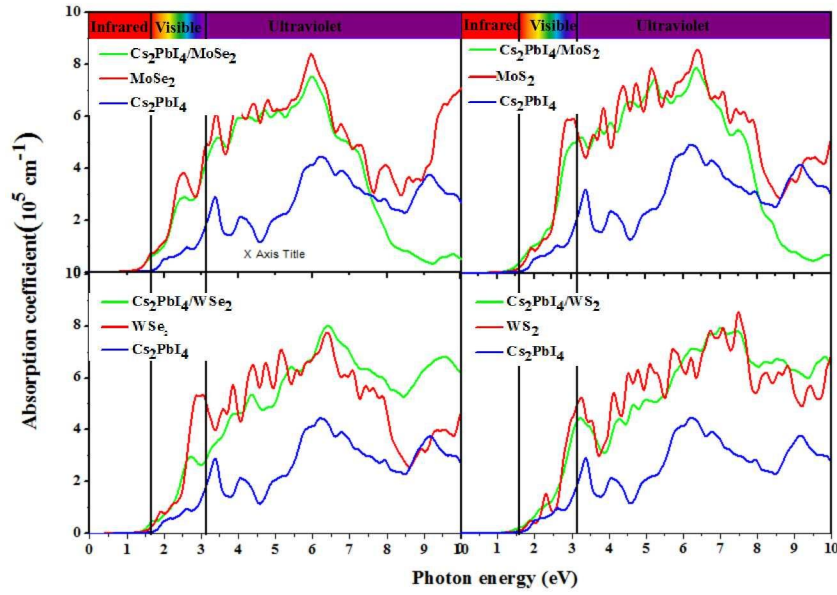


Fig. S2. Optical absorption of monolayer Cs_2PbI_4 , MX_2 ($\text{M}=\text{Mo}, \text{W}; \text{X}=\text{Se}, \text{S}$) and four stable heterostructures $\text{Cs}_2\text{PbI}_4/\text{MoSe}_2$, $\text{Cs}_2\text{PbI}_4/\text{WSe}_2$, $\text{Cs}_2\text{PbI}_4/\text{MoS}_2$ and $\text{Cs}_2\text{PbI}_4/\text{WS}_2$.

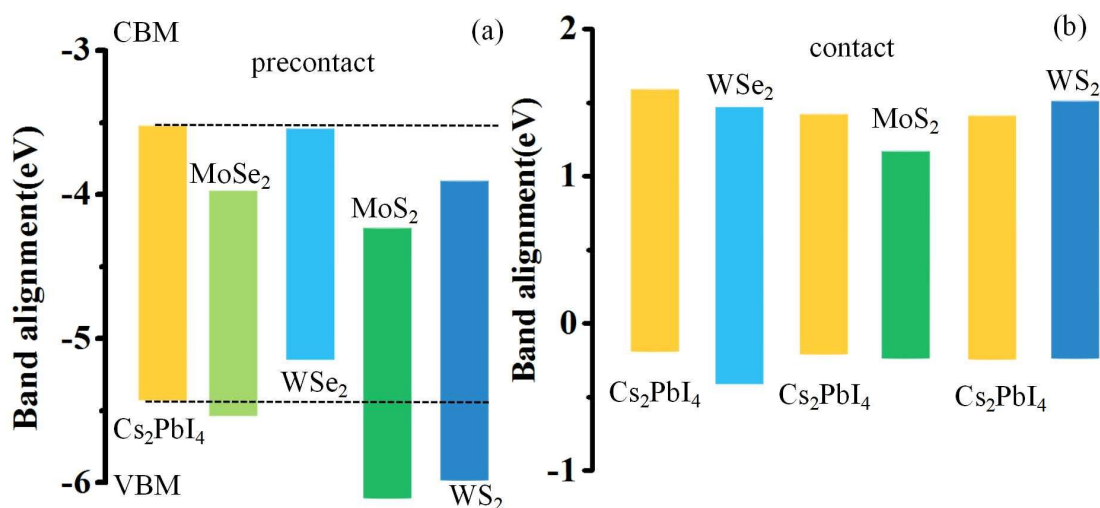


Fig. S3. Band alignments of monolayers Cs_2PbI_4 , WSe_2 , MoSe_2 , and MoS_2 in precontact state (a) and contact state (b) based on HSE+SOC functional.

