

## Supporting information

on

### **Black Anatase Titania Enabling Ultra High-Rate Cycling for Rechargeable Lithium Batteries**

by

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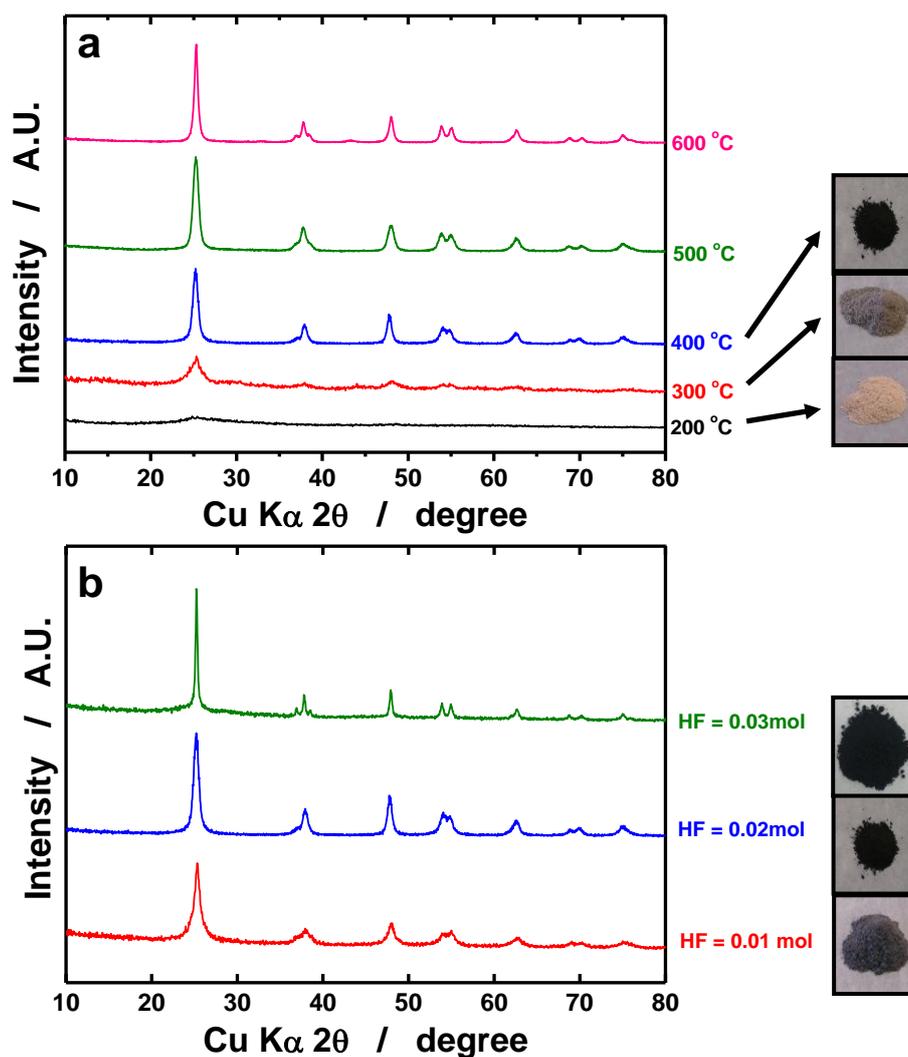
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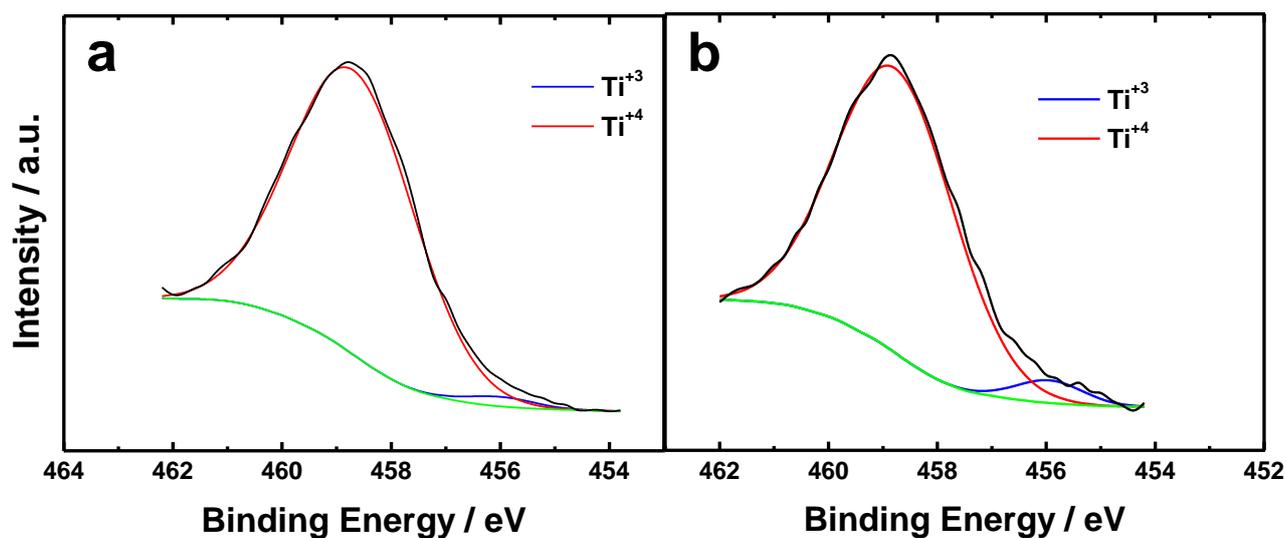
