

Supporting Information

Spin polarized oxygen evolution reaction in chiral amorphous manganese-doped cobalt oxide: Enhancing electrocatalytic activity via lattice oxygen participation

Tae Hyung Kim^a, Hyeon Keun Cho^a, Seung Hyun Jae^a, Su Hwan Lee^a, Jeongu You^b, Youn Jeong Jang^b, Bongjun Yeom^b, Young Moo Lee^{a,} and Young-Hoon Kim^{a,*}*

^a Department of Energy Engineering, Hanyang University, 222 Wangsimni-ro, Seongdong-gu, Seoul 04763, Republic of Korea

^b Department of Chemical Engineering, Hanyang University, 222 Wangsimni-ro, Seongdong-gu, Seoul 04763, Republic of Korea

***Corresponding Author:** ymlee@hanyang.ac.kr, younghoonkim@hanyang.ac.kr

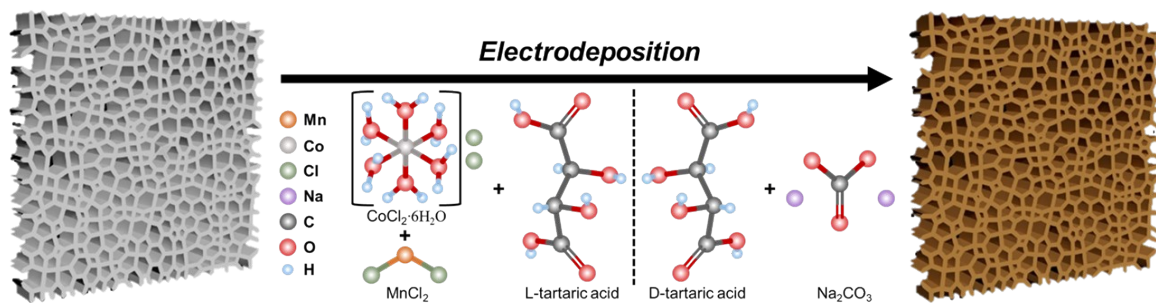


Fig. S1 Schematic illustration of the direct electrodeposition of chiral amorphous manganese-doped cobalt oxide ($\text{Mn}:\text{CoO}_x$) on Ni foam.

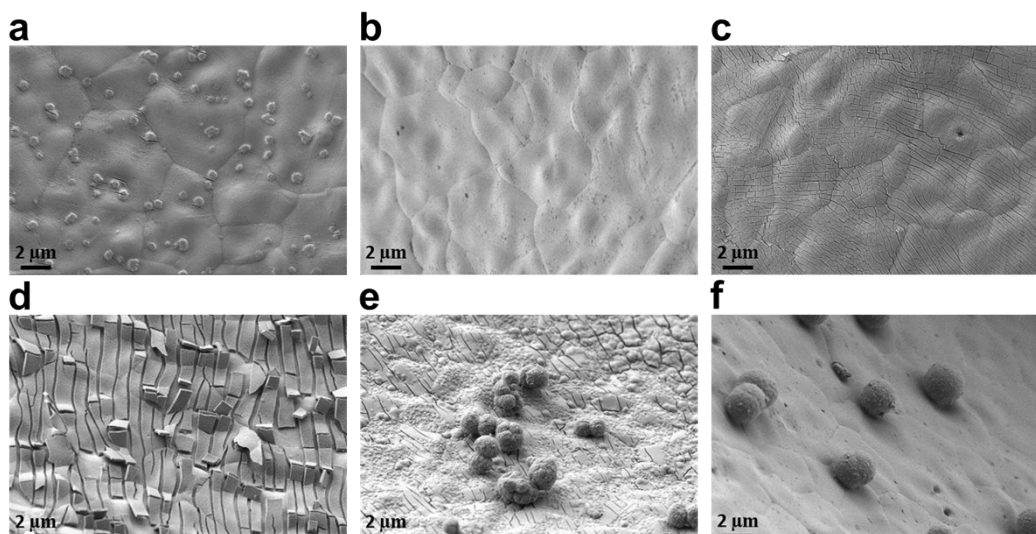


Fig. S2 Scanning electron microscopy (SEM) images of (a) $L\text{-CoO}_x$, (b) $L\text{-Mn}_{0.2}\text{:Co}_{0.8}\text{O}_x$, (c) $L\text{-Mn}_{0.25}\text{:Co}_{0.75}\text{O}_x$, (d) $L\text{-Mn}_{0.4}\text{:Co}_{0.6}\text{O}_x$, (e) $L\text{-Mn}_{0.6}\text{:Co}_{0.4}\text{O}_x$ and (f) $L\text{-MnO}_x$.

