

# Lattice Perturbation by Optically Inert Sc<sup>3+</sup> Grants NaGdF<sub>4</sub> Upconversion Nanocrystals Giant Emission and Excitation-Gated Thermal Response

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# Support Information

## Experimental section

### Materials

Gadolinium (III) chloride ( $\text{GdCl}_3$ , 99.9%), ytterbium (III) chloride hexahydrate ( $\text{YbCl}_3 \cdot 6\text{H}_2\text{O}$ , 99.9%), thulium (III) chloride hexahydrate ( $\text{TmCl}_3 \cdot 6\text{H}_2\text{O}$ , 99.9%), and holmium (III) chloride hexahydrate ( $\text{HoCl}_3 \cdot 6\text{H}_2\text{O}$ , 99.9%) were purchased from Macklin. Erbium (III) chloride hexahydrate ( $\text{ErCl}_3 \cdot 6\text{H}_2\text{O}$ , 99.9%), scandium (III) chloride ( $\text{ScCl}_3$ , 99.9%), sodium hydroxide (NaOH, 98%), ammonium fluoride ( $\text{NH}_4\text{F}$ , 98%), and cyclohexane (AR, 99.5%) were sourced from Sigma-Aldrich. Oleic acid (90%) and 1-octadecene (90%) were acquired from Thermo Fisher, unless otherwise stated. All chemicals were used as received without further purification.

### Synthesis procedure

$\text{Sc}^{3+}$  and lanthanide ion ( $\text{Er}^{3+}$ ,  $\text{Yb}^{3+}/\text{Er}^{3+}$ ,  $\text{Yb}^{3+}/\text{Tm}^{3+}$ , or  $\text{Yb}^{3+}/\text{Ho}^{3+}$ ) co-doped  $\beta\text{-NaGdF}_4$  nanocrystals were prepared via a modified thermal decomposition route. In a typical procedure, 1 mmol of  $\text{LnCl}_3 \cdot 6\text{H}_2\text{O}$  ( $\text{Ln} = \text{Gd}, \text{Sc}$ , and dopants), 6 mL oleic acid, and 15 mL 1-octadecene were combined in a three-neck flask. The mixture was heated to 150 °C under argon for 40 min to yield a clear solution. After cooling, 10 mL of methanol containing 4 mmol  $\text{NH}_4\text{F}$  and 2.5 mmol NaOH was added and stirred vigorously for 40 min. The slurry was then heated to 120 °C under vacuum for 30 min to remove solvents. Under nitrogen, the temperature was rapidly raised to 300 °C and held for 1 h. The resulting nanocrystals were precipitated with ethanol, collected by centrifugation (6,000 rpm), and finally redispersed in cyclohexane or dried at 60 °C for storage.

### HRTEM

TEM samples were prepared by repeated centrifugation and washing (at least 3 times) with an ethanol/cyclohexane mixture. The samples were then ultrasonically dispersed in cyclohexane and drop-cast onto holey copper or molybdenum grids supported by an ultra-thin carbon film, followed by overnight drying. Nano-morphological were carried out using a Talos F200X transmission electron microscope (TEM) equipped with energy-dispersive spectroscopy (EDS), operating at 300 kV.

