Supporting Information for

On-Demand Design of Materials with Enhanced Dielectric Properties *via* Machine Learning-Assisted Materials Genome Approach

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S1. Data sources of dielectric properties

We constructed an extensive dataset comprising dielectric performance data of various polymers. The sources of the collected dielectric performance data are primarily from 70 articles, 3 handbook publications, the PolyInfo database, and the AI plus Polymers database as below.

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S2. Model performance using inputs without dimensionality reduction

It is feasible to directly take all the Mordred descriptors as model inputs for ML modeling. Here, we attempted this operation and found that the dimensionality reduction of descriptors is necessary for training ML models with high accuracy. Taking Gaussian process regression (GPR) and support vector regression (SVR) modeling processes as examples, the Mordred descriptors with or without dimensionality reduction serve as model inputs for training the models of dielectric constant ε , dielectric loss tan δ , and T_{d5} , respectively. For the 100 models trained in each case, the distributions of the R^2 values for the test sets and the consumption time for predicting one structure are presented in Figure S1 and S2, respectively. As can be seen, for the three kinds of targeted properties, the ML models based on the inputs after dimensionality reduction have a higher R^2 and a lower consumption time. For example, for the GPR and SVR models of tan δ , if the dimensionality reduction operation is not conducted for the Mordred descriptors, the R^2 of the ML models is difficult to exceed 0.8, and the consumption time for predicting one structure increases significantly.

As we know, feature engineering is an action of extracting features from raw data. The exact and sufficient features can reduce the difficulty of modeling, and therefore enable the ML models to output results with high quality. Therefore, the dimensionality reduction processing of descriptors, such as the PCA and correlation-based feature engineering methods adopted in this work, can achieve more reliable and effective feature engineering, resulting in the rapid and accurate prediction performance for the ML models of ε , tan δ , and T_{d5} .



Figure S1. Comparison of the model training results using the inputs with and without dimensionality reduction. The box plots show the distributions of the R^2 values of the test set for the GPR and SVR models of ε , tan δ , and T_{d5} . The green and orange boxes correspond to the results of trained models with and without dimensionality reduction, respectively.



Figure S2. Comparison of the consumption time for predicting one structure when adopting the GPR and SVR models of ε , tan δ , and T_{d5} . The green and orange boxes correspond to the results of trained models with and without dimensionality reduction, respectively.

S3. Model performance using various PCA retaining explainabilities

Here, we compared the prediction performance of the trained ML models using the PCA results from the retaining explainability of 85%, 90%, and 95% as inputs. As shown in Figure S3a, retaining 90% of explainability in the GPR modeling for ε , the trained prediction models exhibit a higher R^2 on the test set. When training the SVR models for T_{d5} , choosing 90% or 95% of explainability can enable the trained models to show an R^2 exceeding 0.8 on the test set (Figure S3b). A low PCA retaining explainability leads to the missing of important features, while a high explainability can cause a complex feature space and increase the possibility of model overfitting. Thus, the retaining explainability of 90% or 95% in the present work is reasonable.



Figure S3. Comparison of the training results using the PCA results from the retaining explainability of 85%, 90%, and 95% as inputs. The box plots show the distributions of the R^2 values of the test set for (a) GPR models of ε and (b) SVR models of T_{d5} .

S4. Model performance using various ML modeling methods

For each targeted property, we adopted at least five ML modeling methods for training, including GPR, SVR, decision tree regression (DTR), *etc.* Under different training parameters, the predictive performance of 100 models trained by the above ML modeling methods was evaluated. Then, the ML modeling methods suitable for each targeted property were selected according to their performance.

For example, after performing GPR, SVR, and DTR modeling on the datasets of ε , tan δ , and T_{d5} , respectively, the predictive performance of these trained models is shown in Figure S4. For the training results of ε and tan δ , the GPR models show the R^2 value exceeding 0.8 on the test set, while the SVR and DTR models perform poorly on the test set (Figure S4a-b). For the training results of T_{d5} , both the GPR and SVR models have a good prediction performance on the test set, that is, R^2 can reach 0.8. However, the R^2 value of the training set for the GPR models is close to 1.0, which is significantly higher than that of the test set, indicating a risk of model overfitting. Therefore, the ML prediction models on the datasets of ε and tan δ were established by using the GPR modeling method, while the ML model of T_{d5} was trained by adopting the SVR modeling method, as indicated by the marks in Figure S4.