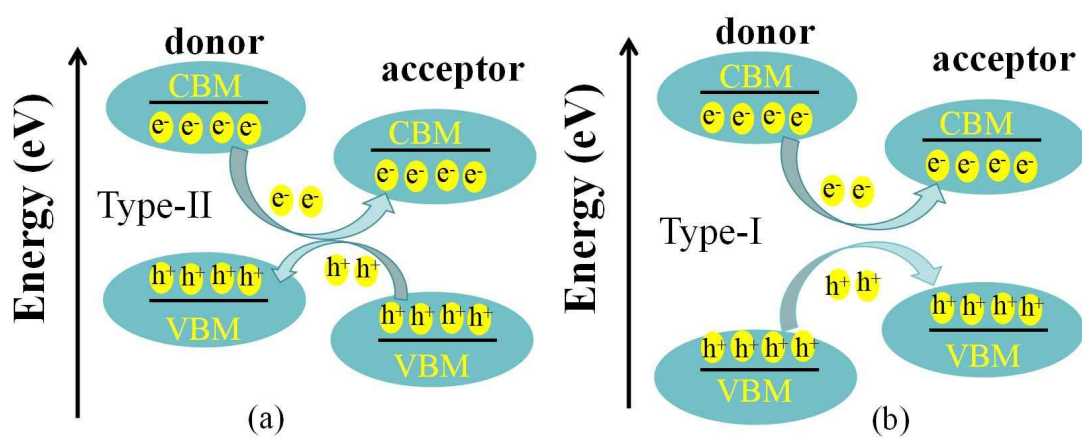


Fig. S4. The mechanism for electron-hole pair separation in heterostructures, (a) Type-II, (b) Type-I.

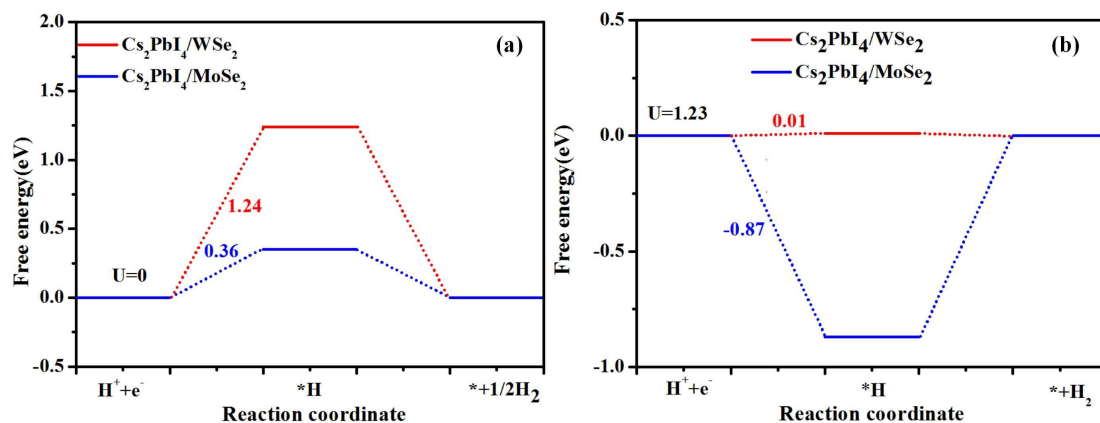


Fig. S5. Gibbs free energy diagram for HER at $pH=0$ under zero cell potential $U=0$ and equilibrium potential $U=1.23$ V.

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