Supplementary Material

Proton Migration Pathways and Hydrogen Evolution Mechanism in g-C$_3$N$_4$/TiO$_2$-B and Li-F Co-Doped Heterostructures From Theoretical Study

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Outline

1. Geometry models

Fig. S1 provides a comprehensive schematic overview of all relevant structural models. In principle, the g-C_3N_4/TiO_2-B(001) heterostructures can be fabricated in several ways, for example, stacking with different contact sites or with different orientations. However, the criterion of heterostructure construction is to have a small lattice mismatch because a favorable interface structure requires a small lattice mismatch, which can maximize the orbital overlap between separated components. Based on this requirement, several possible heterostructure configurations can be constructed. But, after lattice match screening, geometrical models of g-C_3N_4/TiO_2-B(001) heterostructures with the least lattice mismatch are shown in Fig. S1. g-C_3N_4/TiO_2-B(001) heterostructure, $A_1 \times A_2$ g-C_3N_4 supported by $B_1 \times B_2$ TiO_2-B(001) surface, is fabricated by artificial rotation, where $A_1 = 2a_1$, $A_2 = a_1 + a_2$, and $B_1 = b_1 + b_2$, $B_2 = b_1 - b_2$. Here, a_1, a_2 and b_1, b_2 are lattice vectors of clean g-C_3N_4 and TiO_2-B(001), respectively. To examine the compatibility of the interface between g-C_3N_4 and TiO_2-B(001) surfaces, lattice mismatch is calculated. The lattice mismatch is defined as $\delta = (b_0 - a_0)/a_0$, where b_0 is the lattice constant of TiO_2-B(001) surface (i.e., $b_{0,1} = 3.76Å$ and $b_{0,2} = 12.77Å$), and a_0 is the lattice constant of g-C_3N_4 (i.e., $a_0 = 7.17Å$). A positive value indicates that the g-C_3N_4 monolayer is stretched, and a negative value indicates that the g-C_3N_4 monolayer is compressed. Our calculated δ values for lattice constant a_0 and b_0 of the g-C_3N_4/TiO_2-B(001) heterostructure are -0.14% and -1.76%, respectively, indicating that the g-C_3N_4 monolayer is under compressive strain. Note that the lattice vector angle of TiO_2-B(001) is 31.61°, which is slightly larger than that of g-C_3N_4. We adopt a 30° angle between the lattice vectors in order to facilitate the construction of the heterojunction structure. From our calculations, we can see that the lattice mismatch in current calculations is acceptable.

The top- and side-view of g-C_3N_4/TiO_2-B(001) and Li-F co-doped g-C_3N_4/TiO_2-B(001) heterostructures are shown in Fig. S1(c) and (d). To facilitate the presentation of the surface interface structure of the heterojunction, we have re-cut the supercell unit (black dash box), as shown in Fig. S1(c) and (d).
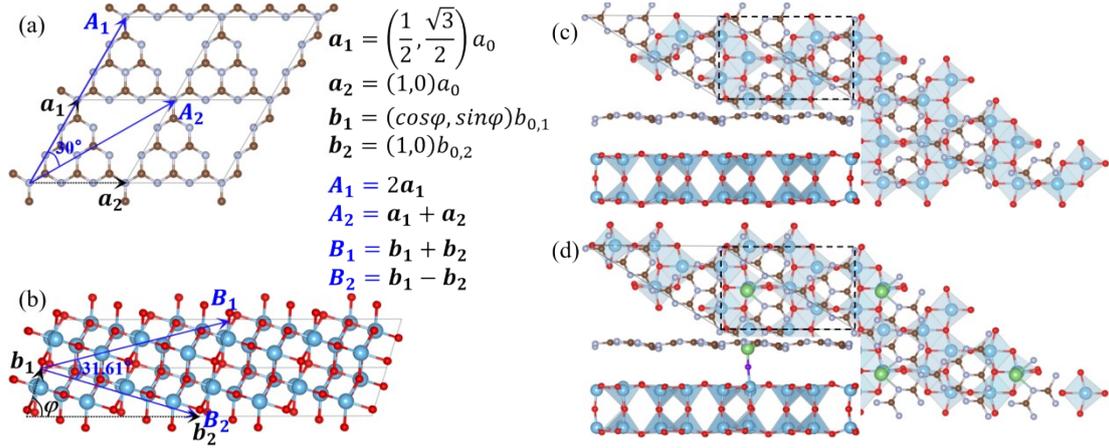
Fig. S1. Schematic diagram of supercell construction for g-C$_3$N$_4$/TiO$_2$-B(001) heterostructure. (a) Geometry and unit cell of g-C$_3$N$_4$. (b) Geometry and unit cell of TiO$_2$-B(001) surface. (c)-(d) Top- and side-view of g-C$_3$N$_4$/TiO$_2$-B(001) and Li-F co-doped heterostructures. Note that the black dash boxes are supercells used in the main body text. Gray, brown, blue, red, green, and purple spheres represent N, C, Ti, O, Li, and F atoms, respectively. The lattice constant of primitive cell g-C$_3$N$_4$ is a$_0$ =7.17Å and the lattice constants of TiO$_2$-B(001) are b$_{0,1}$ =3.76Å and b$_{0,2}$ =12.77Å.

