

Supplementary Information for
**The Role of Sulfur Vacancies on FeS₂(100) for NO Dissociative Adsorption: a
Combined in-situ SR-XPS and DFT Calculation**

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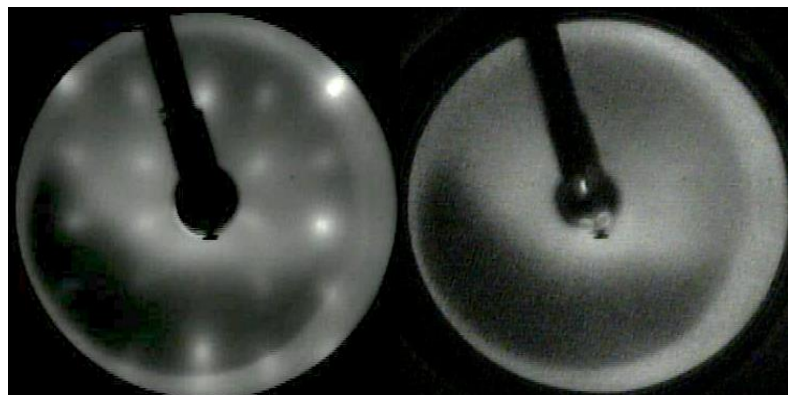


Fig S1. (1×1) LEED pattern of the low- S_{vac} (left), and high- S_{vac} (right) $FeS_2(100)$ surfaces, recorded at an electron beam energy of 110 eV.

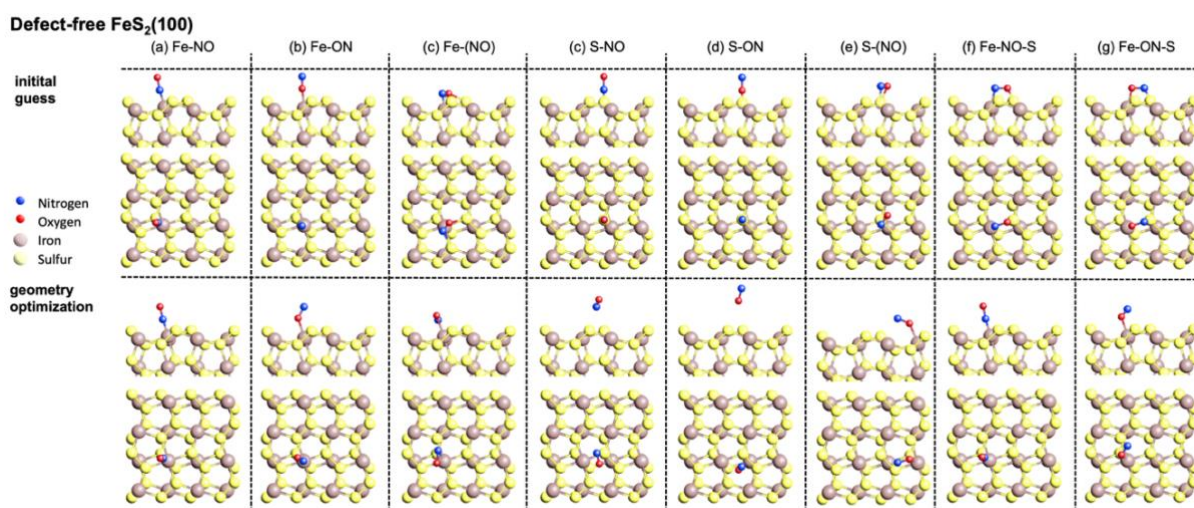


Fig S2. Side view (upper) and top view (lower) of optimized geometries of defect-free $FeS_2(100)$ surfaces for various adsorbate configurations. Top and bottom rows represent the initial adsorption geometries and the structures after geometry optimization, respectively. Color scheme: nitrogen (blue), oxygen (red), iron (brown), and sulfur (yellow).

Table S1. Calculated adsorption energy (ΔE_{ads}), N–O bond length ($d(N-O)$), and charge transfer (ΔQ , Mulliken analysis) for NO adsorption on defect-free $FeS_2(100)$ surfaces with different initial adsorption configurations on defect-free $FeS_2(100)$ surfaces. The most stable configurations are highlighted in red.

