

Supplementary Information

Trioxanes as photodegradable motifs for additive manufacturing

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1. NMR characterization of monomer and reference compound

1.1. Trioxane-containing ene-crosslinker

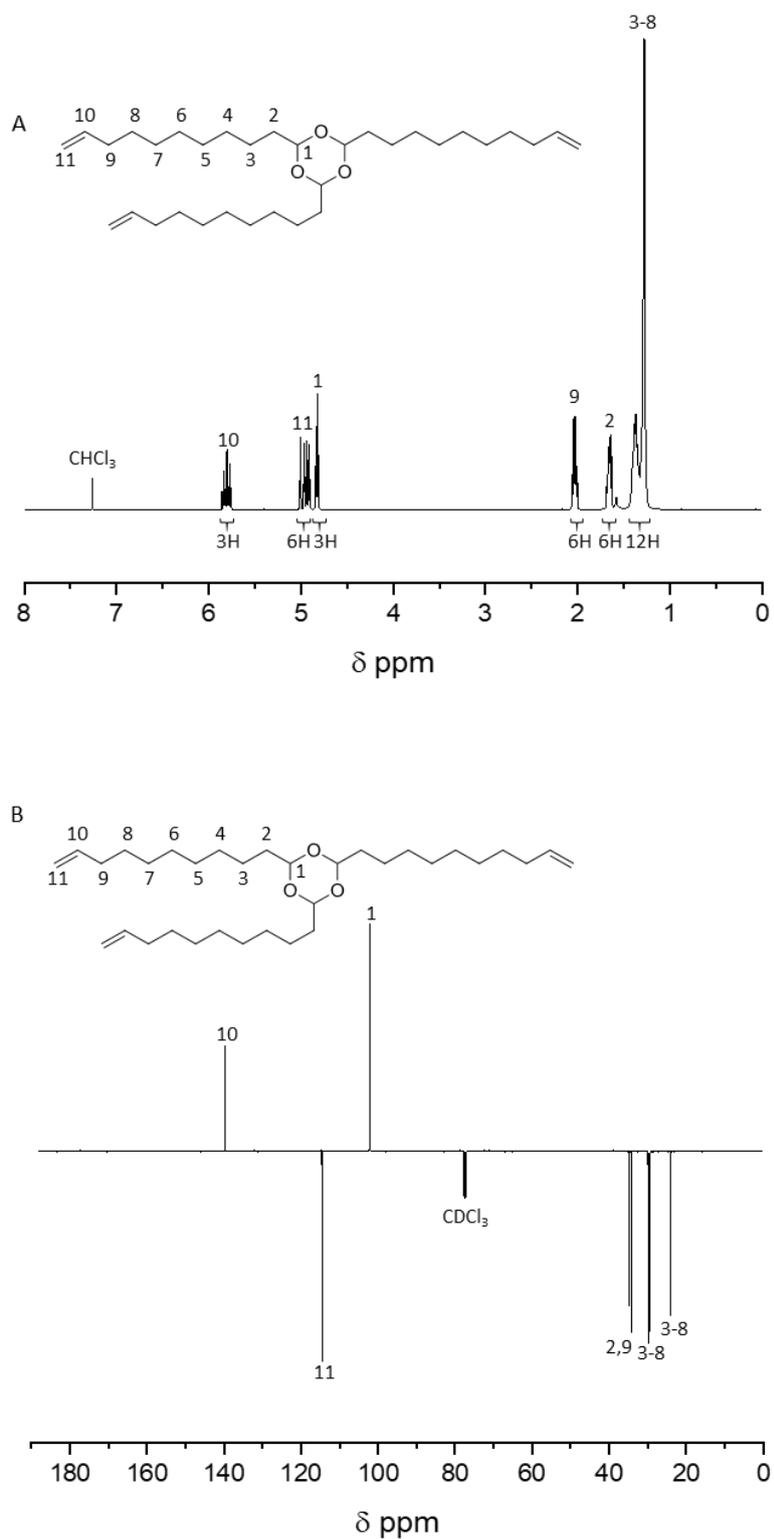


Figure S1: A) ^1H and B) ^{13}C (APT)- NMRs of the product (400 MHz, CDCl_3)

