

## *Supporting Information*

### **Diamond-like $\text{Ag}_x\text{Cu}_{3-x}\text{PS}_4$ ( $x = 1, 2$ ) Synthesized by Metal Oxide-Boron-Chalcogen (MOBQ) Routine: Local High-Entropy Engineering for Balanced Infrared Nonlinear Optical Performance**

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### 4. References

# 1. Experimental Section

## Syntheses

All starting materials were used as received without further purification. Ag<sub>2</sub>S (Aladdin, 99.9 %) , Cu<sub>2</sub>O (Aladdin, 99.9 %) , P (Macklin, 98.5 %) , P<sub>2</sub>S<sub>5</sub> (Macklin, 99 %) , S (Aladdin, 99.5 %) , B (Aladdin, 99.9 %) and KI (Macklin, 99 %). Single crystals of AgCu<sub>2</sub>PS<sub>4</sub> (**1**) and Ag<sub>2</sub>CuPS<sub>4</sub> (**2**) were obtained by the facile metal oxide-boron-chalcogen (MOBQ) solid-state reactions with KI as flux. This method was firstly discovered by Prof. Jin-Shun Huang in 1997, which was then well developed by us from 2008 to now.<sup>1-10</sup>

**1:** The raw-materials were weighed according to the molar ratios of Ag<sub>2</sub>S : Cu<sub>2</sub>O : P : S : B = 3 : 6 : 6 : 24 : 4.

**2:** The raw-materials were weighed according to the molar ratios of Ag<sub>2</sub>S : Cu<sub>2</sub>O : P<sub>2</sub>S<sub>5</sub> : S : B = 6 : 3 : 3 : 9 : 2, and P<sub>2</sub>S<sub>5</sub> was weighted in glove box filled with Ar to prevent oxidation and deliquescence.

Each sample has a total mass of 500 mg and additional 400 mg KI.

Each mixture was ground into fine powder in an agate mortar and further pressed into a pellet, followed by being loaded into a quartz tube. The quartz tubes were evacuated to be  $1 \times 10^{-4}$  torr and flame-sealed and then were placed into a muffle furnace, heated from room temperature to 1173 K with several intermediate homogeneous temperatures and kept for 2 days, finally cooled down to 573 K in 5 days, and powered off. The yellow crystals of **1** and **2** were obtained, which were then washed with water and alcohol, followed by being ultrasonic cleaned. These crystals are stable in room temperature for months.

## Structure determination

Single-crystal X-ray diffraction (SXRD) data of **1** and **2** were collected by the Bruker D8 QUEST X-ray diffractometer with graphite-monochromated Mo-K $\alpha$  radiation ( $\lambda = 0.71073 \text{ \AA}$ ) at 293 K. The structures were solved by Direct Methods and SHELXL-2014/7 program package within Olex2 v1.5 software were used to solve and refine the crystal structures.<sup>11,12</sup> Crystal data, structure refinement parameters, atomic coordinates and bond lengths are summarized in **Table S1–S3**. Their CIF documents were also deposited with the CCDC numbers of 2539219 (**1**) and 2539220 (**2**).

### **Powder X-ray diffraction (PXRD) analysis**

The PXRD patterns were collected by the Bruker D8 Advance diffractometer at 40 kV and 100 mA for Cu-K $\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ) with a scan speed of  $2^\circ \text{ min}^{-1}$  in room temperature. The simulated PXRD patterns were obtained from single-crystal structure data with the Mercury v3.8 software program provided by the Cambridge Crystallographic Data Center.

### **UV-vis-NIR diffuse reflectance spectra**

The diffuse reflectance spectra were measured on the Cary 5000 UV-vis-NIR spectrometer at room temperature in the wavelength range of 200–2100 nm. The BaSO<sub>4</sub> plate was used as the baseline of scanning, on which the finely ground powdery sample was coated.<sup>13</sup> The absorption spectra were calculated from the reflection spectra by the Kubelka-Munk function to obtain the experimental band gaps.<sup>14</sup>

