

# **A Direct-Band-Gap Planar BeN<sub>2</sub> Monolayer with High Mobility and Bilayer-Enhanced Photovoltaic Efficiency**

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## **Computational details of the CALYPSO method**

Crystal structure searches of Be<sub>x</sub>N<sub>y</sub> ( $1 \leq x, y \leq 4$ ) monolayers were conducted with a maximum of four formula units for each stoichiometry using the Crystal structure AnaLYsis by Particle Swarm Optimization (CALYPSO) method. In the first generation, a population of structures corresponding to certain space group symmetries was randomly generated. Starting from the second generation, 60% of the structures with the lowest enthalpies from the previous generation were selected to generate the next generation using Particle Swarm Optimization (PSO) operators. The remaining 40% of structures in the new generation were generated randomly. A structure fingerprinting technique of bond characterization matrix is applied to the generated structures, so that identical structures are strictly forbidden. These procedures significantly enhance the diversity of the generated structures, which is critical for improving the efficiency of the global structural search. The search was considered converged when approximately 1000 successive structures had been generated after the lowest-energy structure was found.

## **Carrier mobility calculations**

The carrier mobility of two-dimensional materials based on the deformation potential (DP) theory by Bardeen and Shockley (B-S) formula [1-3]:

$$\mu_{2D} = \frac{e\hbar^3 C_{2D}}{k_B T m^* m_d (E_1)^2}$$

Here,  $C_{2D}$ ,  $E_1$  and  $m^*$  represent the elastic modulus of the longitudinal strain, the deformation potential constant of the VBM for holes or CBM for electrons, and the carrier effective mass in the transport direction, respectively. The  $C_{2D}$  is defined as  $C_{2D} = 2(E - E_0) / S_0 (\Delta l / l_0)^2$ , where  $E$ ,  $E_0$ ,  $S_0$ ,  $l_0$  represent the total energy for the compressed/dilated structure, the total energy at equilibrium, the equilibrium lattice area, the equilibrium lattice constant in the transport direction and the deformation of  $l_0$ , respectively. The  $E_1$  equals to  $\Delta E / (\Delta l / l_0)$ , in which  $E$ ,  $l$ , and  $\Delta l$  are donated as the energy change of CBM or VBM under compression or tensile strain, the lattice constants in the transport direction, and the deformation of  $l_0$ .  $m^*$  is the effective mass in transport direction, and  $m_d$  is determined by  $m_d = \sqrt{m_x^* m_y^*}$  ( $m_x^*$  is perpendicular to  $m_y^*$ ). The temperature  $T$  is set to be 300 K, and  $k_B$  is Boltzmann constant.

### Absorption spectra calculations

The linear optical properties of semiconductors can be obtained from the frequency-dependent complex dielectric function:

$$\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega)$$

where  $\varepsilon_1(\omega)$  and  $\varepsilon_2(\omega)$  are the real and imaginary parts of the dielectric function, and  $\omega$  is the photon frequency. The relevant parameters such as the absorption  $\alpha(\omega)$  can be calculated from the real  $\varepsilon_1(\omega)$  and the imaginary  $\varepsilon_2(\omega)$  parts [4]:

$$\alpha(\omega) = \frac{\sqrt{2}\omega}{c} \left( \sqrt{\varepsilon_1^2 + \varepsilon_2^2} - \varepsilon_1 \right)^{1/2}$$

### Polar diagrams for Young's modulus and Poisson's ratio

Using the calculated elastic constants, the Young modulus and Poisson ratio along an arbitrary direction  $\theta$  ( $\theta$  is the angle relative to the  $x$  direction) are also obtained as [5]

$$E(\theta) = \frac{C_{11}C_{12} - C_{12}^2}{C_{11}s^4 + C_{22}c^4 + \left(\frac{C_{11}C_{12} - C_{12}^2}{C_{44}} - 2C_{12}\right)c^2s^2}$$

$$\nu(\theta) = -\frac{\left(C_{11} + C_{22} - \frac{C_{11}C_{12} - C_{12}^2}{C_{44}}\right)c^2s^2 - C_{12}(c^4 + s^4)}{C_{11}s^4 + C_{22}c^4 + \left(\frac{C_{11}C_{12} - C_{12}^2}{C_{44}} - 2C_{12}\right)c^2s^2}$$

where  $c = \cos\theta$  and  $s = \sin\theta$ .

