

## Supplementary Information of

# Optimizing the piezocatalytic hydrogen production activity of metal-organic frameworks through precise defect engineering

### Author Information

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### 1. Number of missing BDC linker characterization by TGA

Here, we define  $x$  as the number of missing BDC linkers in the material. In this study, we define  $m$  as the ratio of the mass at 450°C to the final mass at 600°C, with subscripts indicating different temperatures. The number of missing BDC linkers ( $x$ ) in the structure can be determined using the following equation<sup>S1</sup>:

$$m_{600} = MW_{\text{HfO}_2} \cdot n_{\text{HfO}_2} \# \text{ (S1)}$$

$$n_{\text{MOF}} = \frac{n_{\text{HfO}_2}}{6} = \frac{m_{600}}{6 \cdot MW_{\text{HfO}_2}} = \frac{100}{6 \cdot MW_{\text{HfO}_2}} \# \text{ (S2)}$$

$$m_{450} = MW_{\text{MOF}_{450}} \cdot n_{\text{MOF}} \# \text{ (S3)}$$

$$MW_{\text{MOF}_{450}} = \frac{6 \cdot m_{450} \cdot MW_{\text{HfO}_2}}{100} = 12.63 \cdot m_{450} \# \text{ (S4)}$$

$$MW_{\text{MOF}_{450}} = 6 \cdot MW_{\text{Hf}} + (6 + x) MW_{\text{O}} + (6 - x) MW_{\text{BDC}^2} \# \text{ (S5)}$$

$$x = \frac{MW_{\text{MOF}_{450}} - 6(MW_{\text{Hf}} + MW_{\text{O}} + MW_{\text{BDC}^{2-}})}{MW_{\text{O}} - MW_{\text{BDC}^{2-}}} = \frac{2151.71 - 12.63 \cdot m_{450}}{148.13} \#(\text{S6})$$

## 2. Defect concentration calculations of UiO-66 models in MD simulations

To quantify the defect concentration of UiO-66 models considered in MD simulations, we adopted the same definition as commonly used in <sup>1</sup>H-NMR experimental analysis, by comparing the relative amounts of terephthalate linkers (BDC) and formate groups (HCOO<sup>-</sup>). In the *fcu* configuration, each Hf node is fully coordinated by 6 BDC linkers and contains no formate group. In the *reo* configuration, missing-linker defects reduce the coordination to 4 BDC linkers per Hf node, while the remaining coordination sites are compensated by 4 HCOO<sup>-</sup> groups.<sup>S2</sup>

Thus, as for a simulation supercell consisting of  $N$  units, where  $n$  units adopt the *reo* configuration and the remaining  $(N-n)$  units adopt the *fcu* configuration, the corresponding total numbers of BDC and HCOO groups, namely  $n_{\text{BDC}}$  and  $n_{\text{HCOO}}$  can be calculated as:

$$n_{\text{BDC}} = 6(N - n) + 4n \#(\text{S7})$$

$$n_{\text{HCOO}} = 4n \#(\text{S8})$$

The defect concentration  $c$  can be correspondingly defined as follows:

$$c = \frac{n_{\text{HCOO}}}{n_{\text{BDC}}} \#(\text{S9})$$

Based on Equation S9, the defect concentration of UiO-66 models with different numbers of the defective *reo* units ranging from 0 to 10 varies from 0% to 28% correspondingly.

## 3. The carrier concentration ( $N_{\text{D}}$ ) and conduction band ( $C_{\text{B}}$ ) position calculated from the Mott-Schottky measurements

The space-charge capacitance  $C$  of the electrode semiconductor interface obeys:<sup>S3, 4</sup>

$$\frac{1}{C^2} = \left( \frac{2}{\epsilon_0 \epsilon_r A^2 e N_{\text{D}}} \right) \left( V - V_{\text{FB}, \text{SHE}} - \frac{k_{\text{B}} T}{e} \right) \#(\text{S10})$$

Here,  $C$  is the interfacial capacitance and  $A$  is the geometric electrode area, respectively.  $N_{\text{D}}$  is the number of

carrier concentration,  $V$  is the applied voltage,  $k_B$  is Boltzmann's constant,  $T$  the absolute temperature, and  $e$  is the electronic charge,  $\epsilon_r$  is relative dielectric constant,  $\epsilon_0$  the vacuum dielectric constant. At 298 K,  $k_B T/e \approx 0.0257$  V can be ignored at room temperature.

The slope of the linear fitting to the second half of the  $1/C^2$  versus  $V$  plot is denoted as  $S$ . Therefore,  $N_D$  can be estimated as:

$$N_D = \frac{2}{\epsilon_r \epsilon_0 e S A^2} \#(S11)$$

The flat-band potential is obtained from the x-intercept of the linear fit:

$$V_{FB,SHE} = V_{\text{intercept}} - k_B T/e \#(S12)$$

For an  $n$ -type semiconductor, the conduction band edge is estimated from the flat-band potential with a small band-bending correction  $\Delta\Phi$ , which was taken as 0.2 V here.

$$C_{B,SHE} = V_{FB,SHE} + \Delta E - \Delta\Phi \#(S13)$$

The reference electrode  $\Delta E$  was +0.197V for Ag/AgCl (saturated KCl, vs SHE at 25 °C). Accordingly, the valence band edge was determined as follows:

$$V_{B,SHE} = C_{B,SHE} + E_g \#(S14)$$

where  $E_g$  is obtained from the Tauc plot (Fig. S15).

