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Defect-Passivated InGaZnO/In₂O₃ Stacked Thin-Film Transistor with Visible-Light-Assisted Recovery for Room Temperature ppb-Level NO₂ Detection

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Preparation process of IGZO/In₂O₃-Based TFT



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Cr/Au deposition

- 23 Fig. S1. The preparation process of the $IGZO/In_2O_3$ TFT through a bottom-up process.
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25 **Table S1.** PBS stability of the In_2O_3 TFT.

PBS duration (min)	0	5	10	30	60
Threshold voltage (V)	-15.28 ± 0.60	$\textbf{-14.90}\pm0.62$	$\textbf{-14.46} \pm 0.73$	$\textbf{-13.85}\pm0.82$	$\textbf{-13.03}\pm0.78$
Saturated on-state	96.85 ± 7.92	96.31± 6.78	96.10 ± 6.63	93.64 ± 8.95	91.79 ± 9.13
current (µA)					

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27 Table S2. PBS stability of the $IGZO/In_2O_3$ TFT.

PBS duration (min)	0	5	10	30	60
Threshold voltage (V)	1.70 ± 0.58	1.88 ± 0.55	2.01 ± 0.72	2.19 ± 0.69	2.38 ± 0.67
Saturated on-state	296.47 ± 17.79	291.05 ± 20.61	286.81 ± 19.46	280.57 ± 22.38	277.12 ± 23.20
current (µA)					



30 Fig. S2. The configuration for gas sensing of the prepared In_2O_3 and $IGZO/In_2O_3$ TFTs.



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- 34 (a) 50° C, (b) 75° C and (c) 100° C under 50-500 ppb NO₂ concentrations. Corresponding hysteresis
- 35 curves of IGZO/In $_2O_3$ TFTs at (d) 50°C, (e) 75°C and (f) 100°C.
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38 Fig. S4 Transfer characteristics of In₂O₃ TFT and IGZO/In₂O₃ TFT, tested repeatedly for 500 ppb NO₂

39 gas sensing, showing the response after exposure to 500 ppb NO2 and subsequent recovery following

40 UV or Blue LED stimulation.



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 $43 \quad \text{and IGZO/In}_2O_3 \text{ TFT with } V_G \text{ and } V_D \text{ fixing at } 10 \text{ V and } 1 \text{ V, respectively. (b) Mean and standard}$

 $44 \quad \text{deviation of response values of } In_2O_3 \text{ and } IGZO/In_2O_3 \text{ TFT to } 50 \text{ ppb } NO_2 \text{ for } 5 \text{ consecutive cycles at}$

45 25 °C.

46



48 Fig. S6. (a) In₂O₃, and (b) IGZO/In₂O₃ TFT: A typical sensing cycle in response to 50 ppb NO₂. (c)

49 Response and recovery times statistically calculated from five consecutive 50 ppb NO₂ sensing cycles.



51 Fig. S7. Dynamic response to 500 ppb NO₂ of the In_2O_3 and $IGZO/In_2O_3$ TFT with V_G and V_D fixing at

52 10 V and 1 V, respectively.





55 Fig. S8. The selectivity of the In_2O_3 TFT.



- 58 Fig. S9. The selectivity of the $IGZO/In_2O_3$ TFT.



Fig. S10 The O 1s spectra corresponding to the unannealed In_2O_3 thin film.

	Unexposed	50 ppb NO ₂	100 ppb NO ₂	$200 \text{ ppb } NO_2$	500 ppb NO_2
μ_{FE} (cm ² /Vs)	7.55 ± 0.35	7.26 ± 0.68	6.83 ± 0.45	6.53 ± 0.56	5.82 ± 0.70
Vth (V)	$\textbf{-15.28}\pm0.60$	-14.58 ± 0.52	-14.28 ± 0.50	-13.78 ± 0.57	$\textbf{-10.08} \pm 0.84$
On-state current	7.74 ± 0.36	7.62 ± 0.43	7.12 ± 0.33	6.88 ± 0.71	5.45 ± 0.61
(× 10 ⁻⁵ A)					
Off-state current	1.78 ± 1.03	1.41 ± 0.94	1.29 ± 0.86	0.61 ± 0.59	0.37 ± 0.25
(× 10 ⁻¹¹ A)					