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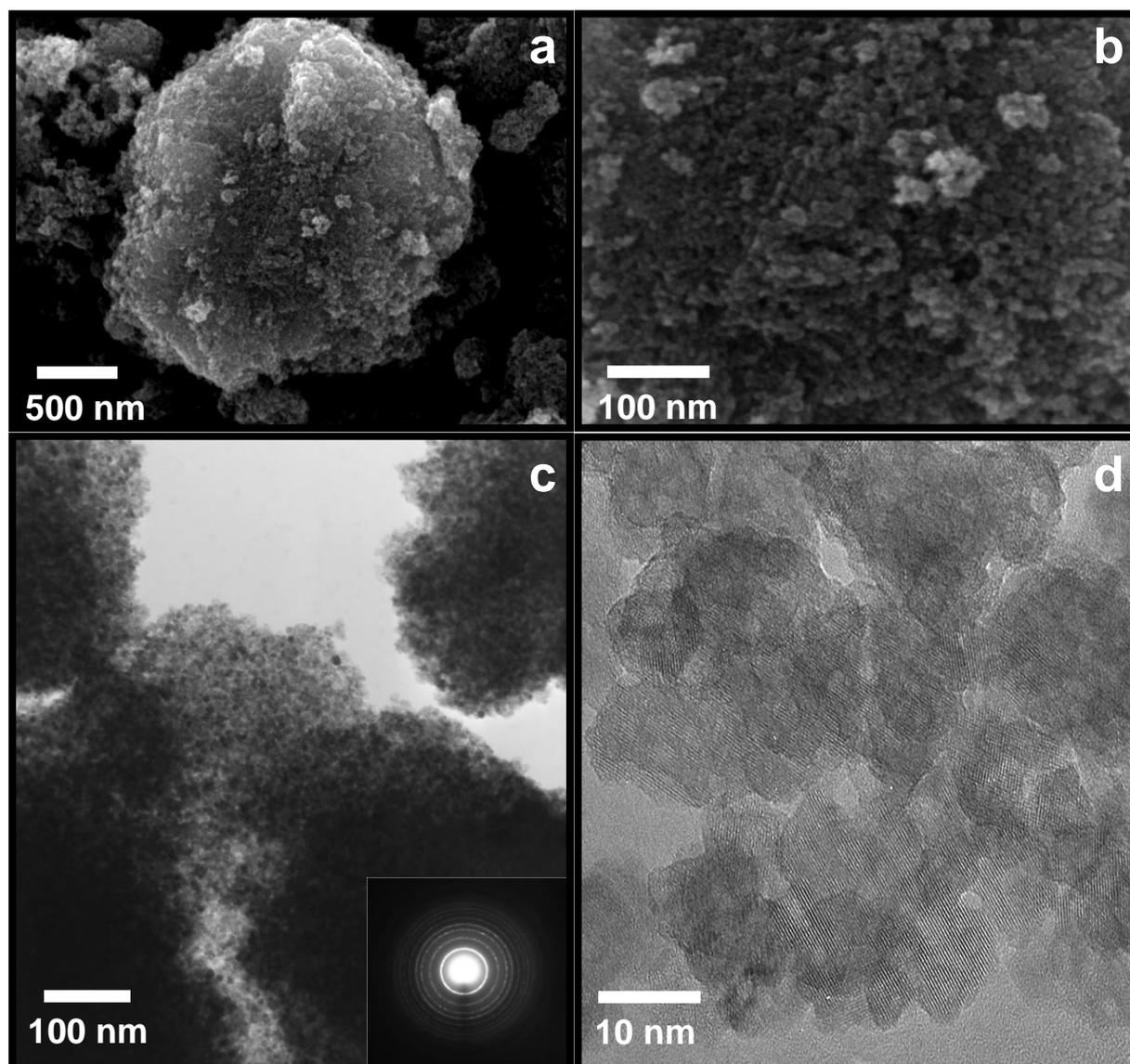
**S-Figure 1.** (a) XRD patterns of light yellow gel heated at various temperatures for 5 h in Ar atmosphere; (b) light yellow gel varying HF concentration heated at 400 °C for 5 h in Ar atmosphere.

