

† SUPPORTING INFORMATION

**Structure and electrochemical properties of novel mixed
Li(Fe_{1-x}M_x)SO₄F (M = Co, Ni, Mn) phases fabricated by
low temperature ionothermal synthesis†**

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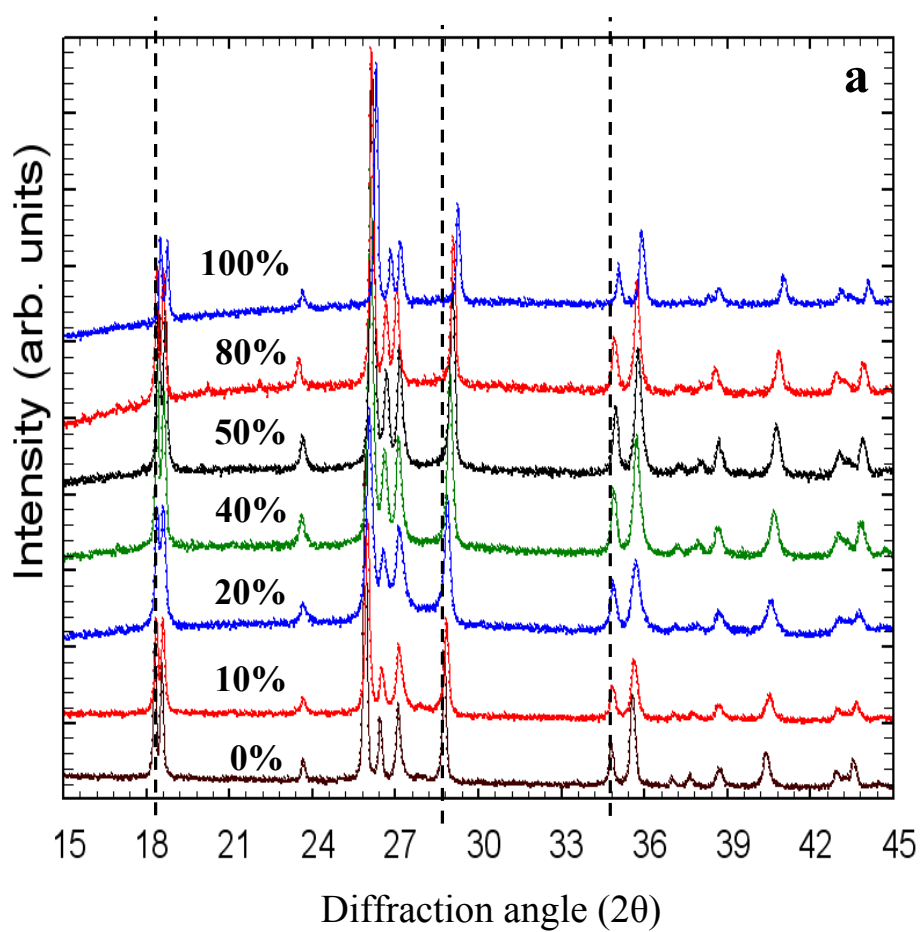
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List of Supporting Information

- Fig. S1** XRD patterns of (a) (Fe_{1-x}Co_x)SO₄.H₂O and (b) (Fe_{1-x}Mn_x)SO₄.H₂O monohydrate precursors with different compositions.
- Table S1** Lattice parameters and cell volume of (Fe_{1-x}Co_x)SO₄.H₂O solid-solution precursors having monoclinic (C2/c) symmetry.
- Table S2** Lattice parameters and cell volume of (Fe_{1-x}Mn_x)SO₄.H₂O solid-solution precursors having monoclinic (C2/c) symmetry.
- Fig. S2** Some additional SEM images of Li(Fe_{1-x}M_x)SO₄F compounds.
- Fig. S3** Some representative EDAX elemental analysis of Li(Fe_{1-x}Mn_x)SO₄F compounds.
- Fig. S4** Galvanostatic electrochemical cycling profiles of (a) Li(Fe_{0.9}Mn_{0.1})SO₄F, (b) Li(Fe_{0.8}Mn_{0.2})SO₄F compounds.