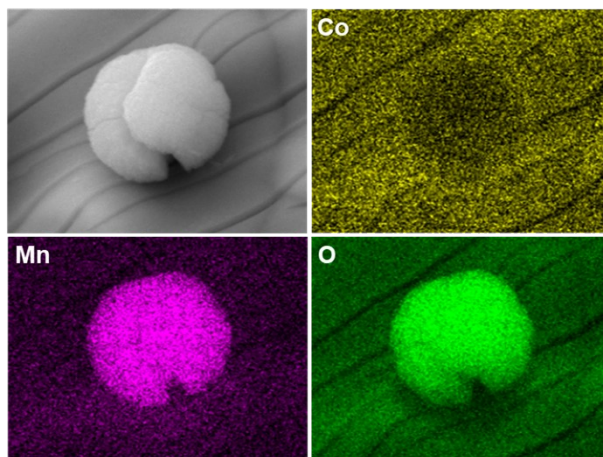


Fig. S3 SEM image and elemental mapping (Co, Mn and O) of $L\text{-Mn}_{0.6}\text{:Co}_{0.4}\text{O}_x$.

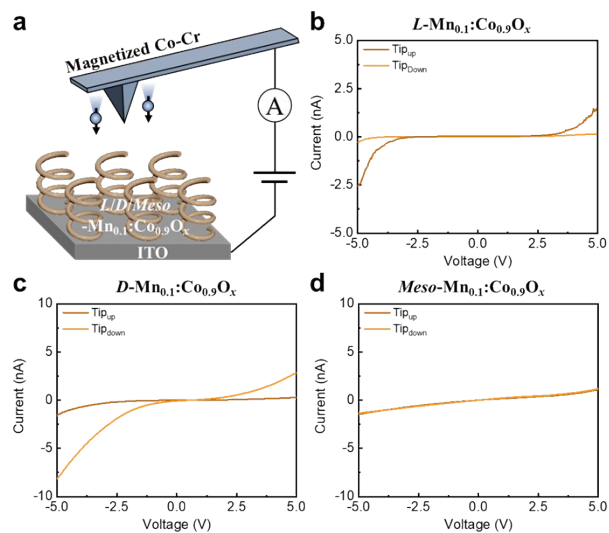


Fig. S4 (a) Schematic diagram of the magnetic conductive probe atomic force microscopy (mCP-AFM) used for thin film measurements. Spin-polarized average I - V curves of chiral (b) $L-Mn_{0.1}:Co_{0.9}O_x$, (c) $D-Mn_{0.1}:Co_{0.9}O_x$ and (d) achiral $meso-Mn_{0.1}:Co_{0.9}O_x$.

