Supplementary Information

Formamidinium-based Sn-rich Mixed-halide Perovskite Solar Cells with Near-optimal Bandgap: FAPb_{1-x}Sn_x(I_{0.8}Br_{0.2})₃

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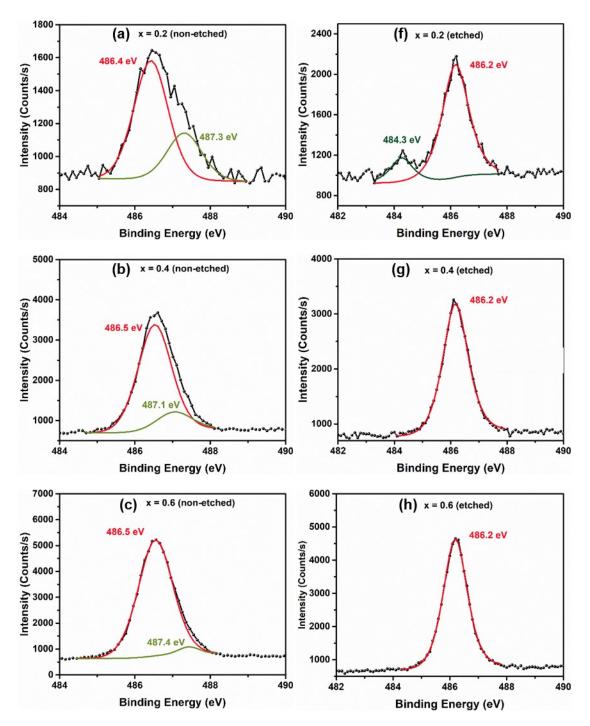


Figure S1 The XPS spectra of Sn 3d5/2 level with fitting for non-etched samples with (a) x = 0.2, (b) x = 0.4, (c) x = 0.6, and for etched samples with (f) x = 0.2, (g) x = 0.4, (h) x = 0.6.

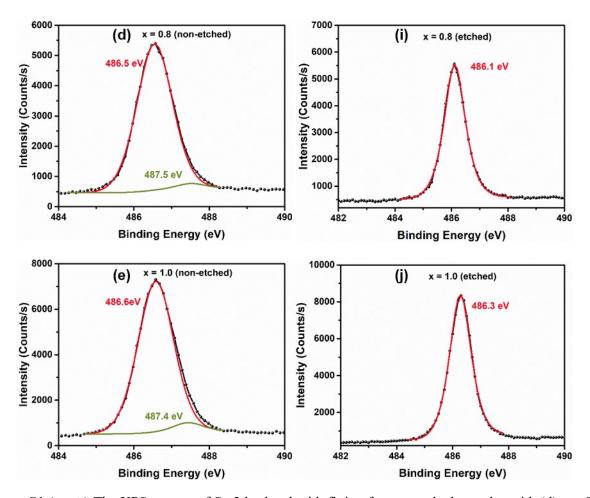


Figure S1 (cont.) The XPS spectra of Sn $3d_{5/2}$ level with fitting for non-etched samples with (d) x = 0.8, (e) x = 1.0 (pure Sn), and for etched samples with (i) x = 0.8, (j) x = 1.0 (pure Sn).

The as-deposited perovskite films were sealed in a N_2 atmosphere bag during transportation and immediately loaded into XPS instrument with minimized air exposure. Nevertheless, the XPS peak of Sn 3d core level for all Sn-containing samples exhibited two chemical states (Sn1 and Sn2) as shown in Figure S2a-e, indicative of air induced oxidation. The chemical state of Sn1 at ~486 eV potentially originates from Sn²⁺ while Sn2 at a higher binding energy of ~487 eV is likely to be Sn-Br oxide. The much lower peak of Sn2 compared to Sn1 implies that Sn oxidation was significantly suppressed owing to the rapid loading process. It is generally known that the elemental composition reflected by XPS correlates with the material within ~10 nm of the surface depth. To further study the elemental composition deeper into the bulk of the samples, argon etching was performed on all the samples until the signal of adventitious Carbon is eliminated. A single chemical state for Sn (at ~486 eV) was identified for all the etched samples with $x \ge 0.4$, as shown in Figure S2f-j. An additional chemical state at ~485 eV corresponding to Sn⁰ was observed in the Sn 3d_{5/2} XPS spectrum of x = 0.2 (Figure S2f). The precipitation of Sn could be ascribed to the unreacted PbI₂ left in the x = 0.2 perovskite film, as revealed by its XRD pattern (Figure 1a).