Dissolution of  $\text{Ti}^{\text{IV}}$  ( $\text{TiCl}_4$  (IV)) readily produces the  $\text{Ti}^{3+}$  ion in acidic media below pH 2 (here, HF was used) [21]. In course of the synthesis, urea ( $(\text{NH}_2)_2\text{CO}$ ) was added to delay hydrolysis of  $\text{Ti}^{4+}$  to  $\text{Ti}(\text{OH})_4$  in the present work. Transparent light yellow gel is received after evaporation of the solution at 80 °C, and consequent heating of the resultant at 200 °C leads to dark yellow powders with low crystallinity (5h in Ar, S-Fig. 1a). Further heat treatments at elevated temperatures, surprisingly, alter the colors of the products, gray at 300 °C and black at 400 – 600 °C, and the resulting crystallinity for X-ray diffraction (XRD) patterns is progressively developed with temperatures. Also, the crystallinity and colors of the powders are dependent on the added amount of HF in the starting solution (S-Fig. 1b). For comparison, the calcination was also performed in air atmosphere (400 °C for 5 h). Products produced in air and Ar exhibit a typical anatase type  $\text{TiO}_2$  as confirmed in X-ray diffraction (XRD) patterns (Fig. 1a and S-Fig. 1). Besides, the powders obtained in air shows the same white color as ordinary anatase  $\text{TiO}_2$ .



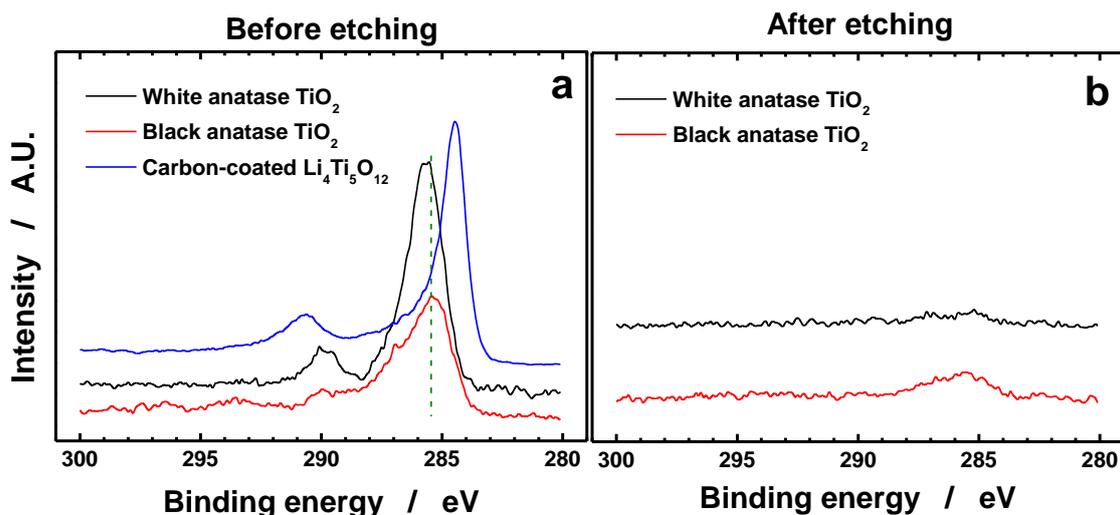
**S-Figure 2.** XPS spectrum of the black anatase  $\text{TiO}_2$  heated at  $400\text{ }^\circ\text{C}$  for 5 h in Ar: HF concentration was (a) 0.1 mol and (b) 0.3 mol for the starting solution.)

