

### Supplemental Section A

**Table A1** Dominant species present at each sample's  $\text{pH}_{\text{PZC}}$  in Table I [1a, 2a-4a].

<i>Sample</i>	<i>Dominant Species</i>	<i>Experimental Point of Zero Charge (pH)</i>
BeO	BeO	10.2
ZnO	ZnO	9.3
Zn(OH) <sub>2</sub>	Zn(OH) <sub>2</sub>	7.8
Al(OH) <sub>3</sub>	Al(OH) <sub>3</sub>	5.0 – 5.2
$\alpha$ -Fe <sub>2</sub> O <sub>3</sub>	$\alpha$ -Fe <sub>2</sub> O <sub>3</sub>	9.04
$\alpha$ -FeO(OH)	$\alpha$ -FeO(OH)	7.4
$\gamma$ -FeO(OH)	$\gamma$ -FeO(OH)	6.7
ThO <sub>2</sub>	ThO <sub>2</sub>	9.0 – 9.3
PuO <sub>2</sub>	PuO <sub>2</sub>	9.0
HgO	HgO	7.3
SnO <sub>2</sub>	SnO <sub>2</sub>	6.6
TiO <sub>2</sub>	TiO <sub>2</sub>	6.7
WO <sub>3</sub>	WO <sub>3</sub>	~ 0.5
SiO <sub>2</sub>	SiO <sub>2</sub>	1.8

**Table A2** Dominant species present at each sample's  $\text{pH}_{\text{PZC}}$  in Table II [1a, 2a-6a] with the presence of nonzero crystal field splitting effect.

<i>Sample</i>	<i>Dominant Species</i>	<i>Experimental Point of Zero Charge</i>
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		<i>(pH)</i>
Co(OH) <sub>2</sub>	Co(OH) <sub>2</sub>	11.4
CuO	CuO	9.4 ± 0.4
Cr <sub>2</sub> O <sub>3</sub>	Cr <sub>2</sub> O <sub>3</sub>	7.0

## References

- 1a. Yoon R. H., Salman, T., Donnay, G., Predicting Points of Zero Charge of Oxides and Hydroxides, *Journal of Colloid and Interface Science*. 1979;70(3):483 - 93.
  
- 2a. Brookins D. G., *Eh-pH Diagrams for Geochemistry*, 1<sup>st</sup> Edition, Springer – Verlag Berlin Heidelberg. 1988.
  
- 3a. Takeno N., Atlas of Eh – pH Diagrams, Geological Survey of Japan Open File Report No. 419, National Institute of Advanced Industrial Science and Technology Research Centre for Deep Geological Environments. 2005.
  
- 4a. Pourbaix M., *Atlas of Electrochemical Equilibria in Aqueous Solutions*, Translator Franklin J. A., National Association of Corrosion Engineers, Houston TX. 1974
  
- 5a. Basharat F., Rana U. A., Shahid M., Serwar M., Heat treatment of electrodeposited NiO films

for improved catalytic water oxidation, RSC Advances. 2015;5:86713 - 22.

6a. Huang L. -F., Hutchison M. J., Santucci Jr. R. J., Scully J. R., Rondinelli J. M., Improved Electrochemical Phase Diagrams from Theory and Experiment: The Ni – Water System and Its Complex Compounds, The Journal of Physical Chemistry C. 2017;121(18):9782 - 89.

### Supplemental Section B

**Table B1** Structure parameters,  $\text{pH}_{\text{PZC}}$  and  $\text{pH}_{\text{PPZC}}$ , values for the metal-oxide/-hydroxide examples in **Table I** [1b, 2b <sup>e</sup>].

Dominant Species	C.N.	$Z_M$	L (Å)	$f_{I\%}$ (%)	$f_i$	$=f_i \times I\%$	Exp. $\text{pH}_{\text{PZC}}$	Cal. $\text{pH}_{\text{PPZC}}$	Ref.
BeO	4	+2	1.49	0.6321	1	0.6321	7.10	7.01	1b, 3b, 4b
ZnO	4	+2	1.978	0.5944	1	0.5944	10.20	10.31	1b, 5b
Zn(OH) <sub>2</sub>	4	+3	1.95	0.5944	1	0.5944	10.3	10.19	1b, 6b
$\alpha$ -Al <sub>2</sub> O <sub>3</sub>	6	+3	1.91	0.6321	1	0.6321	9.5	9.49	1b, 7b
$\alpha$ -Al(OH) <sub>3</sub>	6	+3	1.935	0.6321	1	0.6321	9.50	9.49	1b, 8b-10b
$\alpha$ -Fe <sub>2</sub> O <sub>3</sub>	5	+3	1.966	0.5145	1	0.5145	9.80	9.91	1b, 11b, 12b, 13b
$\gamma$ -FeO(OH)	4	+3	1.89	0.5145	1	0.5145	7.40	7.47	1b, 11b, 12b
$\alpha$ -FeO(OH)	5	+3	1.97	0.5145	1	0.5145	9.80	9.92	1b, 11b, 14b, 15b