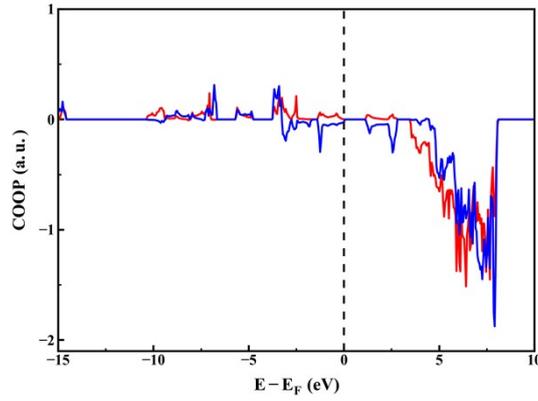


Figure S1. The crystal orbital overlap population (COOP) curves for the shortest Be-N and N-N bonds in the 567-BeN<sub>2</sub> monolayer. Positive and negative values indicate bonding and antibonding states, respectively.

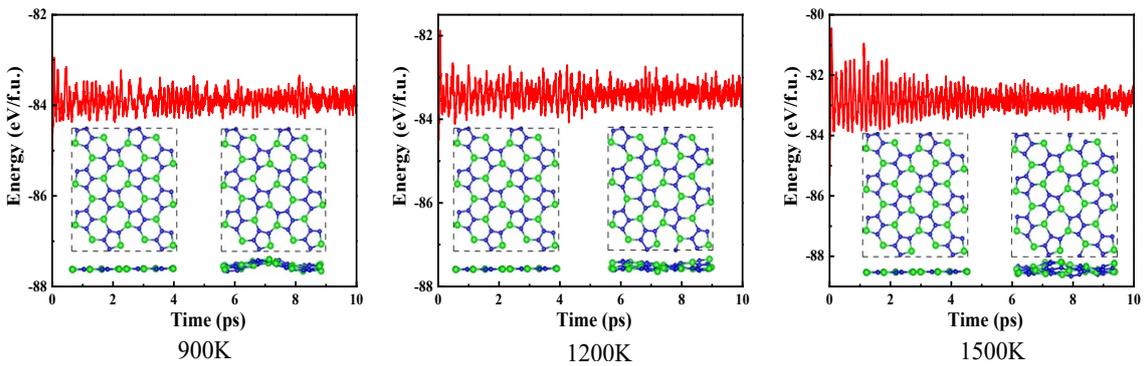


Figure S2. Evolution of energies during AIMD simulations at 900K, 1200 K, and 1500 K of the 567-BeN<sub>2</sub> monolayer. The insets feature structural snapshots captured at the beginning and end of the 10 ps simulation period.

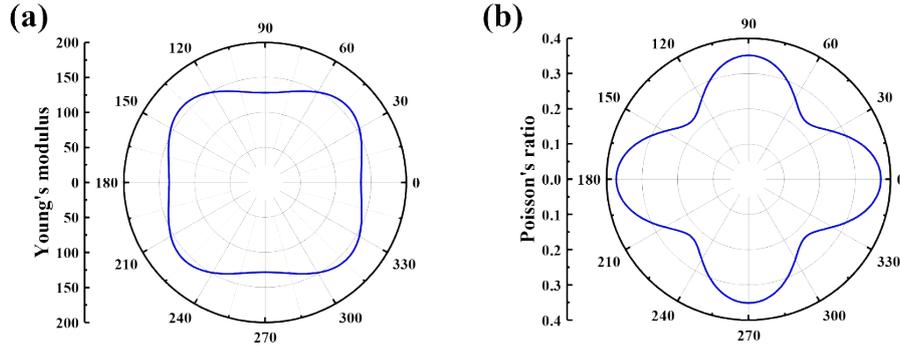


Figure S3. (a) Polar diagrams for Young's modulus ( $E$ ,  $\text{Nm}^{-1}$ ) and (b) Poisson's ratio (Pr) of the 567- $\text{BeN}_2$  monolayer.

