

Supplementary data

Robust conductive mesoporous carbon-silica composite films with highly ordered and oriented orthorhombic structures from triblock-copolymer template co-assembly

Lingyan Song,^a Dan Feng,^b Casey G. Campbell,^a Dong Gu,^b Aaron M. Forster,^d Kevin G. Yager,^c Nathaniel Fredin,^c Hae-Jeong Lee,^c Ronald L. Jones,^c Dongyuan Zhao^{*b} and Bryan D. Vogt ^{*a}

^a *Flexible Display Center, Arizona State University, Tempe, AZ 85287. Fax: 1-480-727-8957; Tel: 1-480-727-8631; E-mail: bryan.vogt@asu.edu*

^b *Department of Chemistry, Shanghai Key Laboratory of Molecular Catalysis and Innovative Materials, and Laboratory of Advanced Materials, Fudan University, Shanghai 200433 (China). Fax: 86-21-51630307; Tel: 86-21-51630205; E-mail: dyzhao@fudan.edu.cn*

^c *Polymers Division, National Institute of Standards and Technology, Gaithersburg, MD 20899*

^d *Building and Fire Research Laboratory, National Institute of Standards and Technology, Gaithersburg, MD 20899*

A. GISAXS data fits and unit cell parameters

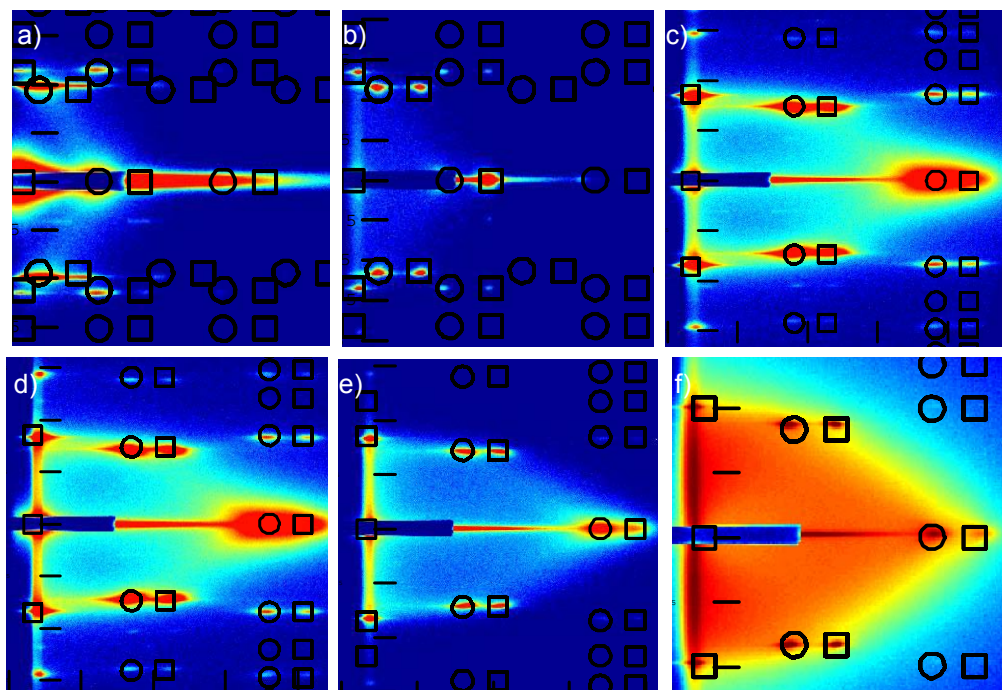


Figure S1. GISAXS patterns of the films. The overlay of simulated spots is from a NANOCELL simulation. The circles and squares represent the transmitted and reflected Bragg peaks, respectively. Simulation results show an (010) oriented *Fmmm* symmetry with many domain oriented normal to the substrate. The unit cell parameters of the films are shown in the table below:

Sample No	Sample Name	Unit Cell Parameters, nm		
		a	b	c
a)	FDU-16-AM	17	20	23
b)	FDU-16-350	17	17	23
c)	FDU-16-800	16	6.8	22.5
d)	CS-20-800	16	7.2	22.8
e)	CS-33-800	16.5	7.6	22.8
f)	CS-43-800	14.5	9.0	19.5

B. FTIR characterization

The FTIR spectra of the mesoporous polymer-silica composite films (350 °C) and carbon-silica composite films (800 °C) are shown in Figure S2. Following recognized peak identification of infrared spectra of organic compound¹⁻⁴ and inorganic compounds, such as silica,⁵⁻⁸ and carbon,^{9,10} the functional groups within these films have been identified by the absorption bands listed in Table S1 and S2. For the mesoporous polymeric films after being calcined at 350 °C, the broad absorbance near 3500 cm⁻¹ is attributed to the O-H stretching of phenolic groups and silanols from the incomplete condensation. Also, the bands at approximately 1226 cm⁻¹ are assigned to C-O stretching of phenolic groups; this absorbance decreases with the increase of silica concentration in the films. As the silica concentration increased up to 38 and 56 wt%, this peak is no longer as well defined due to overlap with the strong Si-O-Si asymmetric stretching at 1100 cm⁻¹. The retained vibrations of phenolic resins and silicates indicate the coexistence of polymer and silica composites in the films. The absorbance at 980 cm⁻¹ for the polymer-silica films with high silica content results from the silanol vibration. Absorbance in the range of 3100 ~ 2800 cm⁻¹ is ascribed to the stretching of carbon-hydrogen bonds.

