

28 The surface morphology of the nanoparticles was examined by field-emission scanning electron
29 microscopy (FE-SEM, TESCAN MIRA3) operated at an accelerating voltage of 5–10 kV. Prior to
30 imaging, the samples were sputter-coated with a thin layer of gold using a Quorum SC7620 sputter
31 coater under high-vacuum conditions.

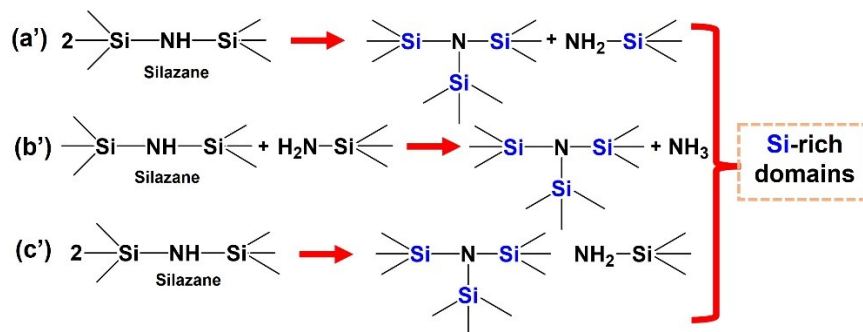
32 Transmission electron microscopy (TEM) was employed to investigate the particle morphology
33 and size distribution using an FEI Tecnai G2 20 S-Twin microscope. For TEM analysis, a dilute
34 nanoparticle suspension (0.0001 wt %) was drop-cast onto a 200-mesh copper grid and allowed to
35 dry under ambient conditions.

36 Ultraviolet-visible absorption spectra were recorded using a Shimadzu UV-2600
37 spectrophotometer. Photoluminescence measurements were carried out on a Cary Eclipse
38 fluorescence spectrophotometer (Agilent Technologies) equipped with a xenon lamp as the
39 excitation source.

40 **Section S3: Transformation of polysilazane to nanostructure**

41 Under solvothermal conditions at 200 °C, polysilazane undergoes a series of coupled reactions
42 leading to the formation of Si-rich amorphous nanoparticles. The transformation is initiated by
43 hydrolysis of the Si–N backbone ($\text{Si–N} + \text{H}_2\text{O} \rightarrow \text{Si–OH} + \text{NH}_3$), accompanied by transamination
44 reactions that facilitate the rearrangement and gradual removal of nitrogen-containing species.
45 Concurrently, reactive Si–H groups undergo dehydrogenation and oxidation, generating additional
46 silanol (Si–OH) functionalities. These silanol groups subsequently participate in condensation
47 (polymerization/crosslinking) reactions ($\text{Si–OH} + \text{Si–OH} \rightarrow \text{Si–O–Si} + \text{H}_2\text{O}$), resulting in the
48 formation of a progressively interconnected siloxane network.¹ In parallel, the organic substituents
49 attached to silicon undergo partial decomposition and rearrangement, contributing to the formation
50 of Si–O–C linkages and disordered carbon domains within the structure. During this polymer-to-
51 ceramic conversion, nitrogen is largely eliminated as volatile species (e.g., NH_3), leading to an
52 oxygen-dominated chemical environment.² The overall process yields an amorphous Si–O–C–N
53 hybrid network, structurally characterized by a dominant Si–O_x framework with residual carbon
54 and trace nitrogen, consistent with the spectroscopic observations.³

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Figure 1S: Reaction pathways of polysilazanes

58 Section S4: Discussion on Defect-Mediated Photoluminescence

59 The photoluminescence observed in the present SiO_x -rich nanoparticles is attributed to defect- and
 60 surface-state-mediated transitions, which are widely reported in amorphous silica and related
 61 silicon-based systems. Previous studies have established that various intrinsic and extrinsic defect
 62 centers within disordered Si–O networks can act as emissive states. For example, Saita et al.
 63 demonstrated that fluorescence in silica-based nanoparticles originates from surface defect pairs
 64 such as dioxasilyrane ($=\text{Si}(\text{O}_2)$) and silylene (Si) centers, with hydrothermal treatment promoting
 65 defect formation and enhancing emission efficiency.⁴

66 Similarly, Guleria et al. reported that photoluminescence in amorphous SiO_2 nanoparticles arises
 67 from multiple defect states, particularly oxygen-related defects, which act as emission centers and
 68 can be tuned to produce emission across the blue-to-green spectral range.^{5,6} The dominance of
 69 oxygen-deficient or oxygen-related defect states has also been correlated with enhanced emission
 70 intensity and sensing functionality in such systems. In polymer-derived systems, Menapace et al.
 71 showed that thermal treatment of polysilazane leads to visible photoluminescence, which is
 72 associated with structural rearrangements, dangling bonds, and the formation of carbon-related
 73 species during the polymer-to-ceramic transition.⁷ These findings highlight the role of structural
 74 disorder and hybrid bonding environments (e.g., Si–O–C) in generating defect states capable of
 75 radiative recombination.