2. Geometries of water on various TiO$_2$-B(001) surfaces

Based on the principle of matching the structure of adsorbates with the substrate, the molecular adsorption and dissociative adsorption configurations of water molecules on different TiO$_2$-B surfaces were constructed. Fig. S2 illustrates the most stable adsorption and dissociation configurations of water on different TiO$_2$-B(001) surfaces. As one can see from Fig. S2, the Ti_{5c}^{*}(Ti13) site exhibits the lowest adsorption energy for water on the pristine and Li-F co-doped TiO$_2$-B(001)/g-C$_3$N$_4$ heterostructures. Consequently, the dissociation adsorption energy of water molecules was optimized and calculated using Ti13 site as the reference configuration in the main text.
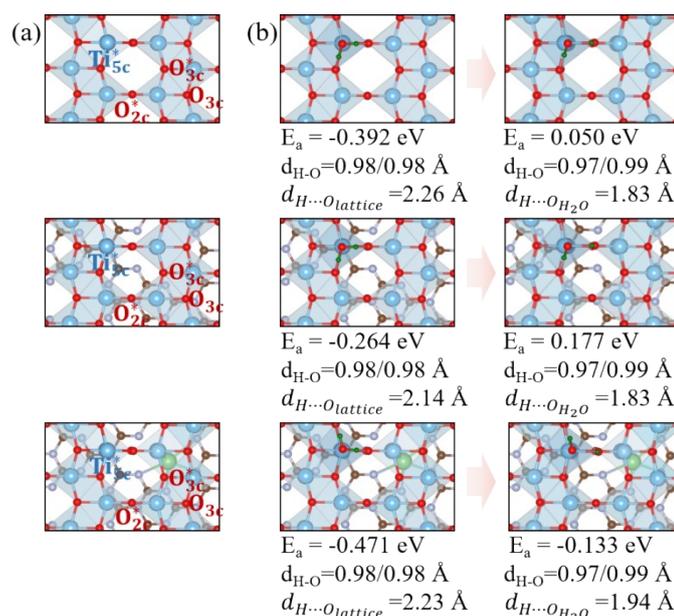

Fig. S2. (a) Surface structural features of TiO$_2$-B(001), TiO$_2$-B(001)/g-C$_3$N$_4$ and Li-F co-doped TiO$_2$-B(001)/g-C$_3$N$_4$. (b) The most stable configurations for molecule adsorption and dissociated adsorption of water on various TiO$_2$-B(001) surfaces. Ti_{5c}^{*} is the primary active site for water adsorption. Adsorption energies and structural parameters are also given below each geometry.