### **Calculation details**

First-principles calculations about the electronic structures and optical properties, and electron localization function (ELF) of **1** and **2** were performed in Materials Studio software base on density functional theory (DFT).<sup>15</sup> The generalized gradient approximation (GGA) by Perdew-Burke-Ernzerhof (PBE) was selected as the exchange-correlation function.<sup>16–17</sup> To ensure the comparability of the results, the computational parameters and convergence criteria for compounds **1** and **2** were maintained identical. The plane-wave cutoff energies for **1** and **2** were both 480.00 eV, and the ultrasoft pseudopotential was selected and the threshold of  $5.0 \times 10^{-7}$  eV was set for the self-consistent-field convergence of the total electronic energy. The electrons in 4d<sup>10</sup>5s<sup>1</sup>, 3d<sup>10</sup>4s<sup>1</sup>, 3s<sup>2</sup>3p<sup>3</sup>, 3s<sup>2</sup>3p<sup>4</sup> orbitals for Ag, Cu, P, S were regarded as valence electrons. The numerical integration of the Brillouin zones were performed using  $4 \times 3 \times 3$  Monkhorst-Pack k-point meshes and the Fermi level at 0 eV was chosen as the reference for **1** and **2**. The real and imaginary parts of the dielectric function were also calculated, which were used to calculated birefringences.<sup>18,19</sup>

### **Ion conductivity measurement**

The electrochemical impedance spectroscopy (EIS) of **2** was measured using an electrochemical workstation (DH7001D) over a frequency range of 1 Hz to 1 MHz. The powder sample was pressed into a pellet with a diameter of 10 mm under a pressure of 380 MPa. The temperature-dependent AC impedance spectroscopy was investigated at variable temperature

from 30 to 80 °C with an increment of 10 °C. The ionic conductivity was calculated via the equation  $\sigma = L/(S \cdot R)$ , where L is the thickness of the pellet sample and S is its area. The total resistance R is equal to the real impedance when the imaginary impedance approaches zero ( $Z'' \approx 0$ ).

## 2. Tables

**Table S1.** Crystal structure refinement parameters for **1** and **2**.

Chemical formula	AgCu <sub>2</sub> PS <sub>4</sub> (1)	Ag <sub>2</sub> CuPS <sub>4</sub> (2)
Fw (g/mol)	394.20	438.49
T (K)	293	
Crystal system	Orthorhombic	
Space group	<i>Pmn</i> 2 <sub>1</sub>	
<i>Z</i>	2	2
<i>a</i> (Å)	7.6319(6)	7.6668(8)
<i>b</i> (Å)	6.2924(4)	6.6063(6)
<i>c</i> (Å)	6.1763(5)	6.3169(5)
$\alpha$ (°)	90	90
$\beta$ (°)	90	90
$\gamma$ (°)	90	90
<i>V</i> (Å <sup>3</sup> )	296.60(4)	319.95(5)
$\rho_{\text{calc}}$ (g cm <sup>-3</sup> )	4.414	4.552
$\mu$ (mm <sup>-1</sup> )	11.906	10.775
F(000)	369.0	404.0
2 $\theta$ range (°)	6.48 to 54.94	6.168 to 49.934
Reflections collected	2292	2302
Indep. reflns/R <sub>int</sub>	647/0.0435	595/0.0317
GOF on <i>F</i> <sup>2</sup>	1.010	1.124
R <sub>1</sub> , <sup>a</sup> wR <sub>2</sub> <sup>b</sup> ( <i>I</i> > 2 $\sigma$ ( <i>I</i> ))	0.0324, 0.0832	0.0318, 0.0776
R <sub>1</sub> , <sup>a</sup> wR <sub>2</sub> <sup>b</sup> (all data)	0.0326, 0.0833	0.0337, 0.0788
$\Delta\rho_{\text{max}}/\Delta\rho_{\text{min}}$ , e/Å <sup>3</sup>	0.65/-1.33	1.28/-1.44
Flack parameter	0.07(2)	0.01(2)

$${}^a R_1 = \sum ||F_o| - |F_c|| / \sum |F_o|. \quad {}^b wR_2 = [w(F_o^2 - F_c^2)^2] / [w(F_o^2)^2]^{1/2}.$$

**Table S2.** Fractional atomic coordinates ( $\times 10^4$ ) and equivalent isotropic displacement parameters ( $\text{\AA}^2 \times 10^3$ ) for **1** and **2**.  $U_{\text{eq}}$  is defined as 1/3 of the trace of the orthogonalized  $U_{ij}$  tensor.