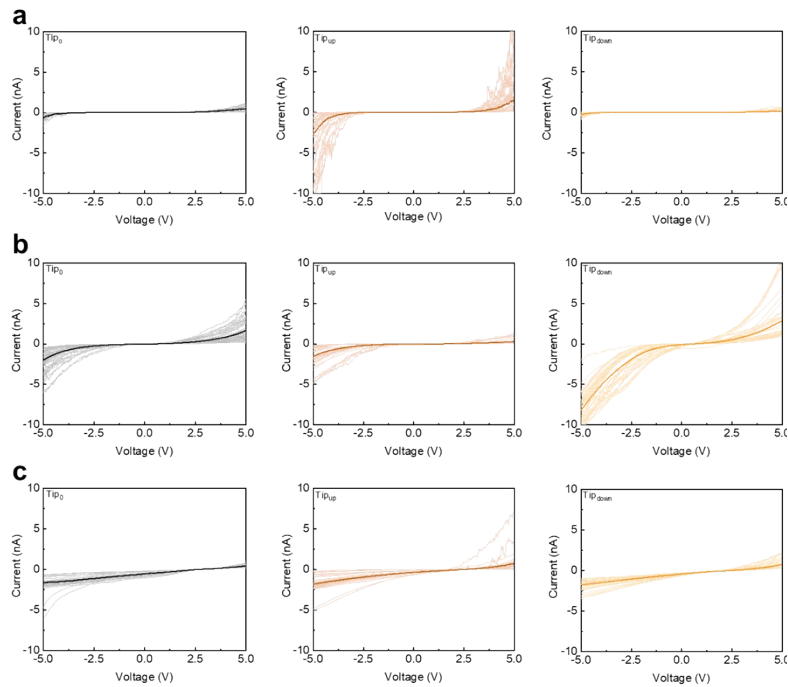


Fig. S5 Raw and averaged (bold) I - V curves of chiral (a) L - $Mn_{0.1}:Co_{0.9}O_x$, (b) D - $Mn_{0.1}:Co_{0.9}O_x$ and (c) achiral $meso$ - $Mn_{0.1}:Co_{0.9}O_x$ thin films, measured using mCP-AFM over a voltage range of -5 V to 5 V. Measurements were conducted with a non-magnetized tip (black), a tip magnetized upward (brown) and a tip magnetized downward (orange).

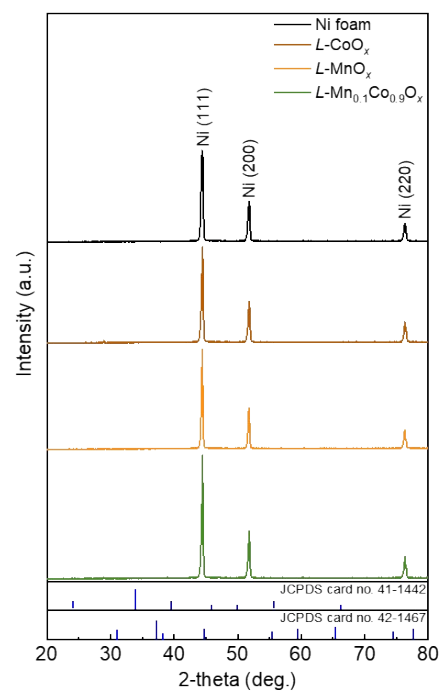


Fig. S6 XRD peaks of Ni foam, $L\text{-CoO}_x$ electrodeposited on Ni foam, $L\text{-MnO}_x$ electrodeposited on Ni foam and $L\text{-Mn}_{0.1}\text{Co}_{0.9}\text{O}_x$ electrodeposited on Ni foam.

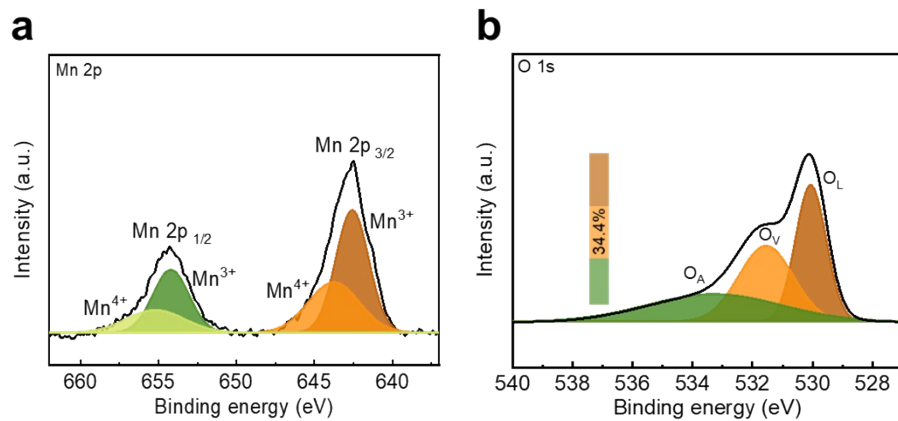


Fig. S7 X-ray photoelectron spectroscopy (XPS) spectra of $L\text{-MnO}_x$ thin film (a) Mn 2p and (b) O 1s peaks.

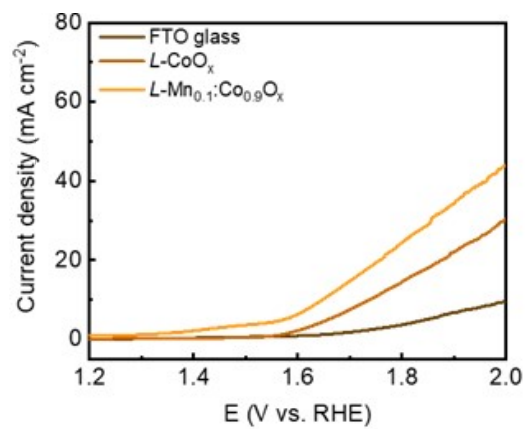


Fig. S8 LSV curves of bare FTO, $L\text{-CoO}_x/\text{FTO}$ and $L\text{-Mn}_{0.1}\text{Co}_{0.9}\text{O}_x/\text{FTO}$ electrodes.