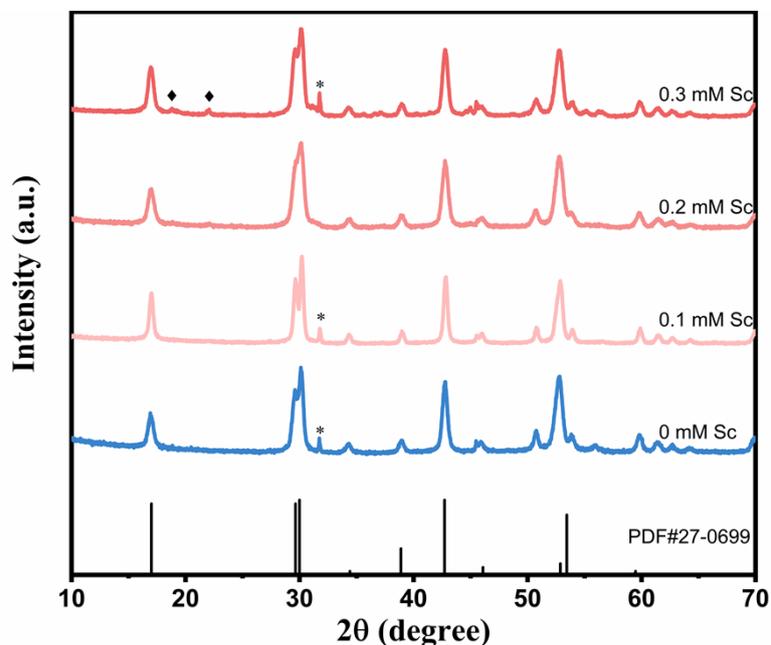
### X-ray characterisation

X-ray diffraction (XRD) was performed for phase analysis of the nanoparticles using a Bruker D8 Advance diffractometer with Cu  $K\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ). Measurements covered a  $2\theta$  range of 10°–80° with a 0.02° step size and 7.27°  $\text{min}^{-1}$  scan rate. Dried nanoparticles were ground into a fine powder, pressed into a flat sample holder to ensure uniform X-ray penetration. Diffraction patterns were analyzed via MDI Jade 6.5 software: phase identification was confirmed against JCPDS reference patterns, and corresponding unit cell parameters were calculated.

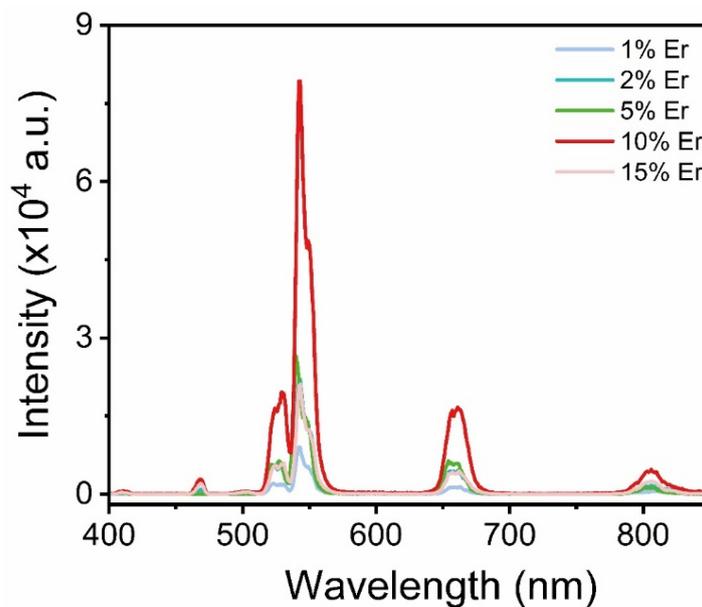
### Optical characterization

The upconversion luminescence (UCL) properties were characterized by steady-state, temperature-dependent, and time-resolved spectroscopy. Steady-state emission spectra were collected using a Horiba FL3-111 fluorescence spectrometer under continuous-wave diode laser excitation at 980 nm (0.296 W) or 1550 nm (0.052 W) with a  $\approx 2$  mm spot at a 45° incidence angle. Emission was detected perpendicularly with a PMT (1 nm resolution) from 400 to 850 nm. Temperature-dependent measurements were performed using a precision temperature-controlled stage (RT–500 K,  $\pm 1$  K). Time-resolved decays were recorded on a Horiba Fluorolog-QM system equipped with pulsed lasers (980/1550 nm, 22 ms pulse width) and a TCSPC module (minimum response time 2666 ns). Decay curves were collected until 10,000 photon counts were reached in the peak channel and were fitted multi-exponentially to extract lifetimes.