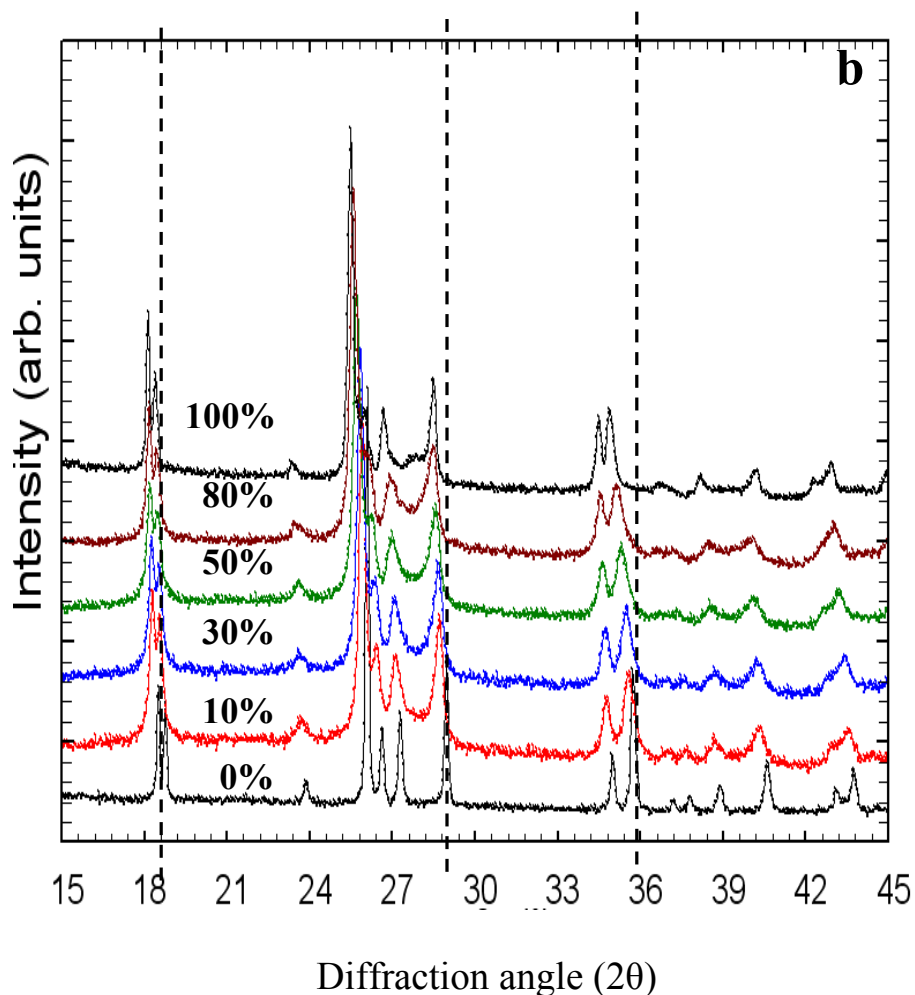


Fig. S1 X-ray diffraction patterns of (a) $(\text{Fe}_{1-x}\text{Co}_x)\text{SO}_4 \cdot \text{H}_2\text{O}$ and (b) $(\text{Fe}_{1-x}\text{Mn}_x)\text{SO}_4 \cdot \text{H}_2\text{O}$ monohydrate precursors varying the Co or Mn doping content from 0 to 100 wt%. The numbers alongside the diffraction patterns present the Co or Mn dopant content (wt%). As shown, increasing Co content gradually leads to lower lattice volume as indicated by peak shift towards higher diffraction angle (or lower d value) due to the lower ionic radii of Co wrt Fe (similar to Ni doping in the manuscript Fig 2). On the other hand, Mn doping shifts the XRD peaks towards lower angle (or higher d value) reflecting the larger ionic radii of Mn wrt Fe. The dashed lines are shown to guide eyes about the gradual peak shift upon varying the precursor composition.

Table S1 Lattice parameters and cell volume of $(\text{Fe}_{1-x}\text{Co}_x)\text{SO}_4\cdot\text{H}_2\text{O}$ solid-solution precursors having monoclinic ($C2/c$) symmetry.