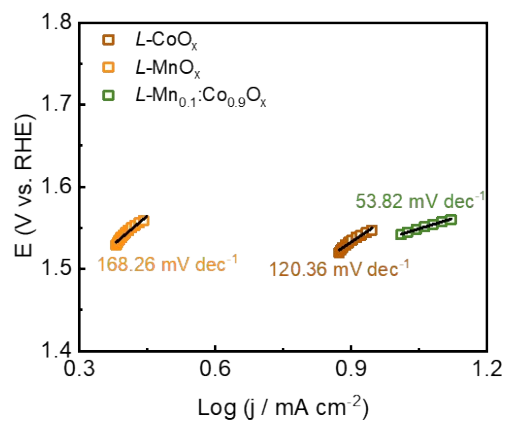


Fig. S9 Tafel plots of $L\text{-CoO}_x$, $L\text{-MnO}_x$ and $L\text{-Mn}_{0.1}\text{:Co}_{0.9}\text{O}_x$ measured 1 M KOH under a half-cell configuration without iR compensation.

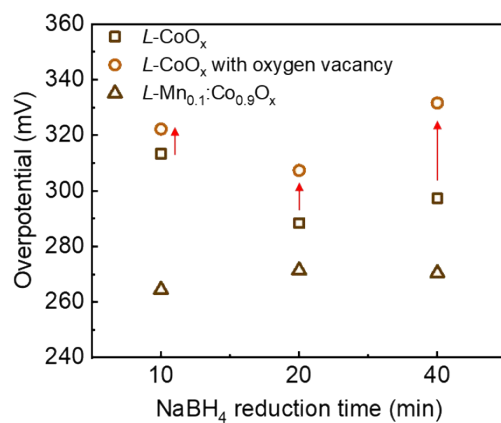


Fig. S10 Overpotential at 10 mA cm⁻² for pristine and NaBH₄-reduced *L*-CoO_x with pristine *L*-Mn_{0.1}Co_{0.9}O_x as a function of reduction time in 0.5 M NaBH₄, measured for independently prepared samples.

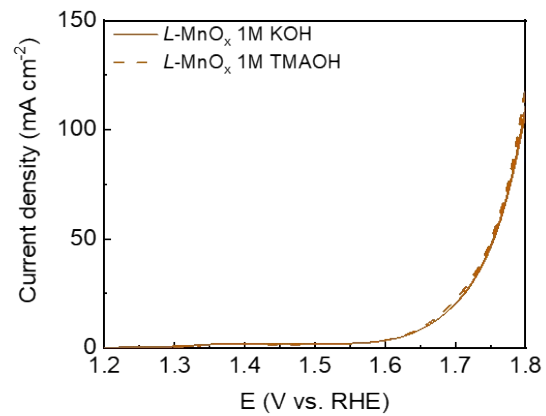


Fig. S11 LSV curves with 90% iR compensation of $L\text{-MnO}_x$ in 1 M tetramethylammonium hydroxide (TMAOH).

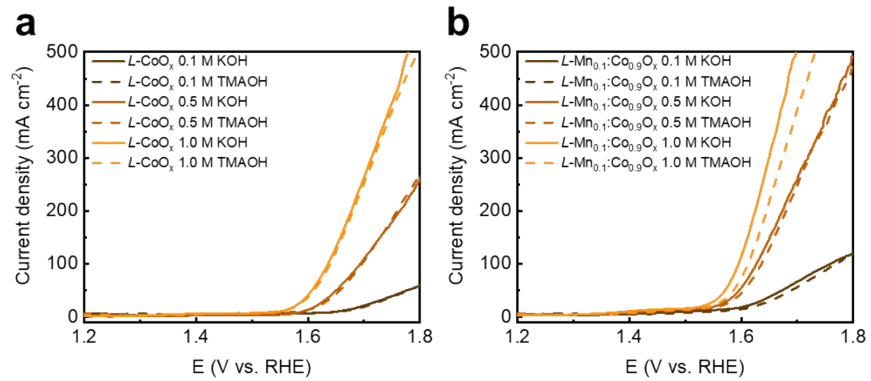


Fig. S12 LSV curves with 90% iR compensation of chiral (a) $L\text{-CoO}_x$ and (b) $L\text{-Mn}_{0.1}\text{:Co}_{0.9}\text{O}_x$ in KOH and tetramethylammonium hydroxide (TMAOH) at different concentrations of 0.1, 0.5 and 1.0 M.

