

Supplementary Information

Investigating Ti-Nb alloys as supports for iridium oxide water oxidation electrocatalysts

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Electrochemistry workflow

A rotating disc electrode (RDE) was used in this study to examine the activities of Ti-Nb supported IrO_x. All experiments were carried out using 0.1 M HClO₄ as the electrolyte. Electrochemical workflow: Chronoamperometry (CA) was firstly carried out by holding the potential at 1.6 V_{RHE} for 1 hour. The current was recorded after being stabilised for activity comparison. The accelerated stress test (AST) was conducted subsequently by cycling in a potential window from 1.2 V_{RHE} to 1.7 V_{RHE} for 3,000 cycles (as shown in **Figure S1**). The electrolyte was taken out twice before and after the AST for measuring the metal dissolution.

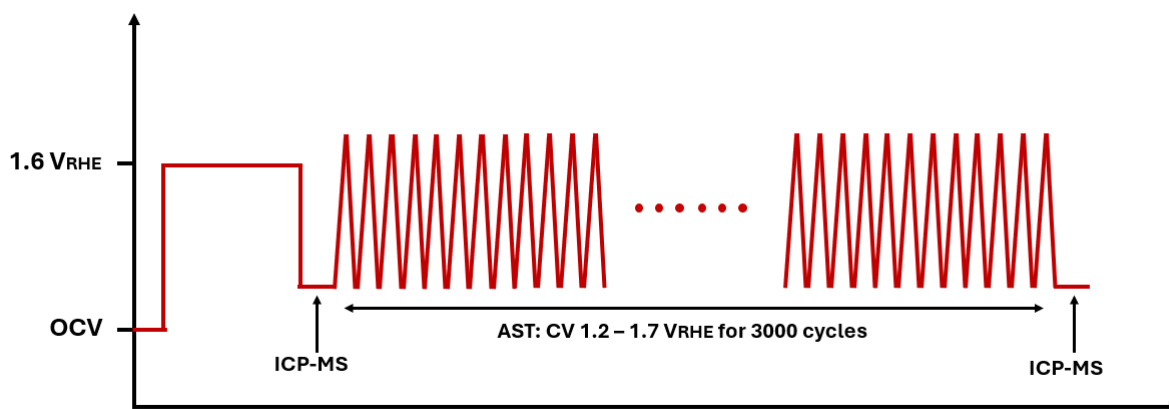


Figure S1. Demonstration of the electrochemistry workflow for the Ti-Nb samples.

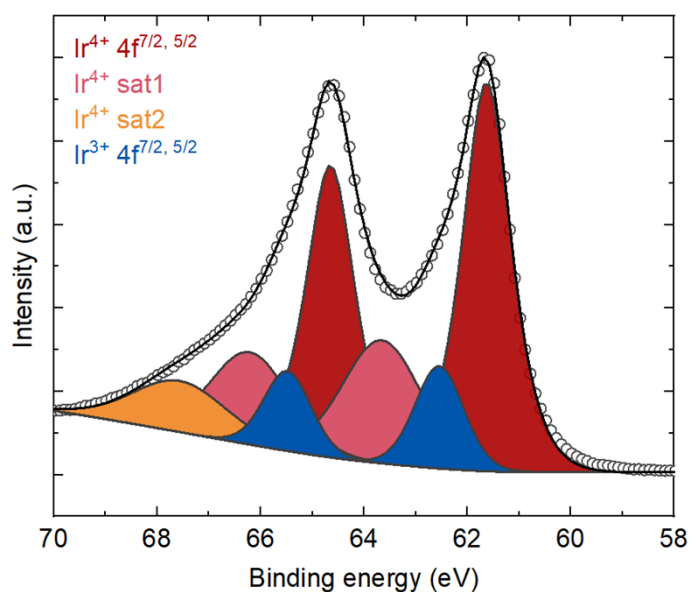


Figure S2. Ir 4f XPS spectrum with fitted components for different Ir oxidation states. The ratio between Ir⁴⁺ to Ir³⁺ is approximately 79% to 21%.

100 nm thick Ti-Nb thin films with varied Nb content were sputtered onto glassy carbon stubs shown in **Figure S3a** using physical vapour deposition under Ar (**Figure S3b**). **Table S1** summarises the sputter conditions.