Figure S4. Comparison of the training results using GPR, SVR, and DTR modeling methods on the dataset of (a) ε , (b) tan δ , and (c) T_{d5} , respectively.

S5. Selection and combination of kernels during the GPR training process

During the GPR training process, the Matern kernel is combined with the constant kernel and the white noise kernel, aiming to improve the prediction performance of the GPR models and avoid overfitting. Compared with the radial basis function kernel, the Matern kernel is more flexible. By using different smoothing parameters, the smoothness degree of the regression results can be adjusted.

For example, during the GPR training processes of ε and tan δ , 100 GPR models were trained using kernel parameters with *nu* values of 0.5, 1.5, and 2.5, respectively. The model performance on the training set and the test set is shown in Figure S5. As can be seen from the figure, when the model kernel parameter *nu*=0.5 or 1.5, the performance of the GPR models in the training set and test set shows significant differences, indicating the overfitting of models. We selected the Matern kernel with *nu* = 2.5 (*i.e.*, M52 kernel), which is suitable for modeling the physical processes with certain irregularities. Compared with the Matern kernel with *nu* =1.5 or 0.5, the M52 kernel can capture more local variations, thereby, it not only fits the data features better, but also effectively avoids model overfitting and improves the generalization ability.



Figure S5. Comparison of the training results using the various kernel parameters of GPR modeling methods on the datasets of (a) ε and (b) tan δ , respectively.

In addition, the product combination of the constant kernel and the Matern kernel enables the GPR model to automatically learn the global scaling of the signal amplitude. The white noise kernel is further combined, achieving the consideration of data noise during the modeling process. Furthermore, the method of maximizing logarithmic marginal likelihood was adopted for the kernel parameter space, achieving the optimization of all parameters of the kernel function.

S6. The 5-fold cross-validation results for the models of targeted properties

Here, the 5-fold cross-validation method has been utilized to evaluate the performance of the trained models of ε , tan δ , and T_{d5} , respectively. The standard deviations from the 5-fold cross-validation for each property are presented in Table S1. It can be seen from Table S1 that the trained models exhibit low standard deviations on the train and validation folds. This indicates that the models have a high prediction accuracy and generalization ability, and model overfitting has been avoided.

		Model of ε	Model of $tan \delta$	Model of T_{d5}
<i>R</i> ²	Train_std	0.0026	0.0046	0.0091
	Validation_std	0.0398	0.0293	0.0940
MAE	Train_std	0.0058	0.0037	0.8932
	Validation_std	0.0069	0.0123	4.2277

Table S1. The standard deviations from the 5-fold cross-validation for each property.

S7. Correlation-based feature engineering for the dielectric loss

The purpose of feature engineering is to capture the structural features of polymers. We utilized SMILES strings to represent the repeating unit structures and employed the RDKit software package to calculate Mordred descriptors for the chemical structures in the datasets. ^[S1, S2] It is imperative to reduce the dimensionality of the descriptors, obtaining a few descriptors that retain as much molecular information as possible for model training. However, due to the complex relationships between features of tan δ and the test conditions (*i.e.*, temperature and frequency), the utilization of the principal component analysis (PCA) dimensionality reduction for all the descriptors results in the loss of some chemical structure feature information that is related to tan δ .

To solve this problem, we extended the correlation-based feature analysis method to the dimensionality reduction and feature extraction in the present work. This method is commonly used in the feature engineering of molecular structures, such as extracting the descriptors of molecular solubility and establishing the corresponding ML prediction model. ^[S3, S4] In such a method, the Pearson correlation coefficient (r) is utilized as the metric. If the absolute value of r between two features is high ($r \ge 0.9$), indicating that the two features can replace each other, one of them can be discarded. Using the correlation-based feature engineering method, we selected the top 30 descriptors that have a higher correlation with tan δ from all the descriptors. Subsequently, the descriptors that can be mutually substituted among the top 30 descriptors were further eliminated, remaining only 17 descriptors that are enough to represent molecular structural features. The 17 molecular Mordred descriptors after correlation-based feature engineering for training the tan δ model are listed in Table S2.