Materials	a (Å)	b (Å)	c (Å)	V (Å ³)	β (°)
$\text{FeSO}_4\cdot\text{H}_2\text{O}$	7.0918(3)	7.5352(2)	7.7822(2)	364.67(2)	118.72(2)
$(\text{Fe}_{0.90}\text{Co}_{0.10})\text{SO}_4\cdot\text{H}_2\text{O}$	7.0845(2)	7.5429(1)	7.7645(1)	364.09(1)	118.66(1)
$(\text{Fe}_{0.60}\text{Co}_{0.40})\text{SO}_4\cdot\text{H}_2\text{O}$	7.0904(2)	7.5864(2)	7.7145(1)	362.36(1)	119.16(5)
$(\text{Fe}_{0.50}\text{Co}_{0.50})\text{SO}_4\cdot\text{H}_2\text{O}$	7.0405(2)	7.5553(2)	7.6875(1)	357.80(1)	118.95(1)
$(\text{Fe}_{0.40}\text{Co}_{0.60})\text{SO}_4\cdot\text{H}_2\text{O}$	7.0246(1)	7.5620(1)	7.6753(1)	356.77(1)	118.86(2)
$(\text{Fe}_{0.20}\text{Co}_{0.80})\text{SO}_4\cdot\text{H}_2\text{O}$	6.9866(6)	7.5498(6)	7.6353(4)	352.93(4)	118.80(1)
$\text{CoSO}_4\cdot\text{H}_2\text{O}$	6.9542(8)	7.5731(1)	7.6242(1)	351.72(3)	118.68(1)

Table S2 Lattice parameters and cell volume of $(\text{Fe}_{1-x}\text{Mn}_x)\text{SO}_4\cdot\text{H}_2\text{O}$ solid-solution precursors having monoclinic ($C2/c$) symmetry.

Materials	a (Å)	b (Å)	c (Å)	V (Å ³)	β (°)
$\text{FeSO}_4\cdot\text{H}_2\text{O}$	7.0918(3)	7.5352(2)	7.7822(2)	364.67(2)	118.72(2)
$(\text{Fe}_{0.90}\text{Mn}_{0.10})\text{SO}_4\cdot\text{H}_2\text{O}$	7.1279(2)	7.5305(2)	7.7921(1)	366.37(2)	118.84(5)
$(\text{Fe}_{0.70}\text{Mn}_{0.30})\text{SO}_4\cdot\text{H}_2\text{O}$	7.1342(2)	7.4689(2)	7.8862(1)	367.19(1)	119.09(5)
$(\text{Fe}_{0.50}\text{Mn}_{0.50})\text{SO}_4\cdot\text{H}_2\text{O}$	7.1592(3)	7.5473(3)	7.8404(2)	370.35(2)	119.05(2)
$(\text{Fe}_{0.20}\text{Mn}_{0.80})\text{SO}_4\cdot\text{H}_2\text{O}$	7.1425(2)	7.4985(2)	7.7248(2)	372.48(2)	115.80(3)
$\text{MnSO}_4\cdot\text{H}_2\text{O}$	7.1241(6)	7.6631(4)	7.7642(7)	381.44(3)	115.85(12)

