

## Supporting Information

### Thermoelectric Bi<sub>2</sub>Te<sub>3</sub>-Improved Charge Collection for High-Performance Dye-Sensitized Solar Cells

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#### 1. Materials and synthesis

**Materials:** Bismuth citrate (99.99% trace metals basis), sodium telluride (99 %), diethyleneglycol, polyethylene glycol, sodium hydroxide and poly(acrylic acid) ( $M_w$ : 1800000) were purchased from Aldrich. Dye sensitizer *cis*-bis(isothiocyanato)(2,2'-bipyridyl-4,4'-dicarboxylato)(2,2'-bipyridyl-4,4'-dinonyl) ruthenium (II) (coded as **Z907**), TiO<sub>2</sub> paste (DSL 18NR-T) and iodide-based liquid electrolyte (HL-HPE) were purchased from Dyesol company. Single-walled carbon nanotube, carbon >90 %,  $\geq 77\%$  (0.7-1.1 nm diameter), and multi-wall carbon nanotube ( $\geq 95\%$ , diam.  $\times$  L = 20-50 nm  $\times$  5-20  $\mu$ m) were also purchased from Aldrich.

**Synthesis of Bi<sub>2</sub>Te<sub>3</sub>:** Bi<sub>2</sub>Te<sub>3</sub> nanoplates with average edge lengths at 300-400 nm and thickness is 50-80 nm of the Bi<sub>2</sub>Te<sub>3</sub> nanoplates were synthesized by a solvothermal method. In detail, 39.88 mg bismuth citrate and 33.74 mg sodium telluride were firstly dissolved in diethylene glycol and polyethylene glycol (15mL/15mL, v/v), followed by adding NaOH (1mL, 5 M) and poly(acrylic acid) (5 mL, 2.5 mg/mL). The mixture was sonicated for 5 min to make a homogenous solution. The solution was thus transferred to autoclave and sealed, heating at 180 °C for 8 hrs. After that, the autoclave was cooled down to room temperature. The product was collect by centrifuging at 10 K rpm, 10 min, in a centrifuge tube (50 mL). To further purify the product, ethanol was added to re-disperse the residue. The product was separated again by centrifugation at 10 K rpm, 10min. This purification process was repeated for another 2 times. The black product was finally dissolved in ethanol for use.

#### 2. Supporting Text

## 2.1 Supporting characterizations

XRD pattern (**Figure S2a**) correspond well with the JCPDS card No. 85-0439 with of hexagonal  $\text{Bi}_2\text{Te}_3$ . **Figure S2d** shows the UV-vis-NIR absorption spectrum of the as-synthesized product. TEM and SEM image shows that the average edge lengths at 150-500 nm and thickness is 20-30 nm of the  $\text{Bi}_2\text{Te}_3$  nanoplates (**Figure S2e** and **f**). It can also be seen from TEM image that the  $\text{Bi}_2\text{Te}_3$  nanoplates are porous structure, which could enhance the thermoelectric properties by decreasing the thermal conductivity.<sup>[1]</sup> The efficiency of converting “waste heat” into useful electricity by thermoelectric materials is usually about 7-8%.<sup>[2]</sup>

To investigate the composition of the anode, we carried out XRD characterization of the  $\text{TiO}_2$  anode. The anode containing either pure  $\text{TiO}_2$  nanoparticle or  $\text{TiO}_2/\text{Bi}_2\text{Te}_3$  hybrid nanoparticles were scraped from the FTO surface. XRD pattern (**Figure S2b**) of the anode of device **1** shows an anatase structure of  $\text{TiO}_2$  (JCPDS card No. 84-1286). Due to the very small content of  $\text{Bi}_2\text{Te}_3$  in the  $\text{TiO}_2$  electrode, it is difficult to obtain XRD patterns of the  $\text{Bi}_2\text{Te}_3$  from the hybrid anode in the main text. We thus prepared a film with high amount of  $\text{Bi}_2\text{Te}_3$  especially for XRD characterization (1/1, w/w). It can be seen that the co-existence of  $\text{Bi}_2\text{Te}_3$  and  $\text{TiO}_2$  (**Figure S2c**).

The cross-section of the anode films were investigated by SEM. For the characterization, the  $\text{TiO}_2$  anodes glasses were broken by a hammer. It can be found that  $\text{Bi}_2\text{Te}_3$  nanoplates can be covered by a dense layer of  $\text{TiO}_2$  (**Figure S3a**, from device **3**). At high concentration of  $\text{Bi}_2\text{Te}_3$  in the  $\text{TiO}_2$  paste (**Figure S3b**, from device **4**),  $\text{Bi}_2\text{Te}_3$  tend to aggregate, which lead to a poor distribution and less “accelerator” loci for electron transport.