Name	Description
SMR_VSA6	MOE MR VSA Descriptor 6 (2.75 < x < 3.05)
RotRatio	rotatable bonds ratio
NssCH2	number of ssCH ₂
ATSC3Z	centered moreau-broto autocorrelation of lag 3 weighted by atomic number
AATS3dv	averaged moreau-broto autocorrelation of lag 3 weighted by valence electrons
BIC2	2-ordered bonding information content
SLogP	Wildman-Crippen LogP
fMF	molecular framework ratio
nS	number of S atoms
BalabanJ	Balaban's J index
C3SP2	SP ₂ carbon bound to 3 other carbons
AATSC1Z	averaged and centered moreau-broto autocorrelation of lag 1 weighted by atomic number
nHBDon	number of hydrogen bond donor
NaaCH	number of aaCH
MATS1Z	moran coefficient of lag 1 weighted by atomic number
TMPC10	10-ordered total path count
Xc-5d	5-ordered Chi cluster weighted by sigma electrons

Table S2. Mordred descriptors used for training the dielectric loss model.

Figure S6 presents the analysis results of the intercorrelations among the aforementioned 17 descriptors and their correlation with $\tan \delta$. The heatmap illustrates the degree of correlation, ranging from blue (indicating negative correlation) to red (indicating positive correlation). Deeper shades of blue signify a higher negative correlation between the descriptors, while deeper shades of red indicate a higher positive correlation. Red lines denote descriptors positively correlated with $\tan \delta$, whereas blue lines denote descriptors negatively correlated with $\tan \delta$. There are several features with a high

correlation with tan δ . For instance, SLogP, which stands for Synthetic Logarithm of Partition Coefficient, serves as an indicator for the hydrophobicity (lipophilicity) of compounds. There exists a negative correlation between SLogP and tan δ , implying that the introduction of lipophilic groups can lead to a reduction in tan δ . Compounds with lipophilic properties tend to contain fewer polar groups, resulting in a decrease in tan δ . The SMR_VSA series of descriptors is utilized to quantify the contributions of different atom types within a molecule to its molecular refraction. In particular, the SMR_VSA6 exhibits a positive correlation with tan δ , indicating that a high value of SMR_VSA6 signifies the presence of a greater number of atomic or functional groups that contribute significantly to molecular refraction. The existence of these groups leads to an increased tan δ of the molecule. Utilizing the 17 descriptors after correlation-based feature engineering, the prediction model of tan δ was established. Compared to the PCA dimensionality reduction method, the correlation-based feature engineering method can offer good interpretability and achieve an accurate prediction of tan δ .



Figure S6. Correlation analysis of descriptors used for training the model of $tan\delta$.

S8. Chemical structures of 130 silane genes

We collected 130 types of silane genes from the open-source PubChem database.^[S5] The molecular structures of the 130 types of silane genes are shown below, where the gene number (No.), the chemical abstracts service (CAS) identifier, or the compound ID (CID) in the database are also provided.





