

## Supporting Information

## 2 *Multifunctional Self-Healing Superhydrophobic Coating with Rapid Sunlight-*

### 3 Induced Recovery and Photothermal Anti-Icing Capability

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21 **Experimental section**

22 *1.1 High-pressure water-jet impingement test*

23 The coating's dynamic impingement resistance was assessed using a high-pressure  
24 water-jetting setup (a programmable rain test chamber, HT-IP9K-1000L). The coated glass  
25 substrate was positioned vertically (90° tilt), and a continuous water jet (25 kPa) was applied from  
26 a distance of 5 cm at a speed of 10 m/s. The surface was exposed to the jet for increasing durations  
27 (0–150 s), after which the WCA and SA were recorded to evaluate changes in wetting behavior.

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29 *1.2. Chemical and physical self-healing tests*

30 To evaluate the self-healing capacity of the coatings, two types of damage were induced:  
31 chemical and mechanical. Chemical degradation was simulated by treating the coatings for 3 min  
32 using an O<sub>2</sub>-plasma system. Mechanical damage was inflicted by creating a surface scratch with a  
33 new razor blade. Following damage, samples were exposed to simulated sunlight from a xenon  
34 lamp solar simulator (CEL-HXF300-T3, Beijing Zhongjiao Jinyuan Technology Co., Ltd., China)  
35 for 5 min to activate the healing process. The recovery from chemical damage was quantified by  
36 measuring the WCA before and after healing to track the loss and subsequent restoration of  
37 superhydrophobicity. Microstructural recovery was assessed by examining scratch morphology  
38 using optical microscopy and FE-SEM, comparing the damaged and healed states.

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40 *1.3. Mechanical and chemical durability tests*

41 The mechanical robustness of the coatings was evaluated through a series of tests designed  
42 to simulate harsh environmental stresses, including sand abrasion, high-pressure water impact, and  
43 sandpaper wear.

44 *Sand abrasion resistance.* Cyclic sand abrasion tests were conducted by releasing 500 g of silica  
45 sand (300–400 µm particle size) from a height of 40 cm onto the coated surface, inclined at 45°.  
46 The sand was delivered in 10 increments of 50 g each. Water contact angles (WCAs) and sliding  
47 angles (SAs) were measured after each full cycle to quantify the retention of superhydrophobicity.

48 *Water impact resistance.* Coatings were subjected to a high-impact water jet spray totaling 5000  
49 L over 24 hours, directed at a 45° angle from 50 cm. The WCAs and SAs were recorded at 4-hour  
50 intervals to monitor degradation under sustained hydraulic pressure.

51 *Sandpaper abrasion Test.* Samples were placed face-down on 800- or 1000-grit SiC sandpaper  
52 and subjected to linear abrasion under a 200 g load. Each 15 cm forward movement constituted  
53 one cycle. WCAs and SAs were measured after every 10 cycles, up to a total of 50 cycles. To  
54 evaluate the self-healing capability post-abrasion, mechanically stressed coatings were irradiated  
55 under simulated sunlight (1.0 sun) for 5 min. Recovery of surface morphology was observed using  
56 optical microscopy, and the corresponding restoration of superhydrophobicity was confirmed  
57 through WCA and SA measurements.

58 *Tape-peeling adhesion test.* Adhesion durability was evaluated using a standard tape-peeling  
59 procedure following ASTM D3359. A 3M #600 adhesive tape (25.4 mm width, 50  $\mu\text{m}$  thickness)  
60 was firmly applied to the coating under a uniform pressure of 2.6 kPa and then peeled off in a  
61 single motion. This process was repeated for multiple cycles, after which the WCA and SA were  
62 recorded to assess any changes in surface wettability.

63 *UV Aging resistance evaluation.* The coatings were placed in a UV-aging chamber equipped with  
64 a 400 W full-spectrum solar simulator, where they were subjected to alternating cycles consisting  
65 of 4 h UVA irradiation at 1000  $\text{W/m}^2$  and 60  $^\circ\text{C}$ , followed by 4 h condensation at 50  $^\circ\text{C}$ . After  
66 designated numbers of cycles, the WCAs and SAs were measured to evaluate the retention of  
67 hydrophobic performance.

68 *Chemical stability* was evaluated by immersing coatings in varying pH levels: acidic (pH 1),  
69 neutral (pH 6), and alkaline (pH 13) solutions for 24 hours. WCA was measured after immersion  
70 to assess superhydrophobicity retention. To evaluate self-healing, chemically treated coatings were  
71 irradiated under simulated sunlight (1 sun) for 5 min. Surface recovery was characterized using  
72 atomic force microscopy (AFM) for topographic analysis, while superhydrophobicity restoration  
73 was confirmed through WCA measurements.

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#### 75 *1.4. Self-cleaning test*

76 The self-cleaning performance of the MESH coating was evaluated by depositing sand  
77 contaminants on both coated and uncoated glass surfaces inclined at 10 $^\circ$ . Water droplets (4 mL)  
78 were introduced to the upper surface, and contaminant removal efficiency was quantified by  
79 comparing residual sand particles after droplet roll-off.

80 *1.5. Dual-mode anti-icing performance*

81       Static anti-icing tests were conducted at  $-20^{\circ}\text{C}$  under 80% relative humidity to simulate cold,  
82 humid environments. The experiments were performed in a programmable constant-temperature  
83 and humidity chamber (HD-ED702-504K70, Haida Instrument Co., Ltd., China; operating range  
84  $-40$  to  $80^{\circ}\text{C}$ ,  $\pm 1^{\circ}\text{C}$ , -100 to 100 % RH). Prior to testing, samples were equilibrated at  $-20^{\circ}\text{C}/80\%$   
85 RH for 1 h to ensure stable environmental conditions.

86 *Static passive anti-icing* performance was evaluated by depositing a 40  $\mu\text{L}$  dyed (methyl red) water  
87 droplet onto both coated and uncoated glass substrates at  $-20^{\circ}\text{C}/80\%$  RH, and recording the  
88 freezing time using digital camera.

