

Supporting Information

Ultrahigh Responsivity UV Photodetector Enabled by Spatially Confined 2D β -Ga₂O₃/PdSe₂ van der Waals Heterojunction

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S1: Raman spectra of pure β -Ga₂O₃, pure PdSe₂ and the heterojunction.

As illustrated in Fig.S1, the Raman spectrum of the β -Ga₂O₃/PdSe₂ heterojunction overlapping region is essentially consistent with that of the pure PdSe₂. Specifically, the characteristic A_{1g} (176 cm⁻¹), B_{1g} (210 cm⁻¹), and A_{2g} (303 cm⁻¹) vibrational modes of PdSe₂ remain well-resolved, showing no obvious peak shift or broadening compared to the pure component. In contrast, the characteristic Raman peaks of β -Ga₂O₃ in the heterojunction region are almost undetectable. This phenomenon likely originates from two core factors and reveals critical interfacial properties of the β -Ga₂O₃/PdSe₂ van der Waals heterojunction:

(1) Weak interfacial interaction without chemical bonding: The negligible variation in PdSe₂'s Raman peaks indicates that the formation of the heterojunction does not induce obvious lattice distortion or strong chemical bonding at the interface. The intrinsic crystal structure of PdSe₂ is well preserved—a hallmark of high-quality van der Waals heterojunctions dominated by weak van der Waals forces. This avoids the generation of interfacial defect states caused by chemical bonding or lattice mismatch, laying a solid foundation for efficient carrier transport;

(2) Optical absorption dominance of PdSe₂: The undetectable Raman peaks of β -Ga₂O₃ in the heterojunction region are mainly attributed to the higher optical absorption coefficient of PdSe₂ at the 532 nm excitation wavelength (used for Raman testing) compared to β -Ga₂O₃. The incident laser is predominantly absorbed by the PdSe₂ layer, leading to the suppression of β -Ga₂O₃'s Raman signal. This is a common optical phenomenon in van der Waals heterostructures with asymmetric optical absorption, rather than a reflection of the interfacial coupling strength^[1].

Notably, the weak van der Waals interfacial interaction observed in Raman characterization is fully consistent with the atomically sharp, defect-free interface revealed by cross-sectional STEM

imaging (Fig. 2a). This clean interface minimizes trap-assisted recombination and provides a smooth channel for carrier transfer, which is directly verified by the device's ultralow dark current and high photo-to-dark current ratio.

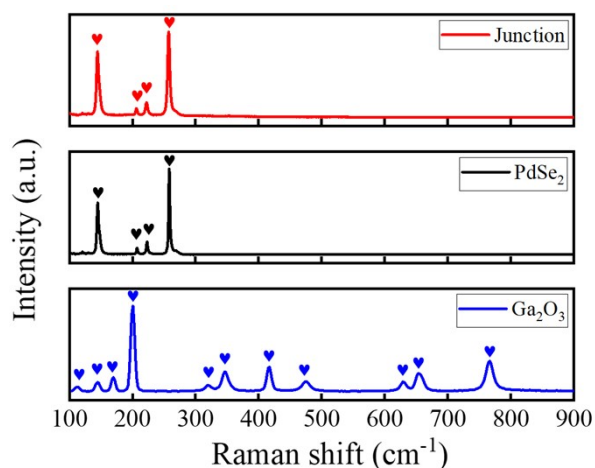


Fig. S1 Raman spectra of β -Ga₂O₃, PdSe₂ and the junction.

S2: Performance comparison of representative β -Ga₂O₃-based photodetectors operating in photoconductive mode.

Table S1 Performance comparison of representative β -Ga₂O₃-based photodetectors operating in photoconductive mode.

Device	Bias (V)	Responsivity (A/W)	Photo-to dark current ratio	Reference
MSM β -Ga ₂ O ₃ micro-flake	30	1.68	1.37×10^3	[2]
MSM β -Ga ₂ O ₃ films	5	14.09	4.9×10^4	[3]
β -Ga ₂ O ₃ /Nb ₂ C	5	28	213	[4]
β -Ga ₂ O ₃ /NSTO	-10	43.31	20	[5]
β -Ga ₂ O ₃ /MoS ₂	-10	0.00721	10^3	[6]
β -Ga ₂ O ₃ /CsCu ₂ I ₃	-1	178.9	5.4×10^4	[7]
β -Ga ₂ O ₃ /Graphene	-10	14.5	3.3×10^3	[8]
β -Ga ₂ O ₃ /PdSe ₂	5	6.17×10^4	2.5×10^6	Our work

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