ThO <sub>2</sub>	8	+4	2.425	0.7018	1	0.7018	11.00	10.58	<b>1b, 16b</b>
PuO <sub>2</sub>	8	+4	2.094	0.7083	1	0.7083	9.00	9.28	<b>1b, 17b</b>
HgO	2	+2	2.231	0.4727	1	0.4727	7.30	7.18	<b>1b, 11b, 18b</b>
SnO <sub>2</sub>	5	+4	1.84	0.5145	1	0.5145	6.60	6.50	<b>1b, 11b, 19b</b>
TiO <sub>2</sub>	6	+4	1.969	0.6321	1	0.6321	6.97	7.15	<b>1b, 20b</b>
WO <sub>3</sub>	6	+6	1.943	0.5551	1	0.5551	3.53	3.26	<b>1b, 21b, 22b</b>
SiO <sub>2</sub>	4	+4	1.64	0.5145	1	0.5145	2.00	1.77	<b>1b, 23b, 24b</b>

### **Beryllium Oxide: (BeO)**

The  $\text{pH}_{\text{PZC}}$  value used by Yoon et al. **[1b]** is for beryllium hydroxide ( $\text{Be}(\text{OH})_2$ ) **[25b]**, not BeO.

Therefore, the  $\text{pH}_{\text{PZC}}$  value for the oxide phase had to be obtained from the literature ( $\text{pH}_{\text{PZC}} = 7.10$ ) **[3b]**. In addition, the average surface bond length for this oxide was incorrect. Typically, surface bond lengths contract by 3% to 4% from those found in the interior bulk structure, for coordination numbers 4 thru 8 **[26b]**. BeO surface bonds are anomalous, as they have been found to contract by 10% from its bulk bond length **[27b]**. Therefore, the surface bond length, adjusted for this contraction, reduced  $L = 1.65 \text{ \AA}$  **[1b]** to  $L = 1.49 \text{ \AA}$ . Entering these structural changes resulted in a predicted  $\text{pH}_{\text{PPZC}} = 7.01$ , a difference of 0.09 pH units from its  $\text{pH}_{\text{PZC}}$ .

### **Gibbsite ( $\gamma\text{-Al}(\text{OH})_3/\alpha\text{-Al}(\text{OH})_3$ )**

There are multiple phases of aluminium hydroxide, with the gibbsite (hydrargillite) structure being the most thermodynamically stable phase **[28b]**. Examination of the Pourbaix diagrams supports the presence of only  $\gamma\text{-Al}(\text{OH})_3/\alpha\text{-Al}(\text{OH})_3$  **[29b-31b]** as the dominant structure across

the pH range for the PZC values of each aluminium hydroxide structure in **Table I [1b]**.

Therefore  $\gamma\text{-Al(OH)}_3/\alpha\text{-Al(OH)}_3$  was the only structure modelled in **Table I**.

The most basic  $\text{pH}_{\text{PZC}}$  for this material is  $\text{pH} = 9.5$ . It was obtained from a source where the gibbsite structure measured had also been confirmed **[32b]**. The average bond length given in **Table I [1b]** is shorter ( $L = 1.89 \text{ \AA}$ ) than that obtained by Saalfeld, Wedde, and Megaw **[8b, 9b]** ( $L = 1.902 \text{ \AA}$ ) for gibbsite. Use of the longer bond length in the model resulted in the  $\text{pH}_{\text{PPZC}}$  value closest to the experimental PZC value. The modelled value **[Eq. 6]** is for a  $\text{pH}_{\text{PPZC}} = 9.45$ , a difference of only 0.05 pH units.

### **Hematite ( $\alpha\text{-Fe}_2\text{O}_3$ )**

The bulk structure for hematite is corundum **[33b]**, where the C. N. = 6. When the structure factors from **Table I [1b]** were placed in **Eq. 6** the  $\text{pH}_{\text{PPZC}} = 11.46$ . This resulted in a  $\text{pH}_{\text{PPZC}}$  of 1.66 pH units above the most basic PZC value found in the literature, a  $\text{pH}_{\text{PZC}} = 9.80$  **[12b]**.

Work by Kraushofer et al. **[13b]** though determined that surface  $\text{Fe}^{+3}$  atoms are all 5-coordinated. Therefore a C. N. = 5 was placed in the model **[Eq. 6]**, along with the calculated bond length for this coordination using Shannon's ionic radii table **[11b]**. The  $\text{pH}_{\text{PPZC}}$  was found to be a  $\text{pH} = 9.91$ , within 0.11 pH units of the experimental value.