3. Geometries of proton on TiO$_2$-B(001) surfaces
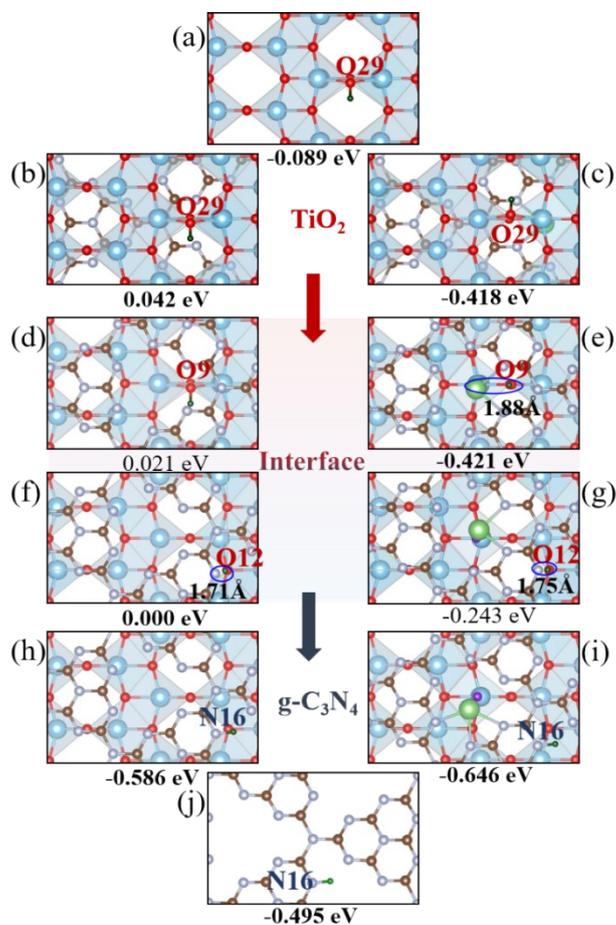
Fig. S3. (a)-(c) The most stable adsorption configurations of proton on O_{2c}^* site (labeled as O29) on different TiO$_2$-B(001) surfaces. (d)-(g) The most stable adsorption configuration candidates of proton on O_{2c}^* site (labeled as O9) or O_{3c} site (labeled as O12) at the interfaces. (h)-(j) The most stable adsorption configurations of proton on pyridine nitrogen (labeled as N16) on g-C$_3$N$_4$ surface. Adsorption energies are also given below each geometry.
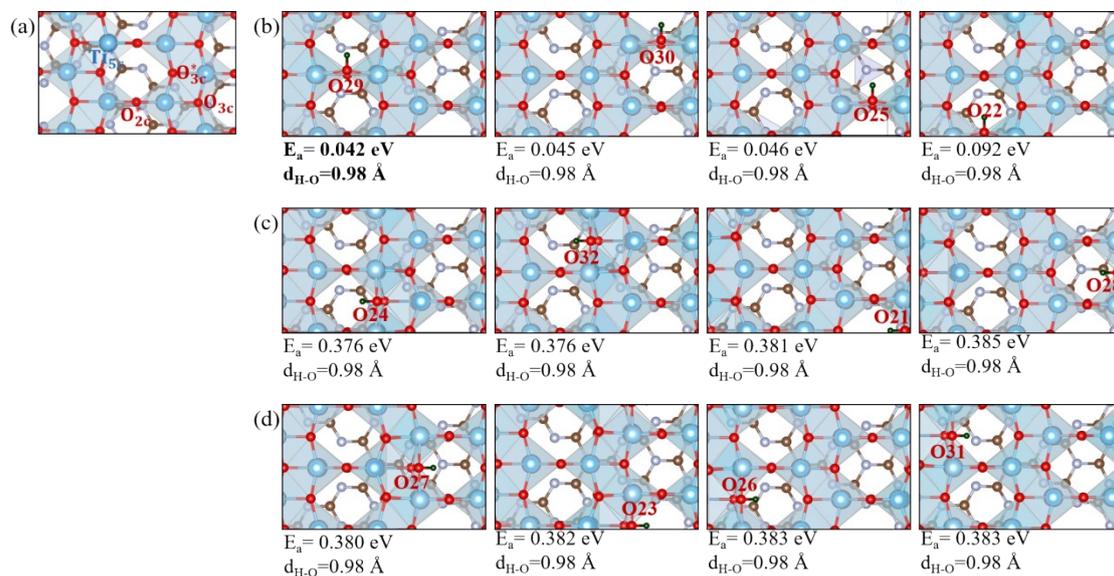
Fig. S4. (a) Some possible adsorption sites of proton on surface of TiO_2-B(001)/g-C_3N_4 heterostructure, i.e., Ti_{5c}^*, O_{2c}^*, O_{3c}^*, and O_{3c} site. (b) Geometries of proton on O_{2c}^* site, i.e., O29, O30, O25, and O22. (c) Geometries of proton on O_{3c} site, i.e., O24, O32, O21, and O28. (d) Geometries of proton on O_{3c}^* site, i.e., O27, O23, O26, and O31. Adsorption energy and bond length are given below each geometry.