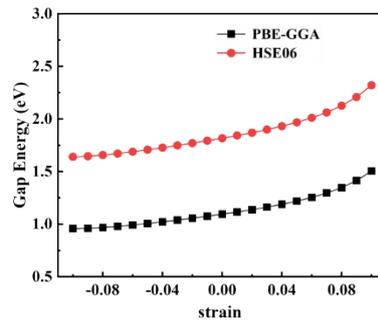


Figure S4. Band gap variation of the 567- $\text{BeN}_2$  monolayer under biaxial strain.

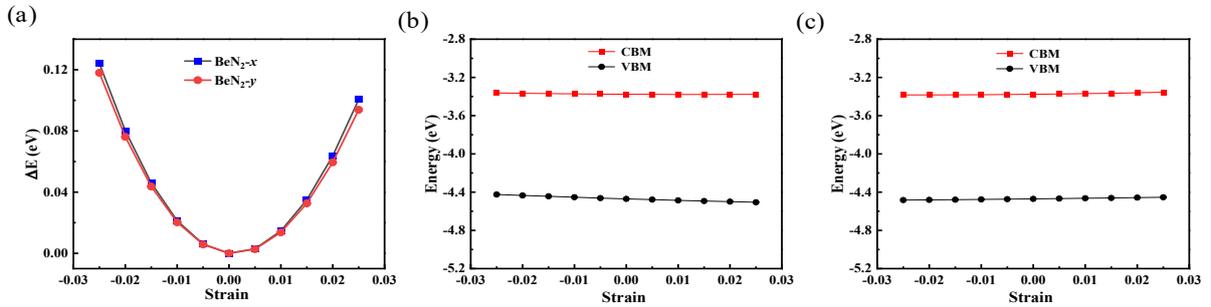


Figure S5. (a) Total energy differences of 567- $\text{BeN}_2$  under uniaxial strain from -3% to 3% applied along the  $x$  and  $y$  directions. (b, c) Strain-induced shifts of the CBM and VBM of monolayer 567- $\text{BeN}_2$  relative to the vacuum level as functions of tensile and compressive lattice deformation along the  $x$  and  $y$  directions.