From the XPS spectrum, compared with the simulated  $\text{Ti}^{3+}$  ( $\text{Ti}^{\text{III}}\text{N}$ , blue line) and  $\text{Ti}^{4+}$  ( $\text{Ti}^{\text{III}}\text{O}_2$ , red line) spectra, the calculated content of the trivalent Ti is found to be 2.1 % (for the 0.1 mol HF starting solution) and 7.2 % (for the 0.3 mol HF starting solution) present in the surface region, implying that much acidic environment from solution gives rise to the further formation of the trivalent Ti.



**S-Figure 3.** SEM images of black anatase  $\text{TiO}_2$  heated at  $400\text{ }^\circ\text{C}$  for 5 h in Ar: (a) low and (b) high magnification; TEM bright-field images: (c) low (inset: SAD pattern) and (d) high magnification.

SEM images display that the synthesized  $\text{TiO}_2$  powders are agglomerates and they are nanoparticles. TEM images also reveals that the particles are high agglomerated showing polycrystalline diffraction pattern.



**S-Figure 4.** Comparison of Cls XPS spectra of white anatase TiO<sub>2</sub>, black anatase TiO<sub>2</sub>, and carbon-coated spinel Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub>: (a) before Ar<sup>+</sup> etching and (b) after Ar<sup>+</sup> etching for 10 min (corresponding to 2.7 nm in depth for the standard SiO<sub>2</sub> powders) in N<sub>2</sub> atmosphere.

The above data represent XPS Cls spectra for both white and black anatase TiO<sub>2</sub>. Carbon-coated Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> was also added to compare the presence of carbon on the surface. Air-formed carbon layers (C with N, S, and O) are mainly observed in range of 284 – 288 eV for both black and white anatase TiO<sub>2</sub> before Ar<sup>+</sup> etching. Another peak observed at 288 – 291 eV would be ascribed to the carbonate layer formed in air. In comparison with the carbon-coated Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> (carbonized from pitch showing high intensity of G-band in Raman spectroscopy), carbon-derived peak is evident at 284.5 eV. Furthermore, carbide-related trace is not found in range of 280.5 – 283 eV both black and white anatase TiO<sub>2</sub> before and after Ar<sup>+</sup> etching for 10 min corresponding 2.7 nm in depth. The above results indicate that carbon is not doped in the black anatase TiO<sub>2</sub> nor not coated on the surface of black anatase TiO<sub>2</sub>.