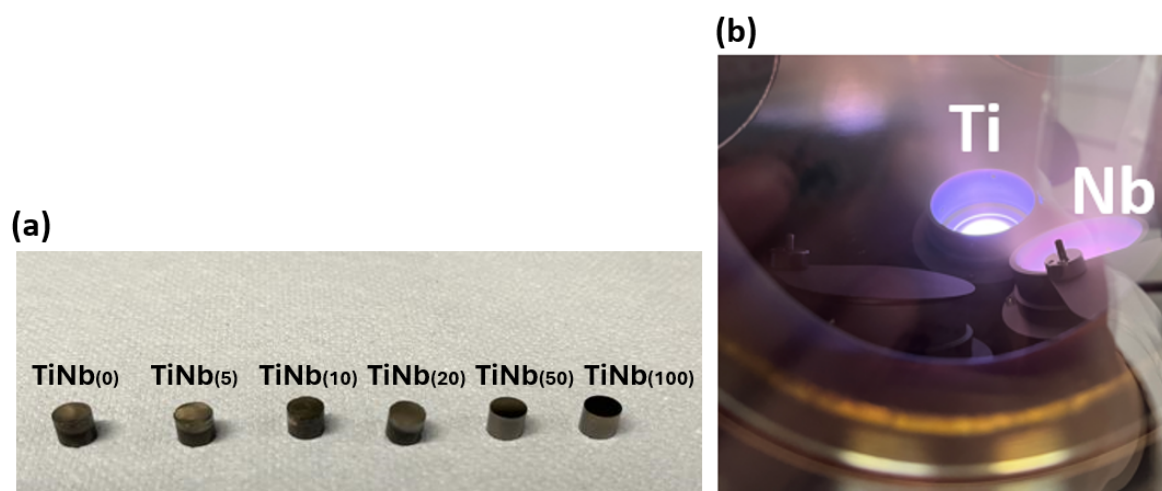


Figure S3. (a) Photograph of a series of TiNb_(x) (x = 0, 5, 10, 20, 50 and 100) samples on glassy carbon substrate (RDE stub) synthesised using a sputter deposition method. (b) Photograph of the co-sputtering of the Ti-Nb samples in an ultra-high vacuum chamber.

Table S1. Nb content of the as-sputtered samples obtained by XPS and the parameters used for the sputter deposition (DC: direct current; RF: radio frequency)

	Nb (at%) (determined by XPS)	Ti growth rate (Å/s)	Nb growth rate (Å/s)	Coating time (s)
TiNb(0)	-	0.77 (150 W DC)	-	1299
TiNb(5)	3.2	0.77 (150 W DC)	0.045 (20 W RF)	1227

TiNb(10)	10.1	0.77 (150 W DC)	0.088 (30 W RF)	1166
TiNb(20)	15.3	0.77 (150 W DC)	0.19 (45 W RF)	1042
TiNb(50)	52.7	0.49 (100 W DC)	0.49 (100 W RF)	1020.5
TiNb(100)	-	-	0.49 (100 W RF)	2041

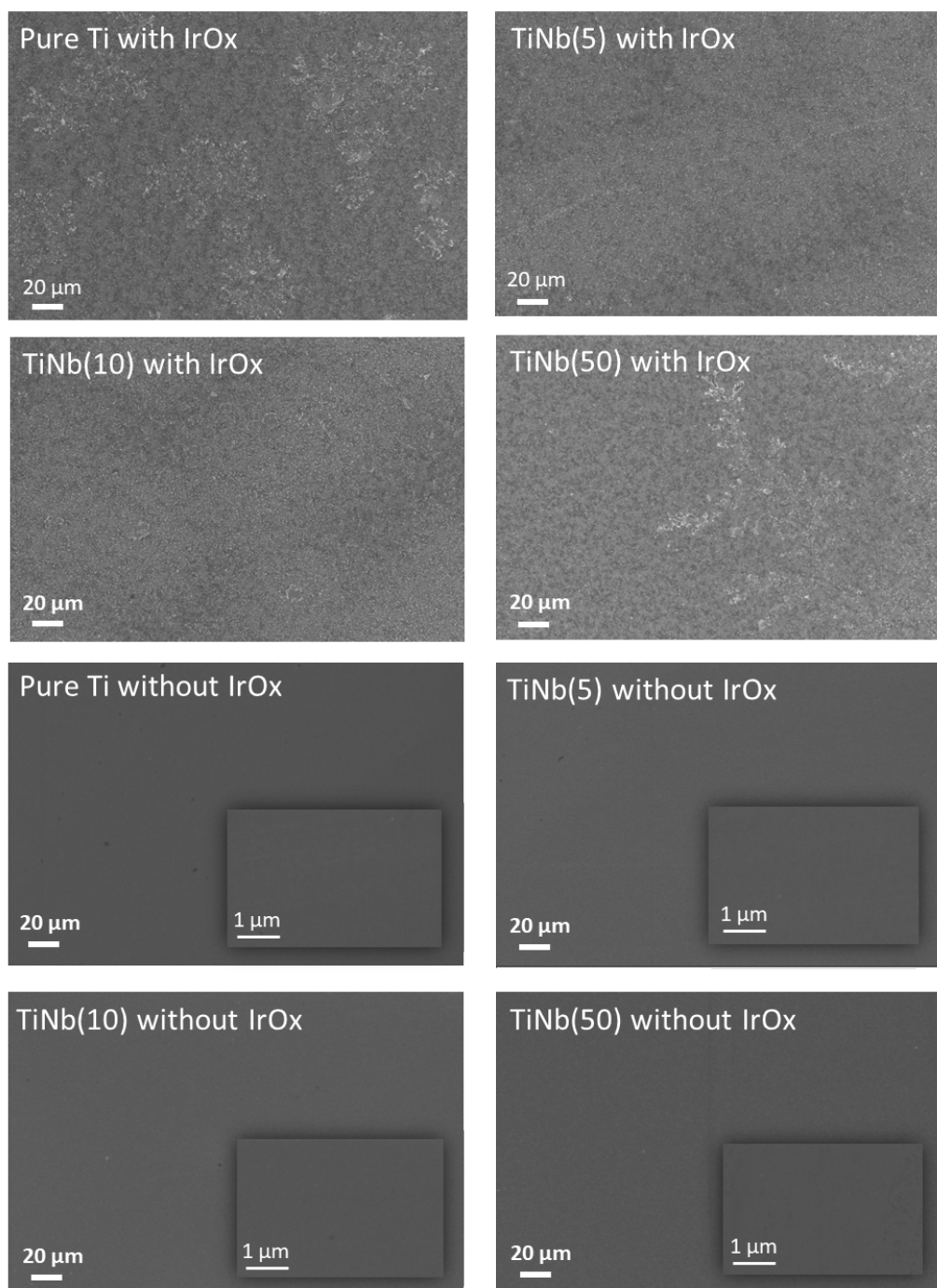


Figure S4. SEM images at 1,000 and 50,000 times magnification showing the TiNb_(x) (x = 0, 5, 10 and 50) supports before and after deposition of the IrO_x-based ink using drop casting.

