

Supplemental information

Figure S1. Photos and schematics of the experimental setup for *in situ* Mössbauer experiments.



Figure S2. Schematic diagram and experimental photo of the cell assembly for in situ XAS experiments.



Figure S3. Configuration of electrolyte circulation for in situ XAS experiments.



Figure S4. Current-time plots recorded during an *in situ* Mössbauer experiment where (a) shows the potential ramp-up and (b) shows the potential ramp-down. The current response is noisier at potentials higher than 1.55 V likely due to oxygen bubble build up and removal. The noise seemed to be reduced by increasing electrolyte circulation up to 150 mL/min, but the Mössbauer spectrum was broadened due to small vibration of the working electrode (carbon paper). Thus, the flow rate was kept at 50 mL/min to minimize peak broadening.



Figure S5. (a) Mössbauer spectrum combining data at low potentials (1.25 V - 1.40 V). (b) Mössbauer spectrum combining data above 1.55 V and back to 1.40 V on the decreasing potential portion of the profile. RFour components are identified and fitted to both spectra.

Table S1 Comparison of experimental conditions and main observations from works of Chen et al.[37], Kuang et al.[38], and this work.

	NiFe	Substrate	BET	Crystallinity	Fe (IV)	Fe(IV) IS,	Relaxation
	hydroxide		Surface	in powder	%	QS, (mm/s)	
	synthesis		area	XRD	(highest		
					potential)		
Chen et	Co-	Carbon	Not	(003), (006),	21%,	IS = -0.25,	Fe (IV)
al. [37]	precipitation	paper	reported	(012)	1.76V	QS = 0	remains
	and			planes,	(vs.	(singlet)	20% at
	hydrothermal			layered	RHE)		1.49V
				structure			during
							cathodic
							step. No
							Fe (IV) at
							OCP.
Kuang	Co-	Carbon	Not	XRD	40%,	IS = -0.25,	No
et al.	precipitation	paper	reported	pattern	1.57V	QS = 0	relaxation
[38]	of NiFe _m -Fe			comparable	(vs.	(singlet)	observed.
	Prussian blue			with α-	RHE)		
	analogues,			Ni(OH) ₂ .			
	followed by			Among			

	transformation			different			
	in alkaline			Ni/Fe ratios,			
	solution.			NiFe _{0.2}			
				shows low			
				crystallinity.			
This	NiFe aerogel	Carbon	382	Very low	90%,	Two	Fe (IV)
work	synthesis with	paper	m²/g	crystallinity	1.75V	doublets:	content
and	supercritical			with no	(vs.	(1) IS =	decreases
Osmieri	CO ₂ drying,			major peak	RHE)	0.001, QS =	at OCP
et al.,	followed by			observable		0.465 and	with half-
[41].	annealing in			in the XRD		(2) IS =	life ~5
	ambient air at			pattern.		0.07, QS =	hours.
	150°C.					0.75.	Return to
							the
							original
							state at
							~ 30 hours.



Figure S6. Representative STEM-EDS maps of the Ni_8Fe aerogel catalyst (a) before *in situ* MS test and (b) after *in situ* MS test.



Figure S7. Pre-edge feature at the Fe K-edge increases at potentials where Fe is oxidized, in agreement with literature for the formation of Fe(IV)⁴⁵. Scans at multiple potentials were merged to decrease noise in this feature due to low absorption intensity.



Figure S8. Comparison of EXAFS spectra collected at APS beamline 10-ID and NSLS-II beamline 7-BM at varied potentials. The APS data was used for EXAFS fitting, but the plots demonstrate that the data are consistent across beamlines and that the spectra collected at 1.0 V and 1.25 V are comparable.



Figure S9. EXAFS fit of the Ni(OH)₂ standard. Coordination numbers / amplitude factors were constrained to known values to determine appropriate Debye-Waller values for sample fits. Separate energy corrections were used for the two paths to allow for adequate fitting of sample fits, and the difference between the two energies (4.2 eV) was used as a standard energy difference. R-factor = 0.0089.



Figure S10. Fit in situ EXAFS spectra from Ni K and Fe K-edge measurements at 1.0 and 1.7 V vs. RHE.



Figure S11. Linear combination fits of (a) Fe and (b) Ni 1.0 V spectra acquired after polarization in the OER potential region (denoted "rev.") with their respective 1.0 V (initial spectrum, before polarization to 1.7 V) and 1.7 V components (fit range: k=3.0-11.0 Å⁻¹).



Figure S12. STEM-EELS measurements of the Ni₈Fe aerogel catalyst showing two representative areas (a) and (b) of the nanosheets of NiFe hydroxide at the edge. (c) and (d) are where the spectrum image was acquired. (e) and (f) are the corresponding EELS thickness map in terms of t/λ .

Supplementary note 1: The effect of carbon substrate is negligible for in situ MS and XAS experiments

The substrate used in this work is carbon paper without PTFE treatment. Although carbon materials are generally not stable at OER potentials, carbon paper appears to be much more stable than other types of carbon. The carbon paper used in the current study resulted in no discoloration of the electrolyte after testing and no loss of Fe reflected in MS, which we observed when using other types of carbon paper, particularly those containing microporous layers. Also, the oxidation state and phase of the Ni₈Fe catalyst reverted to the state observed prior to polarization at >1.45 V, indicating that oxidation of carbon, if it occurred, did not affect the conductivity of the carbon and the utilization of the catalyst. In addition, Figure S13 below compares the MS before and after test. The absorption area (in %effect*mm/s unit) is calculated to be 2.29(2) at the start and 2.28 (2) after relaxation. Within the error, the absorption area is almost identical which suggests the total mass of Fe is consistent before and after test. This not only demonstrates that the carbon substrate is stable but also the chemical state of Fe is reversible before and after OER. We therefore conclude that over the timeframe of both the MS and XAS measurements, the carbon substrate was stable and had a negligible effect on the experimental results.



Figure S13. Comparison of the Mössbauer spectra at the start of the test (before Fe(IV) formation) and after the relaxation at OCP. The area (in %effect*mm/s unit) is calculated to be 2.29(2) at the start and 2.28 (2) after relaxation.



Figure S14. Cyclic voltammograms of Ni8Fe aerogel catalyst of this study (black) and Ni only catalyst prepared using the same synthesis technique (from Osmieri et al. [41]). The two catalysts have comparable BET surface areas: $425 \text{ m}^2 \text{ g}^{-1}$ for Ni only and $382 \text{ m}^2 \text{ g}^{-1}$ for Ni₈Fe.