4. Geometries of proton on interface of TiO$_2$-B(001) heterostructure

(a)

(b)

E_a = 0.000 eV
d_{H-O} = 1.03 Å
$d_{H\cdots N16}$ = 1.71 Å

E_a = -0.006 eV
d_{H-O} = 1.02 Å
$d_{H\cdots N8}$ = 1.74 Å

E_a = -0.005 eV
d_{H-O} = 1.02 Å
$d_{H\cdots N14}$ = 1.74 Å

(c)

E_a = 0.010 eV
d_{H-O} = 0.98 Å

E_a = 0.021 eV
d_{H-O} = 0.98 Å

E_a = 0.033 eV
d_{H-O} = 0.98 Å

(d)

E_a = 0.142 eV
d_{H-O} = 1.00 Å
$d_{H\cdots N12}$ = 2.04 Å

E_a = 0.196 eV
d_{H-O} = 0.98 Å

E_a = 0.444 eV
d_{H-O} = 0.98 Å
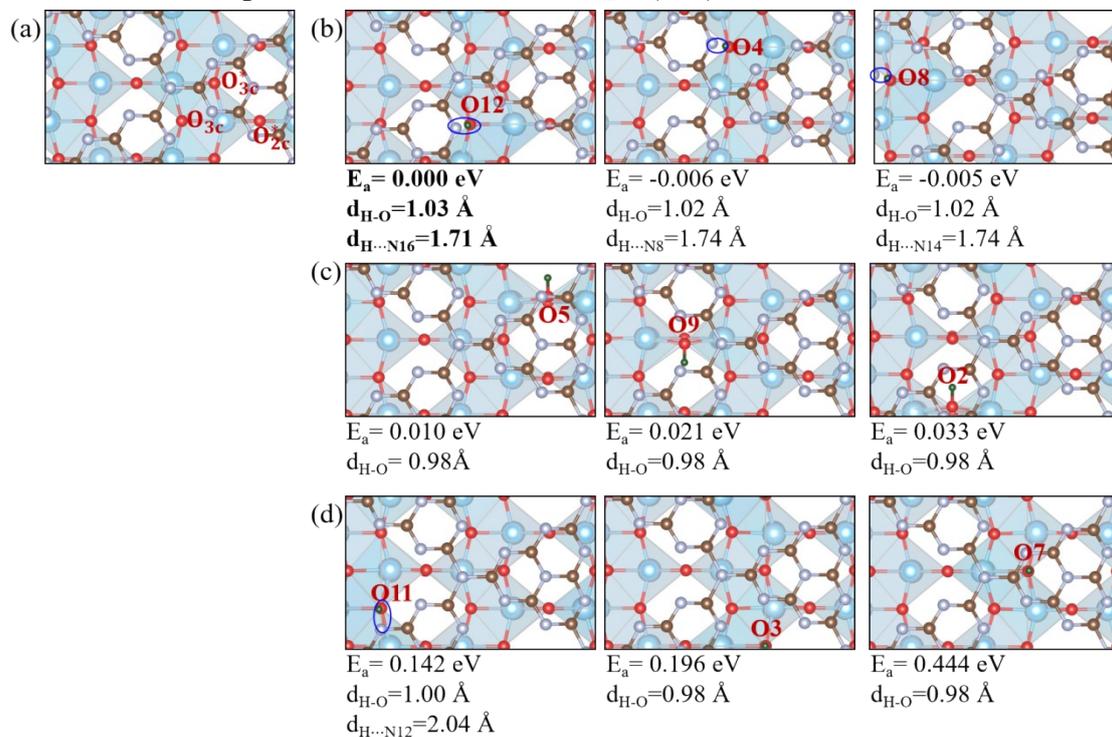
Fig. S5. (a) Some possible adsorption sites of proton at interface of TiO$_2$-B(001)/g-C$_3$N$_4$ heterostructure, i.e., O_{2c}^*, O_{3c}^*, and O_{3c} site. (b) Geometries of proton on O_{3c} site, i.e., O12, O4, and O8. (c) Geometries of proton on O_{2c}^* site, i.e., O5, O9, and O2. (d) Geometries of proton on O_{3c}^* site, i.e., O11, O3, and O7. Adsorption energy and bond length are given below each geometry.