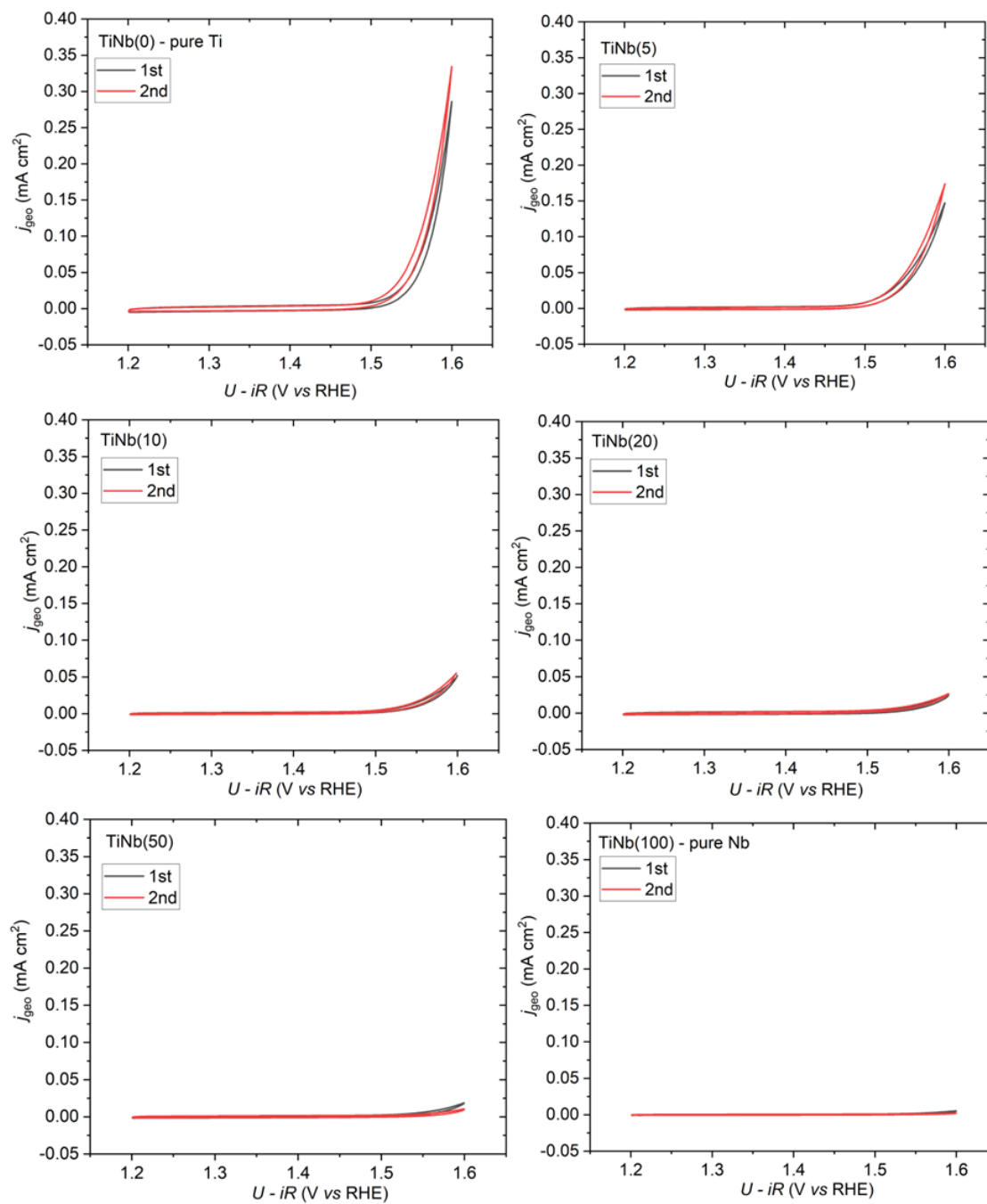


Figure S5. Cyclic Voltammograms comparing the electrochemical water oxidation activity of two independently synthesised samples that contain IrO_x supported on $\text{TiNb}_{(x)}$ ($x = 0, 5, 10, 20, 50$ and 100) on glassy carbon ($1.2 - 1.6 \text{ V}_{\text{RHE}}$, 20 mV/s , 0.1 M HClO_4 , 1600 rpm).