1.2. Reference compound

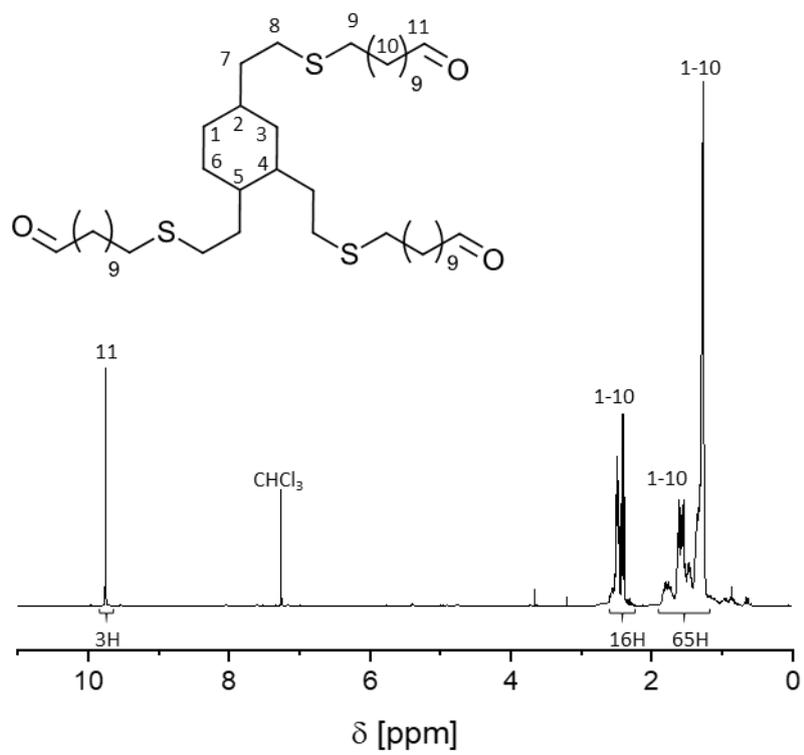


Figure S2: ^1H NMR of the reference aldehyde (400 MHz, CDCl_3)