### **Lepidocrocite ( $\gamma\text{-FeO(OH)}$ )**

Based on the Pourbaix diagram  $\text{FeO(OH)}$  compounds are the **[29b-31b]** stable structures from a  $\text{pH} = 6$  thru 14. Work by Navaro et al. **[34b]** though determined that the stability range for

lepidocrocite is within a temperature range of  $10^{\circ}\text{C}$  to  $60^{\circ}\text{C}$ , from a  $\text{pH} = 4.0$  thru a  $\text{pH} = 8.0$ . The most basic PZC value found within this range is  $\text{pH}_{\text{PZC}} = 7.4$  [35b]. Placing the structure factors from **Table I [1b]** into the model [Eq.6], the  $\text{pH}_{\text{PPZC}} = 11.36$  more than 3 pH units above the range where this structure is stable. Placing a C.N. = 5 and a calculated bond length  $L = 1.97$  [11b] gives a  $\text{pH}_{\text{PPZC}} = 9.92$ , slightly less than 2.0 pH units above its stability range.

Work by Otte et al. [14b] though determined that the Fe<sup>2</sup> sites on the [0 1 0] face of this structure exhibit a four-fold coordination. This suggested that the coordination number should be reduced to 4, with a calculated  $L = 1.89$  [11b]. These structure factors resulted in a  $\text{pH}_{\text{PPZC}} = 7.46$ , only 0.06 pH units different from the experimental value. These modeling results [Eq. 6] support Otte et al.'s [14b] findings indicate that the global coordination number for  $\gamma\text{-FeO(OH)}$  surface Fe<sup>+3</sup> atoms have a C.N. = 4.

### **Goethite ( $\alpha\text{-FeO(OH)}$ )**

The surface anomaly at the hematite surface is also present for goethite. Modelling of the bulk and the [0 1 0] face of the goethite crystal determined that, like hematite, the surface has a C. N. = 5 [14b]. The most basic  $\text{pH}_{\text{PZC}}$  located in the literature was for a  $\text{pH}_{\text{PZC}} = 9.80$  [15b]. Therefore, the Fe – OH bond calculated from ionic radii gave an  $L = 1.97 \text{ \AA}$  [11b]. When placed in the model [Eq. 6] the  $\text{pH}_{\text{PPZC}} = 9.92$ , only 0.12 pH units different from the experimental  $\text{pH}_{\text{PZC}}$  value.

### **Thorium Dioxide ( $\text{ThO}_2$ )**

Thorium dioxide has a wide range of measured PZC values in the literature. Parks [36b] identified a range of values for  $\text{pH}_{\text{PZC}} = 4.2 - 11.0$ . The most basic value for thorium dioxide identified, was derived from an ion exchange experiment at its equilibrium pH point. This is where the material's surface reverses charge, and no further ions are adsorbed. The equilibrium point is also defined as the material's PZC, which for thorium dioxide is a  $\text{pH}_{\text{PZC}} = 11.0$  [35b]. An examination of both individual and compiled PZC values [3b 36b-40b] found no other values above this  $\text{pH}_{\text{PZC}}$ . Results of the modelled value gave a  $\text{pH}_{\text{PPZC}} = 10.58$ , a difference of only 0.46 pH units.

### **Mercuric Oxide (HgO)**

In Yoon et al.'s [1b] work they modelled the  $\text{pH}_{\text{PPZC}}$  using the cinnabar structure for HgS, which has a C.N. = 6 and  $L = 2.57 \text{ \AA}$  [1b, 41b]. The two common structures for HgO though are montroydite and cinnabar which consist of planar O-Hg-O zigzag chains that form these phases, both with a C.N. = 2 [19b]. Using the cinnabar structure (HgS) in Table I results in a  $\text{pH}_{\text{PPZC}} = 13.14$ , almost double the cited experimental  $\text{pH}_{\text{PZC}} = 7.3$  [1b]. Using the structure factors for HgO in the model with a C.N. = 2 and bond length =  $2.231 \text{ \AA}$  [11b] in Eq. 6 results in a  $\text{pH}_{\text{PPZC}} = 7.18$  a difference of 0.12 pH units.

### **Tin Oxide (SnO<sub>2</sub>)**

Using the global structure factors in Yoon et al. [6] with a C.N. = 6 for the Sn atom and a bond length = 2.053, gave a  $\text{pH}_{\text{PPZC}} = 9.73$ , a difference 3.13 pH units, above the experimental  $\text{pH}_{\text{PZC}} = 6.60$  [6]. Work by Merte et al. [19b] though determined that Sn atoms have a C.N. = 5 at the

surface. The Sn – O bond length for a five coordinated Sn atom  $\cong 1.84$  [11b], giving a predicted  $\text{pH}_{\text{PPZC}} = 6.84$ , a difference of 0.28 pH units.