76 More advanced defect-engineered silica systems reported by Chang et al. and Sun et al. further
 77 demonstrate that both intrinsic defects (e.g., oxygen vacancies, non-bridging oxygen centers) and
 78 extrinsic impurities (e.g., carbon- or nitrogen-related species) contribute to a distribution of
 79 emissive states, resulting in complex optical behaviors including fluorescence and long-lived

80 phosphorescence.^{8,9} Such studies emphasize that defect heterogeneity within amorphous silica
 81 networks is a key factor governing their photophysical properties. Additionally, Trillot et al.
 82 highlighted the presence of under-coordinated oxygen atoms in amorphous silica, which not only
 83 influence transport properties but also act as active sites for metal ion coordination.¹⁰ These defect
 84 sites can interact strongly with ions such as Ag⁺ through O–Ag bonding, supporting their dual role
 85 in both optical emission and sensing behavior. Based on these established studies, the emission in
 86 the present system is attributed to a combination of oxygen-related defect centers, Si–O–C
 87 structural disorder, and residual defects generated during the solvothermal polymer-to-ceramic
 88 transformation. The observed excitation-dependent emission further supports the presence of a
 89 distribution of defect states rather than a single well-defined emissive center.

90 **Section S5: Fluorescence Quantum Yield**

91 As a reference, rhodamine 6G, which has a known quantum yield of .95, was used to calculate the
 92 quantum yield (QY) of polysilazane-derived nanoparticles. Both nanoparticles and rhodamine 6G
 93 solutions were made to keep their UV-Vis absorption peaks at or below 0.1. The fluorescence
 94 intensities were calculated by integrating the regions under the PL emission spectra, and the
 95 absorption and fluorescence spectra were taken at an excitation wavelength of 480 nm. The
 96 following (**Equation 1S**) was used to get the quantum yield of nanoparticles. The absorbance of
 97 both solutions was maintained below 0.1 to minimize inner filter effects. The quantum yield of the
 98 polysilazane-derived nanoparticles was calculated to be approximately 26%.

99 The following (**Equation 1S**) was used to get the quantum yield of the particle.

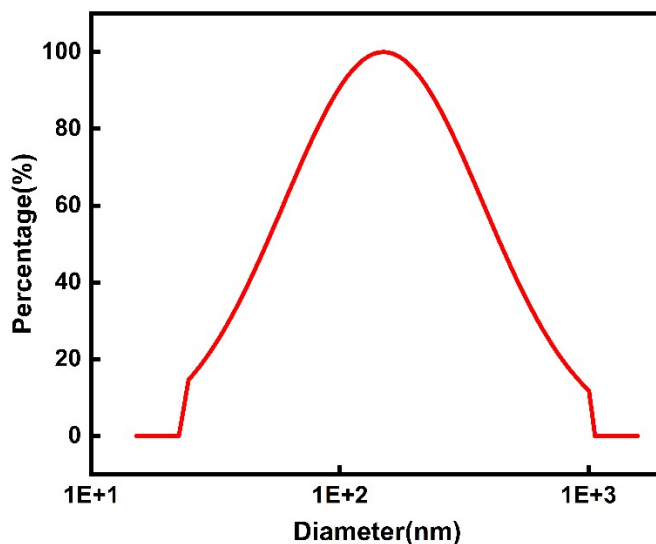
$$100 \quad QY = QY_s \times \frac{I}{I_s} \times \frac{A_s}{A} \times \frac{\eta^2}{\eta_s^2} \times 100\% \quad) \dots\dots\dots (\text{equation 1S})$$

101 *QY* is the quantum yield of the particle. The quantum yield of rhodamine 6G, the reference, is
 102 denoted as *QY_s*. The integral regions of the fluorescence peaks for the reference and the
 103 nanoparticles are denoted by *I* and *I_s*, respectively. The absorbance values for the nanoparticles
 104 and the reference are *A* and *A_s*, respectively. These η represent the solvents' refractive indices.

105 **Section S6: Dynamic light scattering (DLS) measurements**

106 Dynamic light scattering measurements were carried out to evaluate the hydrodynamic size of the
 107 nanoparticles. The average particle size obtained from DLS is 100-105 nm, which is slightly higher

108 than the size observed from TEM (85-90 nm). This difference can be attributed to the
109 hydrodynamic nature of DLS measurements, which include contributions from the solvation layer
110 and surface-associated species, whereas TEM reflects the physical size of dried nanoparticles. The
111 polydispersity index (PDI) of 0.3 indicates a moderately broad size distribution, and the intensity-
112 weighted nature of DLS may also contribute to a slightly higher apparent size. Overall, the results
113 confirm the formation of well-dispersed nanoparticles.