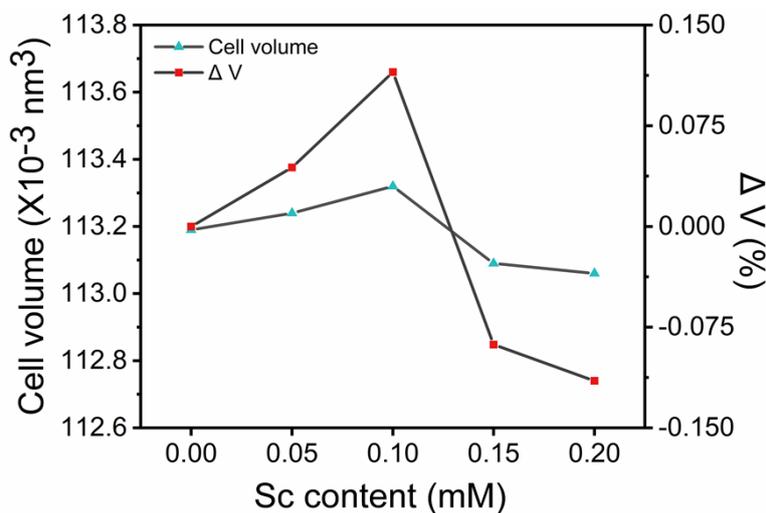
### Supplementary figures:



**Figure.S1** Phase purity of NaGdF<sub>4</sub> nanocrystals with varying Sc<sup>3+</sup> doping concentrations. XRD patterns show that the hexagonal NaGdF<sub>4</sub> phase (JCPDS No. 27-0699) is maintained up to 0.2 mM Sc<sup>3+</sup>, above which ( $\geq 0.3$  mM) characteristic peaks of a Na<sub>3</sub>ScF<sub>6</sub> impurity phase (♦) emerge, defining the doping threshold for host lattice integrity. Peaks marked with \* correspond to residual NaCl (JCPDS No. 05-0628) from synthesis precursors.

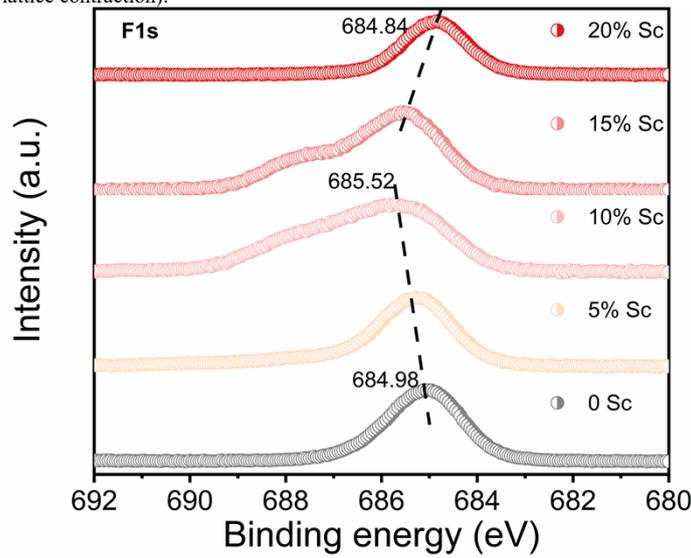


**Figure S2.** Concentration-dependent upconversion luminescence. Emission spectra of NaGdF<sub>4</sub>:20%Sc/x%Er nanocrystals ( $x = 1, 2, 5, 10, 15\%$ ) under 980 nm excitation. The luminescence intensity is enhanced with increasing Er<sup>3+</sup> content up to 10%, but is quenched at 15%, establishing 10% as the optimum for achieving the brightest emission.