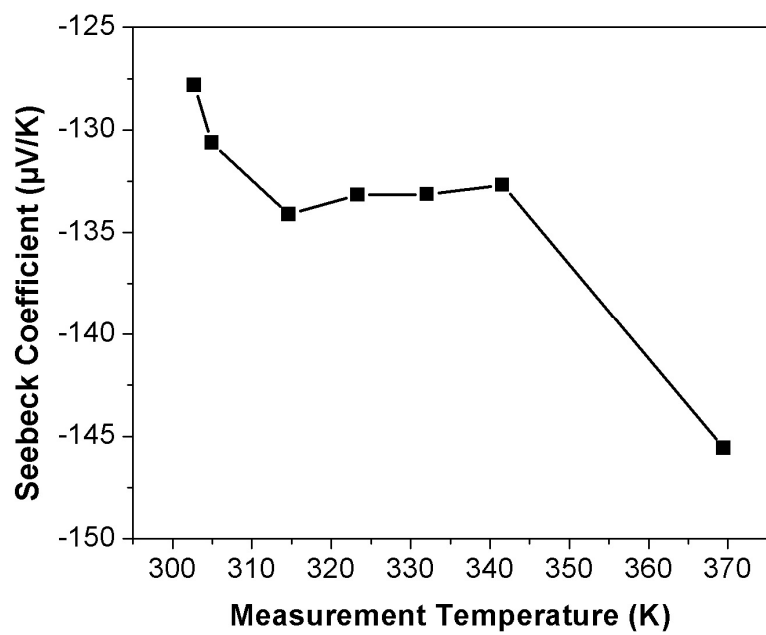
## 2.2 Carbon nanotube incorporated DSCs

Owing to the device temperature increase, one could also expect that the performance increasing in the hybrid  $\text{Bi}_2\text{Te}_3$ -incorporated DSCs is a result of thermal induced better conductivity of  $\text{Bi}_2\text{Te}_3$ . Hence, we employed multi-walled carbon nanotube (MWCNT) to substitute for  $\text{Bi}_2\text{Te}_3$  to perform similar investigation to device **2**, **3** or **4**; MWCNT is a conductor with very small Seebeck coefficient,  $17 \mu\text{V K}^{-1}$ ,<sup>[3]</sup> it was mixed with  $\text{TiO}_2$  commercial paste with weight ratio of 1:4000. Considering the molecular weight of carbon is 1/66.7 of  $\text{Bi}_2\text{Te}_3$ , the amount of MWCNT in the  $\text{TiO}_2$  paste is comparable to  $\text{Bi}_2\text{Te}_3$  in device **3**. The hybrid paste was pasted onto the FTO surface form the bottom layer of the anode.