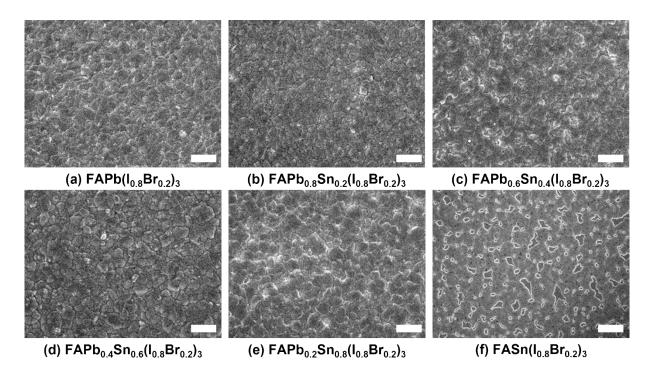


Figure S2 Top-view SEM images of $FAPb_{1-x}Sn_x(I_{0.8}Br_{0.2})_3$ perovskite thin-films depending on the Sn content of: (a) x = 0 (pure Pb); (b) x = 0.2; (c) x = 0.4; (d) x = 0.6; (e) x = 0.8; (f) x = 1.0 (pure Sn). The scale bar in all images is 1 μ m.

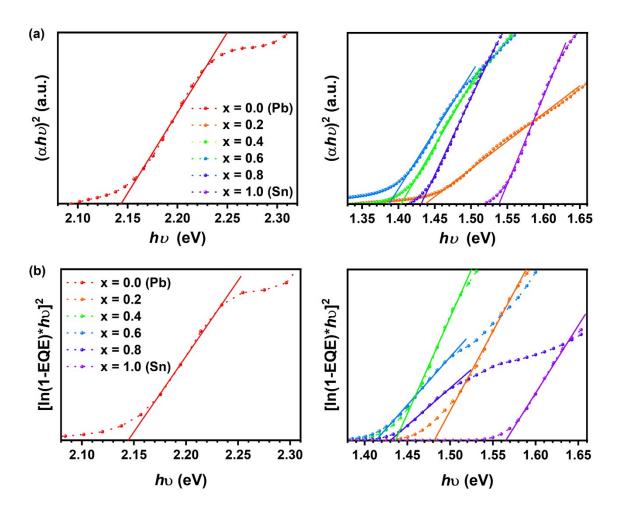


Figure S3 (a) Tauc plot of $(\alpha h v)^2$ *versus* h v and (b) the plot of $[\ln(1-EQE)\times h v]^2$ *versus* h v for $FAPb_{1-x}Sn_x(I_{0.8}Br_{0.2})_3$ perovskites.

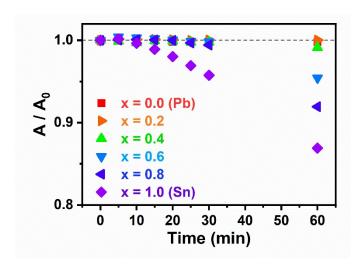


Figure S4 The absorbance at 500nm plotted over time for $FAPb_{1-x}Sn_x(I_{0.8}Br_{0.2})_3$ perovskites. A/A₀ represents the fraction of absorbance remained at 500 nm over time.