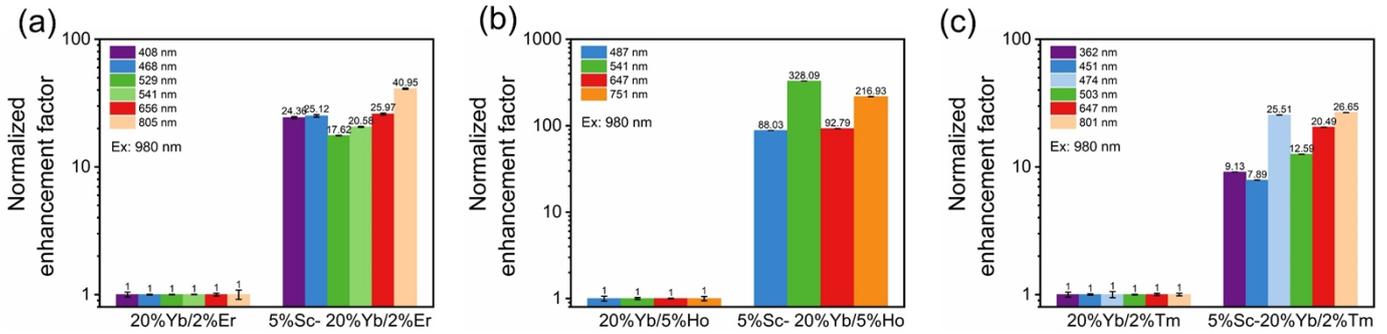


**Figure S3.** Lattice parameters of NaGdF<sub>4</sub>:10%Er as a function of Sc<sup>3+</sup> doping concentration. The unit cell volume exhibits a non-monotonic trend: it initially increases at low Sc<sup>3+</sup> concentrations (<10 mol%) and then decreases at higher concentrations. This behavior supports the proposed dual-site incorporation

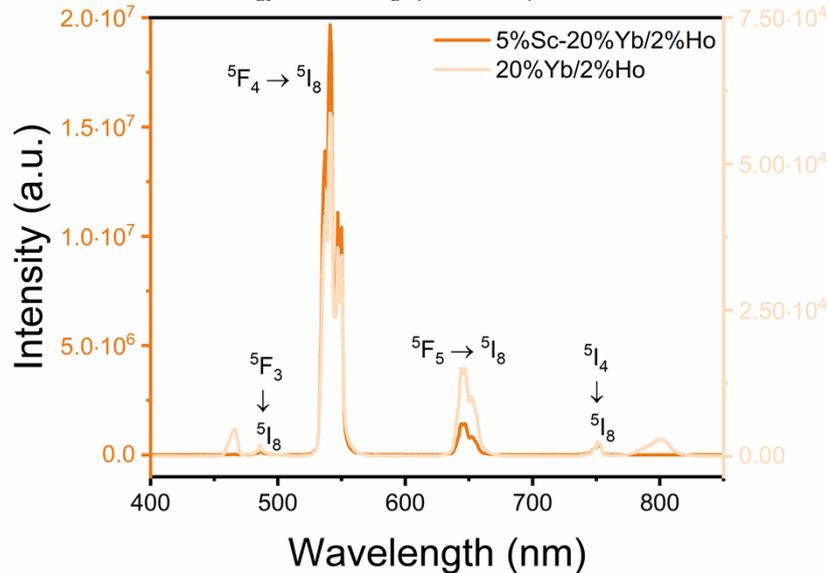
mechanism, where  $\text{Sc}^{3+}$  ions preferentially occupy interstitial sites at low concentrations (causing lattice expansion) and gradually switch to substitutional  $\text{Gd}^{3+}$  sites at higher concentrations (leading to lattice contraction).



**Figure S4.** XPS F 1s spectra of  $\text{NaGdF}_4:10\%\text{Er}$  nanocrystals as a function of  $\text{Sc}^{3+}$  doping concentration. The F 1s binding energy exhibits a non-monotonic shift with increasing  $\text{Sc}^{3+}$  content. The initial increase at low  $\text{Sc}^{3+}$  concentrations ( $<10$  mol%) indicates that interstitial  $\text{Sc}^{3+}$  polarizes surrounding  $\text{F}^-$  ions, withdrawing electron density. The subsequent decrease at higher concentrations ( $>10$  mol%) reflects relaxation of the local distortion as  $\text{Sc}^{3+}$  transitions to substitutional  $\text{Gd}^{3+}$  sites. This trend correlates with the lattice parameter evolution (Fig. S3) and provides direct chemical evidence for  $\text{Sc}^{3+}$ -induced modification of the local crystal field around  $\text{Er}^{3+}$  ions.



**Figure S5.** The enhanced upconversion emission of  $\text{NaGdF}_4:\text{Yb}/\text{Er}$ ,  $\text{Yb}/\text{Ho}$ , and  $\text{Yb}/\text{Tm}$  systems with 5%  $\text{Sc}^{3+}$  doping under 980 nm excitation.  $\text{Sc}^{3+}$  incorporation significantly boosts the emission intensities of all three activator ions (17.62-40.95-fold for Er, 88.03-328.09-fold for Ho, and 7.89-26.65-fold for Tm), confirming that  $\text{Sc}^{3+}$ -induced crystal field modification is a versatile strategy for enhancing upconversion performance across different lanthanide-doped systems.