Initial guess	ΔE_{ads} (eV)	$d(N-O)$	ΔQ of NO (e)
Fe-NO	-2.01	1.19	0.09
Fe-ON	-0.95	1.22	0.15
Fe-(NO)	-0.94	1.19	0.19
S-NO	-0.40	1.20	0.09
S-ON	-0.23	1.20	0.05
S-(NO)	-0.37	1.21	0.08
Fe-NO-S	-2.01	1.19	0.08
Fe-ON-S	-0.93	1.22	0.15

Defective FeS₂(100)

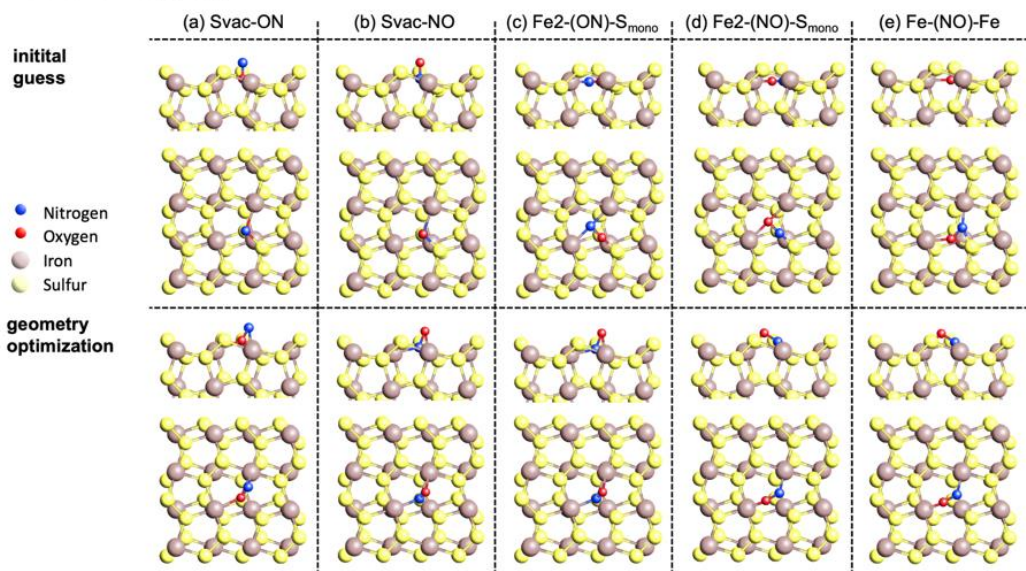
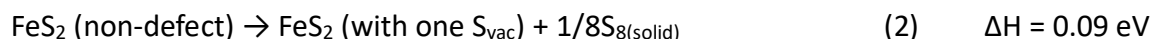
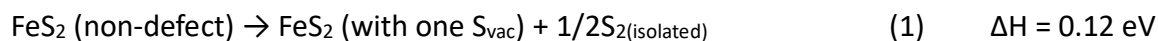


Fig S3. Side view (upper) and top view (lower) of optimized geometries of defective FeS₂(100) surfaces with a sulfur vacancy (Svac) for various NO adsorption configurations. Top and bottom rows represent the initial adsorption geometries and the structures after geometry optimization, respectively. Color scheme: nitrogen (blue), oxygen (red), iron (brown), and sulfur (yellow).

Table S2. Adsorption energy (ΔE_{ads}), N–O bond length ($d(\text{N–O})$), and charge transfer (ΔQ , Mulliken analysis) for NO adsorption on defective FeS₂(100) surfaces with a sulfur vacancy (Svac) in various configurations on defective FeS₂(100) surfaces. The most stable configurations are highlighted in red.

Initial gauss	ΔE_{ads} (eV)	$d(\text{N–O})$	ΔQ of NO (e)
Svac-ON	-1.75	1.29	0.37
Svac-NO	-3.09	1.31	0.40
Fe2-(ON)-S_{mono}	-3.10	1.31	0.39
Fe2-(NO)-S _{mono}	-2.86	1.27	0.35
Fe-(NO)-Fe	-2.84	1.27	0.36

The sulfur-vacancy formation energies for the FeS₂(100) surface were calculated, and we obtained $\Delta H = 0.12$ eV based on eqn (1) and 0.09 eV based on eqn (2). These values suggest that sulfur-vacancy formation is thermodynamically accessible. We note that in our experiments sulfur vacancies were mainly created by Ar⁺ sputtering (non-thermal process).