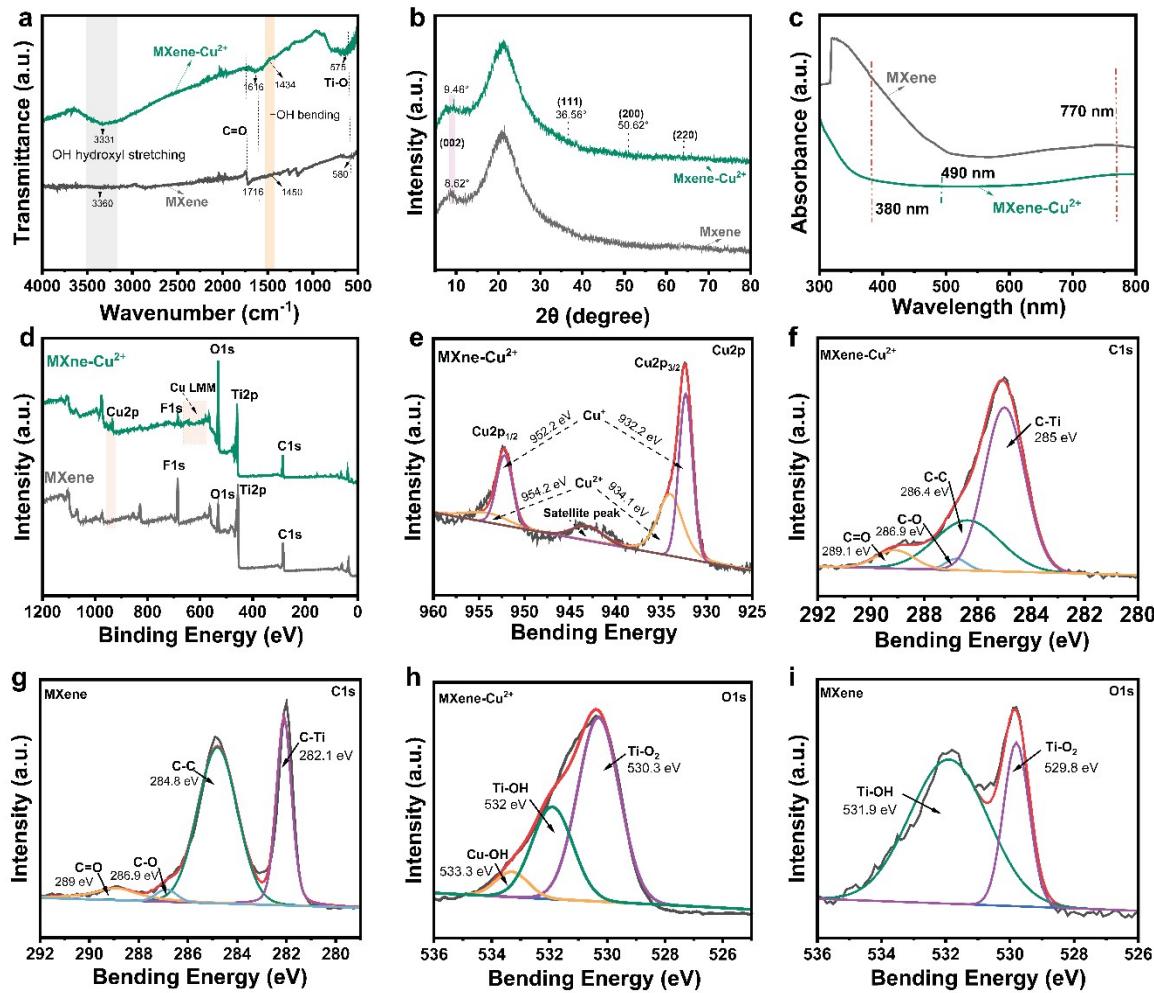
89 *Active anti-icing* was tested by repeating the droplet freezing assay under 1 sun irradiation while  
90 recording the delayed freezing process.

91 *Dynamic passive anti-icing test.* Cold water droplets ( $0^{\circ}\text{C}$ ,  $\sim 40 \mu\text{L}$ ) were continuously dispensed  
92 onto a  $20^{\circ}$ -tilted glass or MESH-coated substrate inside a low-temperature chamber maintained at  
93  $-10^{\circ}\text{C}$  and 80% relative humidity. The onset of freezing, and subsequent ice accumulation were  
94 monitored and recorded using a digital camera to determine the dynamic icing resistance.

95 *Active de-icing test.* A uniform ice layer was first generated on the sample surfaces under  $-20^{\circ}\text{C}$   
96 and 80% RH. The iced substrates, positioned at a  $20^{\circ}$  tilt, were then exposed to 1.0-sun irradiation  
97 while remaining inside the same low-temperature chamber. The melting and detachment of ice  
98 were recorded using a digital camera.

99 *Ice adhesion strength* was measured using a using a lab-built force transducer, which detached  
100 ice from the surface at 0.1 mm/s. The peak force was recorded, and the average of three tests  
101 reported as ice adhesion strength.

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104 **Figure S1.** (a) ATR-FTIR spectra, (b) XRD patterns, and (c) UV-vis spectra of MXene and MXene-Cu<sup>2+</sup>. (d)  
105 XPS survey spectra, and (e–i) high-resolution XPS spectra of Cu 2p, C 1s, and O 1s for MXene-Cu<sup>2+</sup> and  
106 MXene, respectively.

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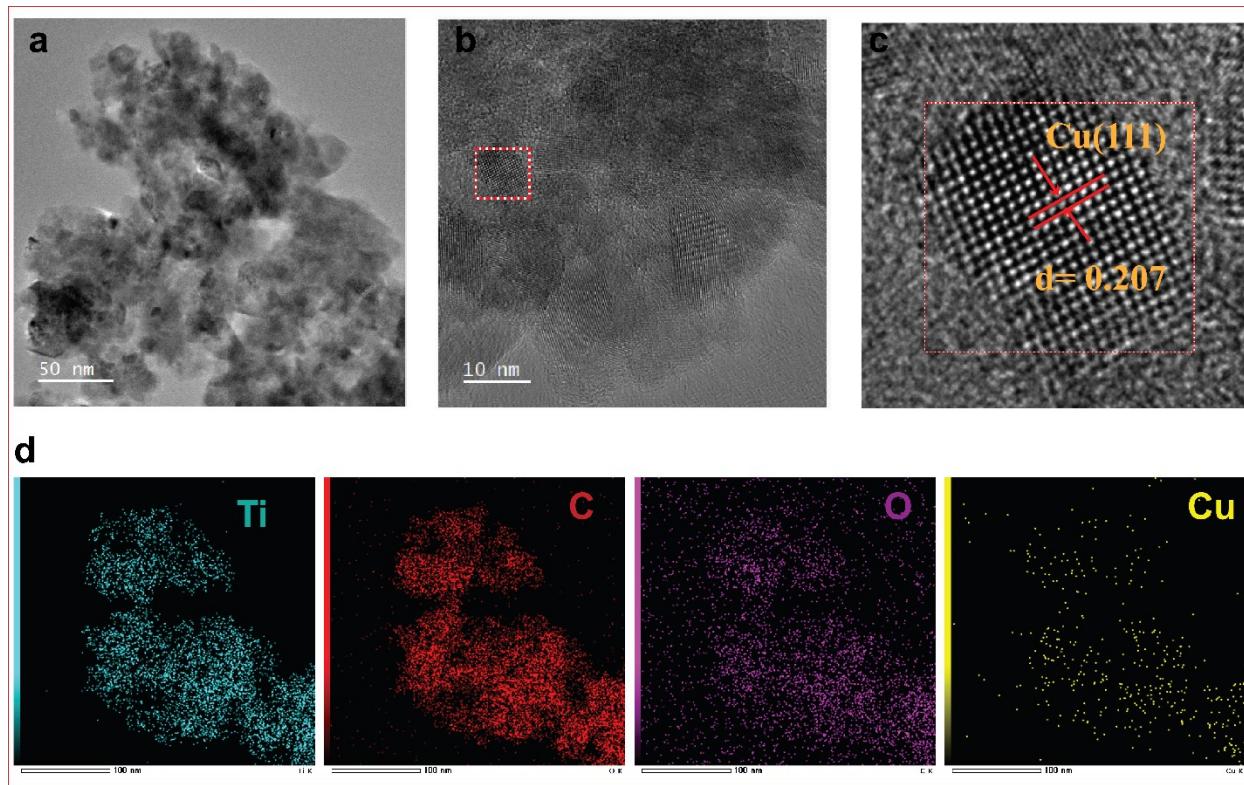
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115 The TEM image (Figure S2a) revealed well-defined nanosheet structures decorated with  
 116 uniformly distributed nanoparticles, while the HRTEM image (Figure S2b) offered a closer view  
 117 of the lattice fringes, confirming the crystalline nature of the embedded copper species. Notably,  
 118 the zoomed-in HRTEM image (Figure S2c) displayed distinct lattice planes corresponding to  
 119 Cu(111), with an interplanar spacing of 0.207 nm, further validating the presence of metallic  
 120 copper domains. Complementary elemental mapping (Figure S2d) showed homogeneous  
 121 distribution of Ti, C, O, and Cu across the nanosheet surface, indicating successful doping and  
 122 intimate integration of Cu<sup>2+</sup> within the MXene nanosheets. These findings collectively confirm the  
 123 structural and compositional integrity of the MXene–Cu<sup>2+</sup> hybrid, supporting its potential for  
 124 enhanced functional performance.