#### 4. Supplementary figures and tables

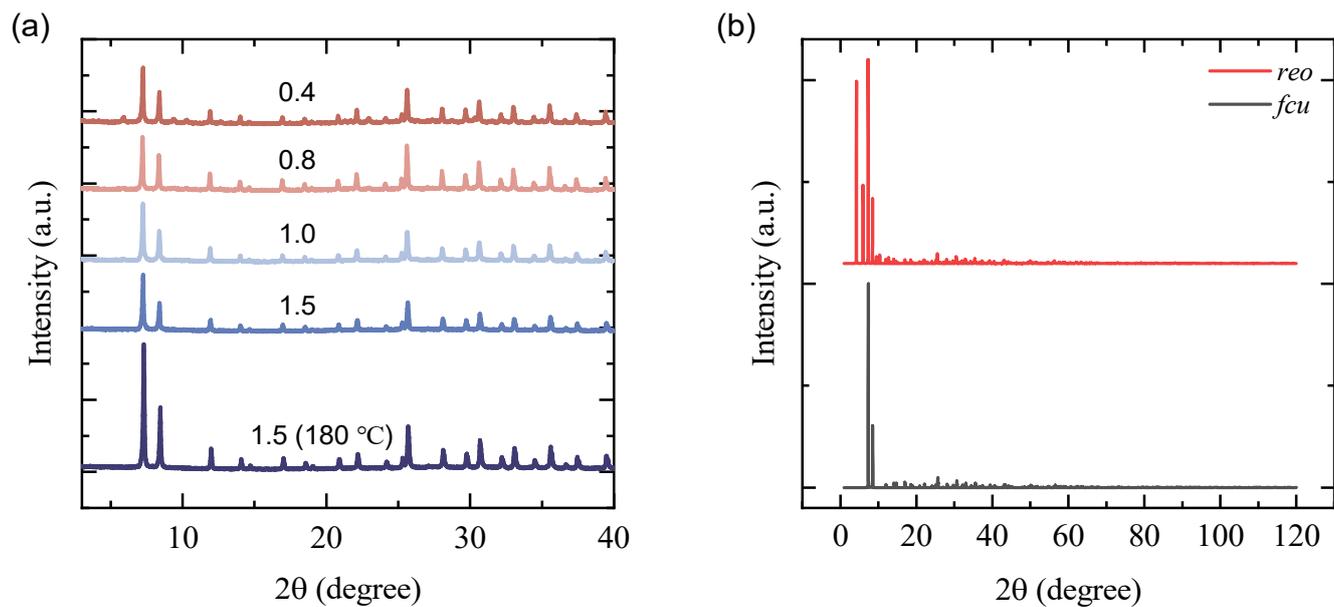


Fig. S1. (a) The PXRD patterns of UiO-66 crystals. (b) The simulated PXRD patterns of UiO-66 with the *fcu* and *reo* topologies.

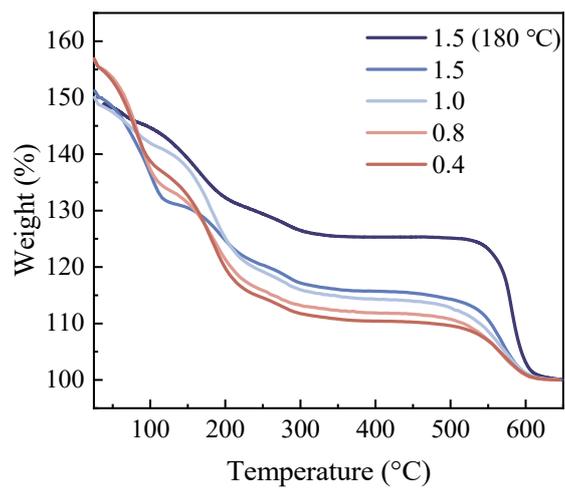


Fig. S2. The TGA patterns of UiO-66 crystals.