### **Tungsten Trioxide ( $\text{WO}_3$ )**

There are seven [42b] different structures of tungsten Trioxide ( $\text{WO}_3$ ). At ambient temperature and pressure  $\text{WO}_3$  exhibits the monoclinic structure [37b]. Therefore, to model the surface structure of  $\text{WO}_3$  it was necessary to choose the most basic PZC value for this crystal structure. The PZC identified for this structure is a  $\text{pH}_{\text{PZC}} = 3.53$  [43b] with an average bond length of  $L = 1.943 \text{ \AA}$  [42b]. The modelled  $\text{pH}_{\text{PPZC}} = 3.26$ , a difference of only 0.26 pH units.

### **Quartz ( $\alpha\text{-SiO}_2$ )**

An examination of the PZC values presented in Parks [36b] work revealed a wide range of results for  $\text{SiO}_2$  with no indication as to whether they were obtained in either Region I or II. Therefore, the structure chosen to model was for  $\alpha\text{-SiO}_2$ , as there is a substantial body of literature for its PZC value and possess a C.N. = 4. Work by Alves Jr. And Baldo [24b] was chosen, as they determined the PZC values for multiple  $\alpha\text{-SiO}_2$  structures with a C.N. = 4. All their results were highly reproducible ( $\text{pH}_{\text{PZC}} \cong 2.0 \pm 0.1$ ).

To confirm the surface structure (i.e. average bond lengths), modelling work based on experimental values were examined. Work by Wang et al. [23b] provided the average surface bond length for  $\alpha\text{-SiO}_2$  as  $L = 1.64 \text{ \AA}$ , slightly longer than the value in **Table I [1b]** of  $L = 1.61$ . The results of this new structure information resulted in a  $\text{pH}_{\text{PPZC}} = 1.77$ , a difference of 0.23 pH units.

**Table B2** Structure parameters for metal-oxides/hydroxide samples in Table II [1b, 2b<sup>e</sup>] with the presence of nonzero crystal field splitting effect.

Dominant Species	C.N.	Z <sub>M</sub>	L (Å)	f <sub>i</sub> (%)	I <sub>%</sub> <sup>ε</sup>	= f <sub>i</sub> x I <sub>%</sub>	Exp. pH <sub>PZC</sub>	Cal. pH <sub>PPZC</sub>	Ref.
Co(OH) <sub>2</sub>	4	+2	1.92	1.0	0.5145	0.5145	11.40	11.07	1b, 11b, 18b
Cr <sub>2</sub> O <sub>3</sub>	5	+3	1.850	1.0	0.5944	0.5944	7.90	8.01	1b, 44b-46b

### Cobalt Hydroxide (Co(OH)<sub>2</sub>)

The pH<sub>PPZC</sub> = 13.68, using the original structure factors (i.e. C.N. = 6) from Table II [6] in Eq. 6 which is two pH units above the experimental pH<sub>PZC</sub> = 11.40 [1b]. There are two known cobalt hydroxide structures, α-Co(OH)<sub>2</sub> with a tetrahedral (C.N. = 4) coordination and β-Co(OH)<sub>2</sub> which exhibits an octahedral (C.N. = 6) coordination [47b]. The most common coordination number for cobalt (Co<sup>+2</sup>) is tetrahedral (C.N. = 4) [18b, 47b]. Using the structure factors for a tetrahedral coordinated Co - OH and calculated bond length, L = 1.93 Å [11b] resulted in a pH<sub>PPZC</sub> = 11.68, a difference of 0.28 pH units.

### Chromium Oxide (Cr<sub>2</sub>O<sub>3</sub>)

The most basic value for Cr<sub>2</sub>O<sub>3</sub> in the literature is for a pH<sub>PZC</sub> = 8.3 [3b]. Entering the materials bulk structure factors [1b] into Eq. 6 gives a pH<sub>PPZC</sub> = 10.80, 2.5 pH units above the most basic pH<sub>PZC</sub> found in the literature. Shortening or lengthening the bond length to allow the model to converge on the experimental pH<sub>PZC</sub> resulted in a non-viable structure.

As  $\text{Cr}_2\text{O}_3$  is the stable structure (i.e.  $Z_M = +3$ ) in each of the Pourbaix diagrams [29b-31b] examined, it eliminated varying the oxidation state as well. To be sure, though the oxidation states of +2 and +4 were modelled resulting in a  $\text{pH}_{\text{PPZC}} = 12.80$  and  $7.73$ , respectively. While the +4-oxidation state is within a half pH unit, the increase in the oxidation state results in a calculated bond length of approximately  $1.79\text{\AA}$  [11b] which reduces the  $\text{pH}_{\text{PPZC}} = 6.52$ , approximately 1.8 pH units below the materials experimental  $\text{pH}_{\text{PZC}}$ .