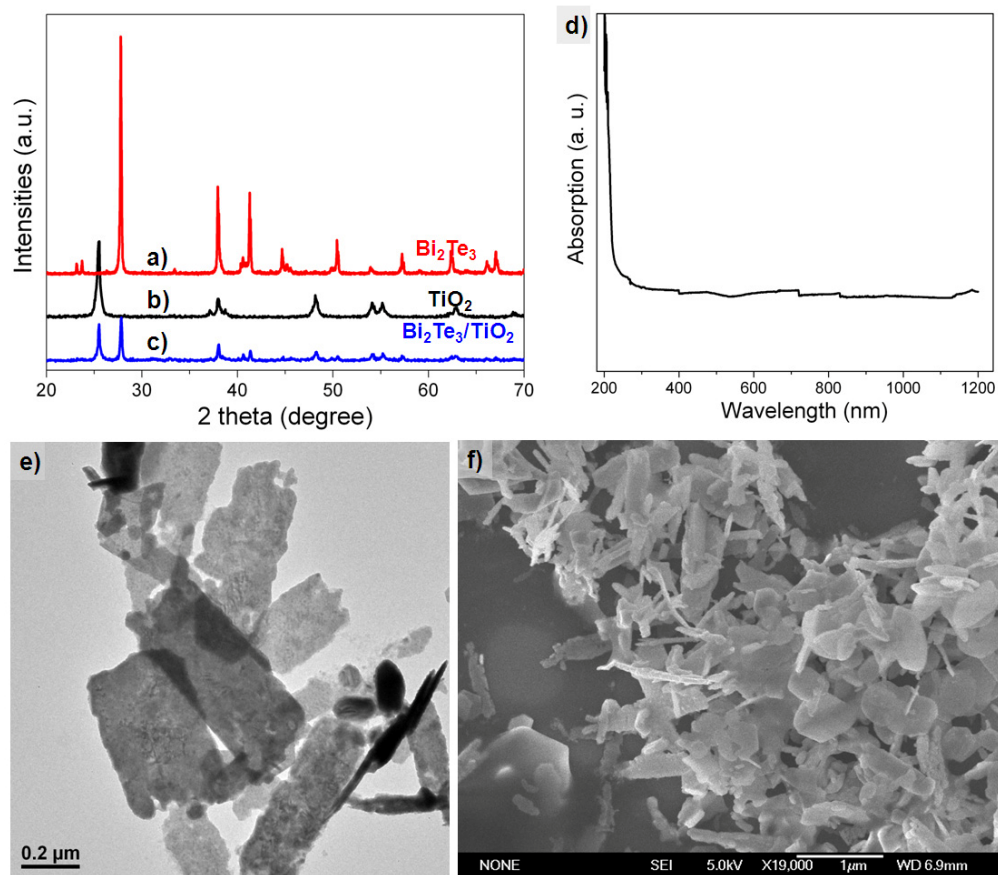
After incorporation of MWCNT, the  $J_{sc}$ ,  $V_{oc}$  and  $FF$  of the hybrid device become  $14.52 \text{ mA cm}^{-2}$ ,  $0.69 \text{ V}$  and  $62 \%$ . The power conversion efficiency was also increased to  $6.26 \%$  (the PCE of normal DSC is  $6.08 \%$  in this work, see **Table 1** in main text), similar to other reports that the conductivity of MWCNT can improve the performance.<sup>[44]</sup> However, after sunlight illumination, the conversion efficiency of MWCNT-incorporated DSCs gradually decreased from  $6.26 \%$  to  $6.08 \%$  at  $150 \text{ min}$  (**Figure S4a**). This observation rules out that the conductivity change of  $\text{Bi}_2\text{Te}_3$  or MWCNT upon temperature variation can induce better energy conversion efficiency. On the other hand, it is also possible that the Seebeck coefficient of MWCNT is too small (about  $1/14$  of  $\text{Bi}_2\text{Te}_3$ ) to improve the performance that can suppress the inherent deterioration of the DSC.

SWCNT possesses higher Seebeck coefficient,  $45 \mu\text{V K}^{-1}$ , than that of MWCNT ( $17 \mu\text{V K}^{-1}$ ), but smaller than that of  $\text{Bi}_2\text{Te}_3$  ( $240 \mu\text{V K}^{-1}$ ).<sup>[3]</sup> Therefore, it can be expected that the incorporation of SWCNT into the  $\text{TiO}_2$  photoanode could suppress the performance degradation and thus improve the efficiency. **Figure S4b** shows the  $J$ - $V$  characteristics of the SWCNT-incorporated DSCs. In the SWCNT-incorporated DSC,  $J_{sc}$  and  $\eta$  increased to  $16.22 \text{ mA cm}^{-2}$  and  $6.54\%$  compared with the normal DSC.  $V_{oc}$  remains unchanged at  $0.73 \text{ V}$ . Upon sunlight illumination,  $J_{sc}$  and  $\eta$  increased gradually and reached their maxima at about  $90 \text{ min}$ ,  $16.74 \text{ mA cm}^{-2}$  and  $6.66 \%$  (**Figure S4**). Although the improvement is very small, this result demonstrates the generality and reliability of utilization of thermoelectric materials for improving the efficiency of the DSCs.