H₂N



Cŀ





Cŀ

-CI

No. 53 CAS: 183798-64-3

No. 59 CAS: 4074-06-0

No. 43 CAS: 18414-38-5

No. 49 CAS: 93917-77-2

No. 44 CAS: 1078-96-2

No. 45 CAS: 17887-45-5

CĮ

ĊΙ

No. 46 CAS: 18035-74-0

No. 47 CAS: 17864-93-6

No. 48 CAS: 71550-63-5



-CI CI

No. 54 CAS: 148718-31-4



No. 60 CAS: 18028-96-1



No. 66 CAS: 18290-79-4



No. 72 CAS: 5578-41-6





No. 84 CAS: 71550-62-4

No. 51 CAS: 151236-68-9

CI-Si-CI



No. 57 CAS: 256343-28-9

No. 58 Compound CID: 22156374

No. 52 CAS: 18276-28-3



=

No. 64 CAS: 145289-11-8



No. 71 CAS: 18395-92-1

N

No. 76 CAS: 1077-57-2



No. 82 CAS: 1009-46-7









SI 19





No. 62 CAS: 18147-18-7

No. 68

CAS: 7282-39-5

CI

No. 74 CAS: 18379-40-3

No. 80 CAS: 168270-65-3

No. 63 CAS: 17869-31-7



No. 69 CAS: 268217-73-8

No. 70 CAS: 14579-04-5



CI





No. 81 CAS: 18290-58-9





No. 78 CAS: 273934-04-6



No. 83 CAS: 18147-23-4

No. 77 CAS: 93037-56-0

No. 65 CAS: 876343-97-4





No. 50 CAS: 18042-67-6

No. 56 CAS: 790234-74-1



No. 55 CAS: 148718-31-4

No. 61 CAS: 216320-36-4

No. 67 CAS: 18243-57-7



No. 73 Compound CID: 89655867

No. 79 CAS: 790234-76-3



CI

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No. 91 CAS: 57831-43-3



No. 97 CAS: 17998-62-8



No. 103 CAS: 64102-92-7



No. 115 CAS: 225940-04-5

No. 116 CAS: 101931-61-7





SI 20







CI cí



No. 120 CAS: 20160-45-6









CI-Si-CI

QH



SI 21

S9. Chemical structures of 287 diynes genes

We collected 287 types of dignes genes from the open-source PubChem database.^[S5] The molecular structures of the 287 types of dignes genes are shown below, where the gene number (No.), the CAS identifier or CID in the database are also provided.