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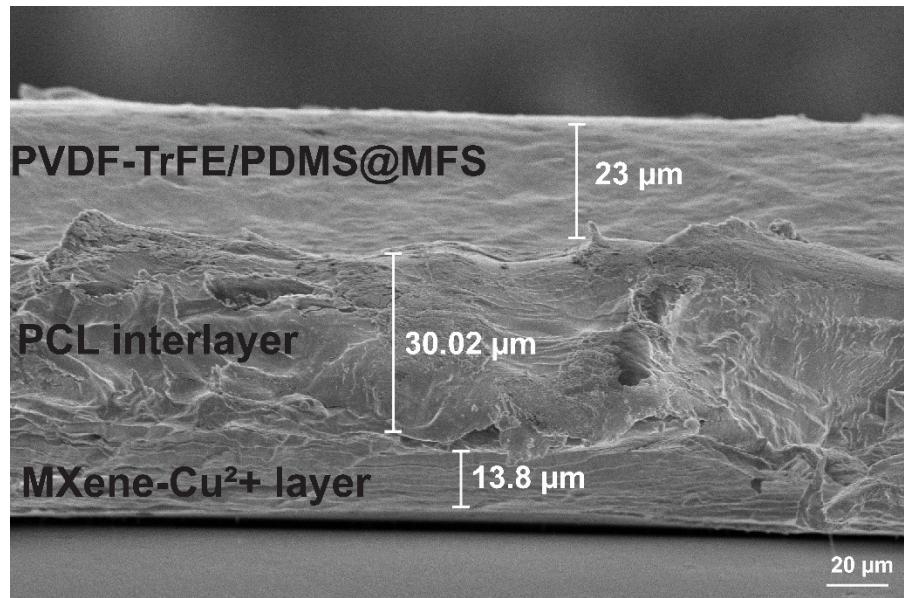
127 **Figure S2.** (a–c) TEM and HRTEM images of MXene–Cu<sup>2+</sup>, and (d) TEM elemental mapping image of MXene–  
 128 Cu<sup>2+</sup>.

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**Figure S3.** Cross-sectional SEM image of the MESH coating

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### 137 **Impalement Resistance Mechanism**

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139 The coating's resistance to liquid impalement under dynamic impact is governed by the  
140 pressure balance between the hammer pressure ( $P_h$ ) of the incoming jet and the capillary  
141 pressure ( $P_c$ ) generated by the surface structure.<sup>1</sup> According to established models' formula S1  
142 and S2:

$$143 \quad P_h \approx 0.2\rho C v \quad (1)$$

144 Here,  $\rho$  is water density,  $C$  is sound velocity,  $v$  is impact velocity

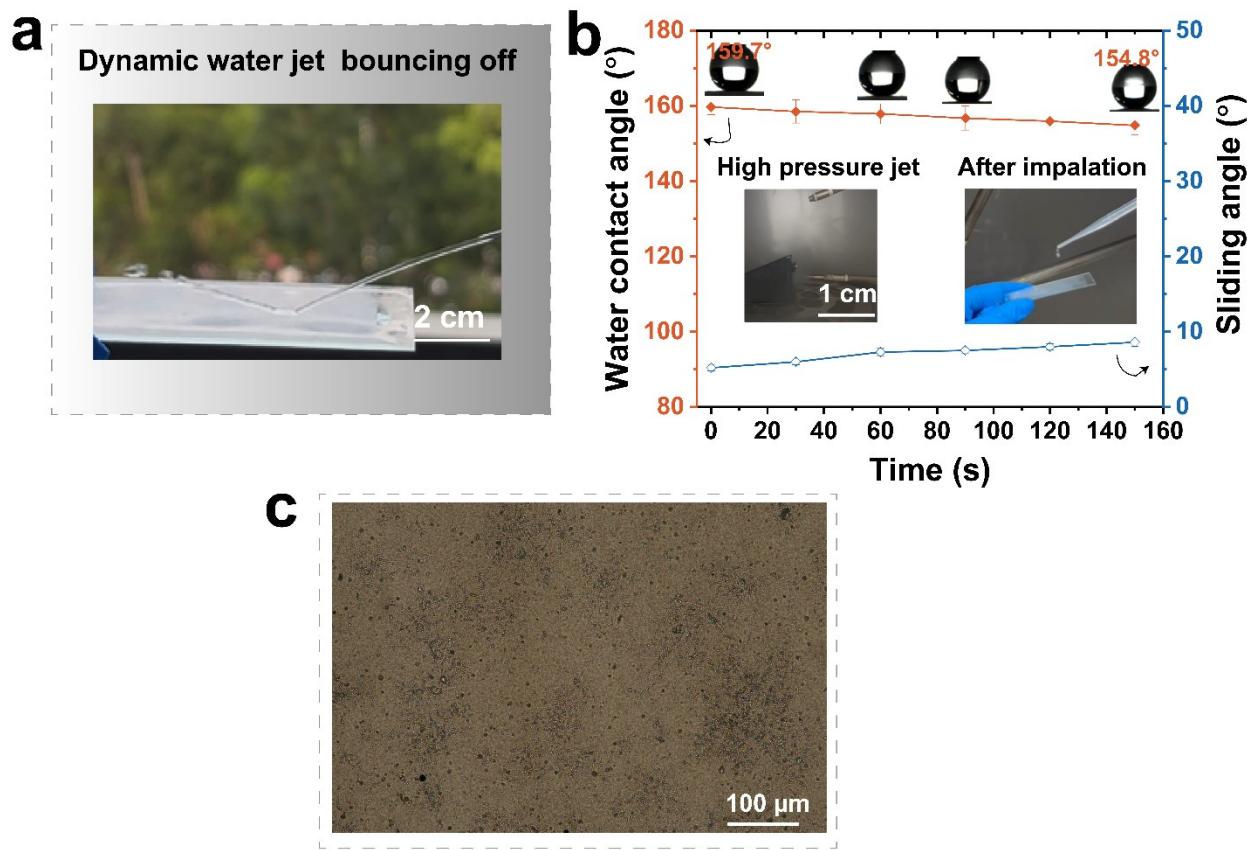
$$P_c \approx \frac{2\gamma r \sin^2\left(\frac{\theta_{adv}}{2}\right)}{d^2} \quad (2)$$

145

146 Where  $\gamma$  is surface tension,  $r$  is particle radius,  $d$  is spacing between protrusions, and  $\theta_{adv}$  is the  
147 advancing contact angle.