2. Photorheology measurements

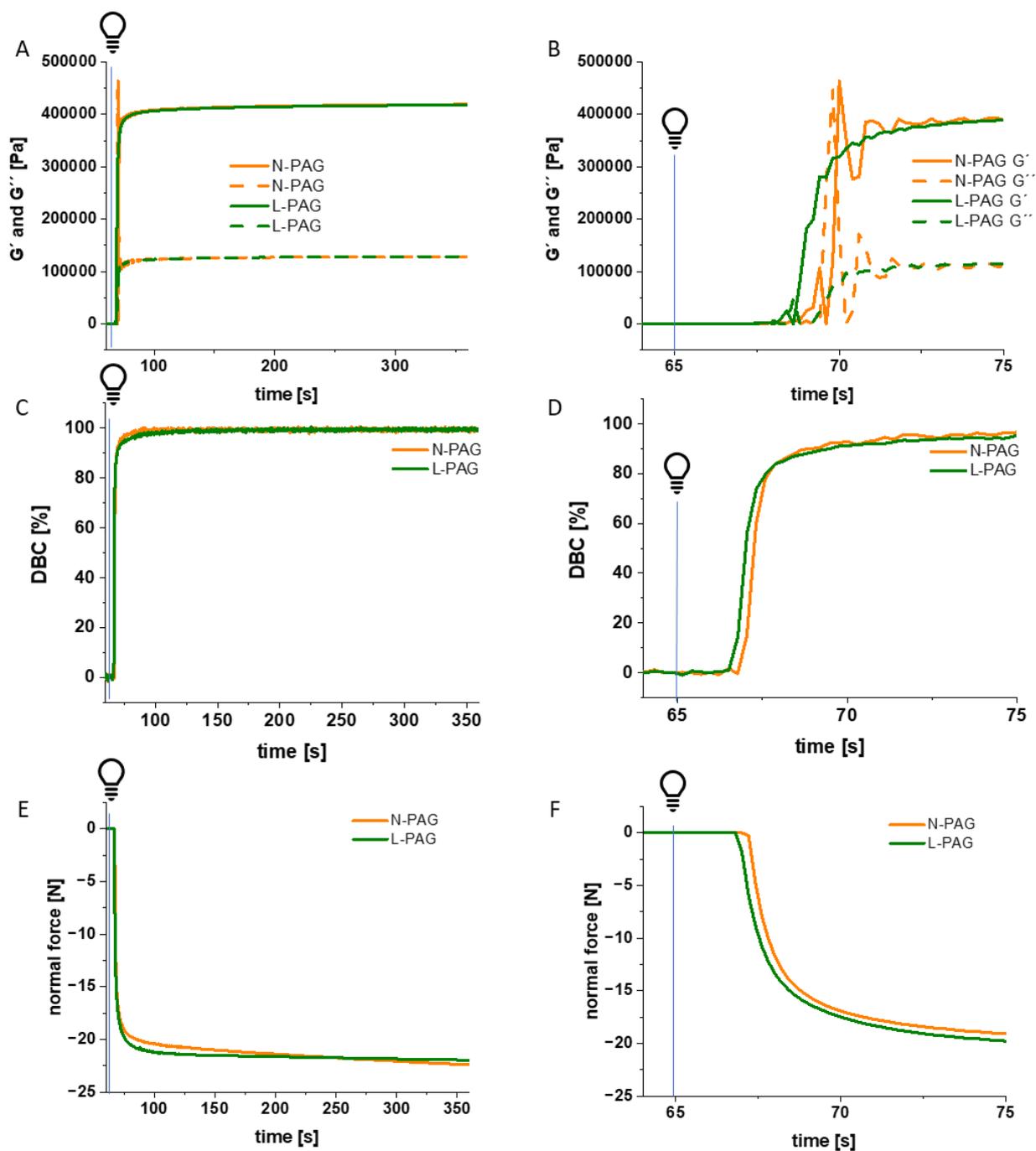


Figure S3: Photoreology curves: storage and loss moduli A) over the entire measured time range and B) zoomed-in version of the steep section, double bond conversion (DBC) C) over the entire measured time range and D) zoomed-in version of the steep section, normal force E) over the entire measured time range and F) zoomed-in version of the steep section. Vertical line at 65 s indicates start of irradiation (385 nm 50 mW cm⁻²)