Atom	Wyckoff site	$x$	$y$	$z$	$U_{\text{eq}}^a / \text{\AA}^2$
<b>1</b>					
Ag(1)	$2a$	5000	3527.1(12)	866.2(4)	30.5(3)
Cu(1)	$4b$	7526.1(9)	8138.4(11)	915.5(10)	18.5(3)
P(1)	$4b$	5000	6875(3)	5820(3)	5.9(4)
S(1)	$2a$	7202(2)	8367(2)	4639(2)	12.2(3)
S(2)	$2a$	5000	3725(3)	4907(3)	11.1(4)
S(3)	$2a$	5000	7171(4)	9165(3)	13.6(5)
<b>2</b>					
Ag(1)	$2a$	5000	6483(3)	4320(4)	55.5(7)
Ag(2)	$4b$	7672.9(17)	1960.3(18)	4020.4(18)	38.1(5)
Cu(1)	$4b$	7672.9(17)	1960.3(18)	4020.4(18)	38.1(5)
P(1)	$2a$	10000	6939(5)	4196(8)	12.0(8)
S(1)	$2a$	5000	2945(7)	5913(7)	19.8(10)
S(2)	$4b$	7816(4)	8418(6)	5253(6)	31.2(8)
S(3)	$2a$	10000	4019(9)	5276(9)	41.2(14)

<sup>a</sup> $U_{\text{eq}}$  is defined as one third of the trace of the orthogonalized  $U_{ij}$  tensor.

**Table S3.** Bond lengths (Å) for **1** and **2**.

Bond	Length/Å	Bond	Length/Å
<b>1</b>			
Ag(1)–S(1)#1	2.5603(17)	Cu(1)–S(2)#5	2.3081(13)
Ag(1)–S(1)#2	2.5603(17)	Cu(1)–S(3)#6	2.2927(13)
Ag(1)–S(2)	2.4988(18)	P(1)–S(1)	2.0588(18)
Ag(1)–S(3)#3	2.522(2)	P(1)–S(2)	2.061(2)
Cu(1)–S(1)#4	2.3451(16)	P(1)–S(3)	2.074(3)
Cu(1)–S(1)	2.3173(18)		
<b>2</b>			
Ag(1)–S(1)	2.544(5)	Cu(1)–S(2)#2	2.422(4)
Ag(1)–S(2)#1	2.577(4)	Cu(1)–S(2)#3	2.469(4)
Ag(1)–S(2)	2.577(4)	Cu(1)–S(3)	2.380(5)
Ag(1)–S(3)#2	2.576(6)	P(1)–S(2)#5	2.050(4)
Ag(2)–S(1)	2.460(3)	P(1)–S(2)	2.050(4)
Ag(2)–S(3)	2.380(5)	P(1)–S(3)	2.046(6)
Cu(1)–S(1)	2.460(3)	P(1)–S(1)#4	2.076(7)

Symmetry transformations used to generate equivalent atoms: For **1**: #1 $3/2-x$ ,  $1-y$ ,  $-1/2+z$ ; #2 $-1/2+x$ ,  $1-y$ ,  $-1/2+z$ ; #3 $+x$ ,  $+y$ ,  $-1+z$ ; #4 $3/2-x$ ,  $2-y$ ,  $-1/2+z$ ; #5 $1/2+x$ ,  $1-y$ ,  $-1/2+z$ ; #6 $1-x$ ,  $+y$ ,  $-1+z$ . For **2**: #1 $-x$ ,  $+y$ ,  $+z$ ; #2 $3/2-x$ ,  $1-y$ ,  $-1/2+z$ ; #3 $+x$ ,  $-1+y$ ,  $+z$ ; #4 $2-x$ ,  $+y$ ,  $+z$ .