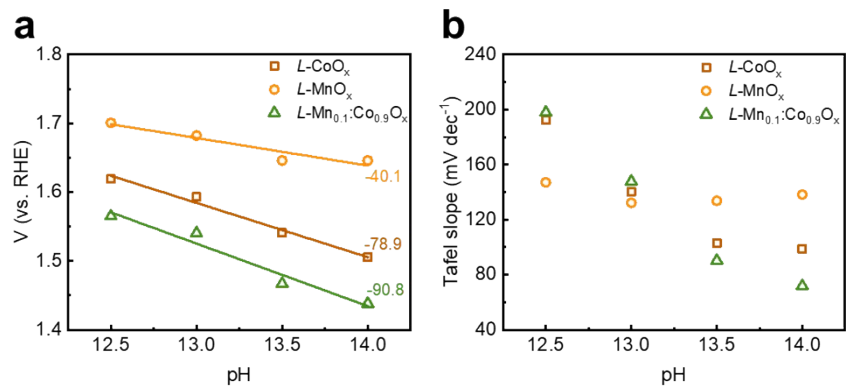


Fig. S13 (a) pH-dependent onset potential on RHE scale and (b) Tafel slope of $L\text{-CoO}_x$, $L\text{-MnO}_x$ and $L\text{-Mn}_{0.1}\text{:Co}_{0.9}\text{O}_x$.

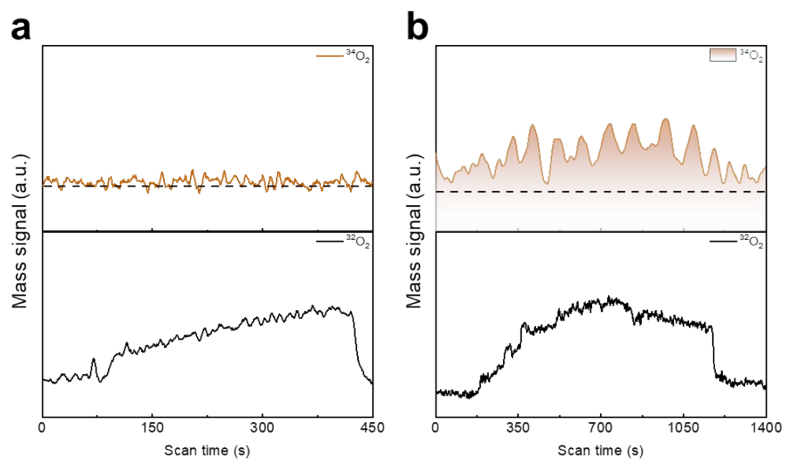


Fig. S14 DEMS signals corresponding to $^{34}\text{O}_2$ and $^{32}\text{O}_2$ evolution during the OER for (a) $L\text{-CoO}_x$ and (b) $L\text{-Mn}_{0.1}\text{:Co}_{0.9}\text{O}_x$.

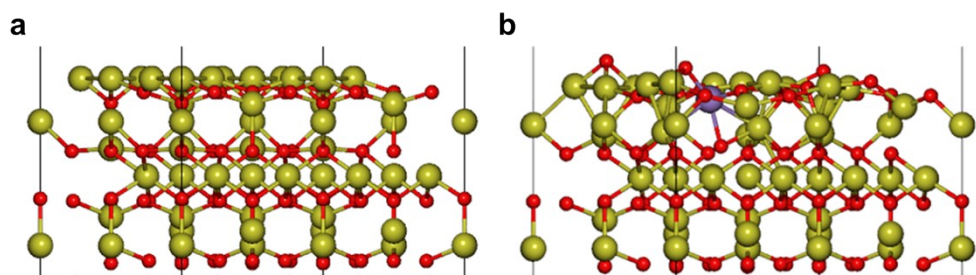


Fig. S15 Theoretical calculation models of (a) $L\text{-Co}_3\text{O}_4$ and (b) $L\text{-Mn:Co}_3\text{O}_4$ slabs.