Therefore, the final factor which could be varied was the C.N. number. Chromium, with an  $Z_M = +3$  has only one common coordination number (C.N. = 6) in the Shannon table of ionic radii [11b]. A search of the literature on  $\text{Cr}_2\text{O}_3$  though revealed its surface structure is an exception, it possesses a C.N. = 5 [44b]. Density Functional Theory modelling of a  $\text{Cr}_2\text{O}_3$  monolayer surface generating hydrogen gave a surface bond length  $L = 1.85\text{\AA}$  [45b]. When entered the model [Eq. 6], a calculated value of  $\text{pH}_{\text{PPZC}} = 8.01$  was obtained, a difference of 0.11 pH units below its most basic average  $\text{pH}_{\text{PZC}}$  value [46b].

## References

- 1b. Yoon R. H., Salman, T., Donnay, G., Predicting Points of Zero Charge of Oxides and Hydroxides, Journal of Colloid and Interface Science. 1979;70(3):483 - 93.
  
- 2b. Pauling L., *The Nature of The Chemical Bond*, 3<sup>rd</sup> Edition, Cornell University Press, Ithaca New York. 1960.

- 3b. Kosmulski M., Compilation of PZC and IEP sparingly soluble metal oxides and hydroxides from literature, *Advances in Colloid and Interface Science*. 2009;152(1-2):14 – 25.
- 4b. Jaffe J. E., Zapol P., Atomic relaxation of the BeO surface, *Surface Science Letters*. 1997;381(1):L563 – 67.
- 5b. Marsalek R., Particle size and Zeta Potential of ZnO, *ICBEE Procedia*. 2014;9:13 – 7.
- 6b. Sun Z. X., Skold R. O., A Multi-Parameter Titration Method for the Determination of Formation pH for Metal Hydroxides, *Minerals Engineering*. 2001;14(11):1429 – 43.
- 7b. Bass B. Editor, *Electrophoretic Deposition (EDP) Advances in Applications and Research*, Nova Science Publishers, Inc. New York. 2017.
- 8b. Saalfeld H., Wedde M, Refinement of the crystal structure of gibbsite,  $\text{Al}(\text{OH})_3$ , *Zeitschrift für Kristallographie*. 1974;139(1-2):129 – 35.
- 9b. Megaw, H. D., The crystal structure of Hydrargillite,  $\text{Al}(\text{OH})_3$ . *Z. Kristallogr*. 1934;87(1-6):185 – 204.

10b. Manning B. A., Goldberg S., Modeling Competitive Adsorption of Arsenate with Phosphate and Molybdate on Oxide Minerals, *Soil Science of America Journal*.

1996;60(1):121 – 31.

11b. Shannon R. D., Revised effective ionic radii and systematic studies of inter-atomic distances in halides and chalcogenides, *Acta Cryst.* 1976;A32:751 - 67.

12b. Klein Wolterink J. K., Koopal L. K., Cohen Stuart M. A., Van Riemsdijk W. H., Surface charge regulation upon polyelectrolyte adsorption, hematite, polystyrene sulfonate, surface charge regulation Theoretical calculations and hematite-poly(styrene sulfonate) system, *Colloids and Surfaces A: Physicochem. Eng. Aspects.* (2006);291:13 – 23.

13b. Kraushofer F., Jakub Z., Bichler M., Hulva J., Drmota P. Weinold M, Schmid M., Setvin M., Diebold U., Blaha P. Parkinson G. S., Atomic-Scale Structure of Hematite  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> “R-Cut” Surface, *The Journal of Physical Chemistry C.* 2018;122(3):1657 - 69.

14b. Otte K., Schmahl W. W., Pentcheva R., Density functional theory study of water adsorption on FeOOH surfaces, *Surface Science.* 2012;606(21-22):1623 – 32.

15b. Wainipee W., Weiss D. J., Sephton M. A., Coles B. J., Unsworth C., Court R., The effect of crude oil on arsenate adsorption on goethite, *Water Research*. 2010;44(19):5673 – 83.

16b. Amphlett C. B., McDonald L. A., Redman M. J., Synthetic inorganic ion-exchange materials – II: Hydrous zirconium oxide and other oxides, *Journal of Inorganic and Nuclear Chemistry*. 1958;6(3):236 – 45.

17b. Coonley M. S., Editor, Carlson S. L., Designer, Lorusso E., Contributing Editor, Archuleta L., Printing Coordination, *Actinide Research Quarterly*, Los Alamos National Laboratory, New Mexico, U.S.A. 2004.

18b. Cotton F. A., Wilkinson G., *Advanced Inorganic Chemistry*, 4<sup>th</sup> Edition, John Wiley & Sons, New York, Chichester, Brisbane, Toronto. 1981.

19b. Merte L. R., Jorgensen M. S., Pussi K., Gustafson J., Shipilin M., Schaefer A., Zang C., Rawle J., Nicklin C., Thornton G. Linday R., Hammer B. Lundgren E., Structure of the SnO<sub>2</sub>(110) - (4 × 1) Surface, *Physical Review Letters*. 2017;119(9):1 – 6.