148 The equations show that reducing the spacing  $d$  between surface features markedly increases  $P_c$ ,  
149 meaning that tightly spaced micro–nano structures are more resistant to impalement.<sup>1</sup>

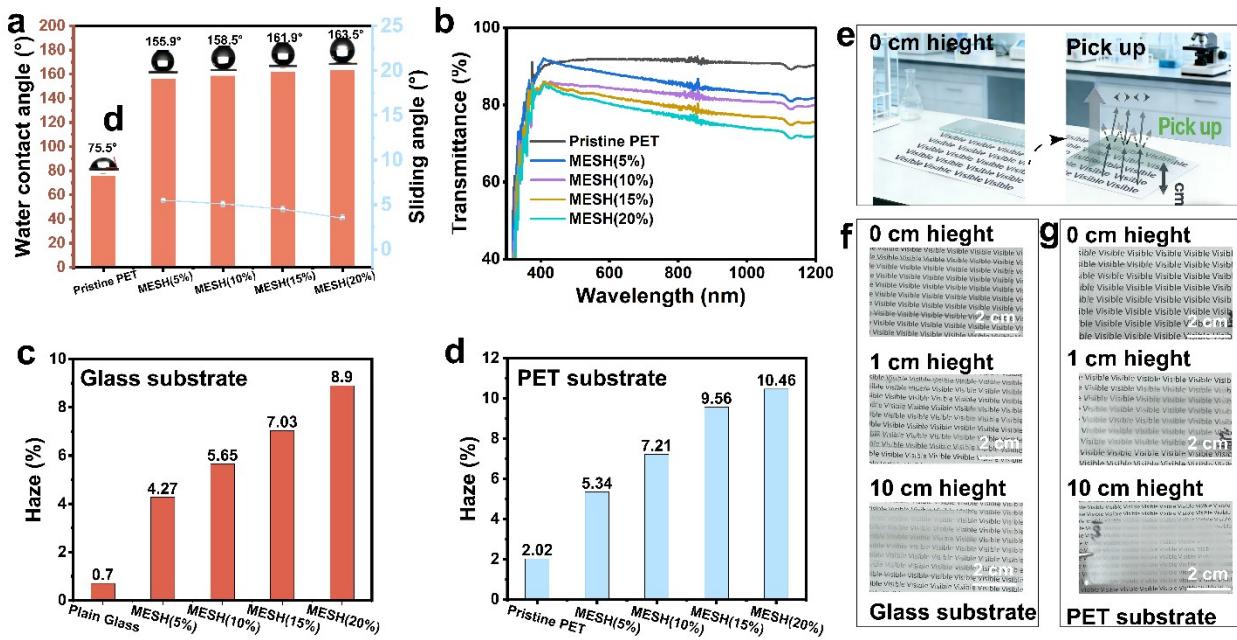
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152 Figure S4. (a) Dynamic water jet bouncing off the MESH coating. (b) Time-dependent evolution of WCA and  
153 SA during continuous high-pressure jetting. (c) Microscopic image of the surface after jet impalement

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156 Figure S5. (a) Wettability of MESH coatings on PET substrate at varying filler ratios, showing WCA and SA.  
 157 (b) Optical transmittance spectra of pristine PET and MESH-coated PET films. (c-d) Haze values at 550 nm for  
 158 MESH coatings on glass (c) and PET (d) substrates (e) Schematic illustration of the measurement configuration  
 159 for transmittance and haze using a paper sheet as background. (f-g) Photographs showing text visibility through  
 160 MESH-coated glass (f) and PET (g) placed directly on the paper sheet (0 cm) and at distances of 1 cm and 10 cm,  
 161 demonstrating the balance between high transmittance and controlled haze.

162

## 163 Photothermal Conversion Efficiency

164 To evaluate the photothermal conversion efficiency ( $\eta$ ) of the MXene–Cu<sup>2+</sup> coating under  
 165 simulated solar illumination, we adopted a modified energy balance approach analogous to the  
 166 method used by Fan et al.<sup>2</sup> Since the system involves a solid coating on a glass substrate without  
 167 phase transition, the stored energy is entirely sensible heat. The efficiency was calculated by  
 168 comparing the thermal energy stored by the sample to the solar energy absorbed during  
 169 illumination.

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173 (1) Thermal energy stored by the coated substrate is define by:

174 
$$Q_{\text{stored}} = m_{\text{eff}} \cdot C_{p,\text{eff}} \cdot (T_{\text{max}} - T_0) \quad (4)$$

175 Where  $Q_{\text{stored}}$  is total sensible heat stored (J),  $m_{\text{eff}}$  the effective mass (g) and  $C_{p,\text{eff}}$  specific heat  
176 capacity of the sample ( $J \cdot g^{-1} \cdot K^{-1}$ ),  $T_{\text{max}}$  maximum temperature during irradiation ( $^{\circ}C$ ),  
177 and  $T_0$  initial temperature ( $^{\circ}C$ )

178 The effective mass is given by:

179 
$$m_{\text{eff}} = \rho \cdot A \cdot d \quad (5)$$

180 2) Solar energy received and absorbed by the coating define by:

181 
$$Q_r = P \cdot S \cdot t_{\text{heat}} \quad (6)$$

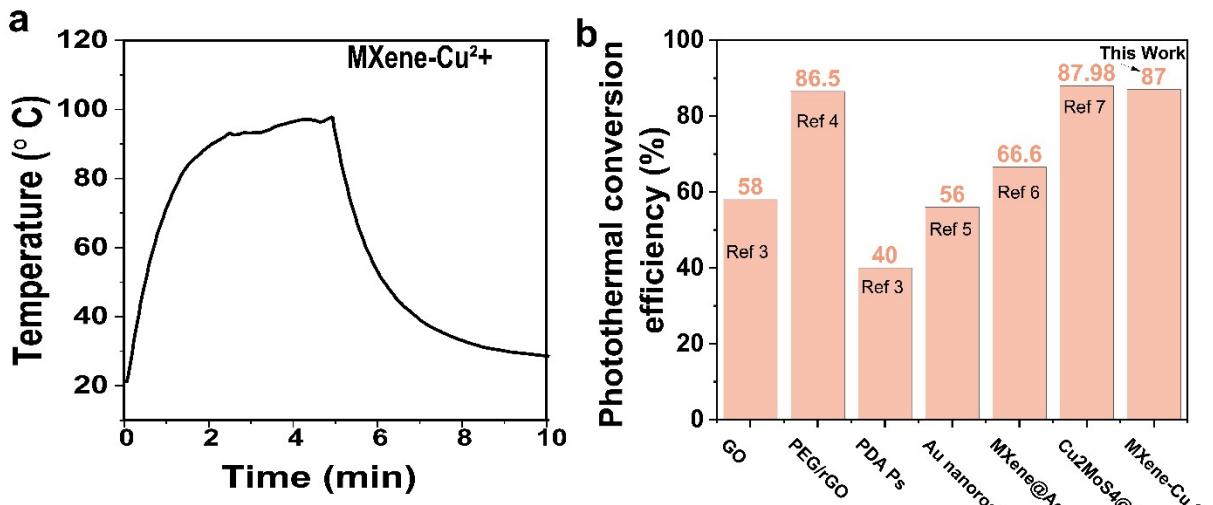
182 Where  $Q_r$  is total incident solar energy (J),  $P$  the solar intensity ( $W \cdot cm^{-2}$ ),  $S$  is illuminated area  
183 ( $cm^2$ ), and  $t_{\text{heat}}$  irradiation time to reach  $T_{\text{max}}(s)$