**Figure.S6** The upconversion fluorescence emission spectrum of  $\text{NaGdF}_4:20\%\text{Yb}/2\%\text{Ho}$  and  $\text{NaGdF}_4:5\%\text{Sc}/20\%\text{Yb}/2\%\text{Ho}$  excited at 980 nm were characterized. Experimental results demonstrate that the introduction of 5% Sc dopant can remarkably boost the intensity of upconversion emission peaks by 2-3 orders of magnitude. Specifically, the emission at 541 nm shows a prominent enhancement effect, while the intensities of other emission peaks are relatively weak, which endows the sample with distinct single-wavelength upconversion emission features.

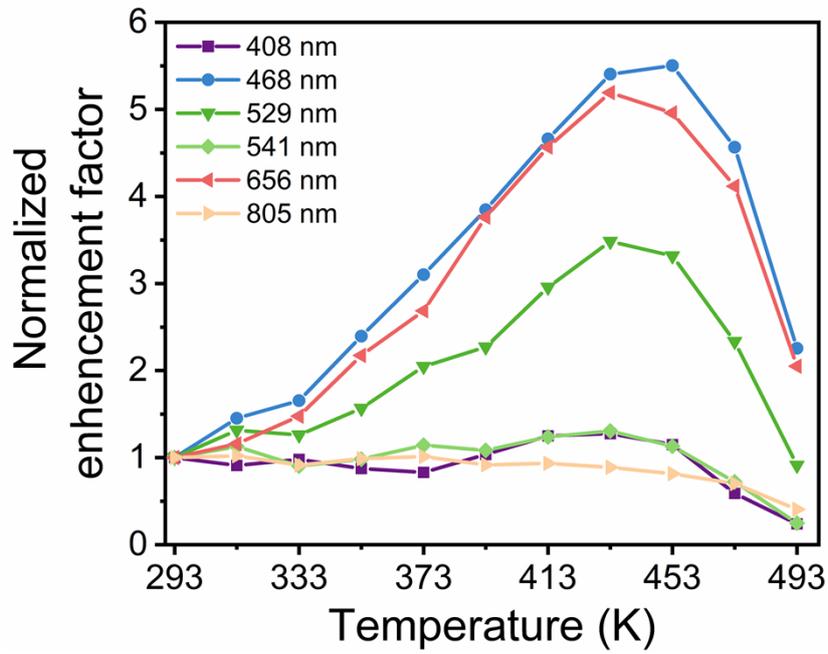


Figure S7. Temperature-dependent upconversion emission intensity of NaGdF<sub>4</sub>:10%Er/5%Sc under 1550 nm excitation.

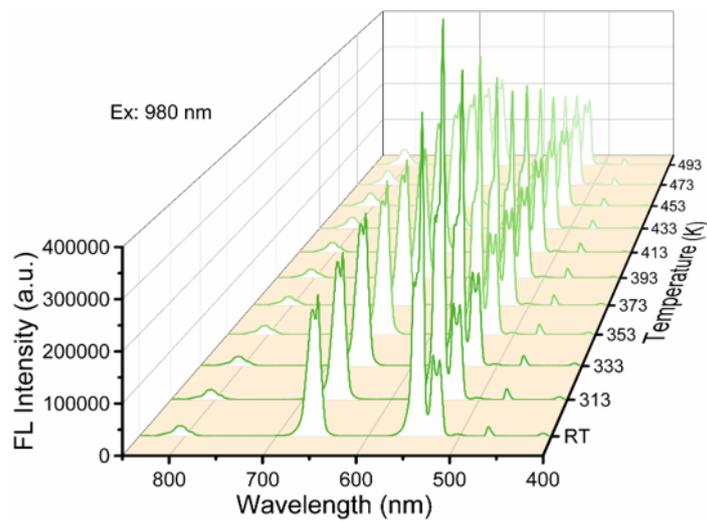


Figure S8. Temperature-dependent (RT-493K) upconversion emission spectra of NaGdF<sub>4</sub>:10%Er/5%Sc nanocrystals under 980 nm excitation.