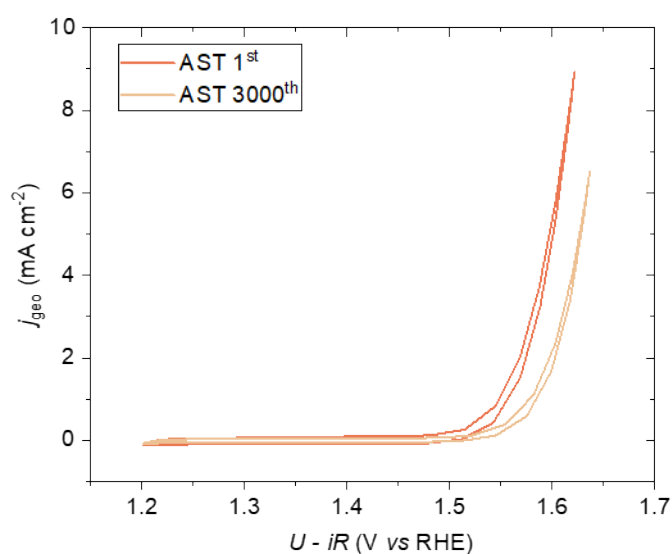


Figure S6. Initial activity of IrO_x supported on a polycrystalline gold disk and the activity after the AST, where the sample was cycled between 1.2 and 1.7 V_{RHE} at 50 mV/s in 0.1 M HClO₄ (pH ~1.1).

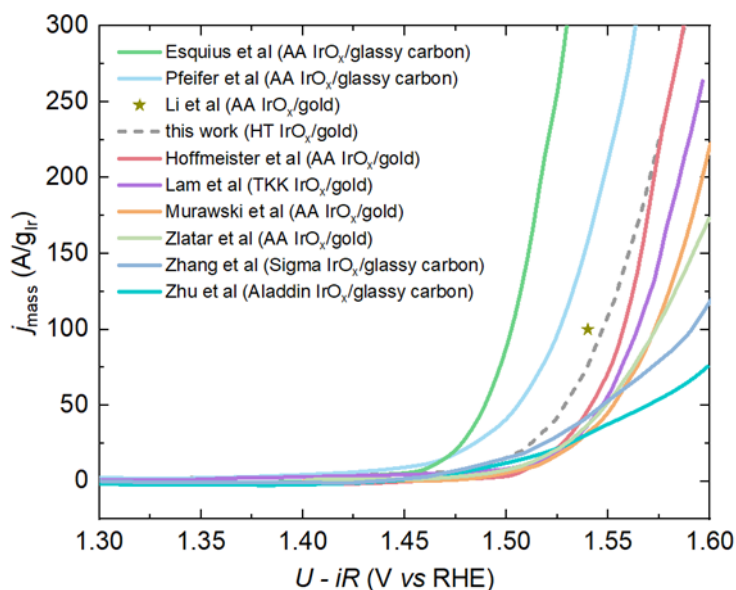


Figure S7. Comparison of the mass-normalised electrochemical water oxidation activity between this work (dotted line) and published literature (coloured lines).¹⁻⁹

Table S2. Experimental conditions of reported IrO_x activities shown in **Figure S7**.

References	Substrate	Electrolyte	Type of IrO _x
Esquius et al, <i>ACS Appl. Energy Mater.</i> , 2020 ⁶	Glassy carbon	0.1 M HClO ₄	Alfa Aesar (AA) IrO _x
Pfeifer et al, <i>Phys. Chem. Chem. Phys.</i> , 2016 ⁷	Glassy carbon	0.5 M H ₂ SO ₄	Alfa Aesar (AA) IrO _x
Li et al. <i>J. Am. Chem. Soc.</i> 2025 ²	Polycrystalline gold	0.1 M HClO ₄	Alfa Aesar (AA) IrO _x
This work	Polycrystalline gold	0.1 M HClO ₄	Hydrothermal IrO _x
Hoffmeister et al, <i>Adv. Sci.</i> ,	Polycrystalline gold	0.1 M HClO ₄	Alfa Aesar (AA) IrO _x

2024 ⁸			
Lam et al, <i>ACS Catal.</i> , 2025 ⁵	Polycrystalline gold	0.1 M HClO ₄	Tanaka Kikinzoku Kogyo (TKK) IrO _x
Murawski et al, <i>Johnson Matthey Technol. Rev.</i> , 2024 ³	Polycrystalline gold	0.1 M HClO ₄	Alfa Aesar (AA) IrO _x
Zlatar et al. <i>ACS Catal.</i> , 2023 ⁹	Polycrystalline gold & Glassy carbon	0.1 M HClO ₄	Alfa Aesar (AA) IrO _x
Zhang et al. <i>Angew. Chem. Int. Ed.</i> , 2025 ¹	Glassy carbon	0.5 M H ₂ SO ₄	Sigma Aldrich IrO ₂
Zhu et al, <i>Nat. Comm.</i> , 2023 ⁴	Glassy carbon	0.5 M H ₂ SO ₄	Aladdin IrO _x