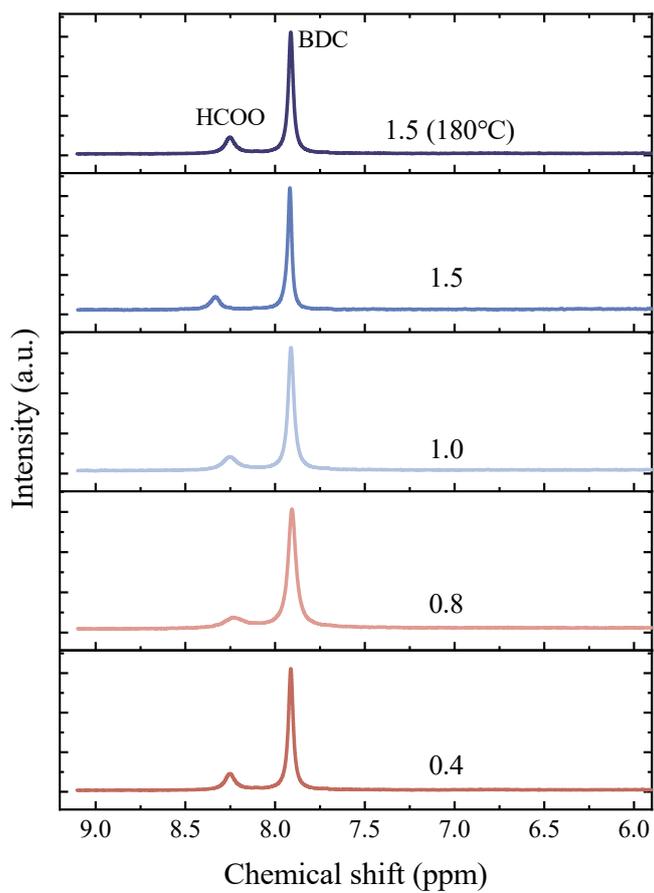


Fig. S3. The <sup>1</sup>H-NMR patterns of UiO-66 crystals.

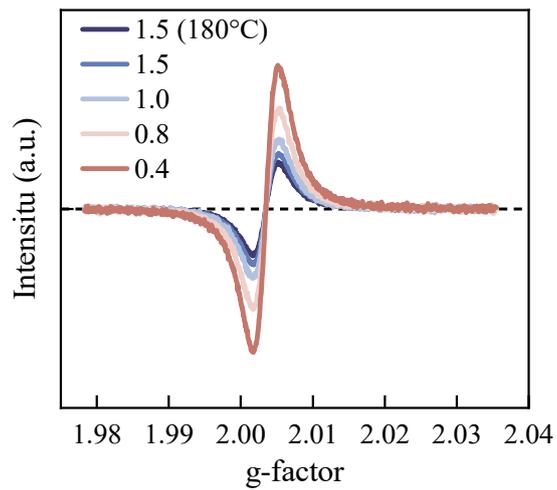


Fig. S4. The EPR diagram of UiO-66 crystals.

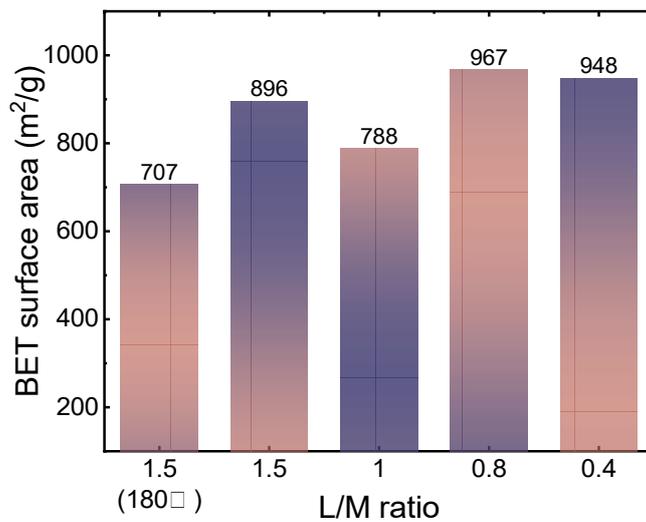


Fig. S5. The BET surface areas of UiO-66 crystals.