20b. Leffler M., Mirich A., Fee J., March S., Suib S. L., Part I: determination of a structure/property transformation mechanism responsible for changes in the point of zero charge of anatase titania with decreasing particle size, *RSC Advances*. 2024;14:30543 – 65.

21b. Lin H., Zhou F., Liu C. – P., Ozolins, Non-Grotthuss Proton Diffusion Mechanism in Tungsten Oxide Dihydrate First-Principal Calculations, *Journal of Materials Chemistry A*. 2014;31:1 – 10.

22b. Zhang S., Li H. Yang Z., Controllable synthesis of  $WO_3$  with different crystalline phases and its applications on methylene blue removal from aqueous solution, *Journal of Alloys and Compounds*. 2017;722(25):555 – 63.

23b. Wang X., Zhang Q., Li X., Ye J., Li L., Structural and Electronic Properties of Different Terminations for Quartz (001) Surfaces as Well as Water Molecule Adsorption on It: A First-Principles Study, *Minerals*. 2018;8(2):58:1 – 16.

24b. Alves Jr. J. A., Baldo J. B., The Behavior of Zeta Potential of Silica Suspensions, *New Journal of Glass and Ceramics*. 2014;4(2):29 – 37.

25b. Heimstra T., Van Riemsdijk W. H., Bolt G. H., Multisite Proton Adsorption Modeling at the Solid/Solution Interface of (hydr)oxides: A New Approach: Model description and evaluation of intrinsic reaction constants, *Journal of Colloid and Interface Science*. 1989;133(1):91 – 104.

26b. Sun C. Q., Li L., Tay B. K., Huang H., An extended 'quantum confinement' theory: Surface-coordination imperfection modifies the entire band structure of a nanosolid, *Journal of Physics D: Applied Physics*. 2001;34(24):3470 – 79.

27b. Jaffe J. E., Zapol P., Atomic relaxation of the BeO surface, *Surface Science Letters*. 1997;381(1):L563 – 67.

28b. Parks G. A., Free Energies of Formation and Aqueous Solubilities of Aluminum Hydroxides and Oxide Hydroxides at 25°C, *American Mineralogist*. 1972;57:1163 –89.

29b. Brookins D. G., *Eh-pH Diagrams for Geochemistry*, 1<sup>st</sup> Edition, Springer – Verlag Berlin Heidelberg. 1988.

30b. Takeno N., Atlas of Eh – pH Diagrams, Geological Survey of Japan Open File Report No. 419, National Institute of Advanced Industrial Science and Technology Research Center for Deep Geological Environments. 2005.

31b. Pourbaix M., *Atlas of Electrochemical Equilibria in Aqueous Solutions*, Translator Franklin J. A., National Association of Corrosion Engineers, Houston TX. 1974.

- 32b. Manning B. A., Goldberg S., Modeling Competitive Adsorption of Arsenate with Phosphate and Molybdate on Oxide Minerals, *Soil Science of America Journal*. 1996;60(1):121 – 31.
- 33b. Pauling L., Hendricks S. B., The Crystal Structure of Hematite and Corundum, *J. Am. Chem. Soc.* 1925;47(3):781 – 90.
- 34b. Navarro G. Acevedo R., Soto A. Herane M., Synthesis and characterization of Lepidocrocite and its potential applications in the adsorption of pollutant species, *Journal of Physics: Conference Series*. 2008;134(012023):1 – 5.
- 35b. Kozin P. A., The Interplay between Surface Structure. Particle Morphology and Counter ion Identity, Ph.D. Thesis, Umea University, Umea Sweden. 2014.
- 36b. Parks G. A., The Isoelectric Points of Solid Oxides, Solid Hydroxides, and Aqueous Hydroxo Complex Systems, *Chem. Rev.* 1965;65(2):177 – 98.
- 37b. Rao S. R., Leja J., *Surface Chemistry of Froth Flotation, Second Edition, Volume 1: Fundamentals*, Springer Science + Business Media, LLC, New York. 2004.
- 38b. Kosmulski M., *Chemical Properties of Material Surfaces*, Marcel Dekker, Inc., New York, Basil. 2001.

39b. Mahal H. S., Venkataramani B., Venkateswarlu K. S., Sorption properties of oxides IX: Effect of anions on the sorption of uranium (VI) and hydrous oxides, Proc. Indian Acad. Science - Chemical Science. 1982;91(4):321 – 7.

40b. Gayer K. H., Leider H., The Solubility of Thorium Hydroxide in Solutions of Sodium Hydroxide and Perchloric Acid at 25°, J. Am. Chem. Soc. 1954;76(23):5938 – 40.

41b. Carter O. W. L., Xu Y., Sadler P. J., Minerals in biology and medicine, RSC Adv. 2021;11(4):1939 – 51.