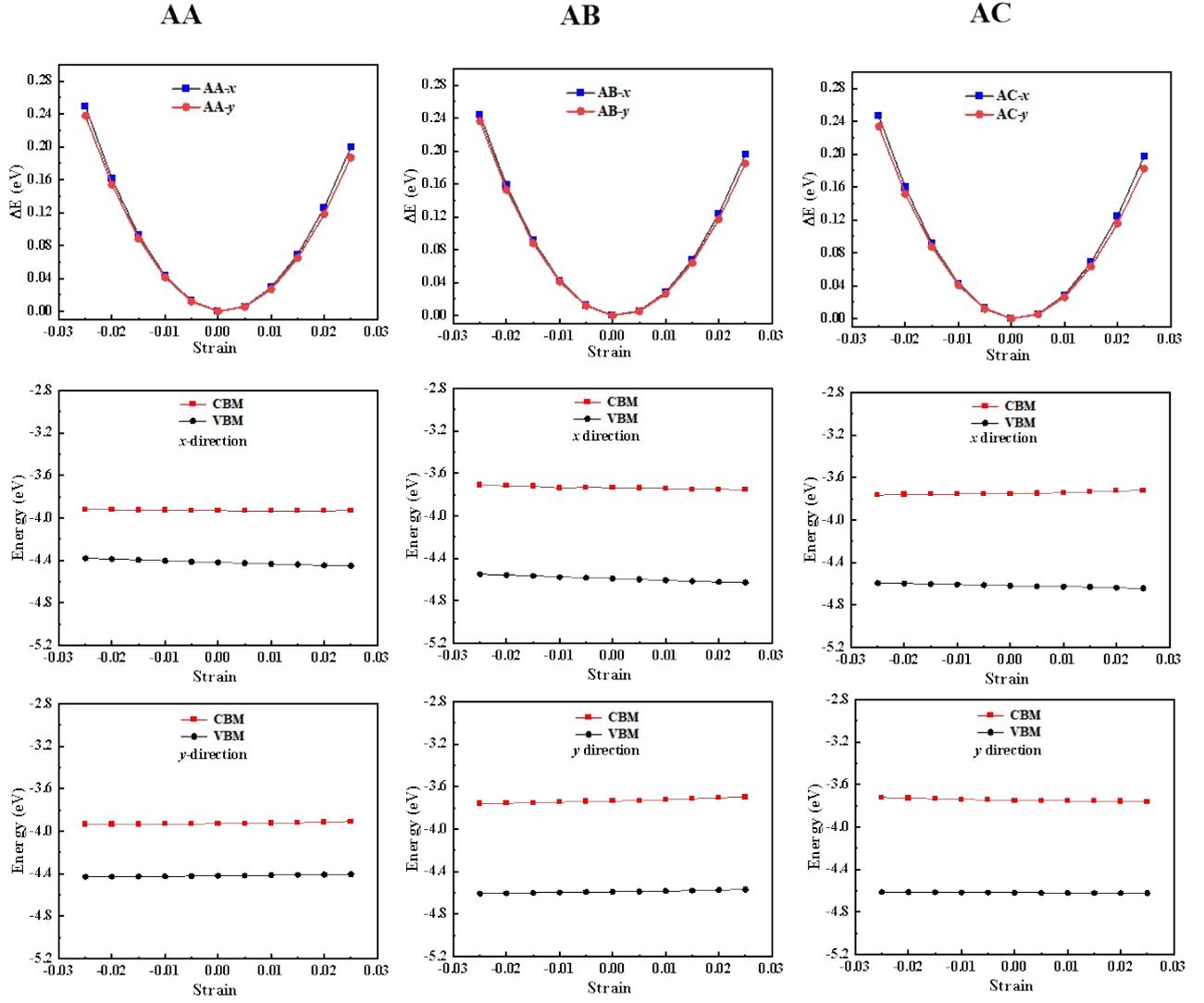


Figure S6. Total energy differences of bilayer 567-BeN<sub>2</sub> under uniaxial strain from -3% to 3% applied along the *x* and *y* directions. Strain-induced shifts of the CBM and VBM of bilayer 567-BeN<sub>2</sub> relative to the vacuum level as functions of tensile and compressive lattice deformation along the *x* and *y* directions.