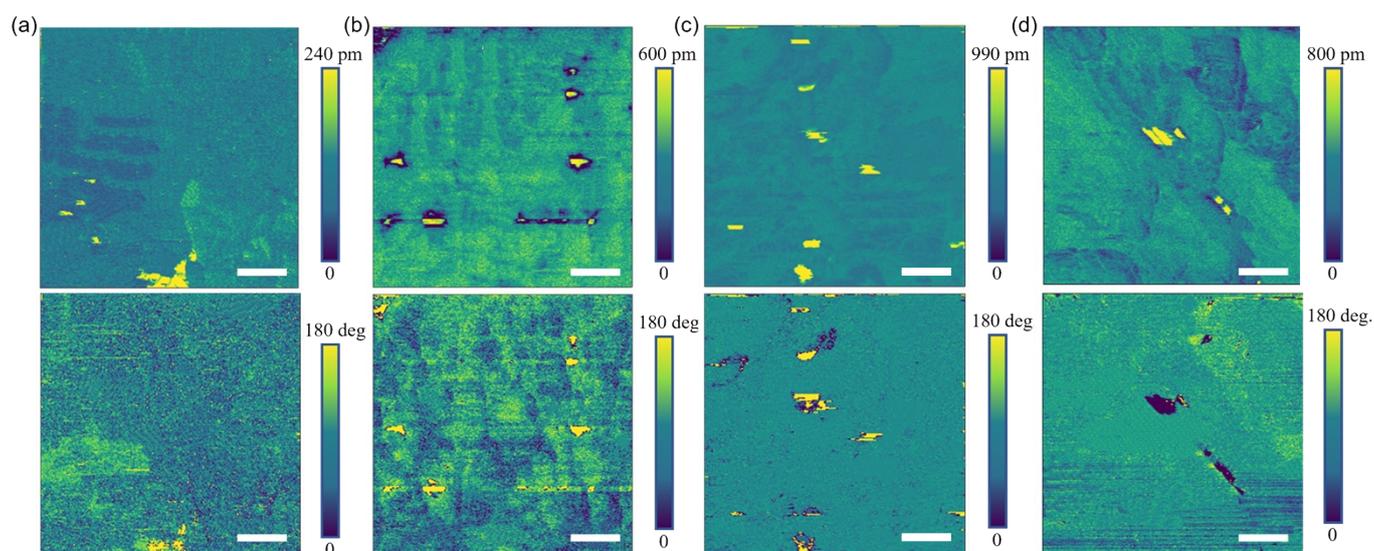


Fig. S6. Piezoelectric amplitudes (top panel) and corresponding phase maps (bottom panel) of UiO-66 with (a) L/M=1.5, (b) L/M=1.0 (c) L/M=0.8 and (d) L/M=0.4. All scale bars are 1 μm.

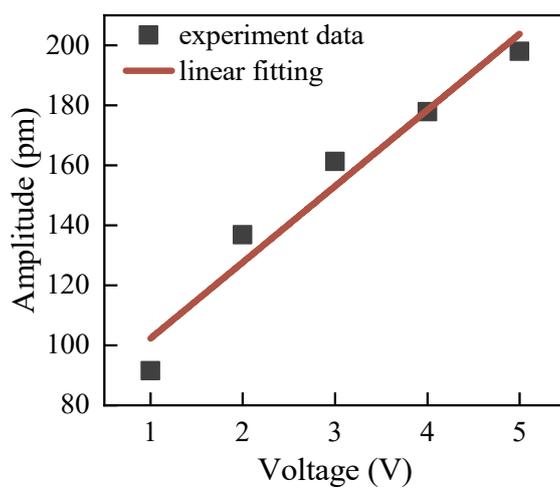


Fig. S7. PFM measurement of the variation of amplitude with applied alternating current voltage for pristine UiO-66 (L/M=1.5, 180 °C).

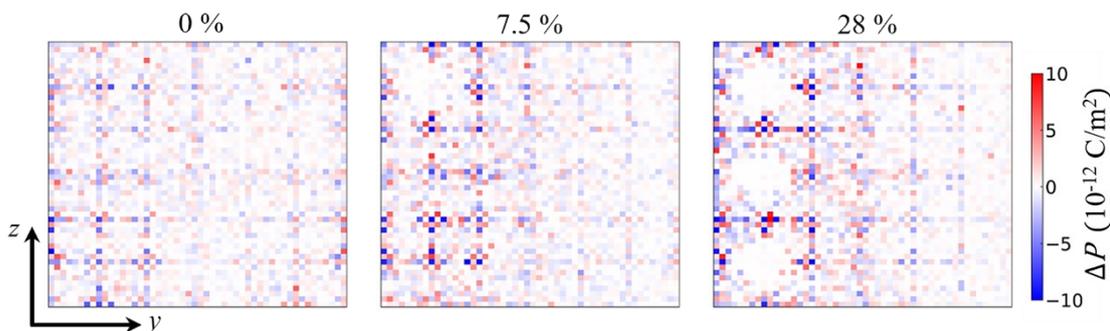


Fig. S8. Distribution of polarization change of pristine UiO-66 and defective UiO-66 with defect concentrations of 7.5% and 28% extracted from MD simulations.