42b. Chatten R. Chadwick A. V., Rougier A., Linday P. J. D., The Oxygen Vacancy in Crystal Phases of WO<sub>3</sub>, J. Phys. Chem. B. 2005;109(8):3146 – 56.

43b. Zhang S., Li H. Yang Z., Controllable synthesis of WO<sub>3</sub> with different crystalline phases and its applications on methylene blue removal from aqueous solution, Journal of Alloys and Compounds. 2017;722(25):555 – 63.

44b. York S. C., Abee M. W., Cox D. F.,  $\alpha$ -Cr<sub>2</sub>O<sub>3</sub> (1 0  $\bar{1}$  2): surface characterization and oxygen adsorption, Surface Science. 1999;437(3):386 – 96.

45b. Zhao Z., Liu C., Tsai H. – S., Zhou J., Zhang Y., Wang T., Ma G., Qi C., Huo M., The strain and transition metal doping effects on monolayer Cr<sub>2</sub>O<sub>3</sub> for hydrogen evolution reaction: The first principal calculations, International Journal of Hydrogen Energy. 2021;47(88):37429 – 37.

46b. Blesa M. A., Magaz G., Salfity A., Weisz A. D., Structure and reactivity of colloidal metal oxide particles immersed in water, Solid State Ionics. 1997;101 – 103(Part 2):1235 – 41.

47b. Zheng W., Liu M., Yoon L., Lee Y. S., Electrochemical Instability of Metal-Organic Frameworks: In Situ Spectroelectrochemical Investigation of the Real Active Sites, ACS Catalysis. 2020;10(1):81 – 92.

### Supplemental Section C

**Table C1** Model structure factors, experimental pH<sub>PZC</sub> and calculated pH<sub>PPZC</sub> values for C.N.= 3

=> C.N.= 4 => C.N.= 5 [1c<sup>ε</sup>, Avg. of A site bond lengths [2c<sup>θ</sup>]

Intervals (Minutes)	C.N.	M	L (Å)	$f_i$	$I_{\%}^{\epsilon}$	$=f_i \times I_{\%}$	Exp. pH <sub>PZC</sub>	Cal. pH <sub>PPZC</sub>	References
Freshly Fractured Surface	3	+3	1.88	1.0	0.5145	0.5145	3.85	3.89	3c, 4c-7c

<b>80</b>	3	+3	1.88	0.55	0.5145	0.283	5.60	5.65	<b>3c, 4c-7c,8c, 2c, 9c-10c</b>
	4	+3	1.93 <sup>0</sup>	0.45	0.5145	0.232			
<b>120</b>	3	+3	1.88	0.16	0.5145	0.082	7.15	7.16	<b>3c, 4c-7c, 2c, 9c,10c</b>
	4	+3	1.93 <sup>0</sup>	0.84	0.5145	0.432			
<b>280</b>	4	+3	1.93 <sup>0</sup>	0.54	0.5145	0.278	8.80	8.82	<b>3c, 4c, 2c, 9c-13c</b>
	5	+3	1.966	0.46	0.5145	0.237			
<b>480</b>	4	+3	1.93 <sup>0</sup>	0.11	0.5145	0.0772	9.80	9.79	<b>3c, 14, 2c, 9c-13c</b>
	5	+3	1.966	0.89	0.5145	0.437			

**Table C2** Model structure factors, experimental pH<sub>PZC</sub> and calculated pH<sub>PPZC</sub> values for C.N.= 3

=> C.N.= 5 [1c<sup>€</sup>].

<b>Intervals (Minutes)</b>	<b>C.N.</b>	<b>M</b>	<b>L (Å)</b>	<b><math>f_i</math></b>	<b><math>I_{\%}^{\epsilon}</math></b>	<b><math>=f_i \times I_{\%}</math></b>	<b>Exp. pH<sub>PZC</sub></b>	<b>Cal. pH<sub>PPZC</sub></b>	<b>References</b>
<b>Freshly Fractured Surface</b>	3	+3	1.88	1.0	0.5145	0.5145	3.85	3.89	<b>3c, 4c-7c</b>
<b>80</b>	3	+3	1.88	0.715	0.5145	0.368	5.60	5.65	<b>3c, 4c-7c, 9c- 13c</b>
	5	+3	1.966	0.285	0.5145	0.147			
<b>120</b>	3	+3	1.88	0.470	0.5145	0.242	7.15	7.16	<b>3c, 4c-7c, 9c- 13c</b>
	5	+3	1.966	0.530	0.5145	0.273			
<b>280</b>	3	+3	1.88	0.200	0.5145	0.103	8.80	8.82	<b>3c, 4c-7c, 9c- 13c</b>
	5	+3	1.966	0.80	0.5145	0.412			
<b>480</b>	3	+3	1.88	0.04	0.5145	0.0283	9.80	9.80	<b>3c, 14c, 4c- 7c, 9c-13c</b>
	5	+3	1.966	0.96	0.5145	0.486			