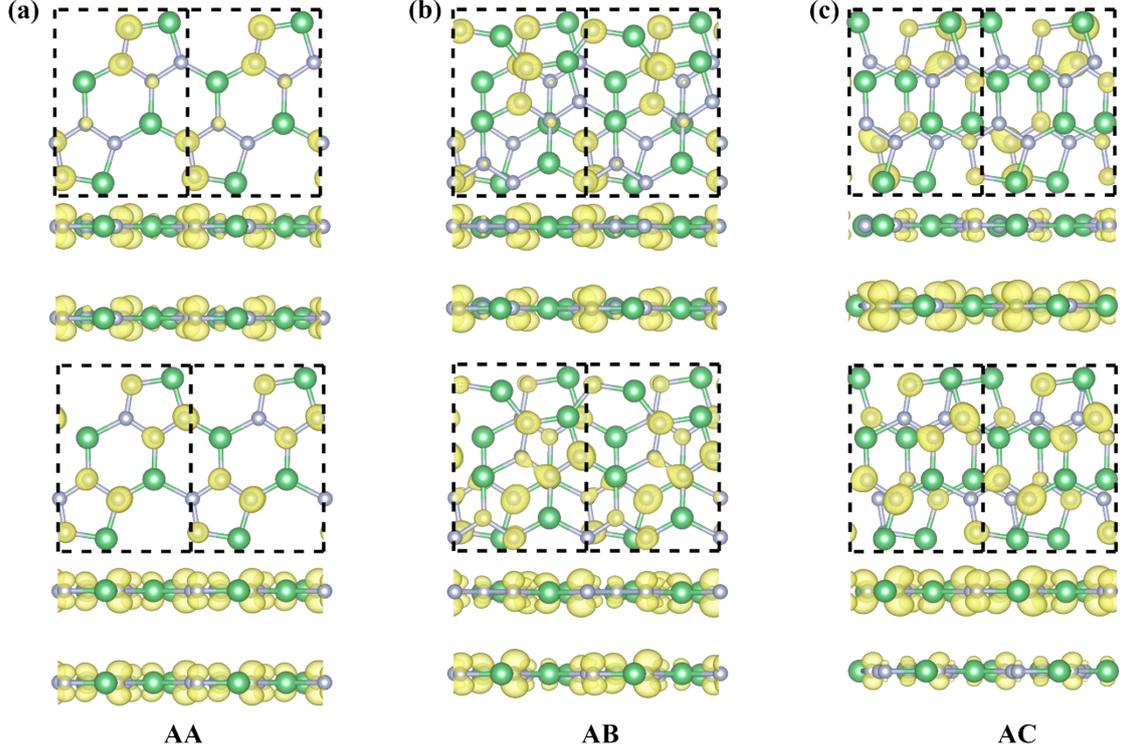


Figure S7. Partial charge density distributions at the conduction-band minimum (CBM) (top panel) and valence-band maximum (VBM) (bottom panel), with an isovalue of 0.007 e/bohr<sup>3</sup> for 567-BeN<sub>2</sub> with (a) AA, (b) AB, and (c) AC stacking configurations.

Table S1. The energy differences among AA-, AB-, and AC- stacking were calculated by the nonlocal optB86b-vdW density functional and DFT-D3 method.

Methods	Energy (eV/atom)		
	$E_{AA}$	$E_{AB}$	$E_{AC}$
optB86b	0.011	0.002	0
DFT-D3	0.010	0.002	0

Table S2. The calculated carrier effective masses ( $m^*$ ,  $m_0$ ), deformation potential constants ( $E_1$ , eV), elastic moduli ( $C$ , J m<sup>-2</sup>), and carrier mobilities ( $\mu$ , 10<sup>3</sup> cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>) along the  $x$  and  $y$  directions for bilayer 567-BeN<sub>2</sub> with different stacking configurations.

Compounds	Carrier	$m_x^*$	$m_y^*$	$E_{1x}$	$E_{1y}$	$C_{2Dx}$	$C_{2Dy}$	$\mu_x$	$\mu_y$
AA	Electron	0.71	0.55	0.24	0.47	359.05	339.79	308.99	96.38
	Hole	0.77	0.93	1.42	0.51	359.05	339.79	5.86	36.17
AB	Electron	0.96	0.62	0.89	1.16	352.08	336.81	12.93	11.23
	Hole	1.11	0.73	1.57	0.78	352.08	336.81	3.06	18.16
AC	Electron	0.70	0.69	0.78	0.04	355.27	333.03	23.98	28.86
	Hole	0.66	1.58	0.98	0.54	355.27	333.03	12.35	97.61

Table S3. The calculated elastic constants ( $C_{ij}$ , N/m), Young's modulus ( $E_Y$ , N/m) and Poisson's ratio ( $P_r$ ) of 567-BeN<sub>2</sub> with AA, AB, and AC stacking configurations.

Stacking configurations	$C_{11}$	$C_{22}$	$C_{12}$	$C_{66}$	$E_{Yx}$	$E_{Yy}$	$P_{rx}$	$P_{ry}$
AA	312.88	296.10	111.06	129.21	273.46	254.44	0.375	0.355
AB	306.49	293.19	108.14	127.76	268.33	253.31	0.369	0.353
AC	312.88	296.10	111.06	129.21	273.46	254.44	0.380	0.356

### POSCAR file for 567-BeN<sub>2</sub>

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0.0000000000000000 0.0000000000000000 30.0000000000000000

Be N

4 8

Direct

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0.2836985688066775 0.9885294841035019 0.5000000000000000

### References

- [1] Bruzzone, Samantha, and Gianluca Fiori. "Ab-initio simulations of deformation potentials and electron mobility in chemically modified graphene and two-dimensional hexagonal boron-nitride." *Applied Physics Letters* 99.22 (2011).
- [2] Fiori, Gianluca, and Giuseppe Iannaccone. "Multiscale modeling for graphene-based nanoscale transistors." *Proceedings of the IEEE* 101.7 (2013): 1653-1669.
- [3] Takagi, Shin-ichi, et al. "On the universality of inversion layer mobility in Si MOSFET's: Part I-effects of substrate impurity concentration." *IEEE Transactions on Electron Devices* 41.12 (1994): 2357-2362.
- [4] Fox, Mark. *Optical properties of solids*. Vol. 3. Oxford university press, 2010.
- [5] Zhang, Chenxi, Jun Lou, and Jizhou Song. "A cohesive law for interfaces in graphene/hexagonal boron nitride heterostructure." *Journal of Applied Physics* 115.14 (2014).