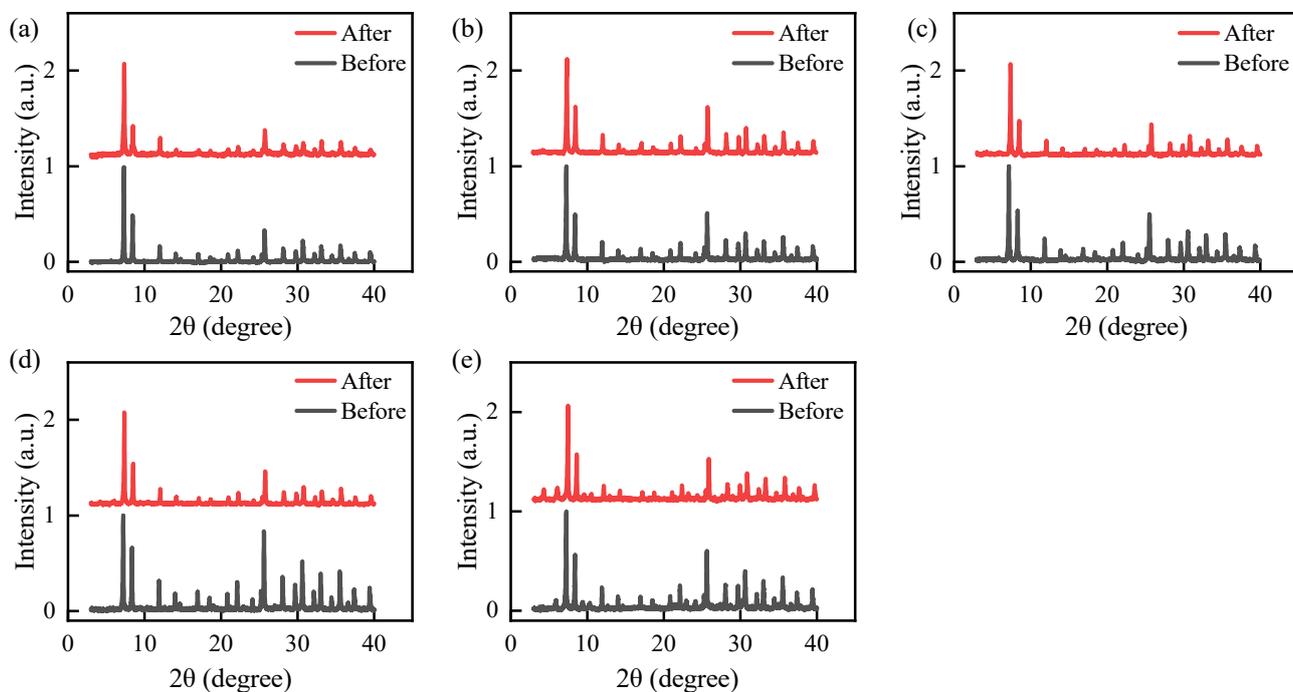


Fig. S9. The PXRD diffraction patterns before ultrasonic vibration and after two hours of ultrasonic vibration for samples with (a)  $L/M=1.5$  (180 °C), (b)  $L/M=1.5$ , (c)  $L/M=1.0$  (d)  $L/M=0.8$  and (e)  $L/M=0.4$ .

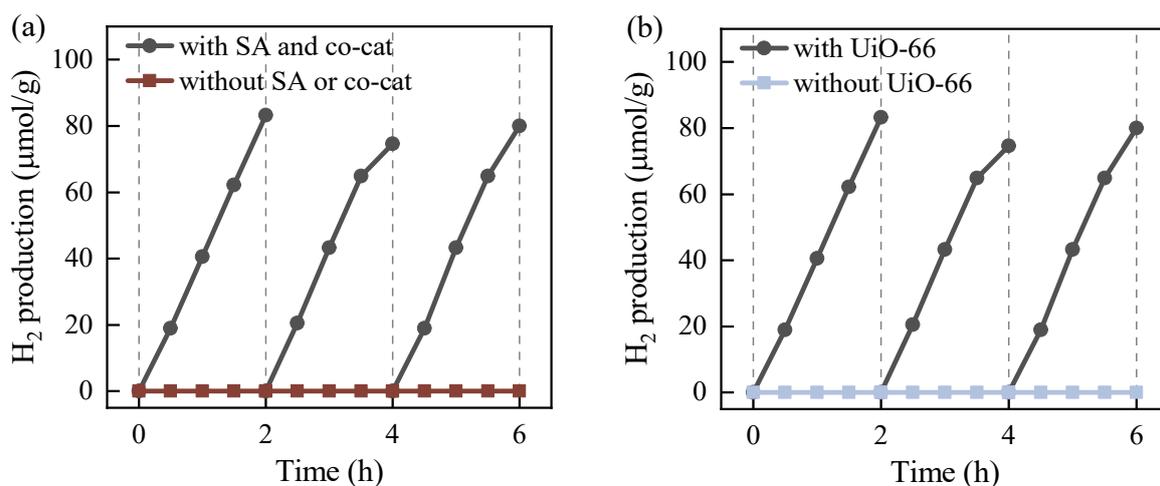


Fig. S10. (a) Hydrogen yields of UiO-66 (L/M = 1.0) under ultrasonic vibration with and without sacrificial agent (SA) and co-catalyst (co-cat). (b) Hydrogen yields in the reaction systems containing H<sub>2</sub>O, sacrificial agent (SA), and co-catalyst (co-cat) under conditions with and without UiO-66 (L/M = 1.0).

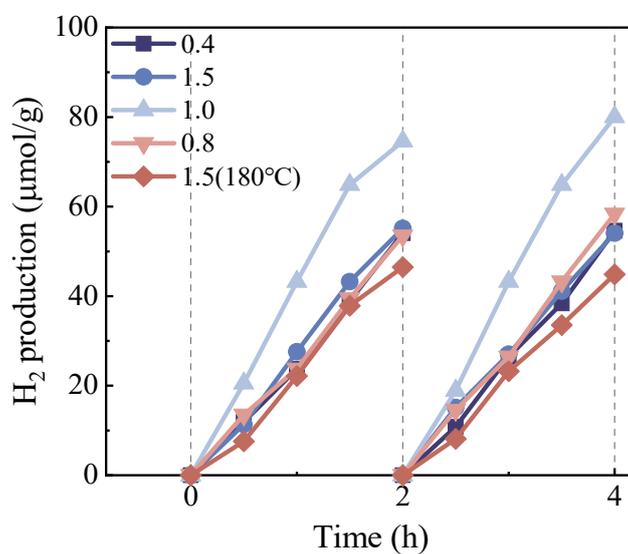


Fig. S11. Two independent replicate tests on hydrogen yields of UiO-66 crystals.