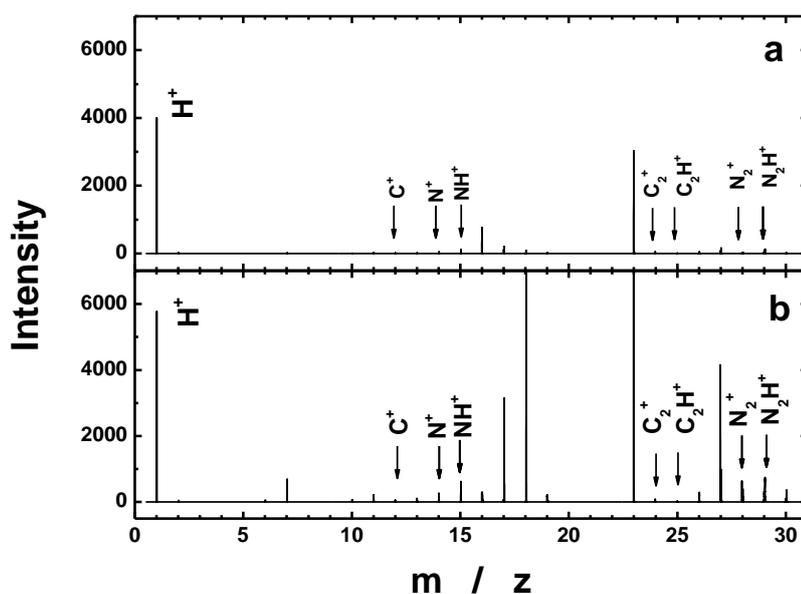
**S-Table I.** Rietveld refinement results of XRD pattern for TiO<sub>2</sub> calcined at 400 °C for 5 h in Air (white).

Formula		TiO <sub>2</sub>				
Crystal system		Tetragonal				
Space group		<i>I4<sub>1</sub>/amd</i>				
Atom	Site	<i>X</i>	<i>Y</i>	<i>Z</i>	<i>g</i>	<i>B</i> / Å <sup>2</sup>
Ti	4 <i>b</i>	0	0.25	0.375	1	0.8
O	8 <i>e</i>	0	0.25	0.165(3)	1	0.8
<i>a</i> -axis / Å		3.7887(2)				
<i>c</i> -axis / Å		9.4894(7)				
Ti-O1 / Å		1.9327(7)				
Ti-O2 / Å		1.9956(16)				
<i>R</i> <sub>wp</sub> / %		11.1				
<i>R</i> <sub>p</sub> / %		8.90				

**S-Table II.** Rietveld refinement results of XRD pattern for TiO<sub>2</sub> calcined at 400 °C for 5 h in Ar (black).

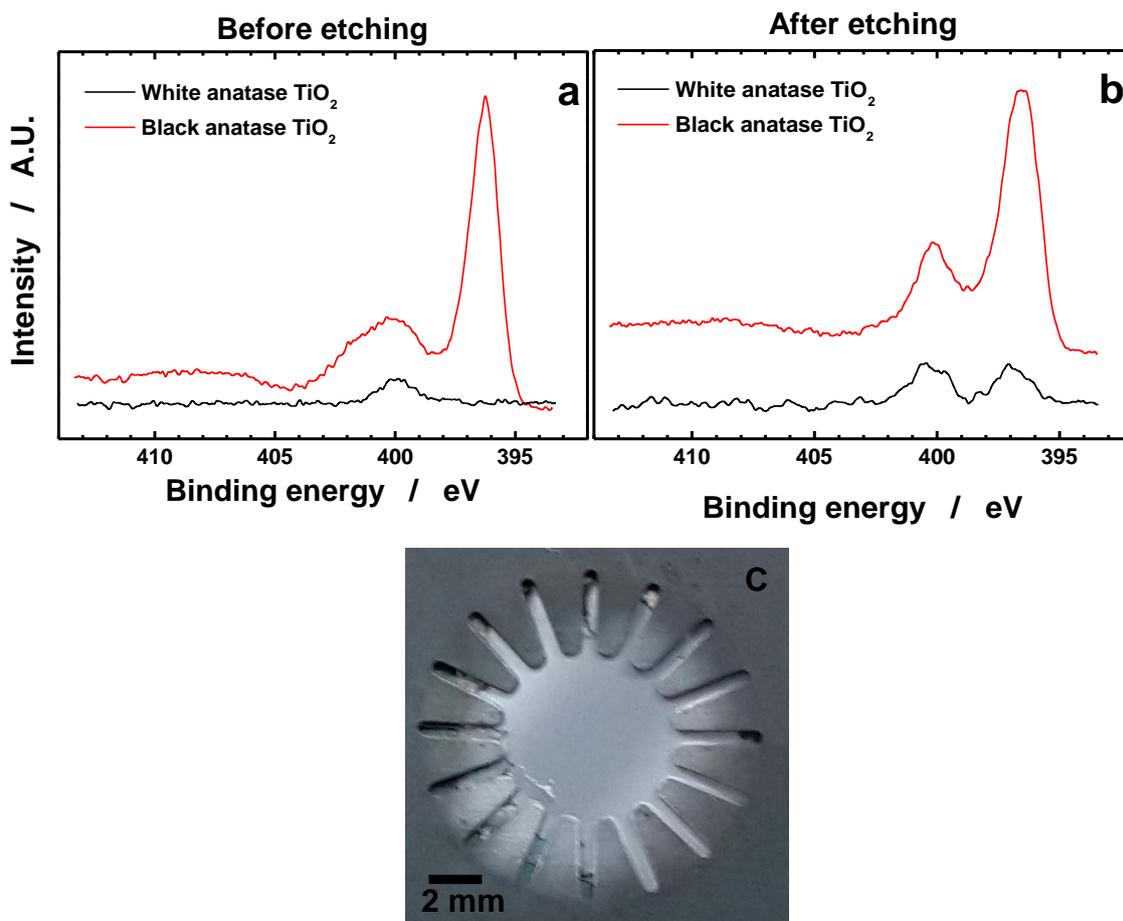
Formula		TiO <sub>2</sub>				
Crystal system		Tetragonal				
Space group		<i>I4<sub>1</sub>/amd</i>				
Atom	Site	<i>X</i>	<i>Y</i>	<i>Z</i>	<i>g</i>	<i>B</i> / Å <sup>2</sup>
Ti	4 <i>b</i>	0	0.25	0.375	1	0.8
O	8 <i>e</i>	0	0.25	0.164(4)	1	0.8
<i>a</i> -axis / Å		3.8038(3)				
<i>c</i> -axis / Å		9.5148(7)				
Ti-O1 / Å		1.9382(7)				
Ti-O2 / Å		1.9991(16)				
<i>R</i> <sub>wp</sub> / %		12.9				
<i>R</i> <sub>p</sub> / %		9.29				