**Table S4.** Bond angles for **1** and **2**.

Atom	Atom	Atom	Angle/°	Atom	Atom	Atom	Angle/°
<b>1</b>							
S(1) <sup>#1</sup>	Ag(1)	S(1) <sup>#2</sup>	113.02(7)	S(3) <sup>#6</sup>	Cu(1)	S(2) <sup>#4</sup>	115.19(6)
S(2)	Ag(1)	S(1) <sup>#2</sup>	108.60(4)	S(1) <sup>#7</sup>	P(1)	S(1)	109.43(12)
S(2)	Ag(1)	S(1) <sup>#1</sup>	108.60(4)	S(2)	P(1)	S(1)	109.98(9)
S(3) <sup>#3</sup>	Ag(1)	S(1) <sup>#2</sup>	107.45(4)	S(2)	P(1)	S(1) <sup>#7</sup>	109.98(9)
S(3) <sup>#3</sup>	Ag(1)	S(1) <sup>#1</sup>	107.45(4)	S(3)	P(1)	S(1)	108.18(8)
S(3) <sup>#3</sup>	Ag(1)	S(2)	111.77(7)	S(3)	P(1)	S(1) <sup>#7</sup>	108.18(8)
S(2) <sup>#4</sup>	Cu(1)	S(1)	112.75(7)	S(3)	P(1)	S(2)	111.03(13)
S(2) <sup>#4</sup>	Cu(1)	S(1) <sup>#5</sup>	108.25(6)				
S(3) <sup>#6</sup>	Cu(1)	S(1)	113.25(6)				
S(3) <sup>#6</sup>	Cu(1)	S(1) <sup>#5</sup>	99.49(7)				
<b>2</b>							
S(1)	Ag(1)	S(2) <sup>#1</sup>	111.41(11)	S(2)	P(1)	S(1) <sup>#2</sup>	107.9(2)
S(1)	Ag(1)	S(2)	111.41(11)	S(2) <sup>#4</sup>	P(1)	S(1) <sup>#2</sup>	107.9(2)
S(1)	Ag(1)	S(3) <sup>#2</sup>	105.90(19)	S(2) <sup>#4</sup>	P(1)	S(2)	107.9(2)
S(2) <sup>#1</sup>	Ag(1)	S(2)	113.79(17)	S(3)	P(1)	S(1) <sup>#2</sup>	111.6(3)
S(3) <sup>#2</sup>	Ag(1)	S(2) <sup>#1</sup>	106.91(11)	S(3)	P(1)	S(2)	109.9(2)
S(3) <sup>#2</sup>	Ag(1)	S(2)	106.91(11)	S(3)	P(1)	S(2) <sup>#4</sup>	109.9(2)
S(1)	Cu(1)	S(2) <sup>#3</sup>	97.73(15)	S(1)	Ag(2)	S(2) <sup>#3</sup>	97.73(15)
S(2) <sup>#2</sup>	Cu(1)	S(1)	112.07(13)	S(2) <sup>#2</sup>	Ag(2)	S(1)	112.07(13)
S(2) <sup>#2</sup>	Cu(1)	S(2) <sup>#3</sup>	102.66(11)	S(2) <sup>#2</sup>	Ag(2)	S(2) <sup>#3</sup>	102.66(11)
S(3)	Cu(1)	S(1)	108.14(16)	S(3)	Ag(2)	S(1)	108.14(16)
S(3)	Cu(1)	S(2) <sup>#3</sup>	113.79(15)	S(3)	Ag(2)	S(2) <sup>#2</sup>	113.79(15)
S(3)	Cu(1)	S(2) <sup>#2</sup>	120.16(18)	S(3)	Ag(2)	S(2) <sup>#3</sup>	120.16(18)

Symmetry transformation used to generate equivalent atoms: For **1**: #1  $-1/2+x$ ,  $1-y$ ,  $-1/2+z$ ; #2  $3/2-x$ ,  $1-y$ ,  $-1/2+z$ ; #3  $+x$ ,  $+y$ ,  $-1+z$ ; #4  $1/2+x$ ,  $1-y$ ,  $-1/2+z$ ; #5  $3/2-x$ ,  $2-y$ ,  $-1/2+z$ ; #6  $1-x$ ,  $+y$ ,  $-1+z$ ; #7  $1-x$ ,  $+y$ ,  $+z$ ; For **2**: #1  $1-x$ ,  $+y$ ,  $+z$ ; #2  $3/2-x$ ,  $1-y$ ,  $-1/2+z$ ; #3  $+x$ ,  $-1+y$ ,  $+z$ ; #4  $2-x$ ,  $+y$ ,  $+z$ .

**Table S5.** Distortion degrees of tetrahedral units in **0–2**.

<b>Unit</b>	<b>DI (S–M–S)</b>	<b>DI (M–S)</b>	<b>DI (S–S)</b>
<b>0</b>			
<b>Cu(1)S<sub>4</sub></b>	0.010000420	0.01207394	0.00667115
<b>Cu(2)S<sub>4</sub></b>	0.03405344	0.00357646	0.02245589
<b>PS<sub>4</sub></b>	0.00691533	0.00247555	0.02442735
<b>1</b>			
<b>AgS<sub>4</sub></b>	0.01123334	0.009762351	0.011905912
<b>CuS<sub>4</sub></b>	0.0410516	0.00658534	0.023787853
<b>PS<sub>4</sub></b>	0.007920026	0.002605113	0.004967297
<b>2</b>			
<b>AgS<sub>4</sub></b>	0.0257334	0.004671078	0.016512573
<b>(Cu/Ag)S<sub>4</sub></b>	0.0572758	0.01315519	0.034699644
<b>PS<sub>4</sub></b>	0.0095943	0.004804768	0.00519388