3. IR-comparison trioxane monomer and 10-undecenal

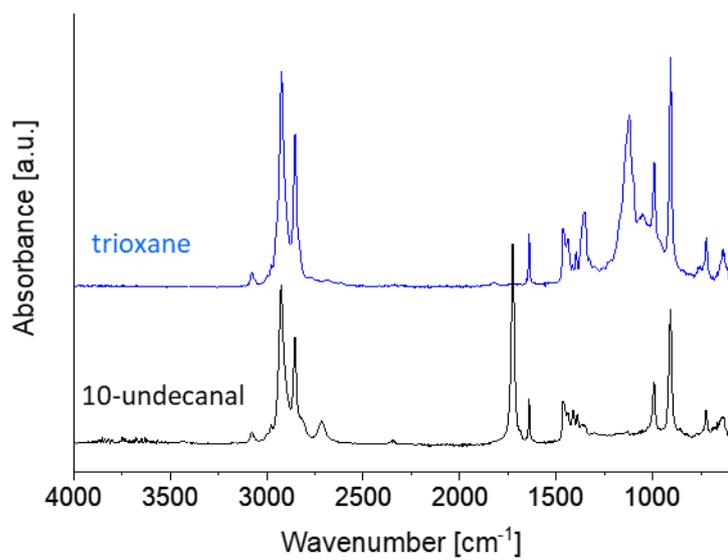


Figure S4: IR spectra of trioxane and reference aldehyde 10-undecenal

4. IR comparison different stages from formulation to degraded polymer

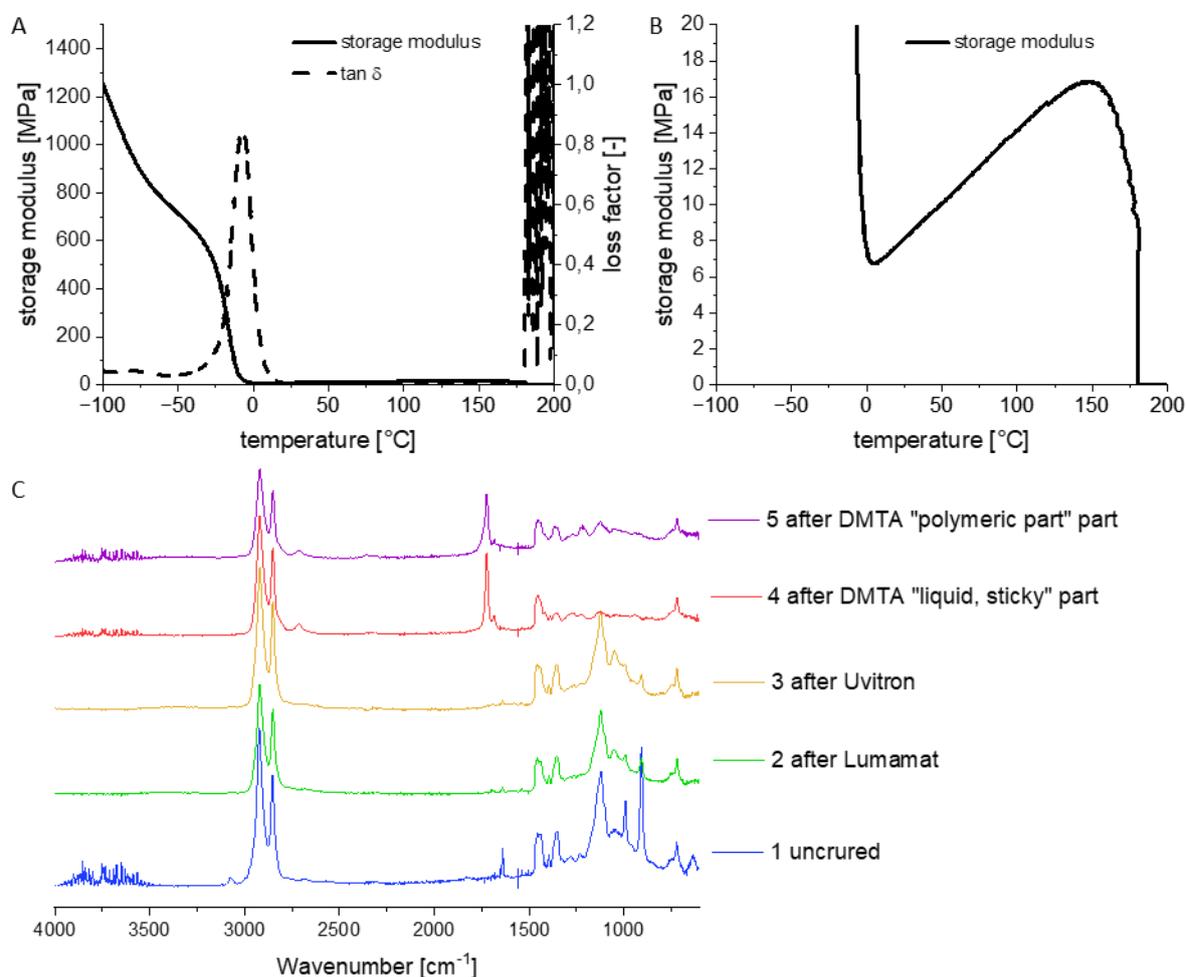


Figure S5: ATR-IR tracking of polymerization and degradation (bottom to top): 1 (blue): uncured formulation; 2 (green): formulation after Lumamat (400 nm- 500 nm) curing to photopolymerize the formulation; 3 (orange): polymer after UV-oven (Uvitron; 320 nm – 500 nm); DMTA specimen melted partly in the DMTA and therefore there consisted of a liquid (4, red) and a polymeric part (5, purple).