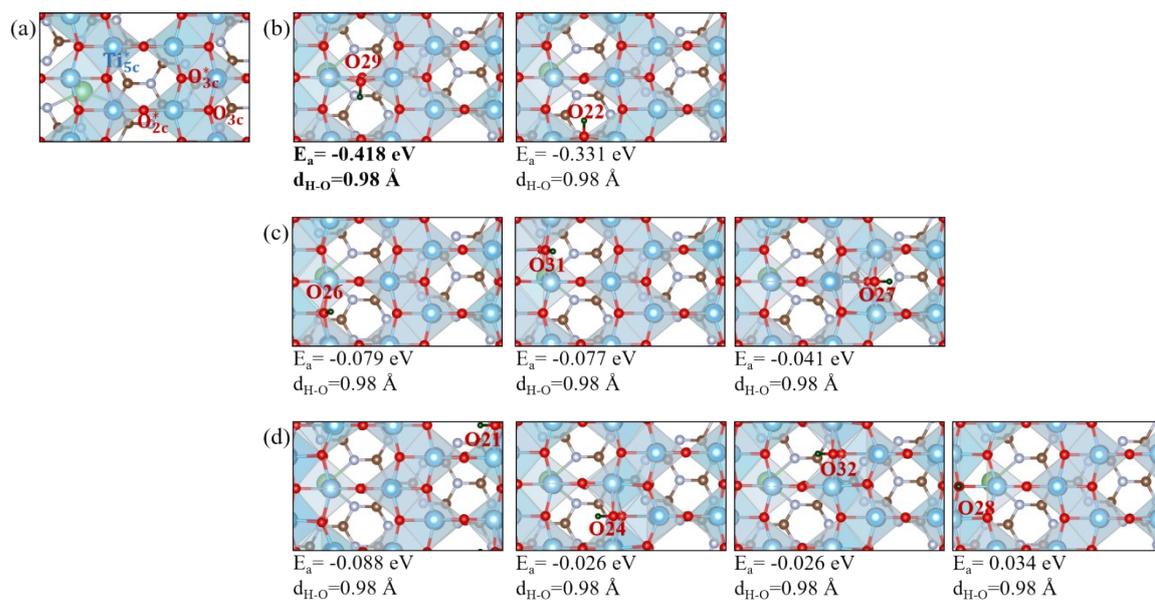## 5. Geometries of proton on TiO$_2$-B surface in Li-F co-doped heterostructure

(a)

(b)

O29

E_a= -0.418 eV
d_{H-O}=0.98 Å

O22

E_a= -0.331 eV
d_{H-O}=0.98 Å

(c)

O26

E_a= -0.079 eV
d_{H-O}=0.98 Å

O31

E_a= -0.077 eV
d_{H-O}=0.98 Å

O27

E_a= -0.041 eV
d_{H-O}=0.98 Å

(d)

O21

E_a= -0.088 eV
d_{H-O}=0.98 Å

O24

E_a= -0.026 eV
d_{H-O}=0.98 Å

O32

E_a= -0.026 eV
d_{H-O}=0.98 Å

O28

E_a= 0.034 eV
d_{H-O}=0.98 Å

Fig. S6. (a) Some possible adsorption configurations of protons on the surface of Li-F co-doped TiO$_2$-B(001)/g-C$_3$N$_4$ heterostructures, i.e., Ti_{5c}^*, O_{2c}^*, O_{3c}^*, and O_{3c} site. (b) Geometries of proton on O_{2c}^* site, i.e., O29 and O22. (c) Geometries of proton on O_{3c}^* site, i.e., O26, O31, and O27. (d) Geometries of proton on O_{3c} site, i.e., O21, O2, O32, and O28. Adsorption energy and bond length are given below each geometry.