184 To account for the fraction of light absorbed by the MXene-  $Cu^2^+$  layer, the absorbance at  $\lambda$   
185 ( $A_{\lambda}$ ) was used:

186 
$$Q_{r,abs} = Q_r \cdot (1 - 10^{-A_{\lambda}}) \quad (7)$$

187 (3) Photothermal conversion efficiency is calculated by:

188 
$$\eta = \frac{Q_{\text{stored}}}{Q_{r,abs}} = \frac{m_{\text{eff}} \cdot C_{p,\text{eff}} \cdot (T_{\text{max}} - T_0)}{P \cdot S \cdot t_{\text{heat}} \cdot (1 - 10^{-A_{\lambda}})} \quad (8)$$

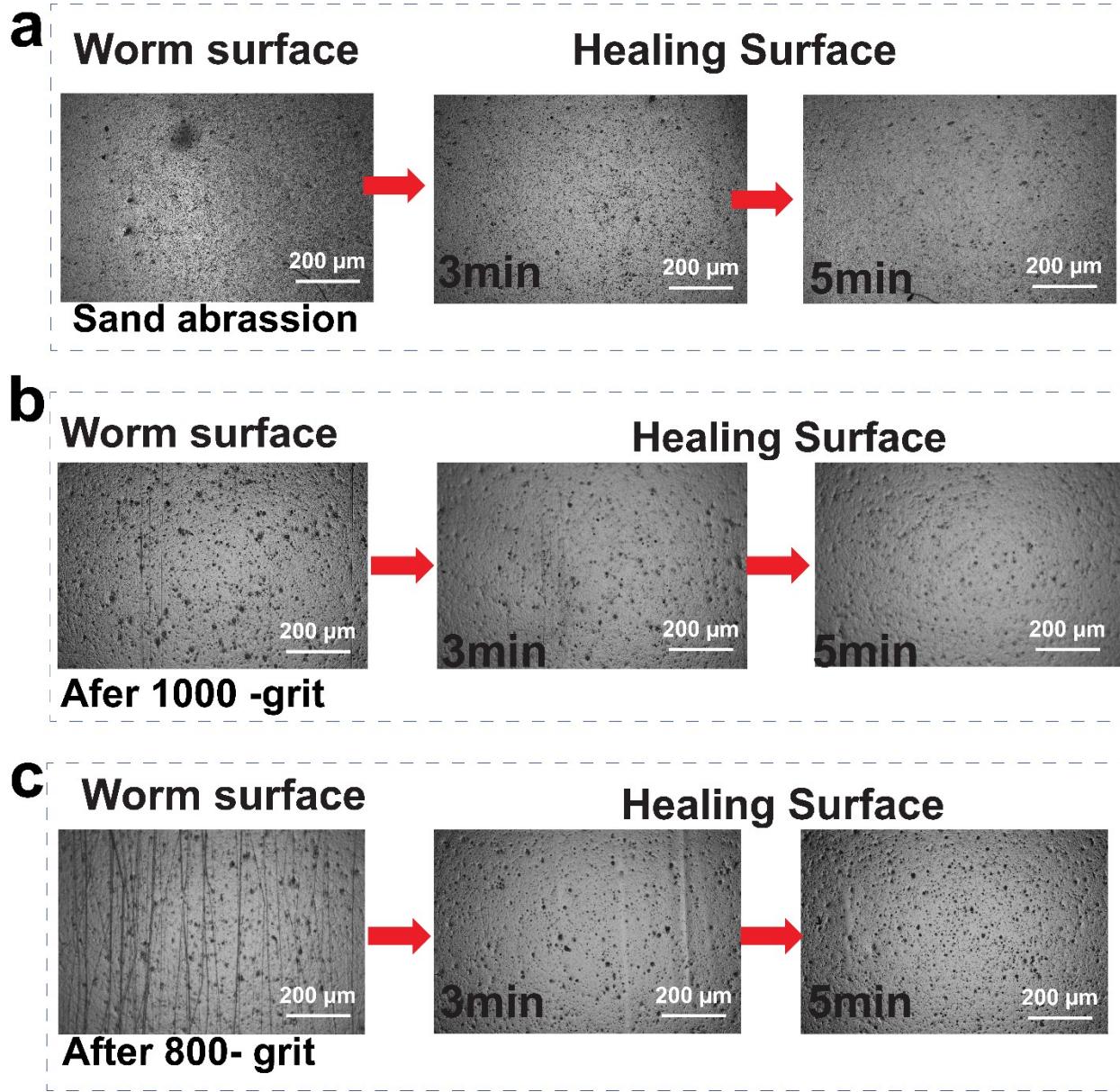


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190 **Figure S6.** (a) Temperature–time profile of MXene–Cu<sup>2+</sup> under simulated solar irradiation (1 sun, 5 min heating  
 191 and 5 min cooling). (b) Comparison of photothermal conversion efficiency ( $\eta$ ) for MXene–Cu<sup>2+</sup> and benchmark  
 192 materials including GO,<sup>3</sup> PEG/rGO,<sup>4</sup> PDA Ps,<sup>5</sup> Au nanorods,<sup>3</sup> MXene@AgAu@PDA,<sup>6</sup> and Cu<sub>2</sub>MoS<sub>4</sub>@MXene.<sup>7</sup>

