## **Electronic Supplementary Information**

# M[Y(BH<sub>4</sub>)<sub>4</sub>] and M<sub>2</sub>Li[Y(BH<sub>4</sub>)<sub>5</sub>Cl] (M=Rb, Cs): New Borohydride Derivatives of Yttrium and Their Hydrogen Storage Properties

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Table S1. Description of the samples and the results of quantitative phase analysis. Dominating yttrium-containing products are bolded. + - "present"; — - "absent". M = Rb: x = 2.28; M = Cs: x = 1.8 (refined for samples Rb A165 and Cs B175, respectively).

Sample	Rb 0	Rb A	Rb A165	Rb A175	Rb A180	Rb A185	Rb A190	Rb A225	Rb A300
Method of preparation	RbCl + LiBH <sub>4</sub> milled 60 min. + $Y(BH_4)_3 \& 3LiCl$	RbBH <sub>4</sub> + Y(BH <sub>4</sub> ) <sub>3</sub> & 3LiCl milled 60 min.	<b>Rb A</b> heated to 165 °C and rapidly cooled	<b>Rb A</b> heated to 175 °C and rapidly cooled	<b>Rb A</b> heated to 180 °C and rapidly cooled	<b>Rb A</b> heated to 185 °C and rapidly cooled	<b>Rb A</b> heated to 190 °C and rapidly cooled	<b>Rb A</b> heated to 225 °C and rapidly cooled	<b>Rb A</b> heated to 300 <sup>o</sup> C and rapidly cooled to RT
	milled 20 min.		to RI	to RI	to RI	to RI	to RI	to RI	
Phase analysis (normalised to 3 mol of CI) [molar fractions]									
Rb[Y(BH <sub>4</sub> ) <sub>4</sub> ]	0.66	0.90	0.67	+	+				
$Rb_2Li[Y(BH_4)_{6-x}Cl_x]$			0.078	+	+	+	+	+	
$Y(BH_4)_3$	0.03								
LiCl	3.00	3.00	2.82	+	+	+	+	+	+
Remarks		very weak diffraction peaks from unidentified phase(s)		diffraction peaks from unidentified phase(s)	diffraction peaks from unidentified phase(s)	very <b>strong</b> diffraction peaks from unidentified phases	very <b>strong</b> diffraction peaks from unidentified phase(s)	diffraction peaks from unidentified phase(s) slightly extinct	diffraction peaks from <b>RbBH</b> <sub>4</sub> , <b>Rb(BH<sub>4</sub>)<sub>1-x</sub>Cl<sub>x</sub></b> ; weak diffraction peaks from unidentified phase(s)

Sample	Cs 0	Cs A	Cs A130	Cs A215	Cs A300	Cs B	Cs B175
Method of preparation	CsCl + LiBH <sub>4</sub> milled 60 min. + Y(BH <sub>4</sub> ) <sub>3</sub> & 3LiCl milled 30 min.	CsBH <sub>4</sub> + Y(BH <sub>4</sub> ) <sub>3</sub> & 3LiCl milled 60 min.	Cs A heated to 130 °C and rapidly cooled to RT	Cs A heated to 215 °C and rapidly cooled to RT	Cs A heated to 300 °C and rapidly cooled to RT	$2CsBH_4 +$ Y(BH_4) <sub>3</sub> & 3LiCl milled 30 min.	<b>Cs B</b> heated to 175 <sup>o</sup> C and rapidly cooled to RT
Phase analysis (normalised to 3 mol of Cl) [molar fractions]							
$Cs[Y(BH_4)_4]$	+	0.36	0.82	0.50			
$Cs_2Li[Y(BH_4)_{6-x}Cl_x]$	+	0.26	0.057	0.21		+	0.82
Y(BH <sub>4</sub> ) <sub>3</sub>	+	0.16					
LiCl	+	2.54	2.90	2.63	+	+	1.53
Remarks					diffraction peaks from <b>CsBH</b> <sub>4</sub> ; weak diffraction peaks from unidentified phase(s)	poorly crystalline sample	improved crystallinity in respect to <b>Cs B</b> ; very weak diffraction peaks from unidentified phase(s)

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Figure S1. PXD patterns of the sample **Rb** A heated to specified temperatures and quickly cooled to RT.