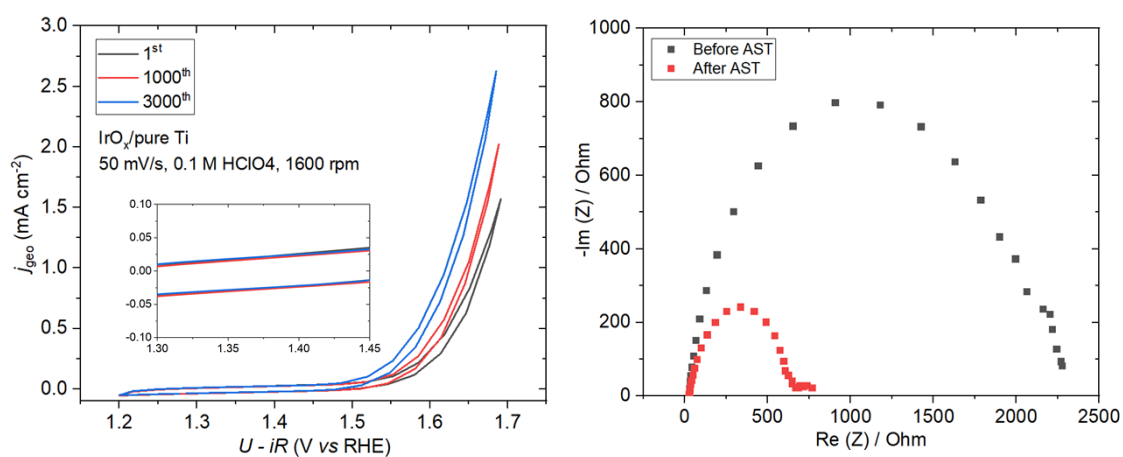


Figure S8. Cyclic voltammogram (**left**) of IrO_x supported onto TiNb(0) – pure Ti support during the AST. Cycle1, cycle 1000 and cycle 3000 are coplotted. EIS (**right**) before and after the AST.

Table S3. Calculation of Ti/Nb dissolution during AST (based on ICP-MS measurements)

NOTE: Surface area of RDE stub 0.196 cm²; Density of Ti metal: 4.5 g/cm³, 8.82 × 10⁻⁸ g/nm; Density of Nb metal: 8.6 g/cm³, 1.68 × 10⁻⁷ g/nm

Nb at%	Ti dissolved (μg)	Nb dissolved (μg)	Ti dissolved (nm)	Nb dissolved (nm)
0	0.0496	-	0.56	-
5	0.045	0.0135	0.50	0.08
10	0.032	0.016	0.36	0.095
100	-	0.057		0.34

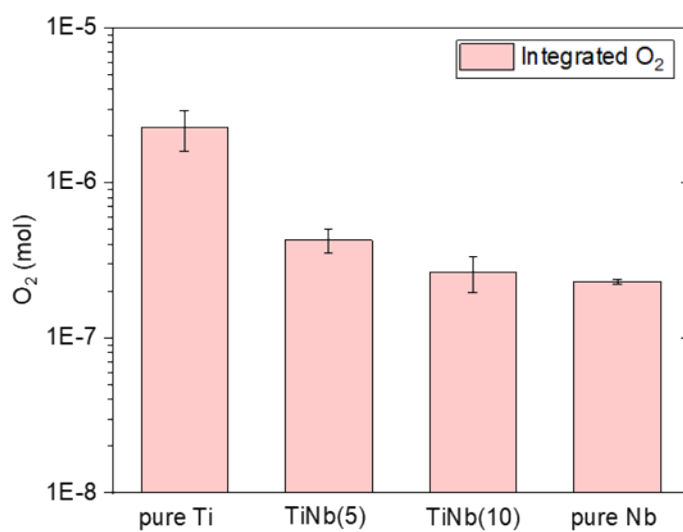
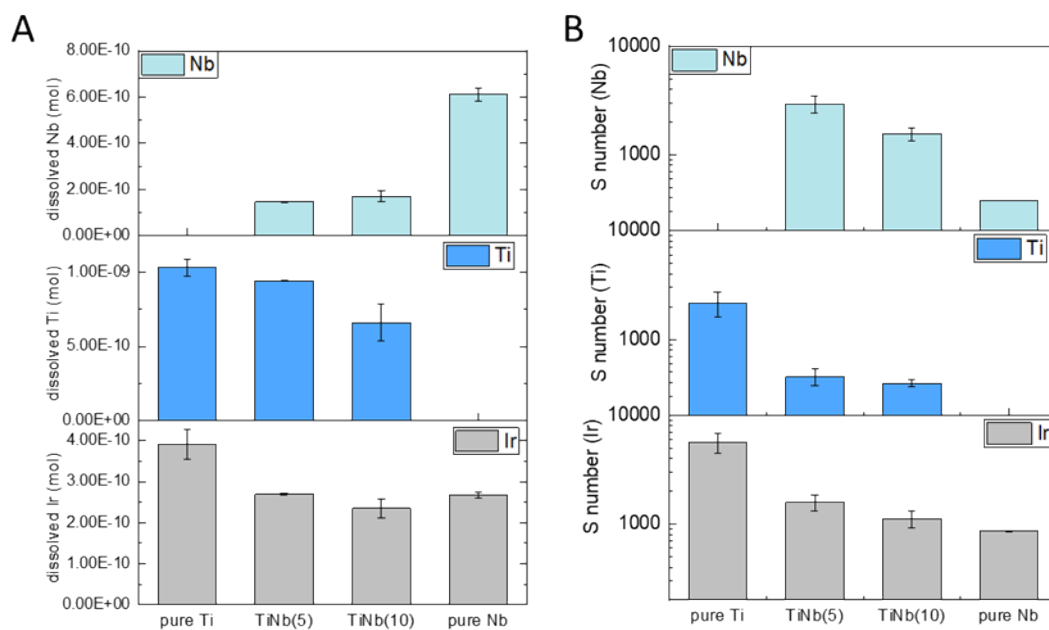


Figure S9. Integrated oxygen produced during the ASTs by integrated charges of CV scans assuming a 100% Faradaic efficiency.

Figure S10. (a) dissolved metals during the AST (RDE, 1.2 – 1.7 V_{RHE}, 100 mV/s, 3000 cycles, 0.1 M HClO₄, 1600 rpm). **(b)** S-numbers of dissolved metals (Ti, Nb and Ir) during the AST.