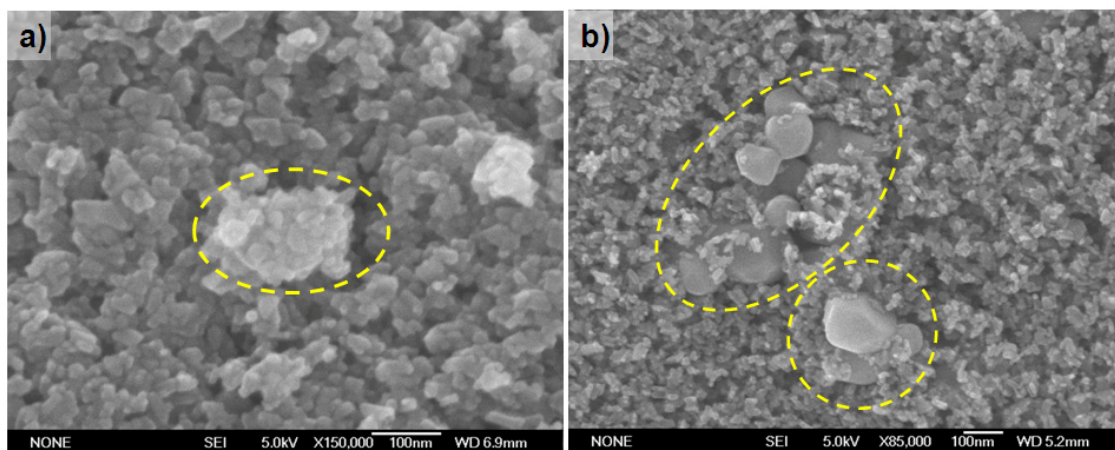
### 3. Supporting figures



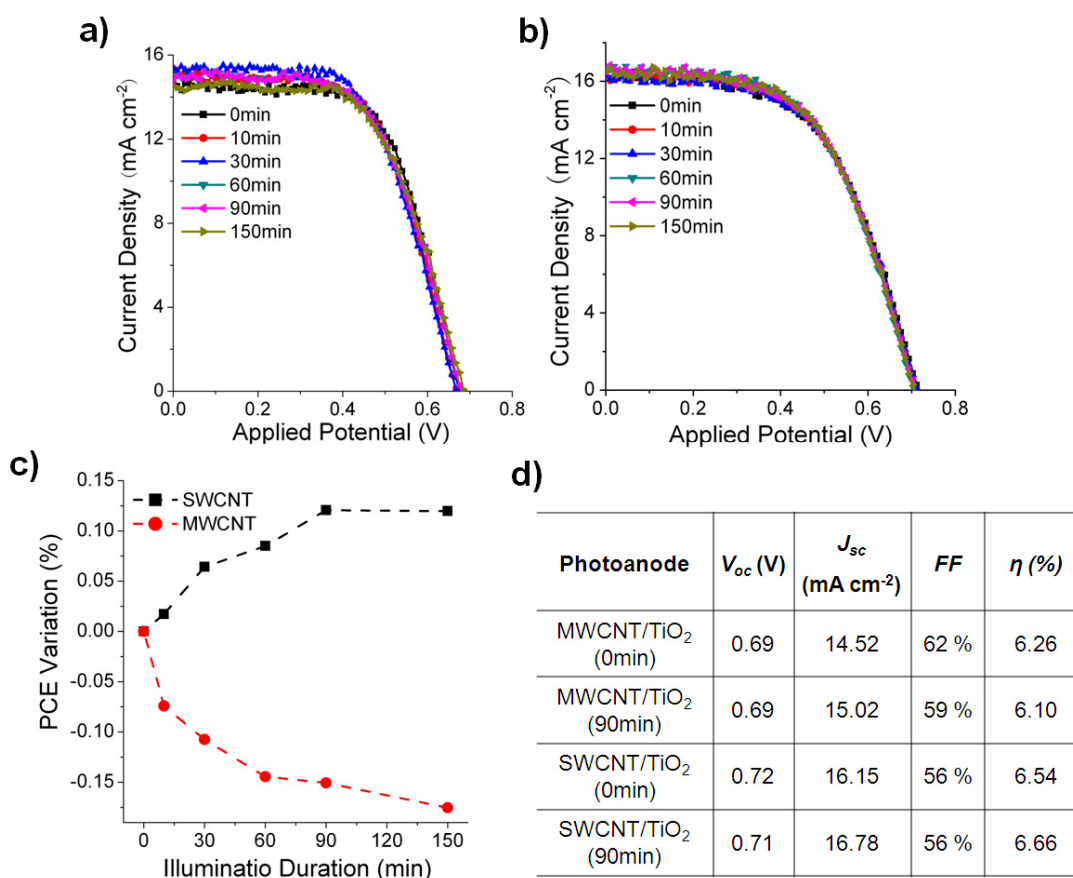
**Figure S1.** Seebeck coefficients measured on the film sample of  $\text{Bi}_2\text{Te}_3$  nanoplates in the temperature range of 298-373 K.