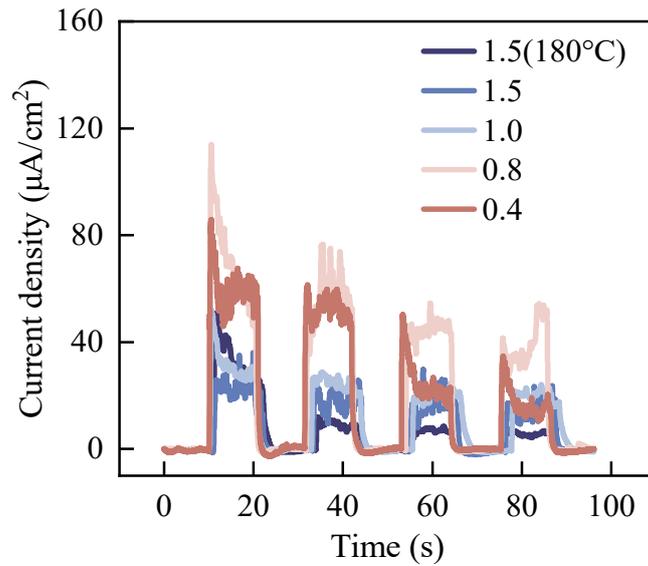


Fig. S12. Transient piezocurrent responses of UiO-66 crystals.

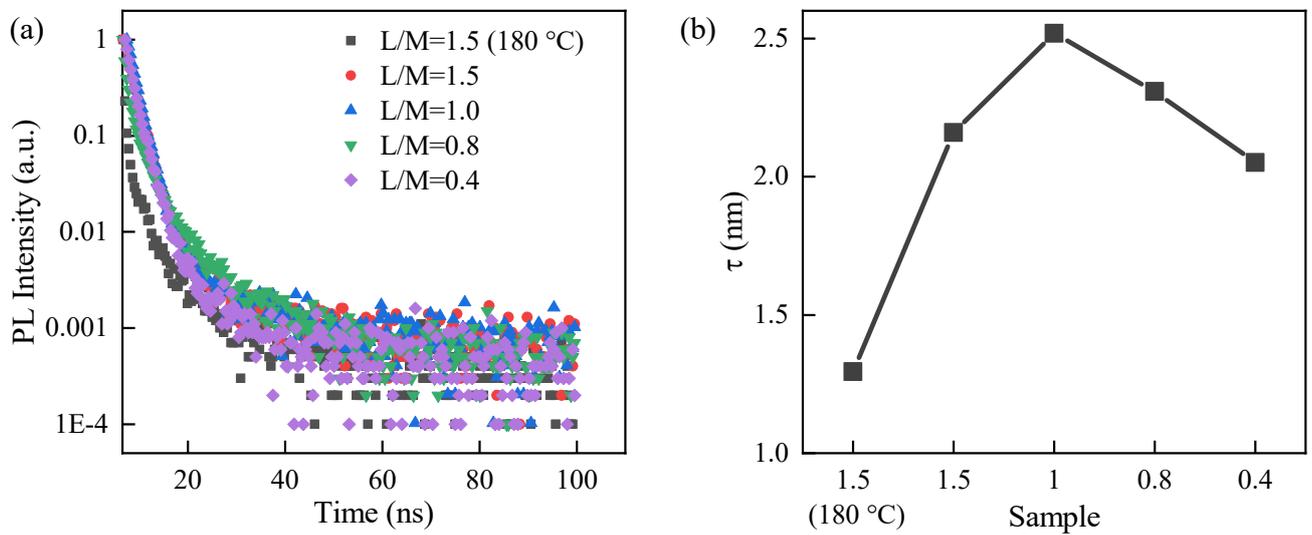


Fig. S13. (a) TRPL decay curves of all synthesized UiO-66 crystals. (b) Average carrier lifetimes ( $\tau$ ) extracted through the bi-exponential fitting of the TRPL curves.