**Table S6.** Calculated dipole moment results within unit-cells of **0–2**.

Compound	Dipole moment					
	<i>x</i>	<i>y</i>	<i>z</i>	Magnitude		
				Debye	$\times 10^{-4}$ esu·cm·Å <sup>-3</sup>	
<b>0</b>	Cu(1)S <sub>4</sub>	0	-4.5890	7.1034	8.4568	0.0603
	Cu(2)S <sub>4</sub>	-0.6992	46.7586	38.0879	60.3120	0.4302
	PS <sub>4</sub>	-0.0004	-7.1905	-10.3163	12.5749	0.0897
<b>1</b>	AgS <sub>4</sub>	-0.0006	-6.6509	15.9535	17.2844	0.1166
	CuS <sub>4</sub>	-42.7020	-36.3740	64.0035	85.1057	0.5739
	PS <sub>4</sub>	0.0008	-14.2563	-11.0708	18.0500	0.1217
<b>2</b>	AgS <sub>4</sub>	0	1.3955	-13.9125	13.9823	0.0874
	(Cu/Ag)S <sub>4</sub>	-5.2170	-37.6707	-74.8507	83.9579	0.5248
	PS <sub>4</sub>	0	18.6971	66.539	69.1160	0.4320

**Table S7.** The temperature-dependent ion conductivity for **2**.

Temperatures / °C	Ion conductivity / mS/cm
30	$9.35 \times 10^{-5}$
40	$9.00 \times 10^{-5}$
50	$1.34 \times 10^{-4}$
60	$4.60 \times 10^{-4}$
70	$7.08 \times 10^{-4}$
80	$9.30 \times 10^{-4}$

**Table S8.** Comparison of the SHG intensities of the reported  $M_3PS_4$  ( $M = Ag, Cu$  and  $Ag/Cu$ )<sup>20</sup>, and  $AgCu_2PS_4$ , and  $Ag_2CuPS_4$  reported here.

<b>Compound</b>	<b>SHG intensity (<math>\times</math> AGS)</b>
$Cu_3PS_4$	0.5 (NPM)
$Ag_{1.5}Cu_{1.5}PS_4$	0.8
$Ag_3PS_4$	1.3
$AgCu_2PS_4$ ( <b>1</b> , this work)	0.7
$Ag_2CuPS_4$ ( <b>2</b> , this work)	1.1