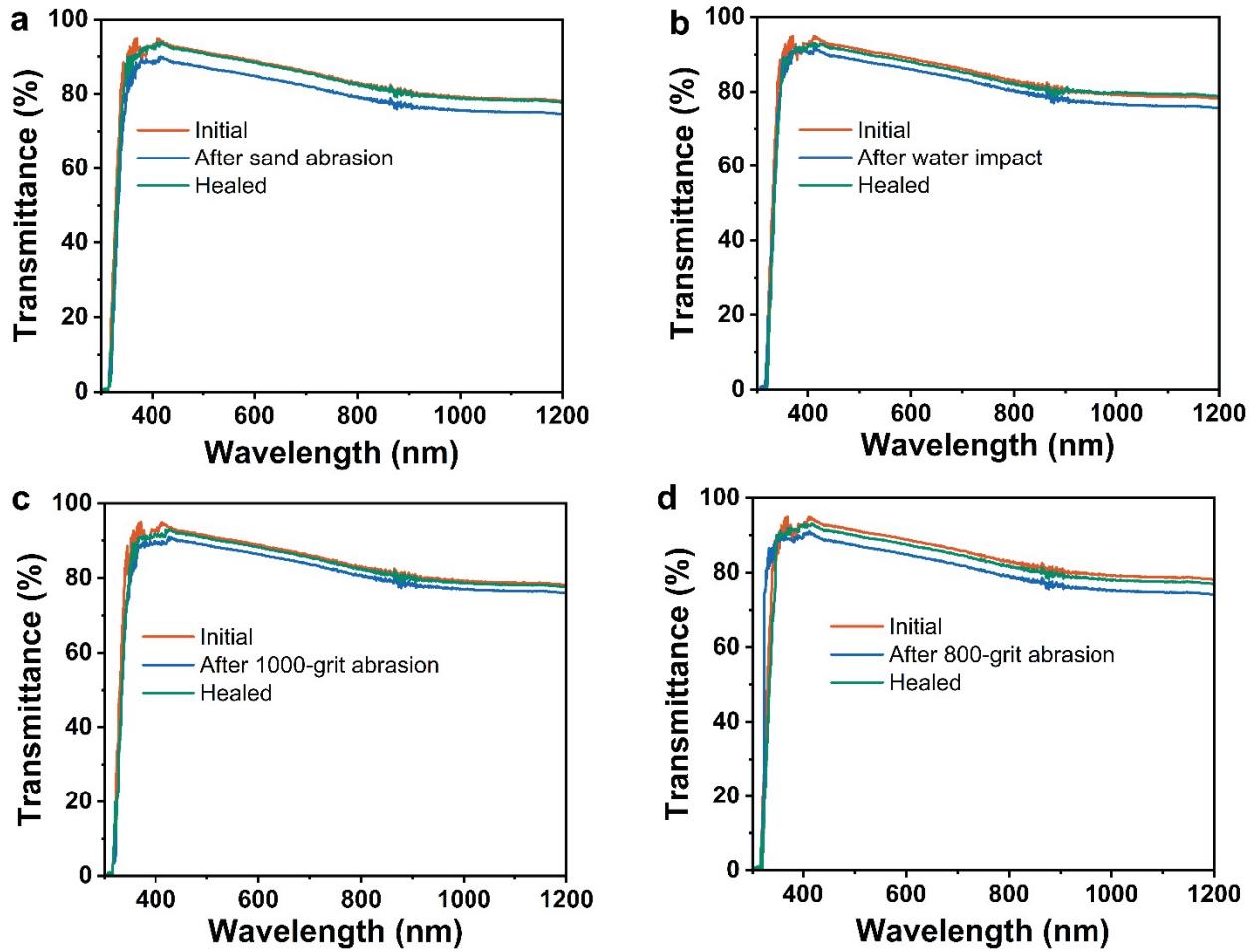
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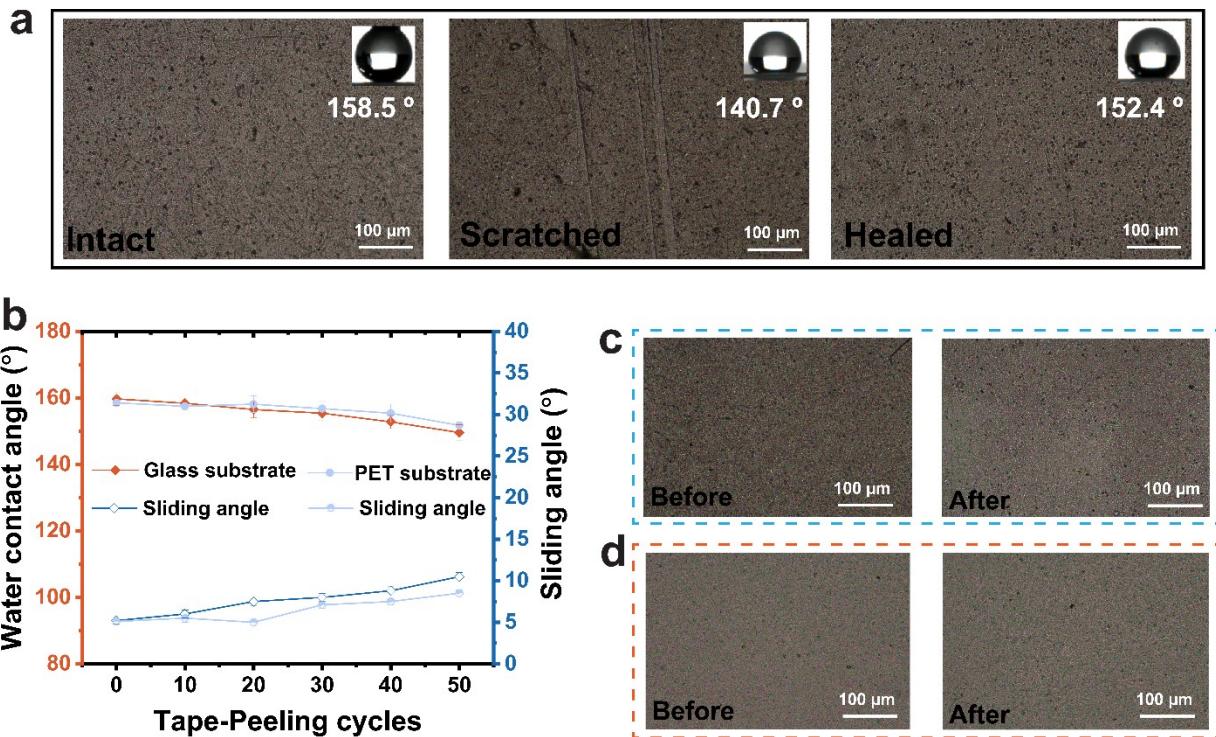
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196 **Figure S7.** a-c) post-impact healed micrograph images of MESH coating. Sand falling cycle (a), Sandpaper  
 197 abrasion cycle under 1000 grit (b) and 800 grit(c)