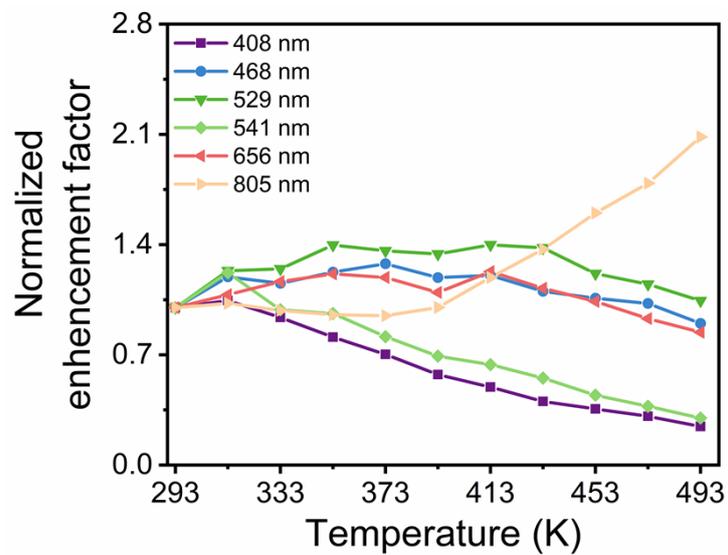


Figure S9. Temperature-dependent upconversion emission intensity of NaGdF<sub>4</sub>:10%Er/5%Sc under 980 nm excitation.

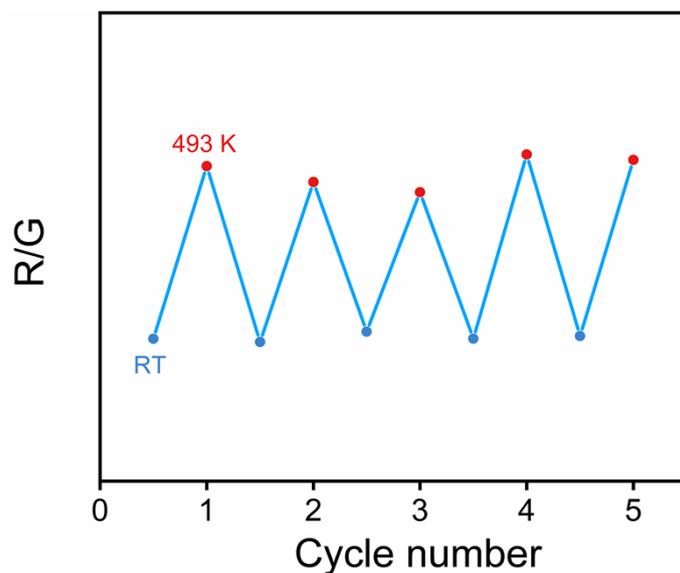


Figure S10. Stability assessment of the colorimetric response under 1550 nm excitation.

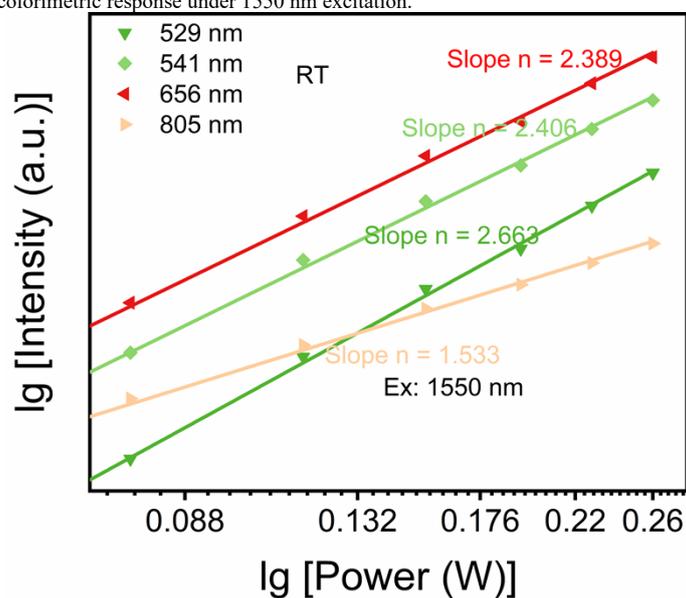


Figure S11. Double-logarithmic plots of upconversion intensity ( $I_{up}$ ) versus laser power ( $P$ ) under 1550 nm excitation at room temperature.