### 3. Figures

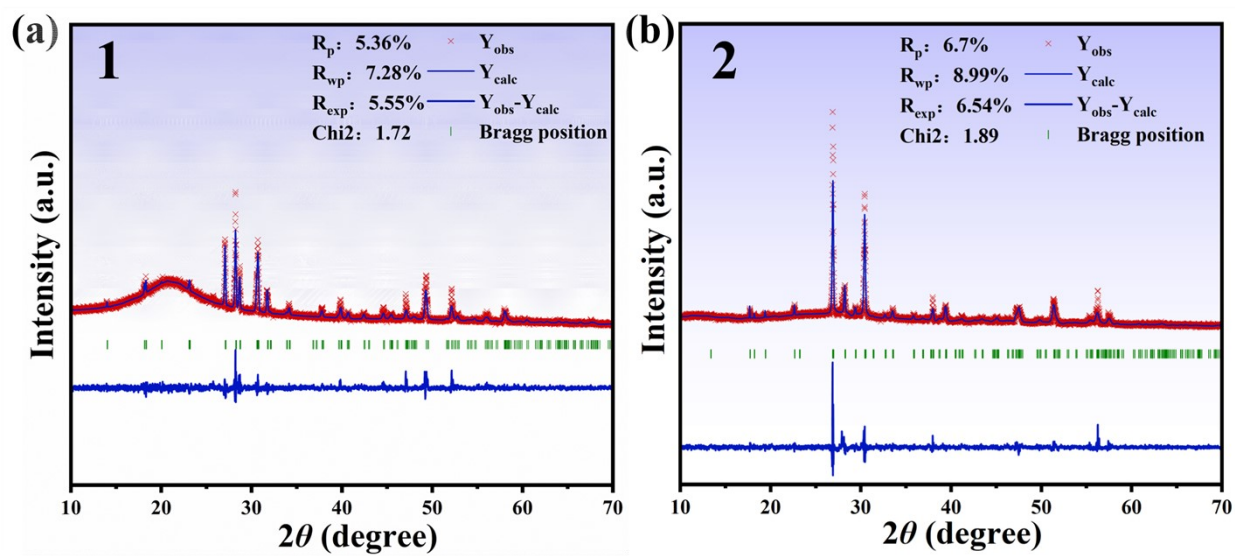
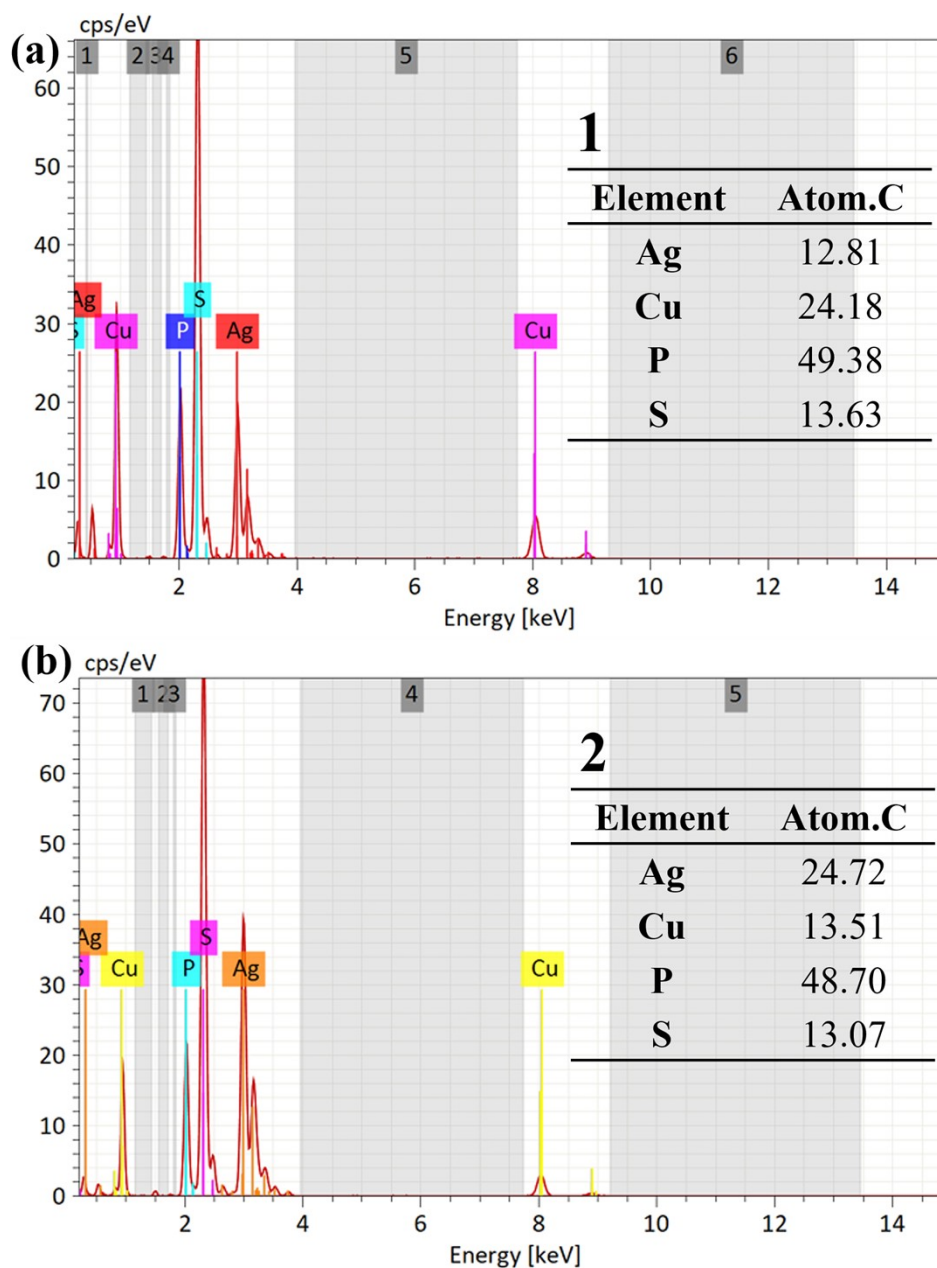
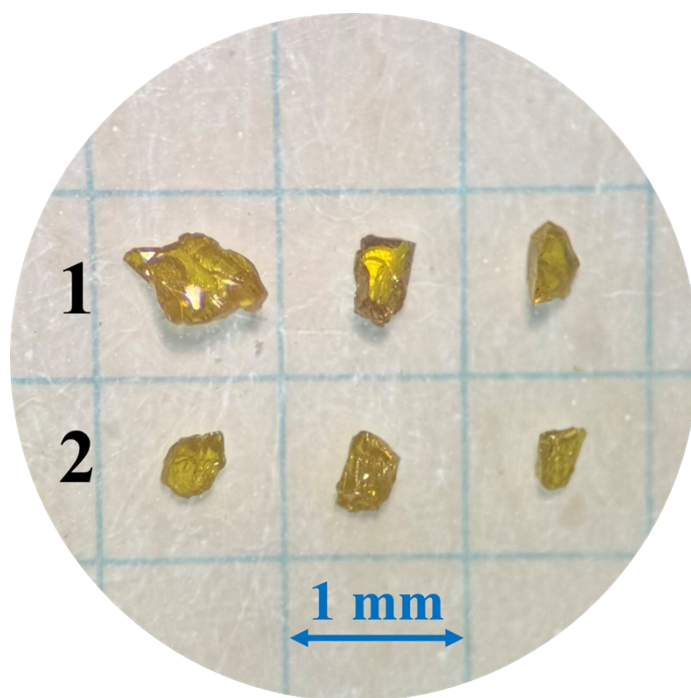