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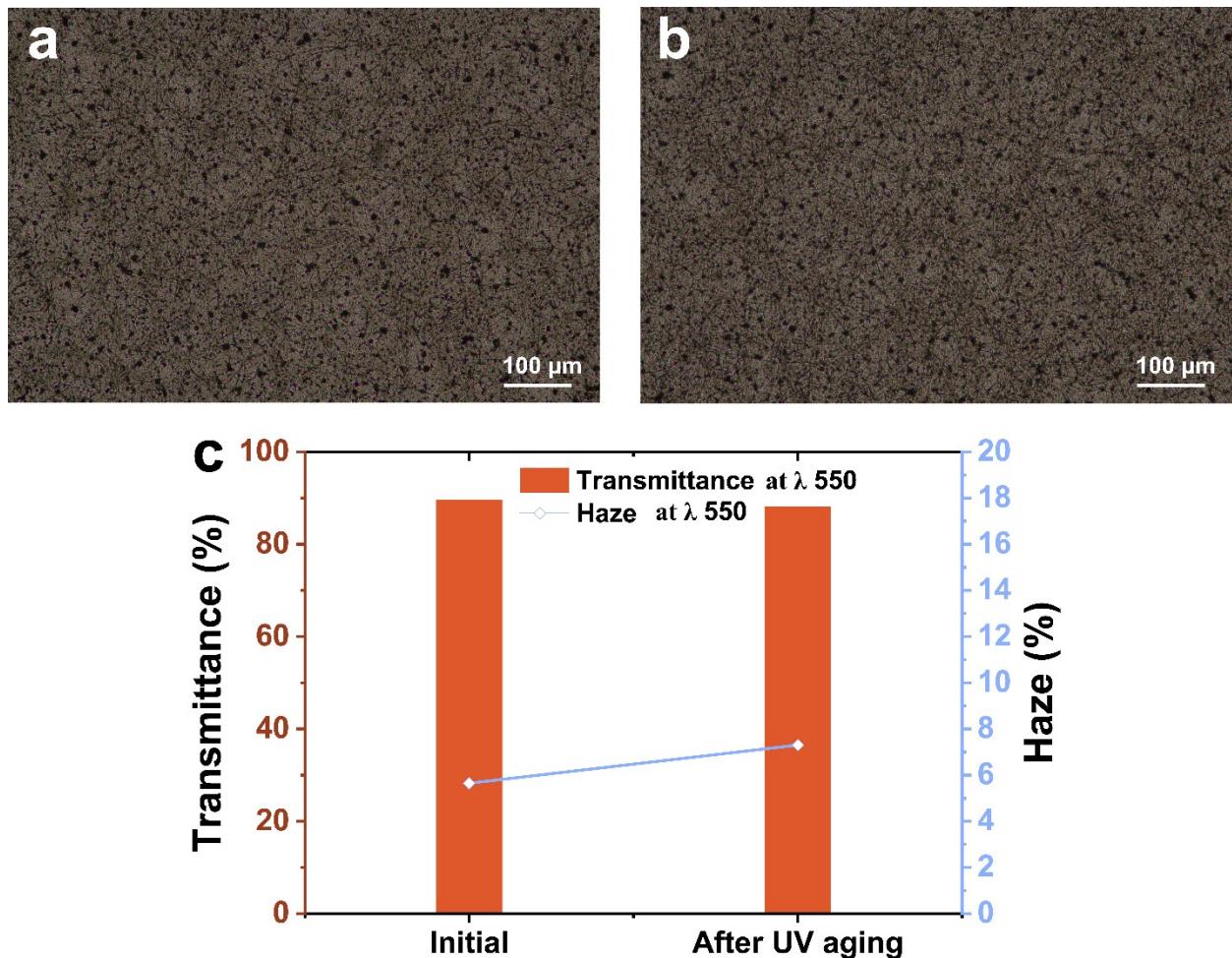
199 **Figure S8.** Optical transmittance spectra of the MESH coating at the initial state and after post-mechanical  
 200 testing and healing, including (a) sand falling test, (b) water impact test, and sandpaper abrasion tests on (c)  
 201 1000-grit and (d) 800-grit surfaces, respectively.



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203 **Figure S9.** a. Microscopic images of MESH- coated PET substrate at initial, and after scratched and healing  
 204 process under 1 sun (5 min) (inset is corresponding WCA). b- Relationship between WCA and SA of MESH  
 205 coating on Glass and PET substrate as a function tape peeling cycles. c- d Microscopic images of MESH coating  
 206 before and after tape-peeling test on PET and glass substrates, respectively.

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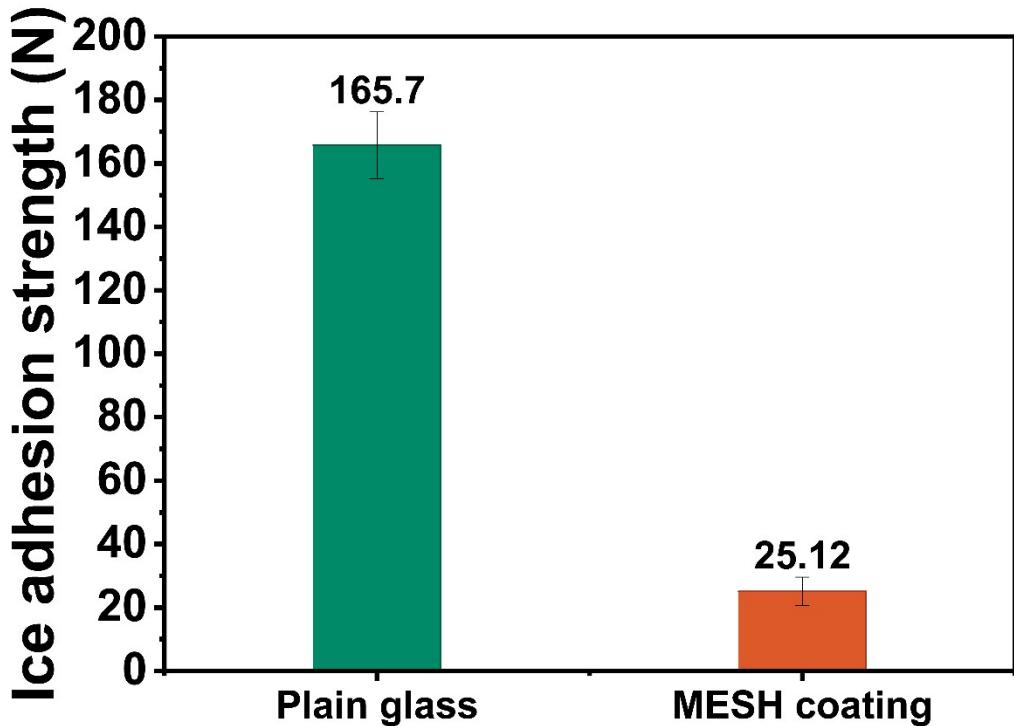


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209 **Figure S10.** Micrograph images of MESH coating before and after UV aging test. Transmittance and haze value  
 210 of MESH coating before and after UV aging test.

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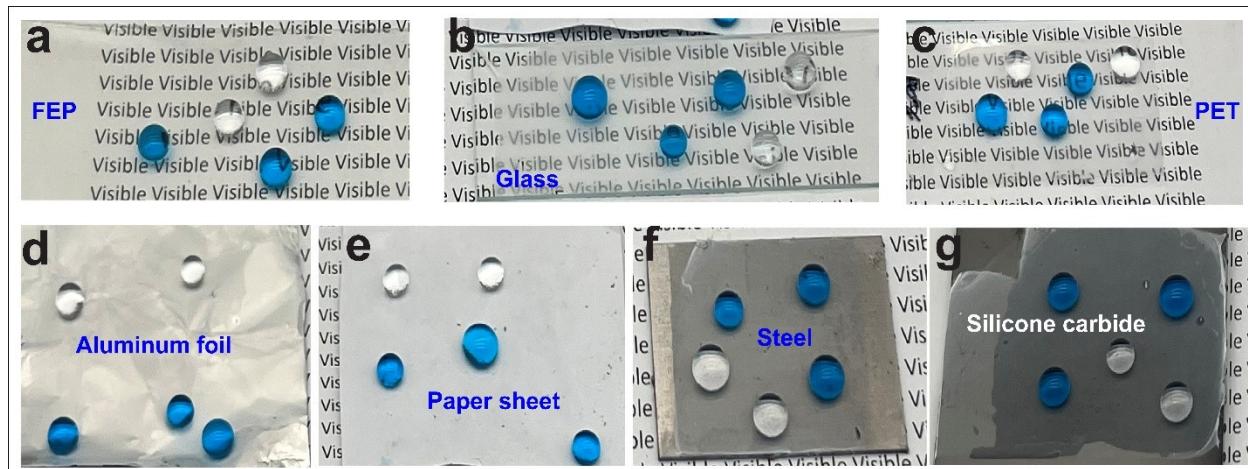
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214 **Figure S11.** Ice adhesion strength of plain glass and the MESH coating.