No. 31 CAS: 2417377-17-2

No. 32 CAS: 2417377-73-0

No. 33 CAS: 2417378-09-5

No. 34 CAS: 2417377-58-1

No. 35 CAS: 2417377-77-4 No. 36 CAS: 2417377-68-3





No. 39 CAS: 2417377-45-6



No. 40 CAS: 2417378-06-2



No. 42 CAS: 2417377-96-7



No. 37 CAS: 2417377-68-3





No. 41 CAS: 2409061-50-1



No. 48 CAS: 2417377-64-9

No. 43 CAS: 2417377-83-2

No. 44 CAS: 2417377-60-5

No. 45 CAS: 27559-94-0



No. 47 CAS: 2417377-51-4



No. 49 CAS: 2417378-18-6



No. 55 CAS: 2417377-79-6

No. 61 CAS: 1293326-48-3

No. 56 CAS: 2417377-76-3

No. 50 CAS: 2417378-30-2



No. 57 CAS: 2417378-20-0



No. 58 CAS: 1397958-88-1

=



No. 60 CAS: 514790-50-2





No. 62 CAS: 914673-80-6







No. 64 CAS: 2417378-29-9

No. 66 CAS: 2417377-61-6



No. 59 CAS: 2417378-00-6

No. 65 CAS: 2417377-19-4

No. 51 CAS: 2417377-94-5

No. 52 CAS: 2417377-31-0

No. 53 CAS: 209254-89-7







No. 97 CAS: 2417378-27-7

No. 98 CAS: 2126816-74-6

No. 99 CAS: 507276-82-6

No. 100 CAS: 113705-27-4

No. 101 CAS: 173678-77-8







No. 103 CAS: 960223-15-8

No. 105 CAS: 2126816-75-7

No. 106 CAS: 2417377-80-9

No. 108 CAS: 2417378-23-3











No. 109 CAS:765911-39-5

No. 110 CAS: 2348521-77-5

No. 111 CAS:2417377-13-8

No. 112 CAS: 1610942-28-3

No. 113 CAS: 2417377-48-9











No. 107 CAS: 2417377-85-4



No. 115 CAS:2417378-14-2

No. 116 CAS: 918778-82-2



No. 118 CAS:2417377-21-8

No. 119 CAS: 951791-07-4

No. 120 CAS: 1629749-69-4



No. 121 CAS: 2417378-35-7



No. 128 CAS:914477-80-8

No. 134 CAS:27594-21-4

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No. 123 CAS:627892-84-6







No.126 CAS: 530097-70-2



No. 127 CAS:627892-82-4



No. 129 CAS:1276668-76-8



No. 130 CAS:2417377-95-6





No. 131 CAS: 2417377-95-6



No. 132 CAS: 909342-66-1



No. 133 CAS: 2417377-70-7



No. 135 CAS: 1240453-62-6

No. 136 CAS: 21368-80-9





No. 138 CAS: 2417377-56-9

No. 125 CAS: 951791-09-6













No. 139 CAS2417377-81-0

No. 145 CAS: 2417377-57-0

No. 141 CAS: 27594-30-5

No. 142 CAS: 7535-21-9

No. 143 CAS: 2417377-55-8

No. 144 CAS: 2417377-43-4

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No. 140 CAS: 909342-65-0



No. 147 CAS: 2417377-78-5



No. 148 CAS: 357437-78-6



No. 150 CAS: 6140-83-6





No. 146 CAS:2417378-05-1







No. 151 CAS: 2417378-38-0

No. 152 CAS: 162318-26-5

No. 158 CAS: 2417378-25-5

No. 164 CAS: 158599-68-9

No. 153 CAS: 38215-38-2

No. 159 CAS: 2417377-08-1

No. 154 CAS: 2417378-19-7

No. 160 CAS: 147613-78-3

No. 166 CAS: 214406-87-8



No. 162 CAS: 2417378-22-2



No. 157 CAS: 2417378-13-1

No. 163 CAS: 2417377-71-8



No. 165 CAS: 856000-46-9







No. 168 CAS: 2417378-26-6



No. 169 CAS: 2417378-10-8

No. 170 CAS: 2417378-43-7 No. 171 CAS: 258531-35-0





No. 174 CAS: 47230-46-6

No. 161 CAS: 935-14-8

No. 167 CAS: 87710-28-9



No. 203 CAS: 2058051-39-9

No. 204 CAS: 2251049-43-9

No. 205 CAS: 2417377-18-3

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No. 206 CAS: 2417377-98-9

No. 207 CAS: 1542157-70-9

No. 208 CAS: 2417377-39-8





No. 251 CAS: 75610-48-9

HO

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No. 257 CAS: 47031-58-3

HC



No. 269 CAS: 427879-51-4

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No. 275 CAS: 79109-92-5

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No. 276 CAS: 2417378-21-1



No. 278 CAS: 960223-15-8

No. 279 CAS: 256439-93-7

No. 280 CAS: 918778-82-2

SI 29











No. 281 CAS: 256439-93-7

No. 283 CAS: 38002-32-3

No. 284 CAS: 137000-66-9

No. 285 CAS: 507276-82-6

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No. 286 CAS: 141135-35-5

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No. 287 CAS: 124417-94-3

S10. The list of the promising PSAs for wave-transparent application

To provide various synthetic options for the experimentalists, the screened PSAs containing genes with significant structural differences are considered as promising PSAs. Here, the list of the total 35 promising PSA structures for wave-transparent materials is given in Table S3 (the 5 representative PSAs listed in Table 1 are included).

ID	Structure	3	tan∂	<i>T</i> d5/°C	ScoreT	ScoreF
1		2.39	0.0016	665	0.18	0.19
2		2.33	0.0015	662	0.14	0.20
3		2.35	0.0020	661	0.12	0.19
4		2.40	0.0020	661	0.20	0.16
5		2.30	0.0015	659	0.20	0.18
6		2.31	0.0015	658	0.15	0.21
7		2.34	0.0018	658	0.09	0.23
8	-=-{}-=-{}	2.35	0.0011	658	0.24	0.21
9		2.38	0.0012	658	0.20	0.21

Table S3. The list of the promising PSAs for wave-transparent application*

ID	Structure	3	tan∂	<i>T</i> d5/°C	ScoreT	ScoreF
10		2.27	0.0021	657	0.24	0.20
11		2.39	0.0017	654	0.24	0.17
12		2.29	0.0014	652	0.16	0.21
13		2.32	0.0017	649	0.10	0.21
14		2.35	0.0016	640	0.22	0.20
15		2.30	0.0014	630	0.18	0.22
16		2.28	0.0020	629	0.14	0.24
17		2.28	0.0020	629	0.15	0.24
18		2.24	0.0018	623	0.23	0.22
19		2.30	0.0019	622	0.13	0.24
20		2.26	0.0017	621	0.19	0.24
21		2.40	0.0016	618	0.13	0.24
22		2.35	0.0021	611	0.23	0.21