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Figure 2S: DLS Study of the synthesized nanoparticles

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117 Section S7: Comparison with other systems for the detection of Ag⁺

118 A comparison of the detection limit and linear sensing range of the present system with previously
119 reported nanoparticle-based Ag⁺ sensors. For instance, Luo et al. reported red-emissive carbon
120 dots with a detection limit of 0.37 μM and a linear range of 0–50 μM , while Lu et al. developed
121 dual-functional carbon dots exhibiting a lower detection limit of 0.05 μM .^{11,12} Similarly, Zhang et
122 al. demonstrated ratiometric carbon quantum dots with a detection limit of 0.14 μM and a wide
123 linear range of 22–220 μM . In comparison, Wang et al. reported N-doped carbon dots with a
124 relatively higher detection limit of 4.7 μM and a linear range of 30–210 μM .^{13,14} Additionally, Wu
125 et al. developed bisbenzimidazole-based fluorescent sensors exhibiting rapid and selective Ag⁺
126 detection based on complexation-induced quenching.¹⁵

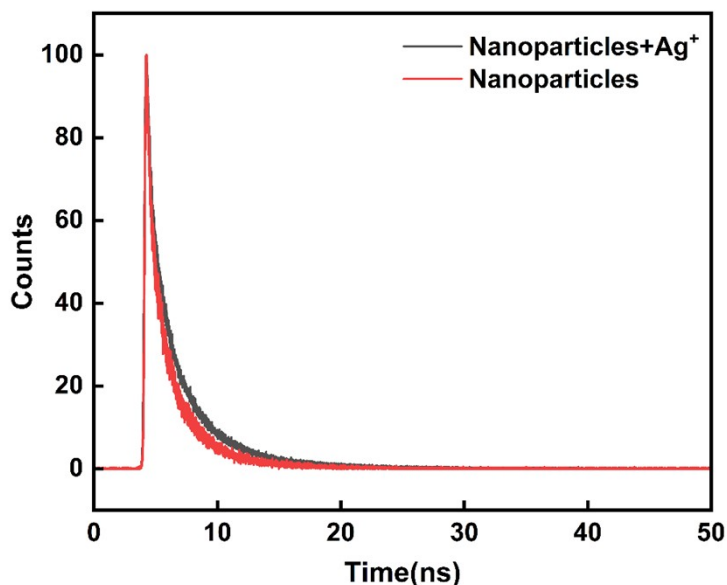
127 These comparisons indicate that the polysilazane-derived nanoparticles in the present study exhibit
 128 a competitive detection limit (0.55 μM) and a comparable sensing range relative to existing
 129 fluorescent nanomaterial-based systems. Notably, unlike conventional carbon-dot or small-
 130 molecule-based probes, the present system is derived from a single-source preceramic polymer via
 131 a one-pot solvothermal route, resulting in an oxygen-rich Si–O–C–N hybrid network. The sensing
 132 mechanism is governed by defect-rich siloxane environments rather than surface functional groups
 133 or molecular fluorophores, offering enhanced structural robustness, long-term stability, and
 134 potential compatibility with coating-based applications.

135 Section S8: Fluorescence lifetime study

136 The fluorescence decay profiles were fitted using a multi-exponential model, yielding three
 137 lifetime components for the pristine nanoparticles ($\tau_1 \approx 0.25$ ns, $\tau_2 \approx 1.11$ ns, and $\tau_3 \approx 2.93$ ns),
 138 with the longest-lived component accounting for $\sim 67\%$ of the total decay. Upon addition of Ag^+
 139 ions, all lifetime components increased ($\tau_1 \approx 0.37$ ns, $\tau_2 \approx 1.95$ ns, and $\tau_3 \approx 4.74$ ns), along with a
 140 marked shift in relative amplitudes toward the longer-lived components ($\tau_2 \approx 48\%$ and $\tau_3 \approx 47\%$).
 141 Consequently, an overall increase in average fluorescence lifetime was observed despite the
 142 simultaneous decrease in emission intensity. This unusual behavior suggests that Ag^+ ions interact
 143 with surface-associated defect states, altering the emissive-state population and suppressing faster
 144 non-radiative decay channels. Together with Raman and XPS results confirming a defect-rich Si–
 145 O–C network containing abundant oxygen-bearing surface functionalities, these findings support
 146 a sensing mechanism dominated by defect-state modulation rather than simple dynamic collisional
 147 quenching. The fluorescence decay profiles were fitted using a multi-exponential model:

$$148 \quad I(t) = B_1 e^{-t/\tau_1} + B_2 e^{-t/\tau_2} + B_3 e^{-t/\tau_3} \dots \dots \dots \text{(equation 2S)}$$

149 where τ_i represent the lifetime components and B_i their relative contributions.



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151 **Figure 3S:** Fluorescence lifetime study showing the slower decay of the excited states of the
 152 nanoparticles in combination of the Ag⁺ ions.

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