Supporting information

Tailoring the room-temperature miscibility gap in ordered spinel LiNi_{0.5}Mn_{1.5}O₄ cathodes by multielement doping

Shunsuke Narumi^{‡a}, Eugenio H. Otal^{‡,a}, Tien Quang Nguyen^{‡,b}, Michihisa Koyama^c, and

Nobuyuki Zettsu* a,b,c,d

^a Department of Materials Chemistry, Faculty of Engineering, Shinshu University, 4-17-1 Wakasato, Nagano 380-8553, Japan

^b Institute for Aqua Regeneration, Shinshu University, 4-17-1 Wakasato, Nagano 380-8553, Japan

^c Research Initiative for Supra-Materials, Shinshu University, 4-17-1 Wakasato, Nagano 380-8553, Japan

^d Energy Landscape Architectonics Brain Bank, Shinshu University, 4-17-1 Wakasato, Nagano 380-8553, Japan



Fig. S1. PXRD patterns of LNMO substituted with Zr, Sn, and Hf.



Fig. S2. *Operando* XRD data of LNMO (a) and STG-LNMO (b) electrodes-based coincell upon electrochemical charging and discharging process alongside corresponding electrochemical curves.



Fig. S3. Temperature–composition diagram of LNMO and miscibility gap (grey region) and solid solution domain (red and blue region) at room temperature.



Fig. S4. The evolution of the lattice constant in LNMO, S-LNMO, ST-LNMO, and STG-LNMO as a function of SOC, determined from the 511 reflection in *operando* XRD measurements.



Fig. S5. Supercell structure of ordered LNMO used in simulation and the Wyckoff positions of all atoms. Li and O atoms are presented in green and red spheres, respectively. While Ni and Mn are displayed by magenta and silver polyhedrons, respectively.



Fig. S6. Calculated XRD profiles of LNMO with 2θ ranging from 56 to 61 degrees (511 reflection) as a function of delithiation level *x* for various Li deintercalation scenarios: (a) scenario 1, (b) scenario 2, (c) scenario 3, and (d) scenarios 4, respectively.



Fig. S7. *Ex-situ* XANES measurements of Mn K edge to SOC-controlled (a) LNMO and (b) STG-LNMO electrodes.



Fig. S8. Evolution of oxidation states of Ni upon electrochemical charging process of LNMO and STG-LNMO.



Fig. S9. Evolution of bond lengths of (a) Mn and (b) Ni to their second coordination shells, and (c) the evolution of bond length of closest Mn–next neighbor shells (Mn–O) upon electrochemical charging process of LNMO and STG-LNMO.



Fig. S10. Chemical structure of 1H,1H,2H,2H-perfluorooctyltriethoxysilane, $F_3C(CF_2)_5(CH2)_2Si(OCH_3)_3$.



Fig. S11. Cycling performance tests of LNMO, S-LNMO, ST-LNMO and STG-LNMO based half-cells at 0.5C charge and 1C discharge, with operating voltage range of 3.5-4.9 V (vs. Li/Li⁺), recorded at room temperature.

Table S1. Stoichiometry of the LNMO and STG-LNMO compounds determinedby ICP-OES.

	Li	Ni	Mn	Si	Ti	Ge
LNMO	0.939	0.500	1.528			
STG-LNMO	0.990	0.500	1.390	0.036	0.034	0.026

Table S2. Refined crystallographic parameters of LNMO derivatives based on XRD results.

LNMO

Rp=2.25% Rpb=35.68% R=3.09% Rwp=2.78% Rexp=2.89% Durbin-Watson d=1.59 1-rho=1.88%

 Wyckoff=8c
 x=0.0110
 y=0.0110
 z=0.0110
 TDS=0.0100

 Wyckoff=12d
 x=0.1250
 y=0.3765
 z=-0.1265
 TDS=0.0032

 Wyckoff=4b
 x=0.6250
 y=0.6250
 z=0.6250
 TDS=0.0068

 Wyckoff=8c
 x=0.3848
 y=0.3848
 z=0.3848
 TDS=0.0099

 Wyckoff=24e
 x=0.1498
 y=0.8575
 z=0.1212
 TDS=0.0099

A=0.8171258 k1=0 k2=3.08e-06 B1=0.002561 GEWICHT=SPHAR4

S-LNMO

Rp=2.29% Rpb=36.56% R=3.01% Rwp=2.85% Rexp=2.88% Durbin-Watson d=1.53 1-rho=2.08%

 Wyckoff=8c
 x=0.0110
 y=0.0110
 z=0.0110
 TDS=0.0100

 Wyckoff=12d
 x=0.1250
 y=0.3746
 z=-0.1246
 TDS=0.0023

 Wyckoff=4b
 x=0.6250
 y=0.6250
 z=0.6250
 TDS=0.0074

 Wyckoff=24e
 x=0.3821
 y=0.3821
 z=0.3821
 TDS=0.0144

 Wyckoff=24e
 x=0.1227
 y=0.8541
 z=0.1437
 TDS=0.0144

A=0.8174585 k1=0 k2=8.3e-06 B1=0.00191 GEWICHT=SPHAR4

ST-LNMO

Rp=2.23% Rpb=34.30% R=2.87% Rwp=2.79% Rexp=2.90% Durbin-Watson d=1.63 1-rho=1.78%