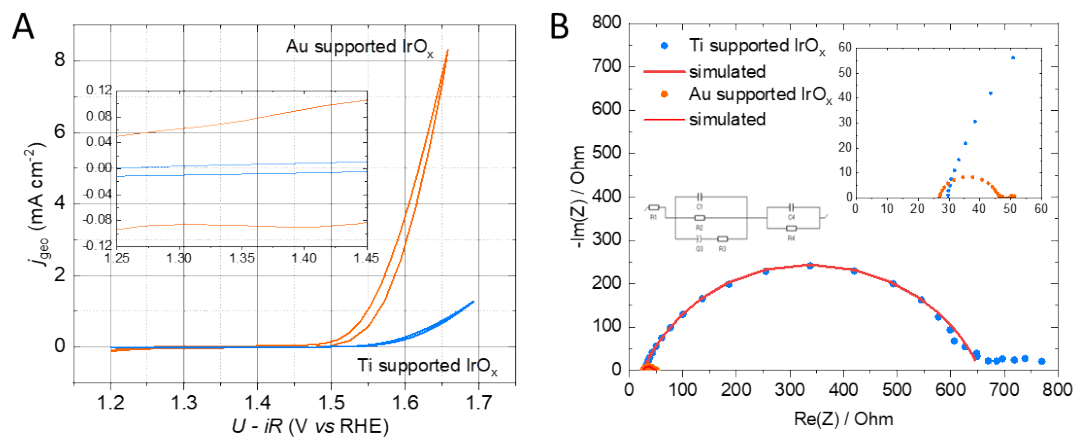


Figure S11. (a) CVs Ti supported and Au supported IrO_x (RDE, 20 mV/s, 0.1 M HClO₄, 1600 rpm). Inset is a close-up at the non-Faradaic region. **(b)** EIS of Ti supported and Au supported IrO_x at 1.6 V_{RHE}.

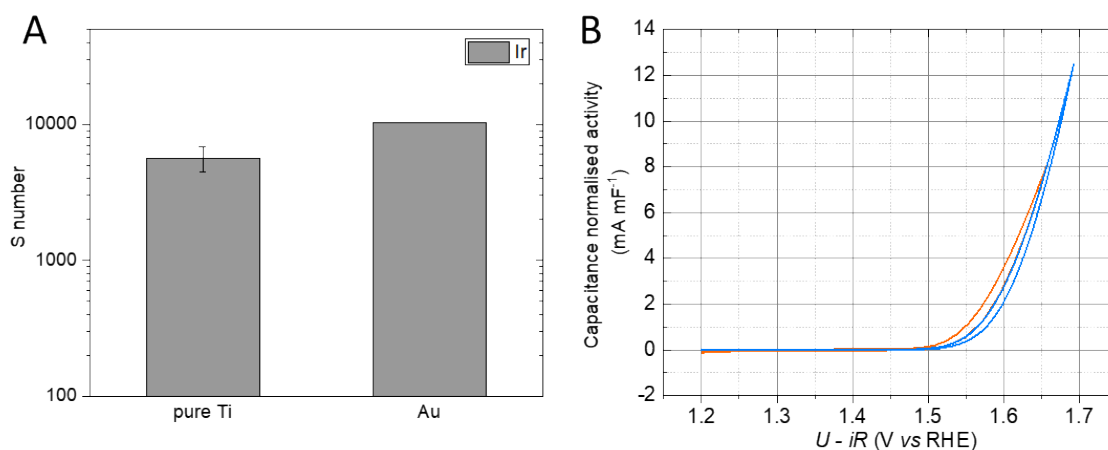


Figure S12. (a) S-numbers of Ti supported and Au supported IrO_x. S-numbers derived from the AST tests (RDE, 1.2 – 1.7 V_{RHE}, 100 mV/s, 3000 cycles, 0.1 M HClO₄, 1600 rpm). **(b)** Capacitance normalised activity (absolute current divided by double layer capacitance) of Ti supported and Au supported IrO_x.

Table S4. XPS fitting parameters (Ti 2p_{3/2} and 2p_{1/2}). FWHM: full width half maximum

Sample name	Line shape	Binding energy (eV)	FWHM
TiNb(0) Ti ⁰	A(0.4,0.4,0)GL(30)	(453.8, 454.1)	(0.9,1.8)
TiNb(0) Ti ⁴⁺	GL(30)	(459.5,459)	(0.9,1.8)
TiNb(50) Ti ⁰	A(0.4,0.4,0)GL(30)	(454.1,453.8)	(0.9,1.8)
TiNb(50) Ti ²⁺	GL(30)	(456,455.5)	(0.9,1.8)
TiNb(50) Ti ³⁺	GL(30)	(457.6,457.3)	(0.9,1.8)
TiNb(50) Ti ⁴⁺	GL(30)	(459,459.5)	(0.9,1.8)

Table S5. XPS fitting parameters (Nb 3d_{5/2} and 3d_{3/2}). FWHM: full width half maximum

Sample name	Line shape	Binding energy (eV)	FWHM
TiNb(0) Nb ⁰	A(0.4,0.4,0)GL(70)	(202.5, 202)	(0.6, 1.8)
TiNb(0) Nb ²⁺	GL(30)	(203.8, 203.2)	(0.6,1.8)