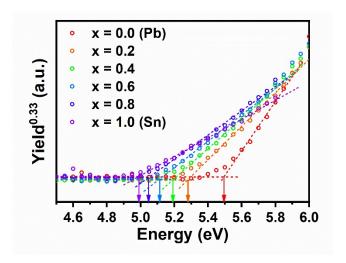


Figure S5 PESA measurement results of $FAPb_{1-x}Sn_x(I_{0.8}Br_{0.2})_3$ perovskites.

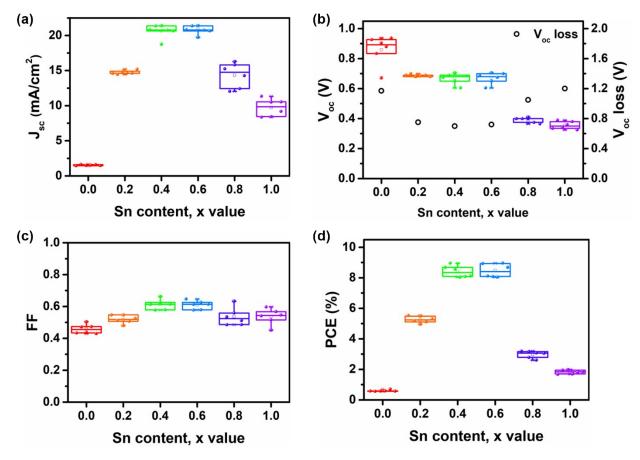


Figure S6 Photovoltaic parameters of $FAPb_{1-x}Sn_x(I_{0.8}Br_{0.2})_3$ based PSCs: (a) J_{sc} , (b) V_{oc} , V_{oc} loss presented as black hollow circles, (c) FF, and (d) PCE.

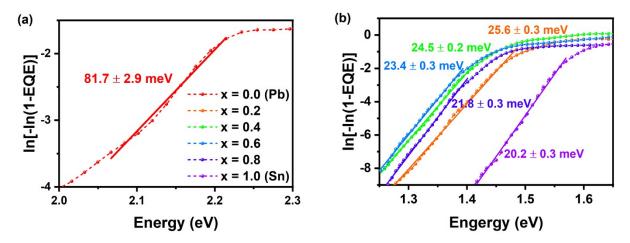


Figure S7 The plot of $\ln \left[-\ln \left(1 - EQE \right) \right]$ against energy for $\text{FAPb}_{1-x} \text{Sn}_x (I_{0.8} \text{Br}_{0.2})_3$ perovskites with (a) x = 0, and (b) x = 0.2 - 1.0. Urbach energy, E_U , was estimated from the slope of the linear portion near the absorption onset.

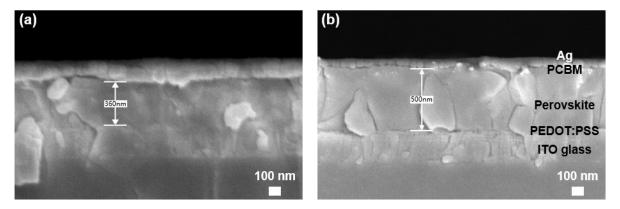


Figure S8 Cross-sectional SEM images of the FAPb_{0.6}Sn_{0.4}($I_{0.8}$ Br_{0.2})₃ PSCs (a) before and (b) after adjusting the concentration of precursor solution. The scale bar in each image is 100 nm.

The thin perovskite layer in Figure S8a was fabricated based on the method described in the Experimental section where the concentration of precursor solution is 1.2 M. Increasing the concentration of precursor solution to 1.5 M resulted in the thick perovskite layer in Figure S8b, while the other processes remained unchanged.

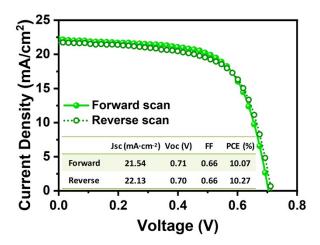


Figure S9 J–V curves of the champion solar cell measured under 100 mW/cm² illumination and under forward (solid line) and reverse (dash line) direction. The insets involved its photovoltaic parameters.