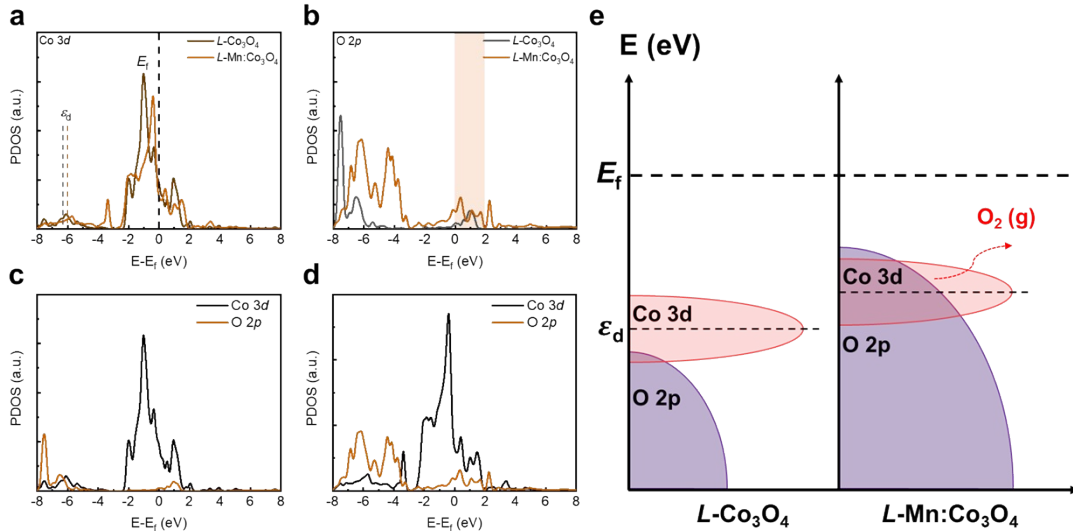


Fig. S16 DFT calculations of $L\text{-Co}_3\text{O}_4$ and $L\text{-Mn:Co}_3\text{O}_4$. Calculated PDOS of (a) Co $3d$ and (b) O $2p$ orbitals. Combined PDOS of Co $3d$ and O $2p$ orbitals for (c) $L\text{-Co}_3\text{O}_4$ and (d) $L\text{-Mn:Co}_3\text{O}_4$ slabs. (e) Schematic band diagrams of $L\text{-Co}_3\text{O}_4$ and $L\text{-Mn:Co}_3\text{O}_4$.

To further understand the electronic structure modulation induced by Mn incorporation, projected density of states (PDOS) calculations were performed using a representative Co_3O_4 (311) slab model with Mn substitution (Fig. S15[†]). As shown in Fig. S16a[†], Mn incorporation shifts the Co d -band center (ϵ_d) closer to the Fermi level (E_f) from -6.31 eV to -5.97 eV, indicating enhanced hybridization between metal–oxygen interactions. In addition, the O $2p$ PDOS exhibits an increased population of states near and above the Fermi level after Mn doping (Fig. S16b[†]), which has been associated with enhanced oxygen redox activity and participation of lattice oxygen in catalytic reactions. Moreover, a more pronounced overlap between Co $3d$ and O $2p$ orbitals is observed upon Mn incorporation (Fig. S16c, d[†]), suggesting increased metal–oxygen covalency. In particular, increased M–O covalency and the presence of O $2p$ states near the Fermi level have been strongly correlated with the transition from the AEM to the LOM pathways (Fig. S16e[†]).^{28–30} Therefore, these results suggest that Mn doping tunes the electronic structure toward a regime favorable for lattice oxygen participation by enhancing metal–oxygen covalency, increasing the availability of O $2p$ states near the Fermi level, facilitating oxygen vacancy formation and oxygen ion mobility.

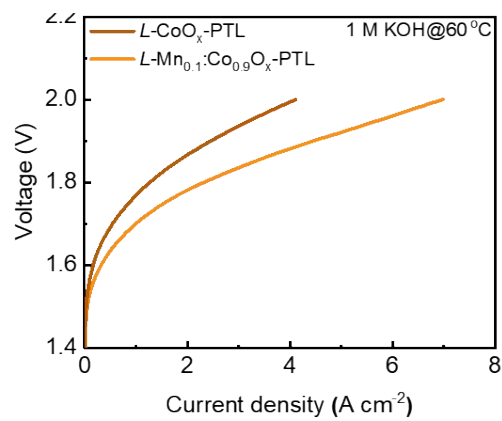


Fig. S17 Polarization curves of AEMWEs with *L*-CoO_{*x*}-PTL and *L*-Mn_{0.1}:Co_{0.9}O_{*x*}-PTL anodes measured in 1 M KOH at 60 °C.