## References

- 1c. Pauling L., *The Nature of The Chemical Bond*, 3<sup>rd</sup> Edition, Cornell University Press, Ithaca New York. 1960.
- 2c. Lu C., Amsler M., Chen C., Unraveling structure and bonding evolution of newly discovered iron oxide FeO<sub>2</sub>, *Phys. Rev. B*. 2018;98(054102):1 – 8.
- 3c. Yoon R. H., Salman, T., Donnay, G., Predicting Points of Zero Charge of Oxides and Hydroxides, *Journal of Colloid and Interface Science*. 1979;70(3):483 - 93.
- 4c. Lutzenkirchen J., Heberling F., Suplijika F., Preocanin T., Kallay N., Johann F., Weisser, Eng P. J., Structure – charge relationship – the case of hematite (0 1 0), *Faraday Discuss.* 2015;180:55 – 79.
- 5c. Jones F., Rohl A. L., Farrow J. B., van Bronswijk W., Molecular modeling of water adsorption on hematite, *Phys. Chem. Chem. Phys.* 2000;14:3209 – 16.
- 6c. Schottner L., Nefedov A., Yang C., Heissier S., Wang Y., Woll C., Structural Evolution of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> Surfaces Under Reduction Conditions Monitored by Infrared Spectroscopy, *Frontiers in Chemistry*. 2019;7(451) 1 – 12.

- 7c. Wasserman E., Rustad J. R., Felmy A. R., Hay B. P., Halley J. W., Ewald methods for polarizable surfaces with application to hydroxylation and hydrogen bonding on the (012) and (001) surfaces of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, *Surface Science*. 1997;385(2 – 3):217 – 39.
- 8c. Koretsky C. M., Sverjensky D. A., Sahai N., A model of surface site types on oxide and silicate minerals based on crystal chemistry: Implications for surface site types and densities, multi-site adsorption, surface infrared spectroscopy, and dissolution kinetics, *American Journal of Science*. 1998;298:349 – 438.
- 9c. Naveas N., Pulido R., Marini C., Hernandez-Montelongo J., Silvan M. M., First-principles calculations of hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) by self-consistent DFT+U+V. *iScience*. 2023;26(2):1 – 17.
- 10c. Tanwar K. S., Lo C. S., Eng P. J., Catalano J. G., Walko D. A., Brown Jr. G. E., Waychunas G. A., Chaka A. M., Tainor T. P., Surface diffraction study of hydrated hematite (1  $\bar{1}$  0 2) surface, *Surface Science*, 2007;601(2):460 – 74.
- 11c. Manceau A., Gates W. P., Surface Structural Model of Ferrihydrite, *Clays and Clay Minerals*. 1997;45(3) 448 – 60.
- 12c. Russell B., Payne M., Ciacci L. C., Density functional theory study of Fe(III)

adsorption and oxidation on goethite surfaces, *Phys. Rev. B, Condensed Matter*.  
2009;79(16):1 – 35.

13c. Gagne O. C., Hawthorne F. C., Bond-length distributions for ions bonded to oxygen: results for the transition metals and quantification of the factors underlying bond-length variation in inorganic solids, *IUCrJ*. 2020;7(Part 4):581 – 629.

14c. Klein Wolterink J. K., Koopal L. K., Cohen Stuart M. A., Van Riemsdijk W. H., Surface charge regulation upon polyelectrolyte adsorption, hematite, polystyrene sulfonate, surface charge regulation Theoretical calculations and hematite-poly(styrene sulfonate) system, *Colloids and Surfaces A: Physicochem. Eng. Aspects*. (2006);291:13 – 23.

### Supplemental Section D

**Table D1:** Surface Bond Lengths for Decreasing Particle Size in Region II of anatase titania [**1d**, **2d**].

Particle Diameter (nm)	Surface Bond Length (Å)	Ionic Content of Surface Bonds (%)	Point of Zero Charge (pH)
3.30	2.1000	99.5	1.54
5.307	2.0897	97.6	1.73
12.69	2.0521	91.6	3.04
16.13	2.0345	87.6	4.25
21.11	2.0091	79.8	5.47

23.52	1.9968	72.5	7.15
29.00 <sup>ψ</sup>	1.9689	62.1	7.19

<sup>ψ</sup>Projected Intercept point of Regions I and II

## References

- 1d. Leffler M., Mirich A., Fee J., March S., Suib S. L., Part I: determination of a structure/property transformation mechanism responsible for changes in the point of zero charge of anatase titania with decreasing particle size, RSC Advances. 2024;14:30543 – 65.
  
- 2d. Leffler M., Fee J., March S., Wu Y., Suib S. L., Part II: Superconductivity Observed in Magnetically Separated Nanoscale Anatase Titania at Ambient Temperature and Pressure in an Aqueous Environment at its Point of Zero Charge, RSC Advances. 2024;14:30317 - 35.