**S-Figure 6.** ToF-SIMS spectra of (a) white anatase TiO<sub>2</sub> heated at 400 °C for 5 h in air and (b) black anatase TiO<sub>2</sub> heated at 400 °C for 5 h in Ar.

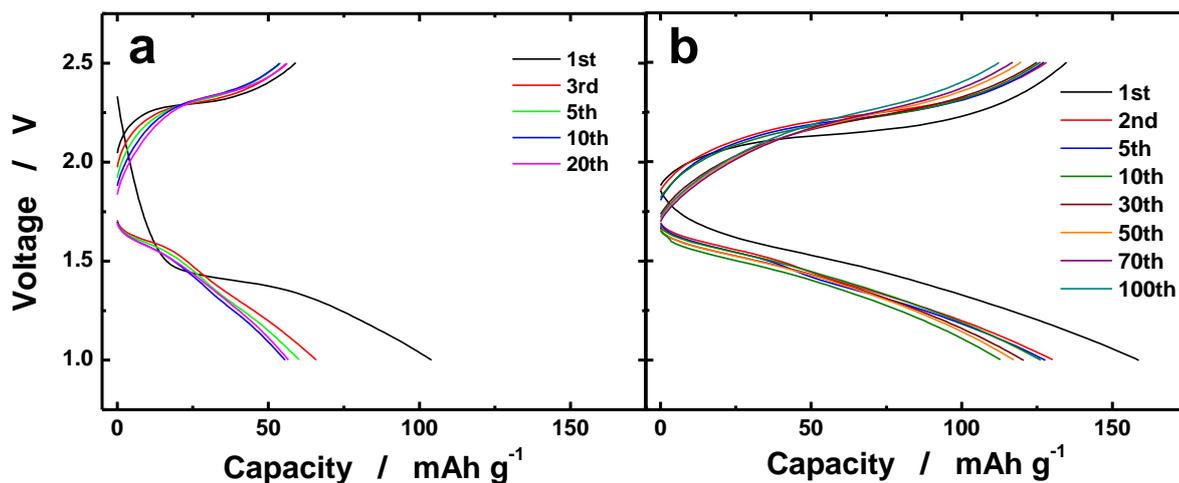
It is further demonstrated by ToF-SIMS data that the present black anatase is carbon-free product, because carbon-related fragments such as C<sup>+</sup> ( $m = 12$ ), C<sub>2</sub><sup>+</sup> ( $m = 24$ ), and C<sub>2</sub>H<sup>+</sup> ( $m = 25$ ) (See Fig. b), are not detected in the spectra (see Figure S-6b). Provided that they present, the resulting spectra should be notable like N-related fragments, N<sup>+</sup> ( $m = 14$ ), NH<sup>+</sup> ( $m = 15$ ), N<sub>2</sub><sup>+</sup> ( $m = 28$ ), N<sub>2</sub>H<sup>+</sup> ( $m = 28$ ) (see Figure S-5b). On the contrary, these fragments are not perceived for the white anatase TiO<sub>2</sub>(see Figure S-6a).