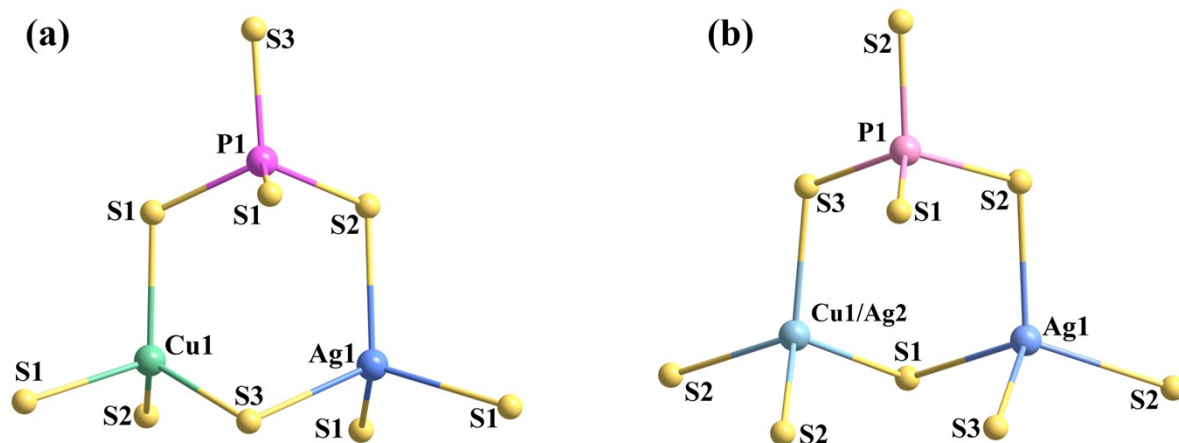
Figure S1. Rietveld refinement of Powder X-ray diffraction patterns for 1 (a) and 2 (b).



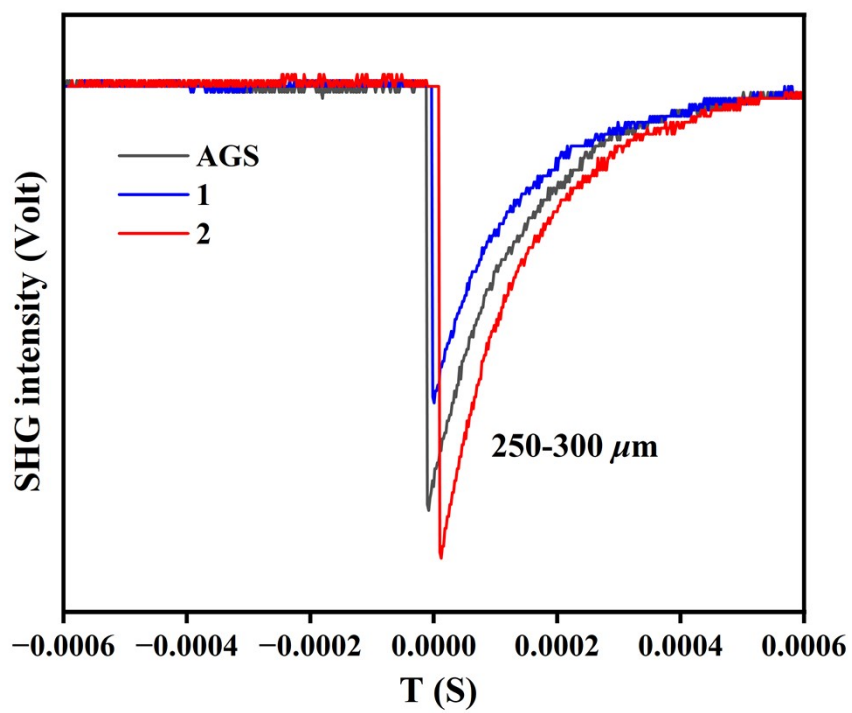
**Figure S2.** Energy dispersion spectroscopies of **1** (a) and **2** (b). Inset: the molar ratios of component elements.