Wyckoff=8c x=0.0110 y=0.0110 z=0.0110 TDS=0.0100 Wyckoff=12d x=0.1250 y=0.3769 z= -0.1269 TDS=0.0021 Wyckoff=4b x=0.6250 y=0.6250 z=0.6250 TDS=0.0073 Wyckoff=8c x=0.3834 y= 0.3834 z= 0.3834 TDS=0.0101 Wyckoff=24e x=0.1519 y=0.8603 z=0.1206 TDS=0.0101

A=0.8179748 k1=0 k2=1.015e-05 B1=0.00162 GEWICHT=SPHAR4

STG-LNMO

Rp=2.27% Rpb=34.87% R=3.05% Rwp=2.83% Rexp=2.93% Durbin-Watson d=1.56 1-rho=1.94%

 $\label{eq:wyckoff=8c x=0.0110 y=0.0110 z=0.0110 TDS=0.0106} \\ Wyckoff=12d x=0.1250 y=0.3747 z=-0.1247 TDS=0.029 \\ Wyckoff=4b x=0.6250 y=0.6250 z=0.6250 TDS=0.0068 \\ Wyckoff=2ke x=0.3906 y=0.3906 z=0.3906 TDS=0.0100 \\ Wyckoff=24e x=0.1205 y=0.8505 z=0.1348 TDS=0.0100 \\ \end{array}$

A=0.8176652 k1=0 k2=9.37e-06 B1=0.00178 GEWICHT=SPHAR4

Table S3. Substitutional energies (in eV) for replacing Ni or Mn with Si, Ge, or Ti in ordered LNMO. Lower substitutional energy indicates a stronger preference for the dopant to occupy a specific site.

Structure	ΔE_{DFT}		ΔE_{PFP}	
	With U	Without U	With U	Without U
LNMO_Si@Mn	0.701	0.722	0.663	0.710
LNMO_Si@Ni	2.006	3.223	1.764	2.841
LNMO_Ge@Mn	0.073	-0.129	0.048	0.100
LNMO_Ge@Ni	1.323	3.158	1.108	2.102
LNMO_Ti@Mn	0.517	0.367	0.520	0.521
LNMO_Ti@Ni	1.726	2.875	1.560	3.496

	C.N.	Bond length (Å)	lonic radii (Å)
Si-LNMO	8.26 +/- 2.34	1.60+/- 0.01	0.4
Ti-LNMO	6.14 +/- 1.66	1.91+/- 0.01	0.605
Ge-LNMO	5.51 +/- 0.98	1.81+/- 0.01	0.56

Table S4. Values for bonding distances and coordination numbers (C.N.) obtainedfrom EXAFS fitting.

Table S5. The peak intensity ratio of I_{580}/I_{620} in FT-IR spectra taken from the LNMO derivatives. Disordered LNMO (*Fd-3m*) was synthesized by solid state synthesis. Stoichiometric mixture of starting materials (1.05 times excess to compensate of Li loss at high temperature) was placed in 30 cm³ crucibles. The mixed powders were directly calcined at 800 °C for 10 h in Muffle furnace under air atmosphere with heating rate of 700 °C·h-1 and cooled down to 500 °C with cooling rate of 200 °C·h⁻¹.

Sample	I ₅₈₀ /I ₆₂₀	Sample	I ₅₈₀ /I ₆₂₀
LNMO(<i>P</i> 4 ₃ 32)	1.08	S-LNMO	1.04
LNMO(<i>F</i> d-3m)	0.86	S,T-LNMO	1.01
		STG-LNMO	1.06

Sample	Discharge capacity (mAh·g-1)
LNMO	130.0
S-LNMO	119.6
ST-LNMO	131.3
STG-LNMO	124.2

Table S6. Typical discharge capacity of LNMO derivatives-based half-cells.

The theoretical specific capacities for LNMO, S-LNMO, ST-LNMO, and STG-LNMO are 146.7 mAh/g, 148.9 mAh/g, 148.1 mAh/g, and 147.1 mAh/g, respectively. Since ST-LNMO exhibited a higher theoretical specific capacity than STG-LNMO, this could partly explain the higher discharge capacity observed for ST-LNMO. Therefore, while multi-element substitution influences the system's configurational entropy, it does not necessarily improve discharge capacity in a straightforward manner.

sample	Retention rate (%)		
LNMO	92.0		
S-LNMO	96.3		
ST-LNMO	96.2		
STG-LNMO	99.9		

Table S7. Discharge capacity retention rates of LNMO, S-LNMO, ST-LNMO andSTG-LNMO between the 10th and 100th cycles.