ID	Structure	3	tanð	<i>T</i> d5/°C	ScoreT	ScoreF
23		2.26	0.0020	602	0.16	0.25
24		2.25	0.0016	593	0.22	0.20
25		2.31	0.0017	593	0.13	0.20
26		2.37	0.0019	593	0.18	0.24
27		2.30	0.0014	592	0.19	0.22
28		2.38	0.0013	589	0.11	0.23
29		2.40	0.0012	588	0.21	0.24
30		2.39	0.0014	578	0.16	0.22
31	Dicio	2.39	0.0015	563	0.13	0.24
32		2.26	0.0007	562	0.22	0.23
33		2.27	0.0007	562	0.23	0.24
34	D'S COLO	2.38	0.0015	561	0.13	0.24
35		2.37	0.0015	560	0.21	0.19

* The values of ε and tan δ given in the table were predicted at 25°C and 1kHz.

S11. The list of the promising PSAs for thin-film capacitor application

For the 36 screened PSAs with high ε , low tan δ , and high T_{d5} , we evaluated their dielectric property stabilities on temperature and frequency. Following the screening criteria of *ScoreT* \leq 0.3 and *ScoreF* \leq 0.3, 12 promising PSA structures were identified to exhibit the application potential as the resin matrix in the field of thin-film capacitors. These promising PSAs are given in Table S4 (the 5 representative PSAs listed in Table 1 are included).

ID	Structure	3	tanδ	<i>T</i> d5 / °C	ScoreT	ScoreF
1		3.74	0.0009	613	0.29	0.18
2		3.78	0.0003	595	0.29	0.27
3		3.79	0.0019	579	0.13	0.28
4		3.73	0.0016	576	0.16	0.29
5	J ^o Sí	3.78	0.0018	575	0.13	0.28
6	She in the	3.74	0.0016	572	0.18	0.29
7	yo~sí	3.77	0.0017	571	0.18	0.28
8		3.74	0.0016	568	0.18	0.30

Table S4. The list of the promising PSAs for thin-film capacitor application*

ID	Structure	3	tan∂	<i>T</i> d5 / °C	ScoreT	ScoreF
9		3.77	0.0017	567	0.18	0.29
10	H ₂ N	3.86	0.0004	558	0.13	0.30
11		3.95	0.0014	547	0.15	0.20
12		3.80	0.0021	547	0.30	0.19

* The values of ε and tan δ given in the table were predicted at 25°C and 1kHz.

S12. Synthesis and characterizations of PSNP-MV resin

To validate the reliability of the structural design method for heat-resistant PSAs with enhanced dielectric performance, we conducted experimental verifications. Taking into account the synthesis accessibility, we chose one kind of PSA resin containing 2,7 diethynylnaphthalene and dichloromethylvinylsilane, named PSNP-MV. The predicted ε and tan δ of the PSNP-MV resin under the condition of 25°C and 1kHz are 2.97 and 0.00075, ranking at the top 2.5% and top 50% of the low ε and tan δ for the PSA candidates, respectively. In addition, the corresponding T_{d5} predicted by the ML model is 673 °C, ranking at the top 0.02% of the high heat resistance for the candidates. Regarding the dielectric property stability, the predicted values of *ScoreT* and *ScoreF* for the PSNP-MV resin are 0.22 and 0.28, which satisfies the screening criteria of *ScoreT* \leq 0.3 and *ScoreF* \leq 0.3.

The 2,7-diethynylnaphthalene monomer was synthesized by the Sonoga-Shira coupling method.^[S6] Then, the PSNP-MV was prepared through a Grignard reaction between dichloromethylvinylsilane and 2,7-diethynylnaphthalene, as shown in Figure S7a. For example, ethyl magnesium bromide (37.5 mL, 37.5 mmol) was added to the three-necked flask with the stirring turned on, and then 2,7-diethynyl naphthalene (3.0 g, 17.04 mmol) in THF (30 mL) was slowly dripped through the funnel in an ice/water bath. The mixture was heated to 70 °C and stirred for 2 hours to produce 2,7-diethynylnaphthalene Grignard reagents. The reaction system was cooled to 20 °C, and then a solution of dichloromethylvinylsilane (1.60 g, 11.36 mmol) in THF (30 mL) was added. The reaction was then refluxed at 70 °C for 2 hours. After cooling down, 8% aqueous hydrochloric acid (50 mL) and methyl tert-butyl ether (100 mL) were slowly added to the flask. After that, the mixture was washed to neutral with deionized water, and then the solution was dried with anhydrous sodium sulfate. Finally, the solvent was distilled off to get a liquid resin (*i.e.*, PSNP-MV) with a yield of 92% (3.47 g).