Figure S2. PXD patterns of the samples of caesium compounds heated to specified temperatures and quickly cooled to RT.

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Figure S3. The FTIR spectra of  $Rb[Y(BH_4)_4]$ , sample **Rb A**, and the reactants,  $RbBH_4$  and  $Y(BH_4)_3$ .



Figure S4. Coordination of  $[Y(BH_4)_4]^{-1}$  anion by  $Rb^+$  cations in  $Rb[Y(BH_4)_4]$ .



Figure S5. Coordination of  $Rb^+$  by  $[Y(BH_4)_4]^-$  anions in  $Rb[Y(BH_4)_4]$ .



Figure S6. The FTIR spectra of the solid decomposition products of  $Rb[Y(BH_4)_4]$ .



*Figure S7. Time-resolved mass spectra of gaseous decomposition products of the sample Rb A. Note the absence of meaningful amount of volatile boron hydrides. Heating rate: 5 K min<sup>-1</sup>.* 



*Figure S8. Time-resolved mass spectra of gaseous decomposition products of the sample Cs A. Note the absence of meaningful amount of volatile boron hydrides. Heating rate: 5 K min<sup>-1</sup>.* 

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Figure S9. Time-resolved mass spectra of gaseous decomposition products of the sample **Cs B**. Note the absence of meaningful amount of volatile boron hydrides. Heating rate:  $5 \text{ K min}^{-1}$ .

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Figure S10. Rietveld plot of the sample **Rb 0**. The Bragg reflections of the three crystalline phases present in the sample have been marked.



*Figure S11. Rietveld plot of the sample Rb A165. The Bragg reflections of the three crystalline phases present in the sample have been marked.* 



*Figure S12. Rietveld plot of the sample Cs A130. The Bragg diffraction peaks of the three crystalline phases present in the sample have been marked.* 



*Figure S13. Rietveld plot of the sample Cs B175. The Bragg reflections of the two crystalline phases present in the sample have been marked.* 

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### Table S2. Symmetrisation of the Y–Rb sublattice of $Rb[Y(BH_4)_4]$

Max deviation	0.0529 Å	
Tolerance used	0.2 Å	
Symmetry group	Pnma	(was $P2_1/c$ )
IT number	62	(was 14)
Option	Origin-1	
a, b, c	12.239 Å, 8.1635 Å, 10.4712 Å	(were 8.1635 Å, 12.2390 Å, 13.2299 Å)
alpha, beta, gamma	90.00, 90.00, 90.00	(were 90.00, 127.678, 90.00)

### Table S3. Symmetrisation of the Y–Cs sublattice of $Cs[Y(BH_4)_4]$

Max deviation	0.000 Å	
Tolerance used	0.001 Å	
Symmetry group	I4 <sub>1</sub> /amd	(was I4 <sub>1</sub> /a)
IT number	141	(was 88)
Option	Origin-1	
a, b, c	7.60845 Å, 7.60845 Å, 18.5560 Å	(no change)
alpha, beta, gamma	90.00, 90.00, 90.00	(no change)

### Remarks on the thermal stability of mixed-cation borohydrides.

As it is well known, metal hydrides (including complex hydrides such as alanates and borohydrides) are thermodynamically and thermally stabilised in a basic environment [see e.g. ref. 18]. In the most simple case of a metal hydride,  $M^{(n+)}H_n$ , Lewis bases decrease the net charge of a cation  $M^{(n+)}$ , which in turn causes stabilisation of compound, i.e. decrease the electrochemical potential – the driving force for  $M^{(n+)}$  reduction to  $M^0$ .