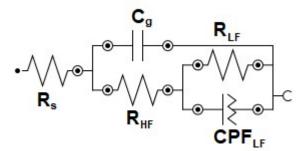


Figure S10 The schematic illustration of the equivalent electrical circuit applied to fit the Nyquist spectra for the FAPb_{1-x}Sn_x(I_{0.8}Br_{0.2})₃ based PSCs with x = 0.4 and 0.6. R_s accounts for the series resistance. R_{HF} and C_g correlate to the resistance and geometrical capacitance at the high frequency range of the applied AC signal. R_{LF} and CPE_{LF} contribute to the impedance response at the low frequency range.

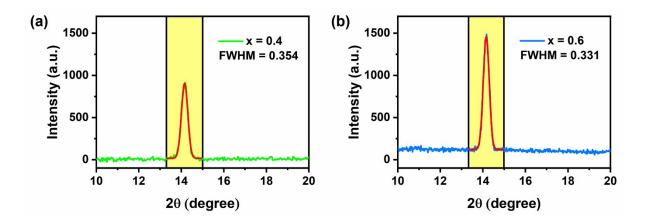


Figure S11 FWHM of the XRD peak at $2\theta \approx 14.12^{\circ}$ for (a) x = 0.4 and (b) x = 0.6.

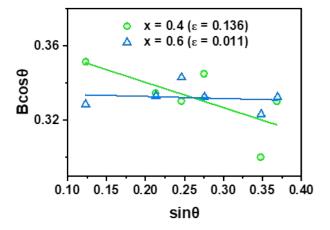


Figure S12 Williamson-Hall plot of $FAPb_{1-x}Sn_x(I_{0.8}Br_{0.2})_3$ perovskites (x = 0.4 and 0.6). The data were calculated using the (100), (110), (200), (210), (222) and (300) peaks from their XRD patterns.

Table S1 Photovoltaic parameters for best-performing PSCs based on FAPb_{1-x}Sn_x($I_{0.8}Br_{0.2}$)₃ (x = 0, 0.2, 0.4, 0.6, 0.8, and 1.0) perovskites.

Sn content / x value	J _{sc} (mA/cm ²)	V _{oc} (V)	FF	PCE (%)	J _{sc} from EQE (mA/cm ²)	V _{oc} loss (V)
0.0 (Pb)	1.58	0.97	0.43	0.66	2.36	1.15
0.2	14.44	0.70	0.55	5.53	14.59	0.74
0.4	19.68	0.70	0.65	8.94	19.03	0.70
0.6	20.39	0.68	0.60	8.36	20.64	0.71
0.8	15.29	0.41	0.51	3.20	16.29	1.02
1.0 (Sn)	11.34	0.34	0.52	1.99	12.17	1.20

Table S2 The capacitive and resistive components (C_g , R_s , R_{HF} , R_{LF} and R_{rec}) obtained from the fitting results of Nyquist spectra for the FAPb_{1-x}Sn_x($I_{0.8}$ Br_{0.2})₃ based PSCs with different Sn contents (x).

x value	C _g (nF)	$R_{s}\left(\Omega\right)$	$R_{\mathrm{HF}}\left(\mathrm{k}\Omega\right)$	$R_{LF}(k\Omega)$	$R_{rec}(k\Omega)$
0.4	8.1	22.9	1.4	6.2	7.6
0.6	8.7	25.6	0.9	4.6	5.5

Table S3 Fitting parameters of TAS spectra for $FAPb_{1-x}Sn_x(I_{0.8}Br_{0.2})_3$ (x = 0.4 and 0.6).

Sn content	λprobe (nm)	Aı	τι (ps)	A 2	τ2 (ps)	τ _{ave} (ps)
x = 0.4	850	0.46	112 ± 20	0.54	2700 ± 200	1500± 200
x = 0.6	870	0.38	18 ± 4	0.62	4300 ± 400	$2700 {\pm}~300$

Reference

1. M.-C. Jung, S. R. Raga and Y. Qi, RSC Advances, 2016, 6, 2819-2825.