Electronic Supplementary Information

Structure-Property Correlation of Hydrogels Obtained via Radical Polymerization Using Central Cores of Multiarm Star Polymers as Crosslinkers

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Fig. S1 Typical ¹H NMR spectrum of linear PNIPAAm (**LP**₅₀) obtained by RAFT polymerization. The DP_n and M_n were calculated from the integral ratio of peaks *d* and *g*. Reaction conditions: see Table 1.



Fig. S2 Typical ¹H NMR spectrum of star PNIPAAm (SP₅₀) obtained by RAFT polymerization. The ratio of the unreacted vinyl groups was calculated from the integral ratio of peaks *i* and *j*. Reaction conditions: see Table 2.



Fig. S3 SEC curves of star PNIPAAms with different amount of unreacted vinyl groups in the core (SP_{100_v0.3}, SP_{100_v1}, SP_{100_v6}, and SP₁₀₀). Reaction conditions: see Table 2.



Fig. S4 ¹H NMR spectrum of **SP**₁₀₀. The inset shows the spectra in range of 6.4-5.6 ppm attributable to vinyl bonds in the core of star PNIPAAms (**SP**_{100_v0.3}, **SP**_{100_v1}, **SP**_{100_v6}, and **SP**₁₀₀).

Table S1. Results of uniaxial tensile tests of the gels prepared with star PNIPAAms with differentamount of vinyl groups in the core. a

Entry	Young's modulus (kPa)	Breaking Stress (kPa)	Breaking Strain (%)
SG ₁₀₀	8.81 ± 0.41	8.20 ± 0.87	204 ± 31
SG _{100_v6}	6.31 ± 0.7	8.04 ± 0.99	322 ± 84
SG100_v1	2.70 ± 0.39	8.69 ± 0.16	420 ± 49

^{*a*} Preparation conditions: see Table 3. The representative stress-strain curves are shown in Fig. 3.

 Table S2. Analysis of SAXS profiles of the gels prepared with star PNIPAAms with different arm

molecular	weights. ^a

Gel	$q_{\max} (\mathrm{nm}^{-1})$	$d_{\rm star} ({\rm nm})^b$
SG _{100_2.5}	0.314	20.0
SG _{200_2.5}	0.307	20.5
SG _{200_5.0}	0.295	21.3
SG _{300_2.5}	0.310	20.3
SG _{300_7.5}	0.298	21.1
SG _{400_2.5}	0.307	20.5

^{*a*} Preparation conditions: see Table 4. The SAXS profiles are shown in Fig. 5. ^{*b*} Calculated by the following equation: d_{star}

 $= 2\pi/q_{\text{max}}.$

Gel	Young's Modulus (kPa)	Breaking Stress (kPa)	Breaking Strain (%)
SG _{50_2.5}	11.9 ± 1.1	11.9 ± 2.7	187 ± 35
SG _{100_2.5}	8.81 ± 0.41	8.20 ± 0.87	204 ± 31
SG150_2.5	9.74 ± 0.88	10.0 ± 1.5	465 ± 65
SG _{200_2.5}	5.58 ± 0.63	12.7 ± 2.1	662 ± 80
SG _{250_2.5}	5.71 ± 0.67	10.4 ± 1.7	738 ± 84
SG _{300_2.5}	4.57 ± 0.45	12.7 ± 1.5	766 ± 110
SG400_2.5	5.00 ± 0.66	4.00 ± 0.67	219 ± 29

Table S3. Results of uniaxial tensile tests of the gels prepared with star PNIPAAms with different arm molecular weights under the same weight fraction (2.5 wt%) of star crosslinkers.^{*a*}

^a Preparation conditions: see Table 4. The representative stress-strain curves are shown in Fig. 6a.

Table S4. Results of uniaxial tensile tests of the gels prepared with star PNIPAAms with differentarm molecular weights under a similar molar concentration of star crosslinkers.^a

Gel	Young's Modulus (kPa)	Breaking Stress (kPa)	Breaking Strain (%)
SG _{50_1.25}	6.15 ± 0.17	8.18 ± 1.32	359 ± 61
SG _{100_2.5}	8.81 ± 0.41	8.20 ± 0.87	204 ± 31
SG _{150_3.75}	16.1 ± 1.5	16.3 ± 2.1	398 ± 56
SG _{200_5.0}	13.0 ± 1.1	11.2 ± 2.3	257 ± 56
SG _{250_6.25}	10.4 ± 1.5	8.65 ± 1.76	210 ± 88
SG300_7.5	8.92 ± 0.72	2.64 ± 0.86	47.8 ± 16.2

^{*a*} Preparation conditions: see Table 4. The representative stress-strain curves are shown in Fig. 6c.



Fig. S5 SEC curves of star PNIPAAm (SP) employed for gel synthesis with a variety of AAm concentration and the linear PNIPAAm precursor (LP) obtained by RAFT polymerization. The preparation conditions were identical with SP₁₅₀ and LP₁₅₀ (LP: DP_{n, NMR} = 163, $M_{n, NMR}$ = 18,900; SP: $R_{C=C} = 9.3\%$).

 Table S5. Results of uniaxial tensile tests of the gels prepared with star PNIPAAms under various monomer concentrations.^a

Gel	Young's Modulus (kPa)	Breaking Stress (kPa)	Breaking Strain (%)
SG _{M20}	46.0 ± 3.4	35.0 ± 4.8	508 ± 147
SG _{M30}	104 ± 5	91.5 ± 17.9	459 ± 90
SG _{M40}	218 ± 16	173 ± 17	368 ± 80

^a Preparation conditions: see the caption of Fig. 8. The representative stress-strain curves are shown in Fig. 9.



Fig. S6 Effect of AAm concentration on the dissipated energy during cyclic tensile tests of starcrosslinked gels at varying strain levels. The dissipated energies were calculated from the area of the hysteresis loop in the stress-strain curves shown in Fig. 10.