After the degradation studies of the trioxane monomer in solution, an **N-PAG** DMTA specimen was prepared by thiol-ene polymerization of the trioxane-based ene-crosslinker with the trithiol. To see if the trioxane ring would open by just UV irradiation, ATR-IR measurements were performed. The UV irradiation is strongest on the surface of the sample, therefore UV-light induced degradation was expected to occur on the surface and ATR-IR was chosen as analysis method.

The absence of a carbonyl peak at 1724 cm^{-1} in the intact polymer network **L-PAG** confirmed that there is no trioxane degradation and that this curing method is suitable for the polymer (Figure S5 2, 3). To check the thermo-mechanical behaviour the specimen was measured on the DMTA with the standard program that ranges from $-100 \text{ }^{\circ}\text{C}$ – $200 \text{ }^{\circ}\text{C}$ with a heating range of 2 K min^{-1} . During this analysis, a decrease in storage modulus starting at $160 \text{ }^{\circ}\text{C}$, with a loss factor starting at $175 \text{ }^{\circ}\text{C}$ was recorded, which was deemed unfeasible. This result can be caused by slipping, breaking or melting of the specimen. Opening the device confirmed partial melting of the sample starting from $160 \text{ }^{\circ}\text{C}$, which can explain the observed decrease in storage modulus.

Both the molten and the solid parts of the DMTA sample were measured on the ATR-IR and confirmed the formation of an aldehyde peak at 1724 cm^{-1} in line with the expected degradation reaction (Figure S5 4, 5).

5. Stability study of cured polymers

IR spectra of three polymers were measured, after storing them at 25 °C for 110 days and compared to the reference aldehyde, which was also taken as benchmark for 100% degraded trioxane groups and compared to a polymer without PAG, that was measured directly after curing it (Figure S5). For evaluation, the area of the carbonyl peak at 1700 cm⁻¹ was compared to the area of the C-H stretch bands at 2900 cm⁻¹ to yield a relative peak area. The relative peak area of the freshly synthesised polymer without PAG was taken as 0% degradation and was subtracted from the relative peak areas before comparing them to the reference compound to obtain a degree of degradation (D, Equation 1).

$$D = \frac{\frac{A_{1700, 110d}}{A_{2900 \text{ cm}^{-1}, 110d}} - \frac{A_{1700, \text{no PAG}}}{A_{2900 \text{ cm}^{-1}, \text{no PAG}}}}{\frac{A_{1700, \text{ref}}}{A_{2900 \text{ cm}^{-1}, \text{ref}}}} \quad [1]$$

A_{1700,110d} ... area of the carbonyl peak in the sample after 110 days (1700 cm⁻¹)

A_{2900,110d} ... area of the peak at 2900 cm⁻¹ in the sample after 110 days (2900 cm⁻¹)

A_{1700,noPAG} ... area of the carbonyl peak of a freshly cured sample without PAG (1700 cm⁻¹)

A_{2900,noPAG} ... area of the peak at 2900 cm⁻¹ of a freshly cured sample without PAG (2900 cm⁻¹)

A_{1700,ref} ... area of the carbonyl peak of the reference aldehyde (1700 cm⁻¹)

A_{2900,ref} ... area of the peak at 2900 cm⁻¹ of the reference aldehyde (2900 cm⁻¹)