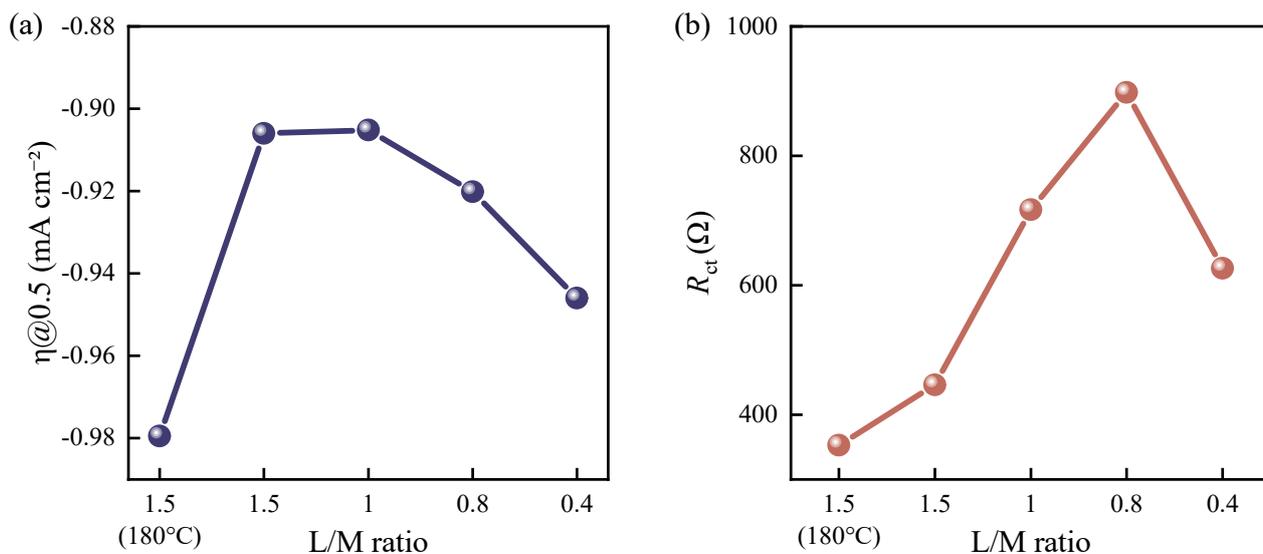


Fig. S14. (a) The overpotential at a fixed current density ( $\eta@0.5$  mA cm<sup>-2</sup>) and (b) the charge transfer resistance ( $R_{ct}$ ) of UiO-66 crystals.

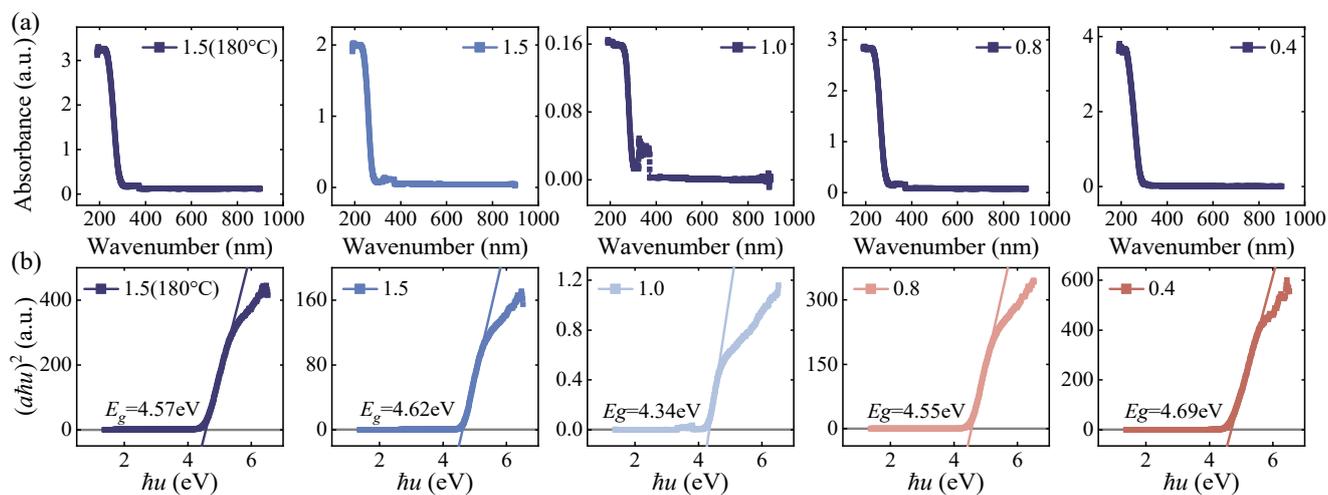


Fig. S15. (a) DR UV-vis spectra and (b) Tauc plots of UiO-66 crystals.

Table S1. Comparative study on hydrogen evolution performance of MOF-based piezocatalysis.