6. Geometries of proton at the interface of Li-F co-doped heterostructure
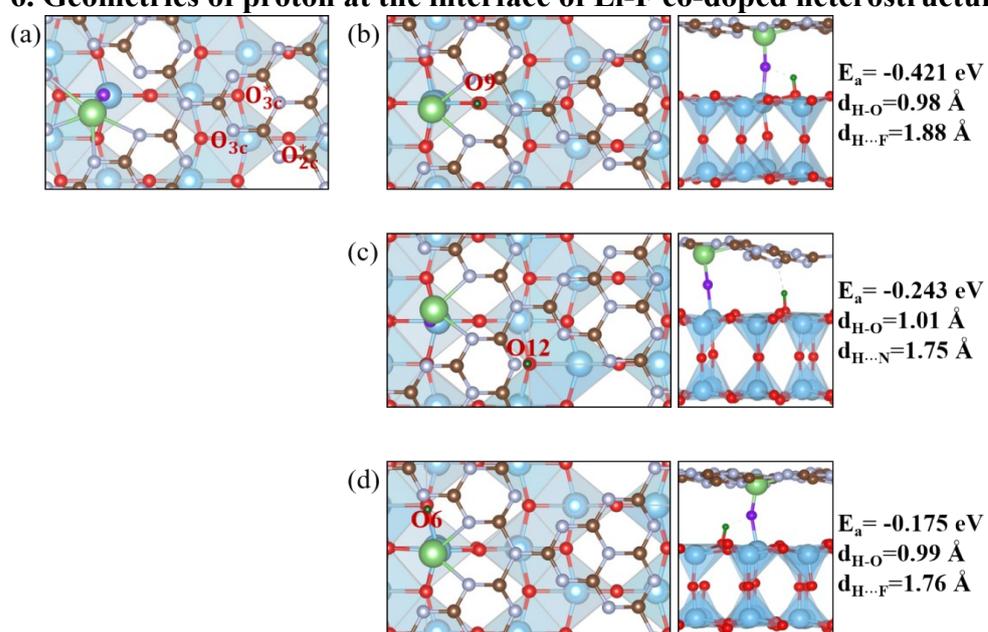
Fig. S7. (a) Some possible adsorption sites of protons at the interface of Li-F co-doped TiO_2-B(001)/g-C_3N_4 heterostructure, i.e., O_{2c}^*, O_{3c}^*, and O_{3c} site. (b) Geometries of proton on O_{2c}^* site, i.e., O9. (c) Geometries of proton on O_{3c} site, i.e., O12. (d) Geometries of proton on O_{3c}^* site, i.e., O6. Adsorption energy and bond length are given below each geometry.