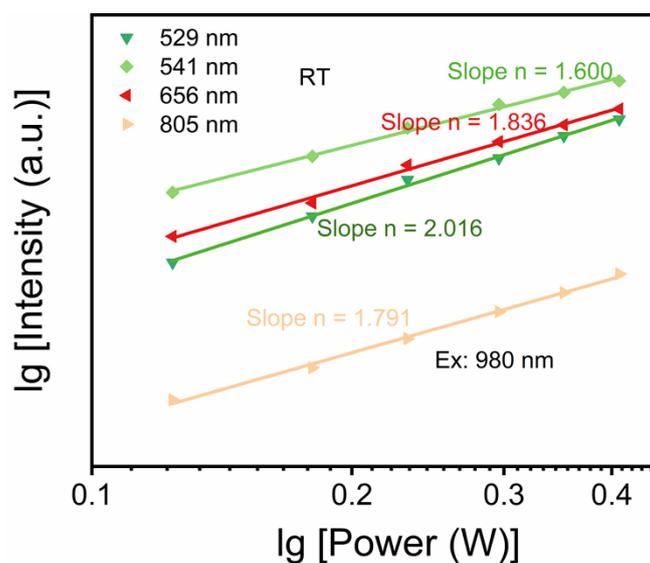
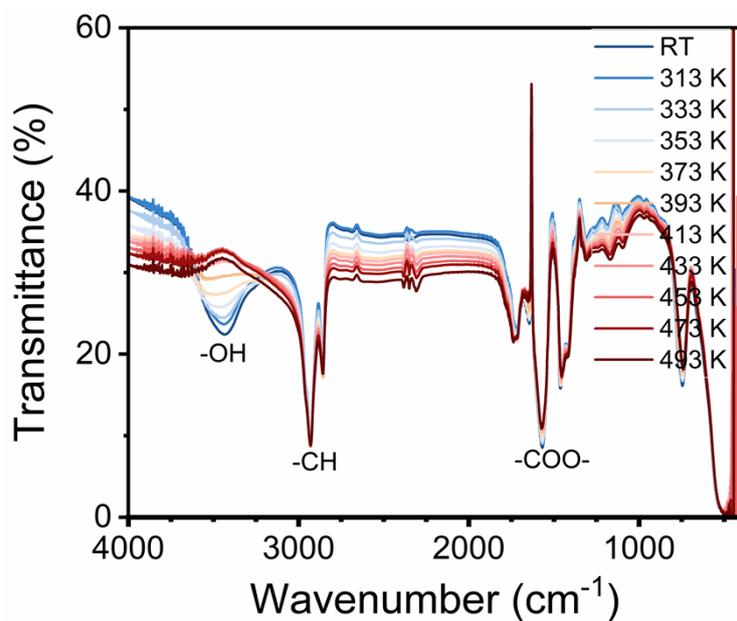
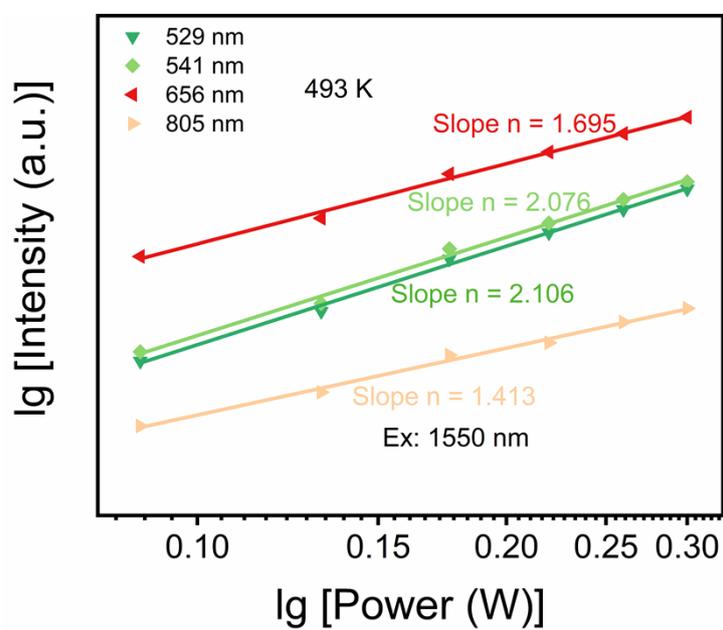