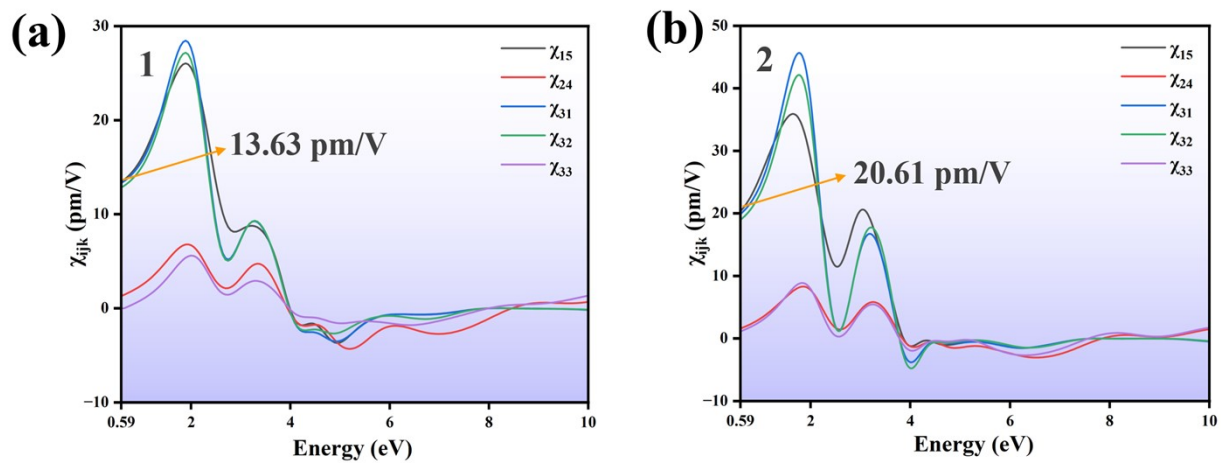
**Figure S3.** Crystal photographs of **1** and **2**.



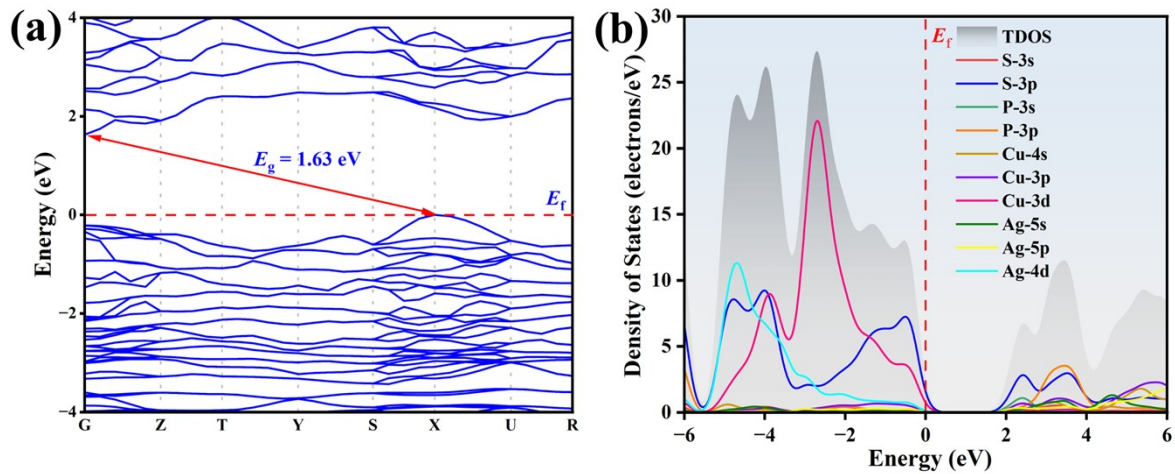
**Figure S4.** Coordination geometries of Ag/Cu and P atoms in **1** (a) and **2** (b).



**Figure S5.** The relative SHG signals of **1**, **2** and AgGaS<sub>2</sub> in the particle size of 250–300 μm



**Figure S6.** Calculated SHG tensors of **1** (a) and **2** (b).



**Figure S7.** Calculated band structure (a) and density of states (b) for **1**. The fermi level is set at 0 eV.

## 4. References

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