Increasing the pyrolysis temperature to 800 °C induces dramatic changes in the FTIR spectra of the mesoporous composite films. Absorption near 3500 and 1226 cm⁻¹ disappears, which suggests the total decomposition of phenol resins at this elevated temperature. The hydrocarbonaceous residues are greatly reduced, as the -CH₃ symmetrical deformation vibration at 1382 cm⁻¹ is nearly vanished. Only aromatic C-H groups in the range of 3100 ~ 2800 cm⁻¹ are detectable. The band multiplet in the range of 1600 – 1480 cm⁻¹ transforms to a single symmetrical C=C band centered near 1600 cm⁻¹ after heating polymeric nanocomposite to 800 °C. These results suggest a formation of carbon materials. With the increase of silica content in the carbon-silica composite films, the Si-O-Si asymmetric stretching bands at 1100 cm⁻¹, and Si-O-Si bands at 468 cm⁻¹ become better resolved.

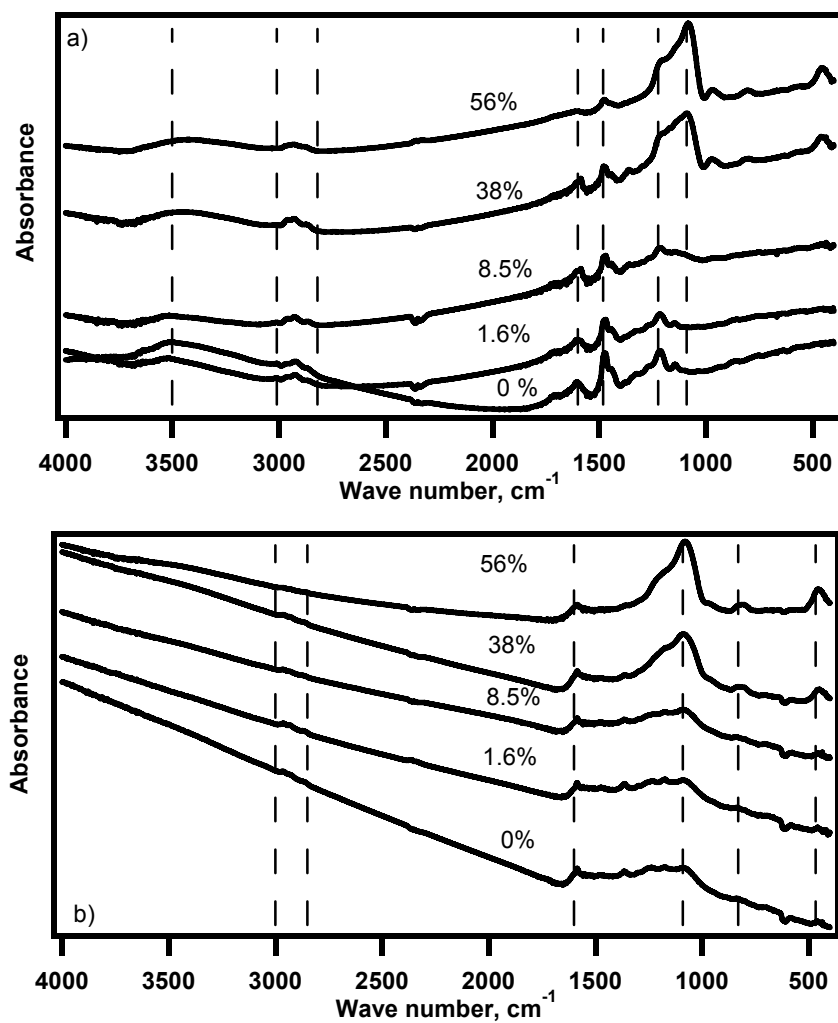


Figure S2. FTIR spectra of mesoporous nanocomposite films of (a) FDU-16, (b) CS-20, (c) CS-33, (d) CS-51 and (e) CS-68, after carbonization at (A) 350 °C and (B) 800 °C.

Table S1 FTIR vibration assignments of mesoporous polymer-silica films after pyrolyzed at 350°C

Wavenumber, cm ⁻¹	Functional group
Near 3500	O-H stretching of phenol group
3100-2800	C-H stretching
1600	C-C aromatic ring stretching
1481	C-H bending of aliphatic bridge structure
1226	C-O stretching of phenol group
1100	Si-O-Si asymmetric stretching
980	-Si-OH vibration
468	Si-O-Si band

**Table S2 FTIR
assignments of
carbon-silica films
800°C**

Wavenumber, cm ⁻¹	Functional group
3100-2800	C-H stretching
1600	C-C aromatic ring stretching
1382	-CH ₃ symmetrical deformation vibration
1100	Si-O-Si asymmetric stretching
468	Si-O-Si band

**Vibration
mesoporous
after pyrolyzed at**

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