7. Geometries of proton on different g-C$_3$N$_4$ surfaces
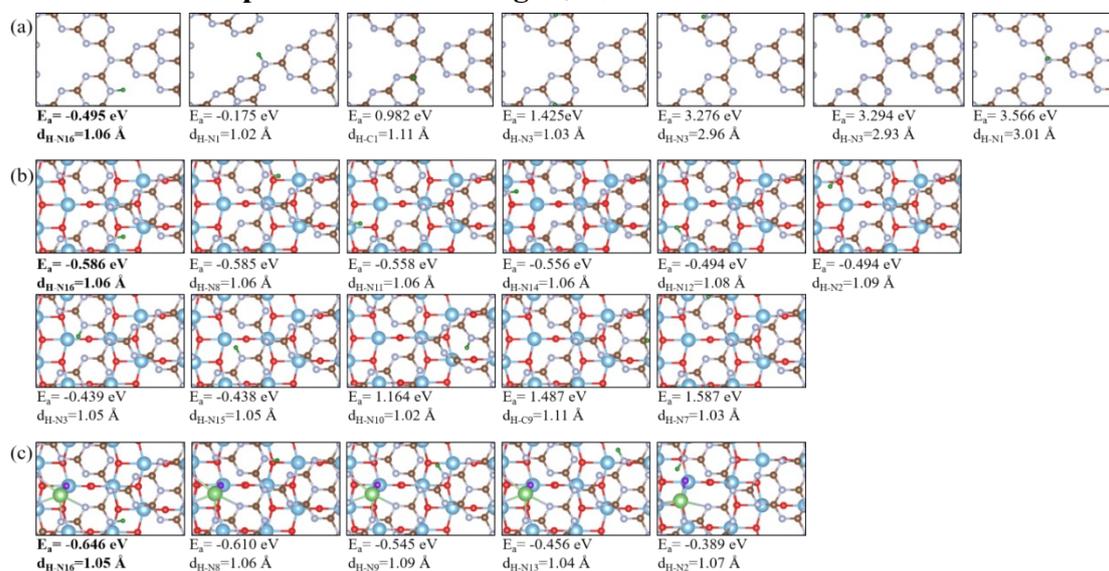
(a)

E_a= -0.495 eV	E_a= -0.175 eV	E_a= 0.982 eV	E_a= 1.425eV	E_a= 3.276 eV	E_a= 3.294 eV	E_a= 3.566 eV
d_{H-N16}=1.06 Å	d_{H-N1}=1.02 Å	d_{H-C1}=1.11 Å	d_{H-N3}=1.03 Å	d_{H-N3}=2.96 Å	d_{H-N3}=2.93 Å	d_{H-N1}=3.01 Å

(b)

E_a= -0.586 eV	E_a= -0.585 eV	E_a= -0.558 eV	E_a= -0.556 eV	E_a= -0.494 eV	E_a= -0.494 eV
d_{H-N16}=1.06 Å	d_{H-N8}=1.06 Å	d_{H-N11}=1.06 Å	d_{H-N14}=1.06 Å	d_{H-N12}=1.08 Å	d_{H-N2}=1.09 Å

E_a= -0.439 eV	E_a= -0.438 eV	E_a= 1.164 eV	E_a= 1.487 eV	E_a= 1.587 eV
d_{H-N3}=1.05 Å	d_{H-N15}=1.05 Å	d_{H-N10}=1.02 Å	d_{H-C9}=1.11 Å	d_{H-N7}=1.03 Å

(c)

E_a= -0.646 eV	E_a= -0.610 eV	E_a= -0.545 eV	E_a= -0.456 eV	E_a= -0.389 eV
d_{H-N16}=1.05 Å	d_{H-N8}=1.06 Å	d_{H-N9}=1.09 Å	d_{H-N13}=1.04 Å	d_{H-N2}=1.07 Å

Fig. S8. Some possible adsorption configurations of proton on various g-C$_3$N$_4$ surfaces. (a) g-C$_3$N$_4$ surfaces. (b) g-C$_3$N$_4$/TiO$_2$-B(001) surface. (c) Li-F co-doped g-C$_3$N$_4$/TiO$_2$-B(001) surface. Pyridine N (labeled as N$_2$) is the primary adsorption site for proton. The typical pyridine nitrogen is at the N16 site. Adsorption energy and bond length are given below each geometry.

8. Confirmation of rate-determining steps

Table S1. Energy barriers for proton migration across the interface or interface diffusion in pristine TiO_2-B/g-C_3N_4 and Li-F co-doped TiO_2-B/g-C_3N_4 heterostructures (at U = 0 V vs. RHE, pH = 0).

Step		Pristine (eV)	Li-F co-doped (eV)
Water dissociation		0.442	0.338
Proton migration	TiO_2-B→interface	1.103	0.999
	Interface diffusion	–	0.701
	Interface→g-C_3N_4	0.168	0.226
Volmer		–	–
Heyrovsky		0.58	0.65

9. Comparison of migration barrier for proton

Table S2. Proton migration/transfer barriers in various systems.

System	Proton migration step	Energy barrier (eV)	Reference
g-C_3N_4/TiO_2-B(001)	TiO_2-B surface → interface	1.103	Current work
Li-F codoped g-C_3N_4/TiO_2-B(001)	TiO_2-B surface → interface	0.999	Current work
Pt10/TiO_2 (rutile)	Pt_{10}/TiO_2 interface	0.30	Ref. 1
$BaZrO_3$	interstitial of ZrO_6 octahedron	0.19~0.30	Ref. 2
$BaFeO_3$	interstitial of FeO_6 octahedron	0.22	Ref. 3
$CaMnO_3$	surface→bulk surface→bulk	0.66 1.57	Ref. 4

10. References

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