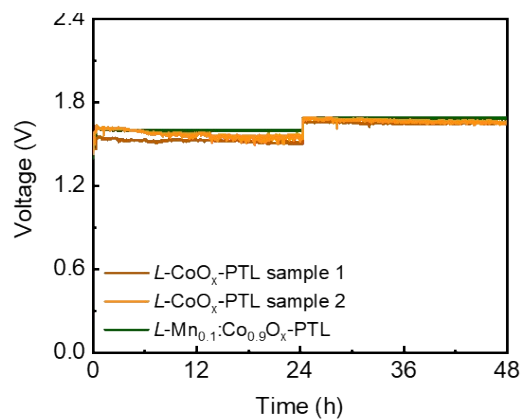


Fig. S18 Stabilization procedure prior to the durability test, involving current holds at 0.25 A cm^{-2} and 0.5 A cm^{-2} for 24 hours each, followed by operating at 1 A cm^{-2} for single-cell AEMWEs with *L*-CoO_{*x*}-PTL sample 1, sample 2 and *L*-Mn_{0.1}:Co_{0.9}O_{*x*}-PTL in 1 M KOH at 60 °C.

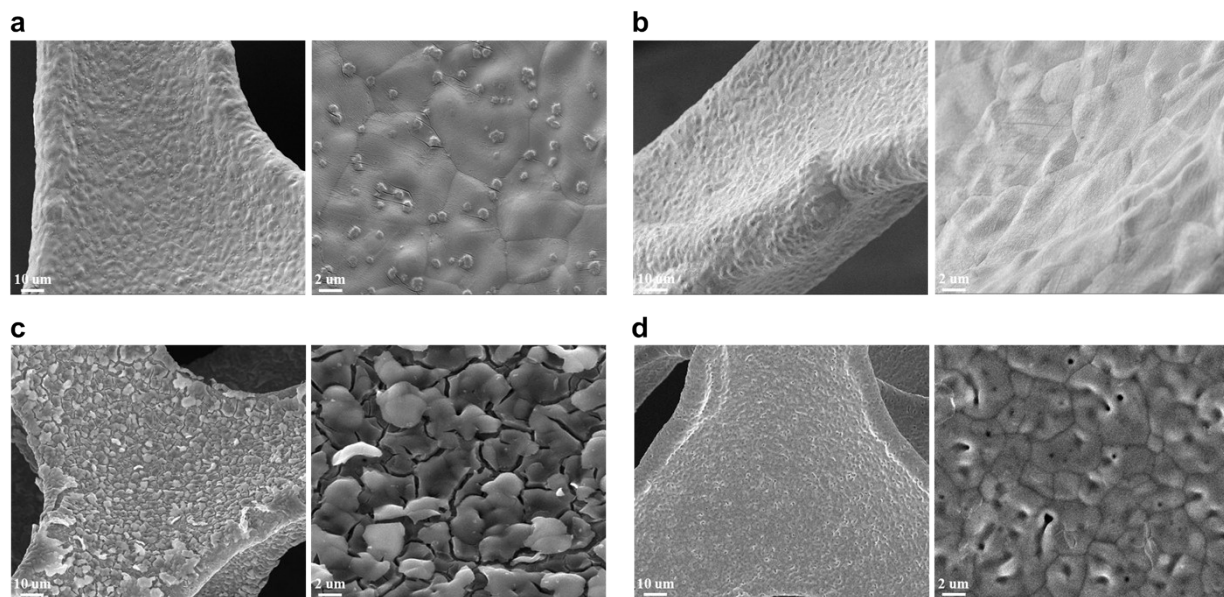


Fig. S19 SEM images of (a) $L\text{-CoO}_x$ -porous transport layer (PTL) and (b) $L\text{-Mn}_{0.1}\text{:Co}_{0.9}\text{O}_x$ -PTL before and (c) $L\text{-CoO}_x$ -PTL and (d) $L\text{-Mn}_{0.1}\text{:Co}_{0.9}\text{O}_x$ -PTL after 50 cyclic voltammetry (CV) cycles between 1 and 2 V (vs. RHE).