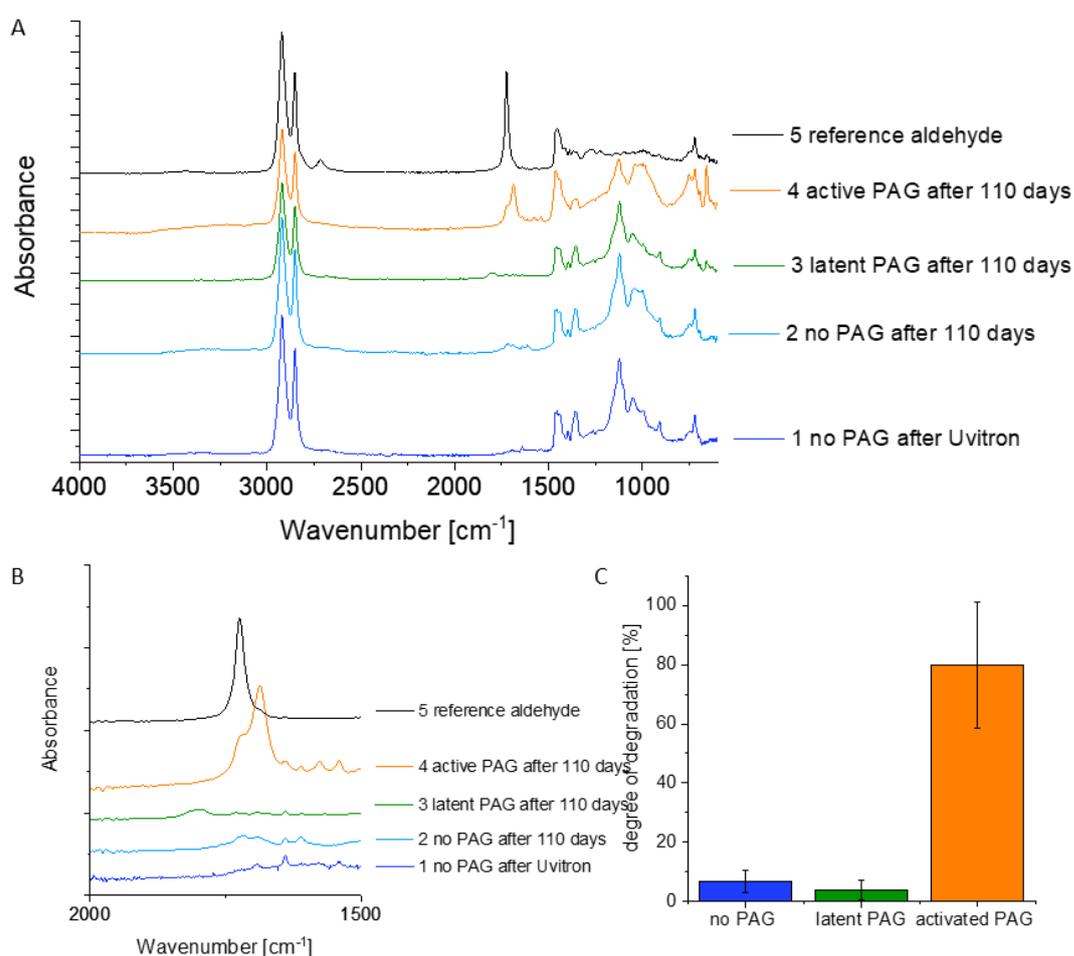


Figure S6: Degradation study of polymers made of the trioxane monomer and the thiol CHTT for specimen containing latent (green, 3) or activated photoacid generator (PAG; orange, 4) compared to the reference aldehyde (5, black) and a freshly cured polymer with (1, dark blue) and without (light blue, 2) PAG: A) full spectrum B) carbonyl peak C) Calculated degree of degradation (D) of trioxane groups after 100 days.

The data shows that the polymer with activated PAG has a high degree of degradation around 80%, even though no heat was applied to this system. The polymers without PAG and with latent PAG do not show a peak in the carbonyl region. The recorded degree of degradation could also correspond to noise in the spectrum and can be seen as minimal for the two polymers.