66 Table S3. Detailed electrical parameters of In₂O₃ TFT after response to 50-500 ppb NO₂.

68 Table S4. Detailed electrical parameters of IGZO/In₂O₃ TFT after response to 50-500 ppb NO₂.

	Unexposed	50 ppb NO ₂	100 ppb NO ₂	$200 \text{ ppb } \mathrm{NO}_2$	500 ppb NO ₂
$\mu_{FE} \left(cm^2/VS \right)$	18.79 ± 0.83	18.06 ± 0.98	17.37 ± 1.26	10.74 ± 1.52	7.14 ± 1.13
Vth (V)	1.70 ± 0.58	2.70 ± 0.50	3.60 ± 0.43	5.11 ± 0.77	6.67 ± 0.91
On-state current	2.99 ± 0.14	2.85 ± 0.17	2.49 ± 0.12	1.25 ± 0.26	0.49 ± 0.06
(× 10 ⁻⁴ A)					
Off-state current	2.51 ± 1.45	2.39 ± 1.33	2.09 ± 1.21	1.04 ± 1.08	0.71 ± 0.35

(× 10⁻¹¹A)



Fig. S11. The transfer characteristics of the (a) In_2O_3 and (b) IGZO/In_2O_3 TFTs under 20-80% RH conditions at room temperature. (c) The statistics of ΔV_T that compared to V_T measured in dry air.



75 Fig. S12. (a) The response of the In_2O_3 TFTs towards 500 ppb NO_2 under dry air, 20% RH, (b) 40%

- 76 RH, (c) 60% RH, and (d) 80% RH.
- 77





- 80 (b) 40% RH, (c) 60% RH, and (d) 80% RH.
- 81



83 Fig. S14. A summary of ΔV_T of the In₂O₃ and IGZO/In₂O₃ TFTs towards 500 ppb NO₂ under 20-

- 84 80% RH at room
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Fig. S15. NBS stability under the condition of fixed V_{GS} = -20 V and V_{DS} =1 V within 60 minutes of (a) In₂O₃ TFTs and (b) IGZO/In₂O₃ TFTs.

	Sensor fabrication	NO ₂ Conc.	Response	Temperature	Ref.
		(ppb)	(Ra-Rg)/Ra	(°C)	
Hydroxy-Rich-Surface	RF magnetron sputtering	500	~2500%	100	[46]
In ₂ O ₃ FET					
O-deficient In ₂ O ₃ TFT	RF magnetron sputtering	100	~21.4%	160	[47]
Floating-gate In ₂ O ₃ FET	RF magnetron sputtering	500	~1800%	100	[48]
Floating-gate In ₂ O ₃ TFT	RF magnetron sputtering	300	~69%	75	[49]
SnO FET	Liquid metal method	500	~20%	RT	[50]
IGZO FET	RF magnetron sputtering	10	~18%	~150	[51]
Ni ₃ (HHTP) ₂ FET	In-situ grown	100	~21.5%	RT	[52]
IGZO/In2O3 TFT	RF magnetron sputtering	50	~18.6%	RT	This work

91 Table S5. Comparison of sensing performance based on metal oxide field effect transistor gas sensors.

93 NBS characteristics of In₂O₃ TFT and IGZO/In₂O₃ TFT. V_{GS} was set to -20 V, and V_{DS} was maintained at 1 V. After 1 hour of NBS stress, the threshold voltage negative 94 shift ΔV_T of In₂O₃ and IGZO/In₂O₃ TFTs were -7.34 V and -2.79 V, respectively, as 95 shown in Figure S9. This indicates that the threshold voltage drift of the IGZO/In₂O₃ 96 TFT under negative bias stress is significantly smaller than that of the In₂O₃ TFT. This 97 suggests that the IGZO/In2O3 TFT experiences less performance degradation and has 98 better stability. Compared to the IGZO/In₂O₃ TFT, the defects in the oxide 99 100 semiconductor film of the In₂O₃ TFT are more easily activated under negative bias. In₂O₃, as a single material, have more intrinsic defects, and these defect states can 101 102 quickly capture charge carriers under negative bias, resulting in a significant shift in

