Electronic Supplementary Information for

Magnetic Anisotropy in a spin 1/2 Quasi-One-Dimensional Antiferromagnetic Copper(II) Complex CuCl₂(pdz) with a Staggered g-tensor

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ESR spectra



Figure S1 Orientation dependence of ESR spectra at room temperature.

Analyzing detail of deconvolution of the local g-tensor

When considering the direction cosines (l, m, n) of the external field (H) with respect to the principal axes of the local *g*-tensor (g_x, g_y, g_z) , observed *g*-value is given by

$$g^2 = g_x^2 l^2 + g_y^2 m^2 + g_z^2 n^2$$

where x, y and z shows the Cl-Cu-Cl, N-Cu-N, and normal to the square plane directions. Meanwhile, we can evaluate the direction cosines of the principal axes of the local g-tensor (l_i, m_i, n_i) , where i = x, y, z, with respect to the crystallographic axes (a^*, b, c) from the geometry of the local crystal field as shown in Figure S2and S3.



Figure S2 The direction cosines the principal axes of the local *g*-tensor (l_i , m_i , n_i), where i = x, y, z, for one sublattice (α -sublattice) with respect to the crystallographic axes.



Figure S3 The direction cosines the principal axes of the local *g*-tensor (l_i , m_i , n_i), where i = x, y, z, for other sublattice (β -sublattice) with respect to the crystallographic axes.

When defined the direction cosines of H with respect to the crystallographic axes (a^*, b, c) as L, M and N, respectively, those are directly given by experiments, l, m and n are shown by the following equations,

$$l = Ll_x + Mm_x + Nn_x$$
$$m = Ll_y + Mm_y + Nn_y$$
$$n = Ll_z + Mm_z + Nn_z$$

Therefore, because observed *g*-value is average of magnetically inequivalent *g*-tensor, diagonal elements of the local *g*-tensor (g_x , g_y , g_z) can be evaluated by fitting the following equation to the orientation dependence of *g*-value.

$$g = \sqrt{\frac{\left(g_x^2 l_{\alpha}^2 + g_y^2 m_{\alpha}^2 + g_z^2 n_{\alpha}^2\right) + \left(g_x^2 l_{\beta}^2 + g_y^2 m_{\beta}^2 + g_z^2 n_{\beta}^2\right)}{2}}$$

where α and β shows the two magnetically inequivalent sublattices,

We used the following value for the fitting of data on a^*b -plane, bc-plane, and a^*c -plane, respectively.

a^*b -plane (H // b when $\theta = 0$)

$$\begin{split} l_{\alpha} &= l_{x\alpha} sin^{[\alpha]}(\theta) + m_{x\alpha} cos^{[\alpha]}(\theta) \\ m_{\alpha} &= l_{y\alpha} sin^{[\alpha]}(\theta) + m_{y\alpha} cos^{[\alpha]}(\theta) \\ n_{\alpha} &= l_{z\alpha} sin^{[\alpha]}(\theta) + m_{z\alpha} cos^{[\alpha]}(\theta) \\ l_{\beta} &= l_{x\beta} sin^{[\alpha]}(\theta) + m_{x\beta} cos^{[\alpha]}(\theta) \\ m_{\beta} &= l_{y\beta} sin^{[\alpha]}(\theta) + m_{y\beta} cos^{[\alpha]}(\theta) \\ n_{\alpha} &= l_{z\alpha} sin^{[\alpha]}(\theta) + m_{z\alpha} cos^{[\alpha]}(\theta) \end{split}$$

bc-plane (H // c when $\theta = 0$)

$$\begin{split} l_{\alpha} &= m_{x\alpha} sin^{[\alpha]}(\theta) + n_{x\alpha} cos^{[\alpha]}(\theta) \\ m_{\alpha} &= m_{y\alpha} sin^{[\alpha]}(\theta) + n_{y\alpha} cos^{[\alpha]}(\theta) \\ n_{\alpha} &= m_{z\alpha} sin^{[\alpha]}(\theta) + n_{z\alpha} cos^{[\alpha]}(\theta) \\ l_{\beta} &= m_{x\beta} sin^{[\alpha]}(\theta) + n_{x\beta} cos^{[\alpha]}(\theta) \\ m_{\beta} &= m_{y\beta} sin^{[\alpha]}(\theta) + n_{y\beta} cos^{[\alpha]}(\theta) \\ n_{\alpha} &= m_{z\alpha} sin^{[\alpha]}(\theta) + n_{z\alpha} cos^{[\alpha]}(\theta) \end{split}$$

ca^* -plane (H // c when $\theta = 0$)

$$\begin{split} l &= l_x sin(\theta) + n_x cos(\theta) \\ m &= l_y sin(\theta) + n_y cos(\theta) \\ n &= l_z sin(\theta) + n_z cos(\theta) \end{split}$$

We used only l_x , l_y and l_z of α -sublattice for fitting the data on the ca^* -plane, because α and β sublattice become equivalent with respect to the magnetic field parallel to the *b*-axis.



Fig. S4 Temperature dependence of magnetic susceptibility in polycrystalline sample of CuCl₂(pdz) at 10000 Oe. Inset shows magnification of the data below 30 K.



Fig. S5 Magnetization curve (per ion) of CuCl₂(pdz) under (a) $H // a^*$ and (b) H // c at 2.0 K.