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217 **Figure S12.** MESH coatings applied to various substrates, including (a) fluorinated ethylene propylene (FEP),  
 218 (b) glass, (c) polyethylene terephthalate (PET), (d) Aluminum foil, (e) Paper-sheet, (f) Steel, and (g) Silicon  
 219 carbide, each shown with its corresponding water droplet (dye methyl blue and none-dye water).

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222 **Table S1. Comparative summary of superhydrophobic self-healing coatings from this work and**  
223 **recent studies.**

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Material	Method	Chemical self-healing		Physical self-healing		Functionality		Ref
		Properties	Scenarios	Properties	Scenarios	Self-cleaning	Anti-icing	
DDA-PDA@CNTs	One-pot	O <sub>2</sub> plasma-repair	150°C, <b>10 min</b>	Surface structure repair	89.9°C, <b>10 min</b>	Yes	Yes	8
BPDI/SiO <sub>2</sub>	Dual cross-linking, spray	plasma etching	Spontaneous room temp	Surface reorganization	UV	Yes	Not stated	9
PU/SiO <sub>2</sub> @HD-POS	Spray-coating, spray	HD-POS migration	room temp			Yes	Yes	10
Al/ZnO-P.I.Z-FAS@PDA	spray	chemical repair	<b>40 min</b> , room temp	scratch repair	<b>5 min</b> , 80°C (photothermal)	Yes	Yes	11
POEG/Fluorinated carbon black	Two-step spray			Surface structure repair	Heat treatment	Not stated	Yes	12
DTMS@PDA@SiO <sub>2</sub> @CNTs/epoxy	One-step spray	chemical corrosion	NIR irradiation	scratch repair		Yes	Not stated	13
ZIF-7@ZnG@PFDS/epoxy	MOF-based, spray	NaCl				Yes	Yes	14
PDA-Cu <sup>2+</sup> -GO/ODA/PDMS	Layered, chelation, spray	O <sub>2</sub> plasma	1-sun irradiation	Scratch repair	1-sun irradiation	Yes	Not stated	15
PU/ZnO@PDA-SA	disulfide-bonded			Scratch repair	70°C, <b>1h</b>	Yes		16
ZnO@MPDA/silicone latex	Blending, stimuli-responsive	controlled release	UV/NIR/acid/base			Yes	Not stated	17
PEG/Fe <sub>3</sub> O <sub>4</sub> -SA/polyamide-epoxy	Hydrogen bonding		4 h room temp, <b>10 min</b> oven			Yes	Yes	18
CNT@SiO <sub>2</sub> /epoxy	Spray	48 h alkali cycles	4 h room temperature, <b>10 min</b> 80 °C oven			Not stated	Yes	19
PDMS-IPDI-TFB/PDA NPs	Spray		Sunlight, room temp			Yes	Yes	20
MXene-Cu <sup>2+</sup> /PCL/PVDF-TrFE/PDMS	Electrospinning assisted coating	O <sub>2</sub> plasma	1-sun, <b>5 min</b>	Scratch repair	1-sun, <b>5 min</b>	Yes	Yes	This work

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230 **References**

231 1. H. Teisala, F. Geyer, J. Haapanen, P. Juuti, J. M. Mäkelä, D. Vollmer and H. J. Butt, *Adv Mater*, 2018, **30**, 1706529.

233 2. R. Fan, N. Zheng and Z. Sun, *Sol. Energy Mater. Sol. Cells*, 2022, **237**, 111572.

234 3. O. A. Savchuk, J. Carvajal, J. Massons, M. Aguiló and F. Díaz, *Carbon*, 2016, **103**, 134-235 141.

236 4. W. Luo, M. Zou, L. Luo, Y. Ma, W. Chen, X. Hu, Q. Li and X. Jiang, *J. Energy Storage*, 237 2024, **78**, 109950.

238 5. Y. Li and D. Wang, *Chin. J. Pharm*, 2021, **19**, 81-86.

239 6. G. Liu, H. Wang, C. Xu, Q. Fang, H. Wang, Y. Xu, M. Sang, S. Xuan and L. Hao, *J. Mater. Chem. B*, 2023, **11**, 10678-10691.

241 7. L. Li, J. Zhang, Y. Lin, Y. Zhang, S. Li, Y. Liu, Y. Zhang, L. Shi, S. Yuan and L. Guo, 242 *Front. Bioeng. Biotechnol.*, 2022, **10**, 902312.

243 8. Y. Li, H. Li, J. Wu, X. Yang, X. Jia, J. Yang, D. Shao, L. Feng, S. Wang and H. S. Song, 244 *Appl. Surf. Sci*, 2022, **600**, 154177.

245 9. J. Tao, L. Dong, Y. Wu, X. Liu, J. Xie, H. Wu and Q. Ran, *Compos. Pt. B-Eng*, 2024, 246 **273**, 111245.

247 10. Y. Li, B. Li, X. Zhao, N. Tian, J. Zhang, *ACS Appl. Mater. Interfaces*, 2018, **10**, 39391-248 39399.

249 11. T. Zhang, J. Deng and L.-Z. Zhang, *Prog. Org. Coat*, 2023, **180**, 107569.

250 12. K. Li, J. Xiang, J. Zhou, X. Su, H. Xie, S. Lin, Z. Xiao, W. Huang, C. Zhao and H. Chen, 251 *Prog. Org. Coat*, 2023, **177**, 107415.

252 13. J. Liu, W. Huang, G. Cui, X. Xing and J. Liu, *Process Saf. Environ. Prot.*, 2025, **195**, 253 106759.

254 14. G. Zhu, J. Su, C. Yin, H. Li, Y. Yao, L. Zhang, X. Yao, X. Zhang and F.-Q. Liu, *Appl. Surf. Sci.*, 2023, **622**, 156907.

256 15. C. Lee, S. Ji, S. Oh, S. Park, Y. Jung, J. Lee and H. Lim, *Nanoscale Adv.*, 2022, **4**, 761-257 771.

258 16. C. Li, P. Wang, D. Zhang and S. Wang, *ACS Appl. Mater. Interfaces*, 2022, **14**, 45988-259 46000.

260 17. X. Ni, C. Li, Y. Lei, Y. Shao, Y. Zhu and B. You, *ACS Appl. Mater. Interfaces*, 2021, **13**, 261 57864-57879.

262 18. X. Wang, L. Tang, S. Fan and W. Fan, *Nanomaterials*, 2024, **14**, 1981.

263 19. Z. He, H. Xu, Y. Zhou and Y. Tan, *J Mater Res Technol*, 2024, **30**, 2609-2619.

264 20. Y. Wu, L. Dong, X. Shu, Y. Zhang and Q. Ran, *ACS Appl. Mater. Interfaces*, 2025, **17**,  
265 18852-18868.

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