MOF	Conditions	Hydrogen evolution rate	Ref.
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( $\mu\text{mol}\cdot\text{g}^{-1}\cdot\text{h}^{-1}$ )			
Defective UiO-66(Hf)	220 W, 40 kHz Methanol + AgNO <sub>3</sub>	UiO-66(L/M=1.5 180 °C): 23.2 UiO-66(L/M=1.0): 39.6	This work
UiO-66(Zr) UiO-66(Zr)-F <sub>4</sub>	110 W, 40 kHz Na <sub>2</sub> SO <sub>3</sub> + H <sub>2</sub> O	UiO-66(Zr): ~0.1 UiO-66(Zr)-F <sub>4</sub> : 35.7	S5
Pt/UiO-66-NH <sub>2</sub> (Zr)	200 W, 53 kHz	~0.1	S6
Pt/UiO-66-NH <sub>2</sub> (Hf)	200 W, 53 kHz	~0.1	S6
NH <sub>2</sub> -MIL-125 Cu-NH <sub>2</sub> -MIL-125	150 W, 50 kHz Acetonitrile, Water H <sub>2</sub> PtCl <sub>6</sub> ·6H <sub>2</sub> O	MIL-125: ~0.1 NH <sub>2</sub> -MIL-125: 5.1 Cu-NH <sub>2</sub> -MIL-125: 5.5	S7

Table S2. Position of conduction band ( $C_B$ ) and position of flat band ( $V_B$ ) relative to SHE and bandgap ( $E_g$ ) of UiO-66 crystals.

Sample	$C_B$ (V)	$V_B$ (V)	$E_g$ (eV)
1.5 (180 °C)	-0.913	3.657	4.57
1.5	-0.773	3.847	4.62
1.0	-1.033	3.307	4.34
0.8	-0.903	3.647	4.55
0.4	-0.803	3.887	4.69

Table S3. Experimental conditions used for pristine and defective UiO-66. The reagents selected were as follows: the metal salt (M) was HfCl<sub>4</sub> (Aldrich, 98%), the organic ligand (L) was 1,4-dicarboxylic acid (BDC, Aldrich, 98%), and the modulator (Mod) was formic acid (FA, Alfa Aesar, 97%), N,N-dimethylformamide

(DMF, Scharlau).

Sample	Temperature (°C)	L/M ratio	Mod/L	M (mmol)	L (mmol)	Mod (mmol)	DMF <sup>S8</sup> (mL)	V <sup>S8</sup> (mL)
1.5 (180 °C)	180	1.5	94.06	0.24	0.36	33.9	4	5
1.5	120	1.5	94.06	0.24	0.36	33.9	4	5
1.0	120	1.0	112.9	0.3	0.3	33.9	4	5
0.8	120	0.8	127.0	0.3	0.3	33.9	4	5
0.4	120	0.4	197.5	0.4	0.2	33.9	4	5

## Reference

- S1. Tatay, S.; Martinez-Gimenez, S.; Rubio-Gaspar, A.; Gomez-Oliveira, E.; Castells-Gil, J.; Dong, Z.; Mayoral, A.; Almora-Barrios, N.; Padial, N. M.; Marti-Gastaldo, C., Synthetic control of correlated disorder in UiO-66 frameworks. *Nature Communications* **2023**, *14* (1), 6962.
- S2. Cliffe, M. J.; Wan, W.; Zou, X.; Chater, P. A.; Kleppe, A. K.; Tucker, M. G.; Wilhelm, H.; Funnell, N. P.; Coudert, F.-X.; Goodwin, A. L., Correlated defect nanoregions in a metal–organic framework. *Nature Communications* **2014**, *5* (1), 4176.
- S3. Sivula, K., Mott–Schottky Analysis of Photoelectrodes: Sanity Checks Are Needed. *ACS Energy Letters* **2021**, *6* (7), 2549-2551.
- S4. Gelderman, K.; Lee, L.; Donne, S. W., Flat-Band Potential of a Semiconductor: Using the Mott–Schottky Equation. *Journal of Chemical Education* **2007**, *84* (4), 685-688.
- S5. Zhao, S.; Liu, M.; Zhang, Y.; Zhao, Z.; Zhang, Q.; Mu, Z.; Long, Y.; Jiang, Y.; Liu, Y.; Zhang, J.; Li, S.; Zhang, X.; Zhang, Z., Harvesting mechanical energy for hydrogen generation by piezoelectric metal–organic frameworks. *Materials Horizons* **2022**, *9* (7), 1978-1983.
- S6. Zhang, C.; Lei, D.; Xie, C.; Hang, X.; He, C.; Jiang, H.-L., Piezo-Photocatalysis over Metal–Organic Frameworks: Promoting Photocatalytic Activity by Piezoelectric Effect. *Advanced Materials* **2021**, *33* (51), 2106308.
- S7. Hu, H.; Li, X.; Zhang, K.; Yan, G.; Kong, W.; Qin, A.; Ma, Y.; Li, A.; Wang, K.; Huang, H.; Sun, X.; Ma, T., Dual Modification of Metal–Organic Frameworks for Exceptional High Piezo-Photocatalytic Hydrogen

Production. *Advanced Materials* **2025**, 37 (20), 2419023.

S8. Zhang, Y.; Feng, K.; Song, M.; Xiang, S.; Zhao, Y.; Gong, H.; Ni, F.; Dietrich, F.; Fulanović, L.; Zhuo, F.; Buntkowsky, G.; Frömling, T.; Zhang, D.; Bowen, C.; Rödel, J., Dislocation-engineered piezocatalytic water splitting in single-crystal BaTiO<sub>3</sub>. *Energy & Environmental Science* **2025**, 18 (2), 602-612.