TiNb(O) Nb ⁴⁺	GL(30)	(206.6, 206)	(0.6,1.8)
TiNb(O) Nb ⁵⁺	GL(30)	(207.9, 207.3)	(0.6,1.8)
TiNb(O) Nb ⁰	A(0.4,0.4,0)GL(70)	(202.5, 202)	(0.6, 1.8)
TiNb(O) Nb ⁴⁺	GL(30)	(206.6, 206)	(0.6,1.8)
TiNb(O) Nb ⁵⁺	GL(30)	(207.9, 207.3)	(0.6,1.8)

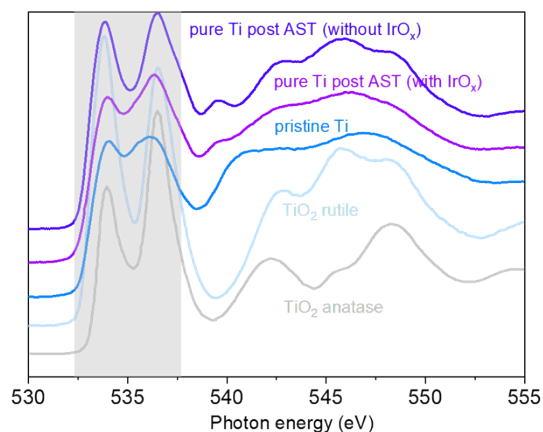


Figure S13. NEXAFS of the O K-edge of pure Ti support samples (pristine, post AST with IrO_x catalyst and without IrO_x).

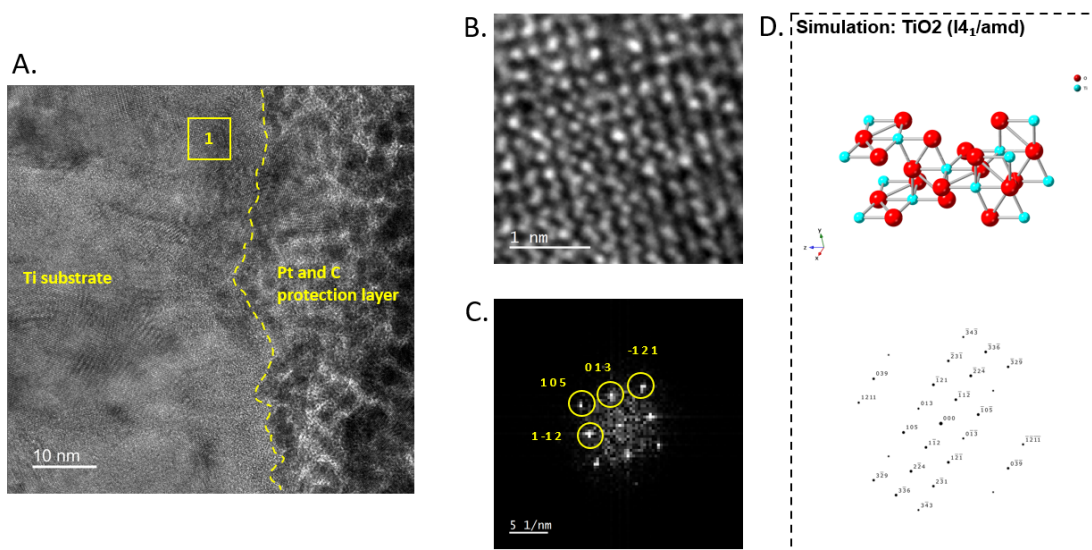


Figure S14. Transmission electron microscopy (TEM) characterisation of a focused ion beam (FIB) milled cross sectional specimen of the pure Ti after the AST tested with IrO_x. **(a)** TEM image where the left of the image is the Pt and C strap used to protect the surface of the Ti during FIB sample preparation. **(b)** Atomic lattice information from the yellow box region (1) in (a). **(c)** The Fourier transform of (b), where the spots are indexed to the anatase TiO₂ phase. **(d)** Crystallmaker atomic structure model and simulated diffraction pattern for the anatase TiO₂ phase (*I4₁/amd*).

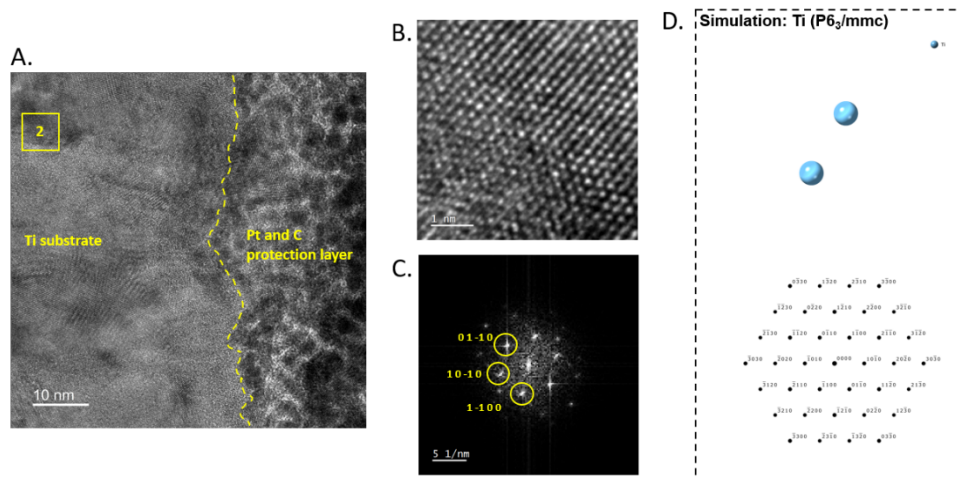


Figure S15. TEM characterisation of the same region but in the bulk shown in Figure 4 for the pure Ti after the AST tested with IrO_x. **(a)** TEM image, **(b)** Atomic lattice information from the yellow box region (2) in (a). **(c)** the Fourier transform of (b), where the spots are indexed to alpha phase metal Ti (hexagonal close packed). **(d)** Crystalmaker atomic structure model and simulated diffraction pattern for metallic alpha phase Ti ($P6_2/mmc$).