Figure S7. (a) Synthesis route of PSNP-MV. The PSNP-MV is prepared through a Grignard reaction between dichloromethylvinylsilane and 2,7-diethynylnaphthalene. (b) ¹H NMR (CDCl₃, ppm) spectra and (c) FTIR spectra of PSNP-MV.

The chemical structures of PSNP-MV were characterized by ¹H NMR and FTIR spectra. Figure S7b shows the ¹H NMR (CDCl₃, ppm) spectra of PSNP-MV. The resonance peaks at 7.57–8.02 ppm are assigned to the aromatic protons (Ph-H), and the peak at 3.15 ppm is caused by the terminal acetylene groups (C=C-H). The vinyl protons on silicon and the silicon methyl protons resonate at 6.20–6.28 ppm and 0.71 ppm, respectively. The ratio of the relative peak integral area is close to the design value, indicating that the designed structure was synthesized. In the present work, the number of repeating units of PSNP-MV is 2.

Figure S7c shows the FTIR spectra of PSNP-MV. The absorption at 3292 cm⁻¹ is attributed to the vibration of the C=C-H group. There is a characteristic absorption peak at 3014 cm⁻¹ for the stretching vibration of the C=C-H₂ bond. The characteristic peak at 2155 cm⁻¹ belongs to the stretching vibration

of the -C=C- bond. The bands of 1593 cm⁻¹ and 1502 cm⁻¹ are stretching vibrations of the naphthalene ring. The above results indicate that the spectra meet the designed structure of PSNP-MV.

The thermal curing process of PSNP-MV was carried out in air conditions, involving 150 °C for 4 hours, 180 °C for 2 hours, 200 °C for 2 hours, and 230 °C for 2 hours. The thermal stability of the cured PSNP-MV resin was examined by a thermal gravimetric analyzer under a nitrogen atmosphere. Figure S8 shows the TGA curve of the cured PSNP-MV resin. The PSNP-MV thermoset has a T_{d5} of 671 °C, exhibiting excellent thermal stability. The experimental T_{d5} value of 671 °C is close to the predicted value of 673 °C, indicating the reliability of the T_{d5} prediction model.



Figure S8. Thermo-gravimetric analysis curve of the cured PSNP-MV resin at a heating rate of $10 \,^{\circ}\text{C}\cdot\text{min}^{-1}$ under a nitrogen atmosphere.

S13. Additional experimental validation of PSA-V resin

Here, we chose another kind of PSA resin and carried out the additional experimental validation. Poly(silicon-containing arylacetylene vinyl)s (PSA-V) resin was synthesized from m-diacetylene benzene and dichloromethylvinylsilane, and its 5% thermal decomposition temperature (T_{d5}) can reach 630 °C (see Figure S9). By utilizing the established ML models, the predicted T_{d5} value of PSA-V is 642 °C, which is consistent with the experimental value. In addition, the PSA-V resin is predicted to exhibit a relatively higher dielectric constant and higher dielectric loss than the PSNP-MV resin. We examined the dielectric properties of the cured PSA-V at various temperatures and frequencies. Figure S10 shows the comparison between experimental and predicted values of the dielectric properties, where the orange symbols correspond to ε and the green symbols represent tan δ . It can be seen that at various test conditions, the experimental values of ε and tan δ are in the ranges of 3.7~4.0 and 0.005~0.01, respectively, which agrees with the predicted values.



Figure S9. Thermo-gravimetric analysis curve of the cured PSA-V resin at a heating rate of 10 $^{\circ}$ C· min⁻¹ under a nitrogen atmosphere.



Figure S10. The variations of dielectric constant and loss with frequency at various test temperatures (a) 25 °C, (b) 100 °C, (c) 200 °C, and (d) 300 °C for the cured PSA-V resins.

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