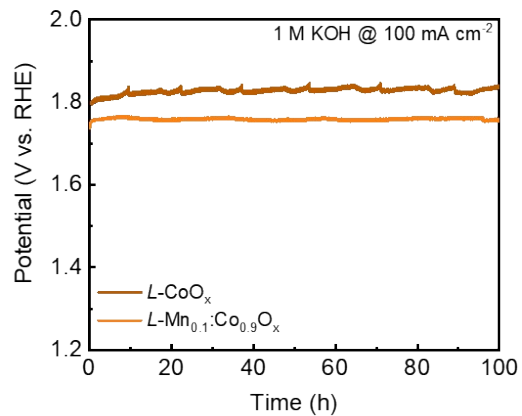


Fig. S20 Durability tests of $L\text{-CoO}_x$ and $L\text{-Mn}_{0.1}\text{:Co}_{0.9}\text{O}_x$ at 100 mA cm^{-2} for 100 hours in a half-cell configuration.

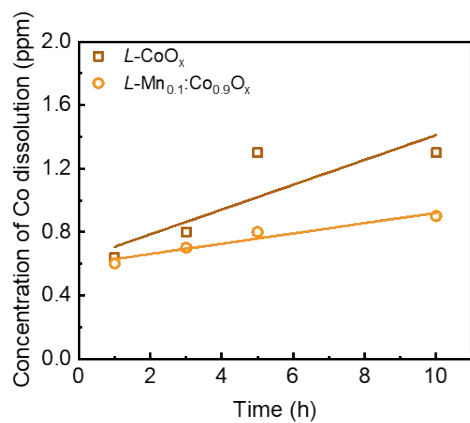


Fig. S21 Time-dependent Co dissolution during a 10 hour durability test at a constant current density of 500 mA cm⁻² for L-CoO_x and L-Mn_{0.1}:Co_{0.9}O_x in 1 M KOH.

Table S1. Summary of Mn fraction (Mn / (Mn + Co)) of chiral Mn:CoO_x samples determined by SEM-EDS.

Sample	Mn (at%)	Co (at%)	Mn / (Mn + Co) (%)
<i>L</i> -CoO _x	73.92	0	0.00
<i>L</i> -Mn _{0.05} :Co _{0.95} O _x	76.74	4.74	5.82
<i>L</i> -Mn _{0.1} :Co _{0.9} O _x	22.1	2.09	8.64
<i>L</i> -Mn _{0.15} :Co _{0.85} O _x	55.96	17.02	23.32
<i>L</i> -Mn _{0.2} :Co _{0.8} O _x	27.83	8.59	23.59
<i>L</i> -Mn _{0.25} :Co _{0.75} O _x	41.09	14.87	26.57
<i>L</i> -MnO _x	0	26.74	100.00

Table S2. Summary of overpotential (10mA cm⁻² without *iR* compensation), double layer capacitance (C_{dl}) results for chiral amorphous *L*-Mn:CoO_x.

Sample	Overpotential (mV)	C _{dl} (mF cm ⁻²)
<i>L</i> -CoO _x	327	1.32
<i>L</i> -Mn _{0.05} :Co _{0.95} O _x	319	1.53
<i>L</i> -Mn _{0.1} :Co _{0.9} O _x	266	1.64
<i>L</i> -Mn _{0.15} :Co _{0.85} O _x	309	1.57
<i>L</i> -Mn _{0.2} :Co _{0.8} O _x	313	1.47
<i>L</i> -Mn _{0.25} :Co _{0.75} O _x	318	1.63
<i>L</i> -MnO _x	426	9.9

Table S3. Summary of the reported properties of PGM-free AEMWEs

Anode catalyst	Cathode catalyst	Membrane	Electrolyte	Operation temperature (°C)	Current density (A cm ⁻² at V)	Ref.
CuCo ₂ O ₄	Pt/C	X37-50	1 M KOH	45	1.4 (1.9)	4
N-doped CoO	Pt/C	piperION A15R	1 M KOH	60	2 (2.06)	5
Ru _{8%} -Co ₃ O ₄	Pt/C	FAA-3-PK-130	1 M KOH	85	1 (2.06)	6
Fe@Co	Pt/C	X37-50	1 M KOH	60	2.44 (1.8)	7
Fe-Ni-Co	PtRu/C	PSTP-20	1 M KOH	80	10.7 (2)	8
Fe-Ni-Co	PtRu/C	Patterned PDTP	1 M KOH	80	14.42 (2)	9
<i>L</i> -Mn _{0.1} :Co _{0.9} O _x -PTL	PtRu/C	PDTP-25	1 M KOH	80	11.14 (2)	Our work

Supplementary References

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