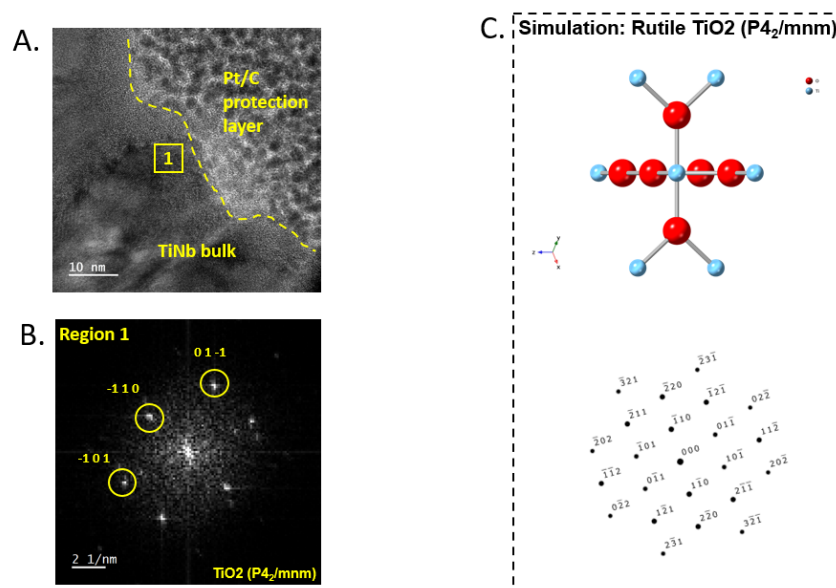


Figure S16. TEM characterisation of a FIB milled cross sectional specimen TiNb₍₅₀₎ after the AST tested with IrO_x. **(a)** TEM image where the protective Pt and C coating is visible in the top right of the image. **(b)** The Fourier transform from the yellow box region (1) in (a), where the spots are indexed to rutile phase TiO₂. **(c)** Crystalmaker atomic structure model and simulated diffraction pattern for rutile TiO₂ phase ($P4_2/mnm$).

References

- 1 W. Zhang, C. Zhu, Y. Wen, M. Wang, Z. Lu and Y. Wang, *Angew. Chem. Int. Ed.*, DOI:10.1002/anie.202418456.
- 2 G. Li, A. Priyadarsini, Z. Xie, S. Kang, Y. Liu, X. Chen, S. Kattel and J. G. Chen, *J. Am. Chem. Soc.*, 2025, **147**, 7008–7016.
- 3 J. Murawski, S. B. Scott, R. Rao, K. Rigg, C. Zalitis, J. Stevens, J. Sharman, G. Hinds and I. E. L. Stephens, *Johnson Matthey Public Limited Company*, 2024, preprint, DOI: 10.1595/205651324X17055018154113.
- 4 W. Zhu, X. Song, F. Liao, H. Huang, Q. Shao, K. Feng, Y. Zhou, M. Ma, J. Wu, H. Yang, H. Yang, M. Wang, J. Shi, J. Zhong, T. Cheng, M. Shao, Y. Liu and Z. Kang, *Nat. Commun.*, DOI:10.1038/s41467-023-41036-9.
- 5 A. Lam, Q. Sun, Z. Polus, C. Buek, K. McCrea, I. Mahoney, L. Wegge, Z. Zeng, M. McGowan, H. Yu, F. Yang, T. Valdez, M. Hamdan and Q. Jia, *ACS Catal.*, 2025, 8414–8425.
- 6 J. Ruiz Esquiú, D. J. Morgan, I. Spanos, D. G. Hewes, S. J. Freakley and G. J. Hutchings, *ACS Appl. Energy Mater.*, 2020, **3**, 800–809.
- 7 V. Pfeifer, T. E. Jones, J. J. Velasco Vélez, C. Massué, M. T. Greiner, R. Arrigo, D. Teschner, F. Girgsdies, M. Scherzer, J. Allan, M. Hashagen, G. Weinberg, S. Piccinin, M. Hävecker, A. Knop-Gericke and R. Schlögl, *Physical Chemistry Chemical Physics*, 2016, **18**, 2292–2296.
- 8 D. Hoffmeister, S. Finger, L. Fiedler, T. C. Ma, A. Körner, M. Zlatar, B. Fritsch, K. W. Bodnar, S. Carl, A. Götz, B. A. Zubiri, J. Will, E. Spiecker, S. Cherevko, A. T. S. Freiberg, K. J. J. Mayrhofer, S. Thiele, A. Hutzler and C. van Pham, *Advanced Science*, DOI:10.1002/advs.202402991.
- 9 M. Zlatar, D. Escalera-López, M. G. Rodríguez, T. Hrbek, C. Götz, R. Mary Joy, A. Savan, H. P. Tran, H. N. Nong, P. Pobodinskas, V. Briega-Martos, A. Hutzler, T. Böhm, K. Haenen, A. Ludwig, I. Khalakhan, P. Strasser and S. Cherevko, *ACS Catal.*, 2023, **13**, 15375–15392.