**S-Figure 7.** Comparison of N1s XPS spectra of white anatase TiO<sub>2</sub> and black anatase TiO<sub>2</sub>, and carbon-coated spinel Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub>: (a) before Ar<sup>+</sup> etching, (b) after Ar<sup>+</sup> etching for 10 min (corresponding to 2.7 nm in depth for the standard SiO<sub>2</sub> powders) in N<sub>2</sub> atmosphere, and (c) digital camera image of white anatase TiO<sub>2</sub> after Ar<sup>+</sup> etching for 10 min in N<sub>2</sub> atmosphere.

It is evident for the white anatase TiO<sub>2</sub> that only contaminates layer, probably NO<sub>x</sub> formed in air, is found in range of 399 – 401 eV. Meanwhile, nitride-related layer (395 – 398 eV) and contaminates layer (399 – 401 eV) with ammonium salt (NH<sub>4</sub><sup>+</sup>, 401 – 404 eV) are found for the black anatase TiO<sub>2</sub> in S-Fig. 7a, confirming the presence of TiN<sub>4</sub>H<sup>+</sup> fragment in ToF-SIMS data (shown in Fig. 2a). Therefore, the absence of TiN<sub>4</sub>H<sup>+</sup> fragment in ToF-SIMS data (S-Figs. 5 and 6) is proved in the XPS spectrum for the white anatase TiO<sub>2</sub> that does not show the binding energy of ammonium salt (NH<sub>4</sub><sup>+</sup>, 401 – 404 eV) in S-Fig. 7b.

Since Ar<sup>+</sup> etching was carried out in N<sub>2</sub> atmosphere, adhesion or incorporation N element is possible so that the white surface was changed to black as shown in S-Fig. 7c. For the reason, nitride-related peak is observed for the white anatase TiO<sub>2</sub>. Also, one can able to understand the ready formation of the trivalent Ti after the Ar<sup>+</sup> etching in N<sub>2</sub> atmosphere (see S-Fig. 7b). Again, the above XPS data clarify the presence of Ti-N-H bond in the black anatase TiO<sub>2</sub>.



**S-Figure 8.** Continuous charge and discharge curves of black anatase TiO<sub>2</sub> heated at 400 °C for 5 h in Ar: (a) starting HF concentration: 0.1 mol and (b) 0.3 mol.

Electronic conductivity is also dependent on the starting HF content, since the acidity of the starting solution affect the formation of trivalent Ti: the lower solution pH, the higher trivalent Ti concentration in the black anatase TiO<sub>2</sub>, as is found in the XPS data (see S-Fig. 2). As a result, the electronic conductivity is improved from  $1.5 \times 10^{-3} \text{ S cm}^{-1}$  for the gray anatase TiO<sub>2</sub> (for the 0.1 mol HF starting solution) to  $8.9 \times 10^{-2} \text{ S cm}^{-1}$  for the black anatase TiO<sub>2</sub> (for the 0.3 mol HF starting solution). Since the formation of the trivalent Ti is favored in much acidic environment, here 0.3 mol HF added (S-Fig. 5b), the black anatase TiO<sub>2</sub>, which exhibits black color in S-Fig. 1b, delivers high capacity and superior retention at 100 C-rates ( $20 \text{ A g}^{-1}$ ) charge and discharge test.