threshold voltage. On the other hand, the IGZO/In₂O₃ TFT uses a composite structure 103 of IGZO and In₂O₃, which typically exhibits better performance by improving the 104 crystal structure or reducing the formation of certain defects. Overall, the IGZO/In₂O₃ 105 TFT shows better NBS performance, mainly due to its composite film structure, which 106 effectively reduces defect formation and the degradation effect of negative bias on 107 device performance, maintaining better electrical stability and long-term reliability. In 108 contrast, the In₂O₃ TFT, with more defect states, experiences more severe performance 109 degradation under negative bias. 110

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The origin of PBS and the significance of PBS testing. Positive bias stress (PBS) is 113 114 of significance in thin-film transistor technology and PBS testing serves as a method to evaluate the stability and reliability of TFT devices over prolonged operational 115 durations [42-44]. It becomes feasible to simulate voltage stresses that the devices 116 might encounter during real-world usage, such as prolonged operation or stresses 117 induced by specific environmental conditions by subjecting TFTs to positive bias stress. 118 These TFTs exhibited a trend of threshold voltage positive shift and gradual decrease 119 120 in drain current and two reasons account for this phenomenon. Firstly, lattice mismatch occurring at the interface between In₂O₃ or IGZO thin film layer and the insulating layer 121 SiO₂, inevitably leads to defect states such as unsaturated dangling bonds. Part free 122 electrons are captured by interface defect states under PBS conditions, resulting in a 123 reduction in the concentration of free electrons in the channel layer [40]. Therefore, 124 additional gate voltage, i.e., an increase in threshold voltage ΔV_T , is required to 125

overcome the effects caused by these captured electrons, thereby mitigating the positive drift of threshold voltage and decrease in drain current. Secondly, the sensing layer In₂O₃ is highly sensitive to oxygen due to its strong electronegativity, and the oxygen adsorbed on the thin film surface captures electrons from the channel layer surface. The reaction equation during low-temperature measurements (below 100°C) is as follows:

$$0_2 + e^- \rightarrow 0_2^- \#(1)$$

Under the combined influence of oxygen in the air and interface defect states, the electron concentration in the channel of the fabricated depletion-type thin-film transistors (DTFTs) is reduced, resulting in a decrease in the additional negative voltage required to deplete electrons, thereby causing the threshold voltage to shift forward.

The influence of temperature on output current and NO₂ sensing. In₂O₃ and 137 IGZO/In₂O₃ TFTs are both n-type transistors. First, the concentration of free electrons 138 increases with rising temperature because the thermal excitation effect causes more 139 electrons to transition from the valence band to the conduction band, thereby enhancing 140 the electrical conductivity. Second, at lower temperatures, the mobility of charge 141 carriers in the semiconductor material is relatively low due to the higher scattering 142 frequency of carriers within the channel, which results in lower conductivity. As the 143 temperature increases, the motion of charge carriers becomes more active, scattering 144 events decrease, and mobility improves. These two factors contribute to an increase in 145 146 leakage current. Additionally, the negative shift in threshold voltage can be attributed to the increased number of free carriers, which allows the device to conduct at a lower gate voltage. NO₂ molecules interact with the surface of n-type semiconductors, capturing surface free electrons and reducing the electron concentration in the channel, which consequently decreases its conductivity. This phenomenon serves as the foundation for NO₂ gas sensing. Specifically, a higher concentration of NO₂ results in the capture of more electrons, leading to a more significant reduction in conductivity. This change manifests in two prominent ways, a positive shift in the threshold voltage and a decrease in the output current.