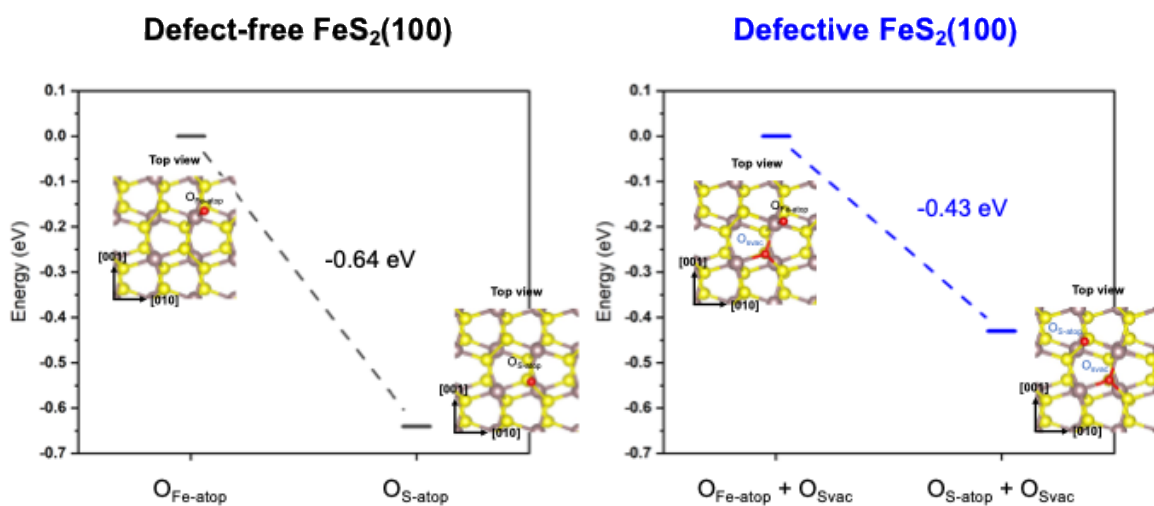


Fig S4. DFT-D3 calculated relative stabilities of atomic oxygen adsorbed at Fe-atop and S-atop sites on defect-free and defective $\text{FeS}_2(100)$ surfaces. On both surfaces, oxygen adsorption at the S-atop site is energetically more favourable than at the Fe-atop site, with energy differences of -0.64 eV for the defect-free surface and -0.43 eV for the defective surface. Representative optimized adsorption geometries are shown as insets.

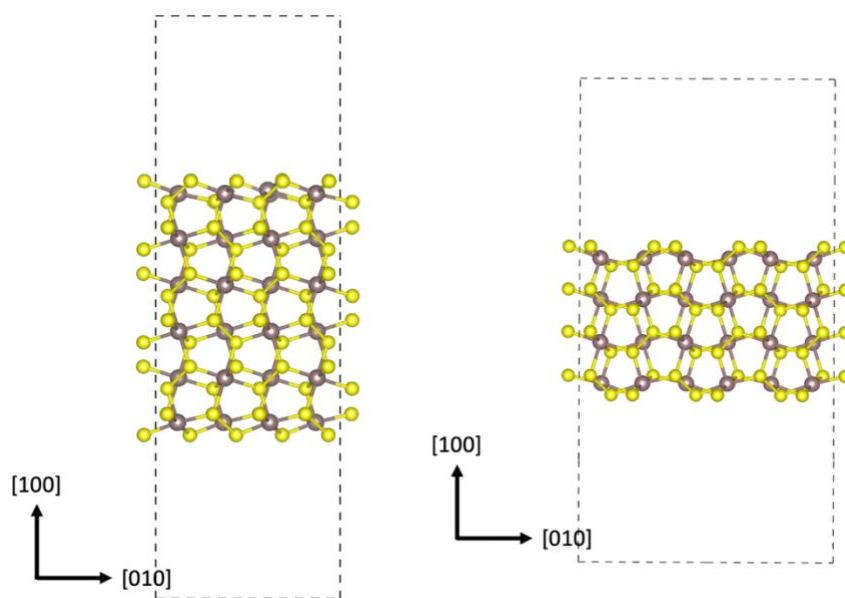


Fig S5. DFT-D3 calculated $(3 \times 2 \times 2)$ slab model for geometry optimizations (**left**) and $(2 \times 3 \times 3)$ slab model for absolute binding energy calculations (**right**)