Such stabilisation may be achieved for metal borohydrides,  $M(BH_4)_n$ , by forming quasi-ternary  $M[Y(BH_4)_4]$  salts:

$$\mathsf{MBH}_4 + \mathsf{Y}(\mathsf{BH}_4)_3 \to \mathsf{M}[\mathsf{Y}(\mathsf{BH}_4)_4]. \tag{S1}$$

Yttrium borohydride, crystallising as a 3D network with bridging borohydride groups forming a distorted octahedron around  $Y^{3+}$  cation, acts as a Lewis acid in reaction with the most basic among the tested MBH<sub>4</sub> substrates, resulting in derivatives which show discrete  $[Y(BH_4)_4]^-$  anions.

Based on systematic investigations for a series of derivatives of yttrium borohydride as model systems we can now qualitatively verify the influence of Lewis acidity of cation  $M^+$  on the thermal stability of  $M[Y(BH_4)_4]$  systems. Although these materials constitute a single-crystalline phase or may be a fine mixture of MBH<sub>4</sub> and Y(BH<sub>4</sub>)<sub>3</sub> (for M=Li, Na), and they may decompose according to various mechanisms, yet their thermal stabilities (as quantified here by the temperature of the fastest mass loss), reveal a simple trend, Fig. S14.



Fig. S14. The temperature of maximum rate of mass loss (DTG) for a series of derivatives of yttrium borohydride, i.e. composites  $MBH_4/Y(BH_4)_3$ , M = Li, Na, and compounds  $M[Y(BH_4)_4]$ , M = K, Rb, Cs,  $(CH_3)_4N - TMA$ ,  $(n-C_4H_9)_4N - TBA$  as compared with parent  $Y(BH_4)_3$ . Heating rate: 5 K min<sup>-1</sup>; every sample was synthesised in similar conditions and contained LiCl.

While the relation is not monotonic (due to a lot of factors influencing the thermal stability of a particular compound or composite), the investigated systems may be divided easily into three groups with increasing thermal stability along with Lewis basicity: Tdec(M=Li) << Tdec(M=Na) ~Tdec(M=K) ~Tdec(M=Rb) ~Tdec(M=Cs) << Tdec(M=TMA) > Tdec(M=TBA).

The highest decomposition temperatures are observed for the two compounds containing tetraalkylammonium cations,  $R_4N^+$  (TMA, TBA) which are the least acidic (in a Lewis sense) among the borohydrides studied. These organic cations are bulky enough to place their positive charge away from the  $[Y(BH_4)_4]^-$  anion and increase  $T_{dec}$  by up to 25 °C as compared to  $Y(BH_4)_3$ . On the opposite extreme, LiBH<sub>4</sub> acts as a strong Lewis acid; its presence decreases the thermal decomposition temperature of LiBH<sub>4</sub>/Y(BH<sub>4</sub>)<sub>3</sub> composite by 10 °C with respect to parent  $Y(BH_4)_3$ . These effects are not large but they qualitatively agree with what is expected based on the Lewis basicity scale. Recall that Li<sup>+</sup> forms KLi(BH<sub>4</sub>)<sub>2</sub> salt with basic KBH<sub>4</sub>, but its Lewis acidity is insufficient to form Li(BH<sub>4</sub>)<sub>2</sub><sup>-</sup> anion with  $Y(BH_4)_3$ . Li-based system behaves unlike the system composed of the heavier alkaline metals, M = Na, K, Rb, Cs which show almost invariant  $T_{dec}$ . Such dissimilarity of lithium compounds to the rest of alkaline metals.

The influence of Lewis basicity of  $M^+$  cation on the decomposition temperature clearly allows for fine property tuning of metal borohydrides. It does not lead to particularly impressive impact on the  $T_{dec}$  value for yttrium salts, but it is a valid concept which could be used for other M-BH<sub>4</sub> systems as well.