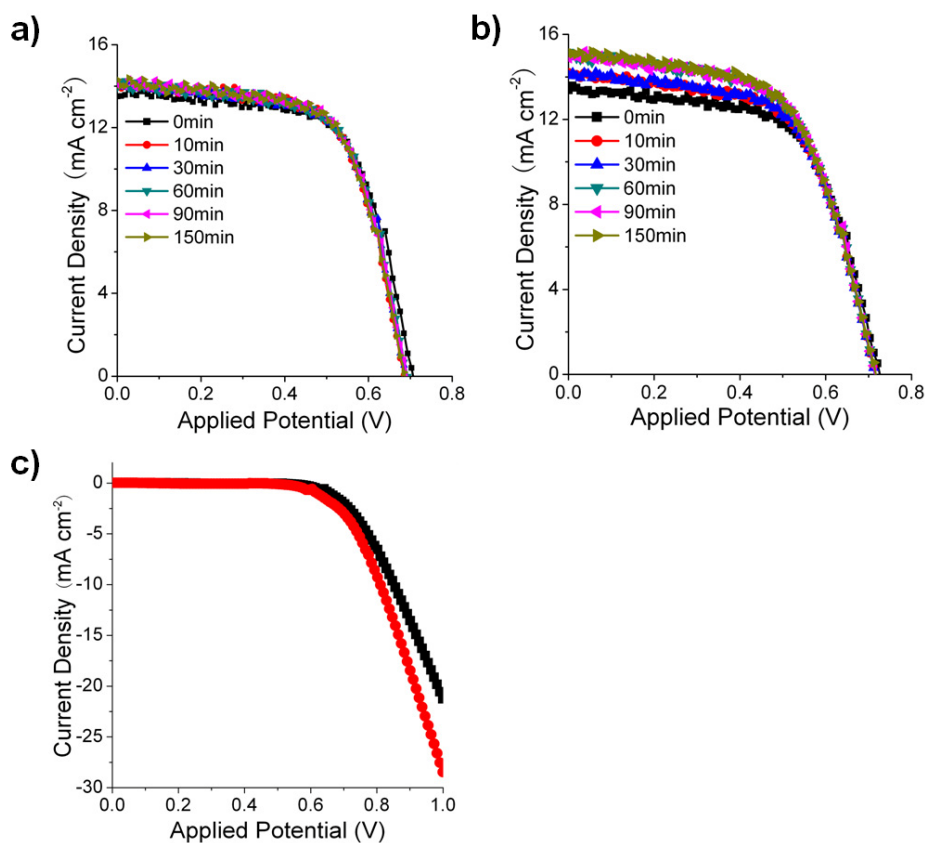
**Figure S2.** (a), (b) and (c) showing the XRD patterns of Bi<sub>2</sub>Te<sub>3</sub>, the anode materials of device 1 (TiO<sub>2</sub>) and hybrid film of TiO<sub>2</sub>/Bi<sub>2</sub>Te<sub>3</sub>; (d) displaying the UV-vis-NIR absorption of the as-synthesized Bi<sub>2</sub>Te<sub>3</sub> nanoplate a. q. solution; (e) and (f) representing TEM and SEM image of the as-synthesized Bi<sub>2</sub>Te<sub>3</sub> nanoplates.



**Figure S3.** SEM image of cross-section of device **3** (a),  $\text{Bi}_2\text{Te}_3$  nanoplate was densely decorated by  $\text{TiO}_2$  nanoparticles; (b) showing the SEM image of the cross-section of device **4**, there are some aggregates of  $\text{Bi}_2\text{Te}_3$  due to the high concentration of  $\text{Bi}_2\text{Te}_3$  nanoplates.



**Figure S4.** (a) and (b) showing the  $J$ - $V$  characteristics of the MWCNT- and SWCNT-incorporated upon long illumination by sunlight; (c) displaying the PCE variation of the corresponding MWCNT- and SWCNT-incorporated hybrid devices; (d) showing the performance of the MWCNT- and SWCNT-incorporated hybrid devices at 0 min and 90 min sunlight irradiation.



**Figure S5.** (a) and (b) showing the  $J$ - $V$  characteristics of device 2 and 4 upon long illumination by sunlight; (c) displaying the  $J$ - $V$  characteristics of device 4 under dark, it can be seen that after heating by sunlight irradiation, dark current is higher than without long time illumination. Similar to that of device 3 in main text.

#### 4. Supporting References

- [1] J. Y. Tang, H. T. Wang, D. H. Lee, M. Fardy, Z. Y. Huo, T. P. Russell, P. D. Yang, *Nano Lett.* **2010**, *10*, 4279.
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- [3] A. M. Rao, X. H. Ji, T. M. Tritt, *MRS Bull.* **2006**, *31*, 218.
- [4] S. Muduli, W. Lee, V. Dhas, S. Mujawar, M. Dubey, K. Vijayamohanan, S. H. Han, S. Ogale, *Acs Appl. Mater. & Interfaces* **2009**, *1*, 2030.