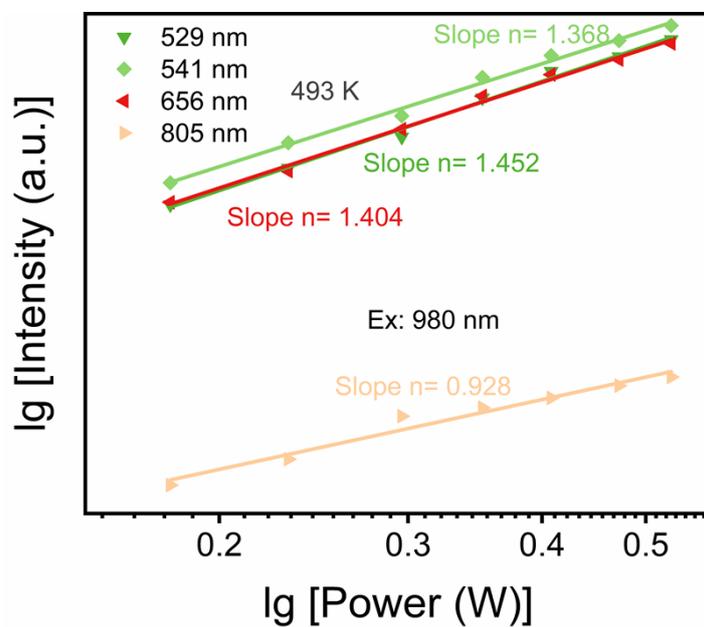
Figure S12. Double-logarithmic plots of upconversion intensity ( $I_{up}$ ) versus laser power ( $P$ ) under 980 nm excitation at room temperature.



**Figure S13.** In situ Fourier transform infrared spectra of NaGdF<sub>4</sub>:10% Er/5% Sc nanoparticles recorded at temperatures ranging from room temperature to 493 K.



**Figure S14.** Double-logarithmic plots of  $I_{\text{up}}$  vs.  $P$  under 1550 nm excitation measured at 493 K.



**Figure S15.** Double-logarithmic plots of  $I_{\text{up}}$  vs.  $P$  under 1550 nm excitation measured at 493 K.

### Supplementary figures:

**Table.S1** The chromaticity coordinates corresponding to the upconversion fluorescence emission of NaGdF<sub>4</sub>:5%Sc/10%Er as a function of temperature were investigated under 980 nm and 1550 nm excitation, respectively. Results show that the sample exhibits a slight color variation under 980 nm excitation, while the upconversion emission color undergoes a remarkably large variation with temperature under 1550 nm excitation.

| Temperature | CIE chromaticity coordinates (980 nm Ex) | CIE chromaticity coordinates (1550 nm Ex) |
|-------------|--|---|
| RT          | (0.290, 0.677)                           | (0.304, 0.661)                            |
| 313 K       | (0.285, 0.682)                           | (0.301, 0.662)                            |
| 333 K       | (0.290, 0.674)                           | (0.320, 0.636)                            |
| 353 K       | (0.287, 0.675)                           | (0.337, 0.617)                            |
| 373 K       | (0.288, 0.672)                           | (0.336, 0.614)                            |
| 393 K       | (0.287, 0.669)                           | (0.360, 0.586)                            |
| 413 K       | (0.291, 0.664)                           | (0.360, 0.585)                            |
| 433 K       | (0.289, 0.665)                           | (0.360, 0.583)                            |
| 453 K       | (0.291, 0.569)                           | (0.363, 0.575)                            |
| 473 K       | (0.289, 0.658)                           | (0.380, 0.553)                            |
| 493 K       | (0.288, 0.657)                           | (0.406, 0.519)                            |