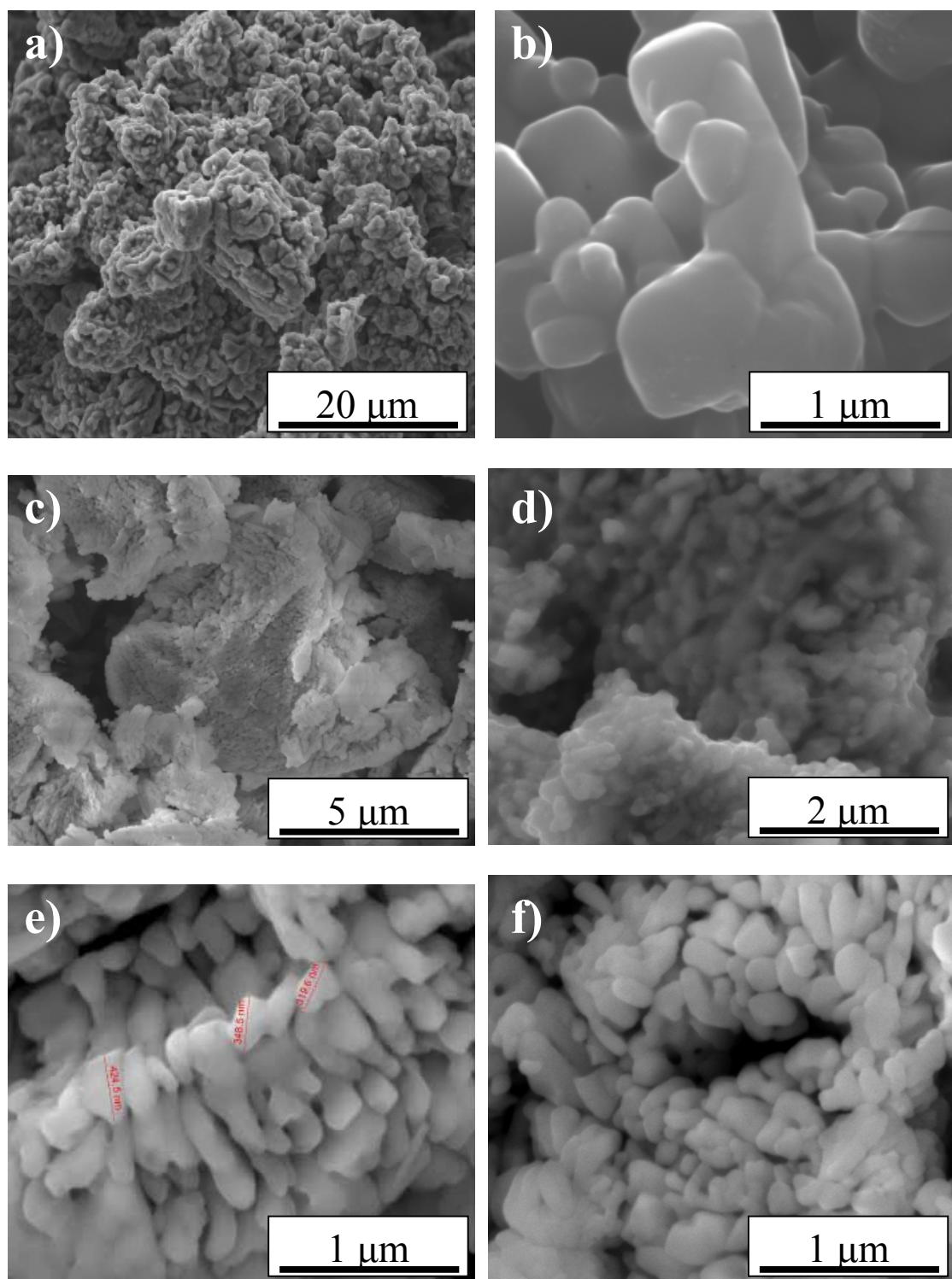


Fig. S2 SEM images showing the topotactic reaction involving initial $(\text{Fe}_{1-x}\text{M}_x)\text{SO}_4 \cdot \text{H}_2\text{O}$ precursor and final $\text{Li}(\text{Fe}_{1-x}\text{M}_x)\text{SO}_4\text{F}$ products with similar morphology. (a) $(\text{Fe}_{0.80}\text{Co}_{0.20})\text{SO}_4 \cdot \text{H}_2\text{O}$, (b) $\text{Li}(\text{Fe}_{0.80}\text{Co}_{0.20})\text{SO}_4\text{F}$, (c) $(\text{Fe}_{0.80}\text{Ni}_{0.20})\text{SO}_4 \cdot \text{H}_2\text{O}$, (d) $\text{Li}(\text{Fe}_{0.80}\text{Ni}_{0.20})\text{SO}_4\text{F}$, (e) $(\text{Fe}_{0.90}\text{Ni}_{0.10})\text{SO}_4 \cdot \text{H}_2\text{O}$ and (f) $\text{Li}(\text{Fe}_{0.90}\text{Ni}_{0.10})\text{SO}_4\text{F}$.

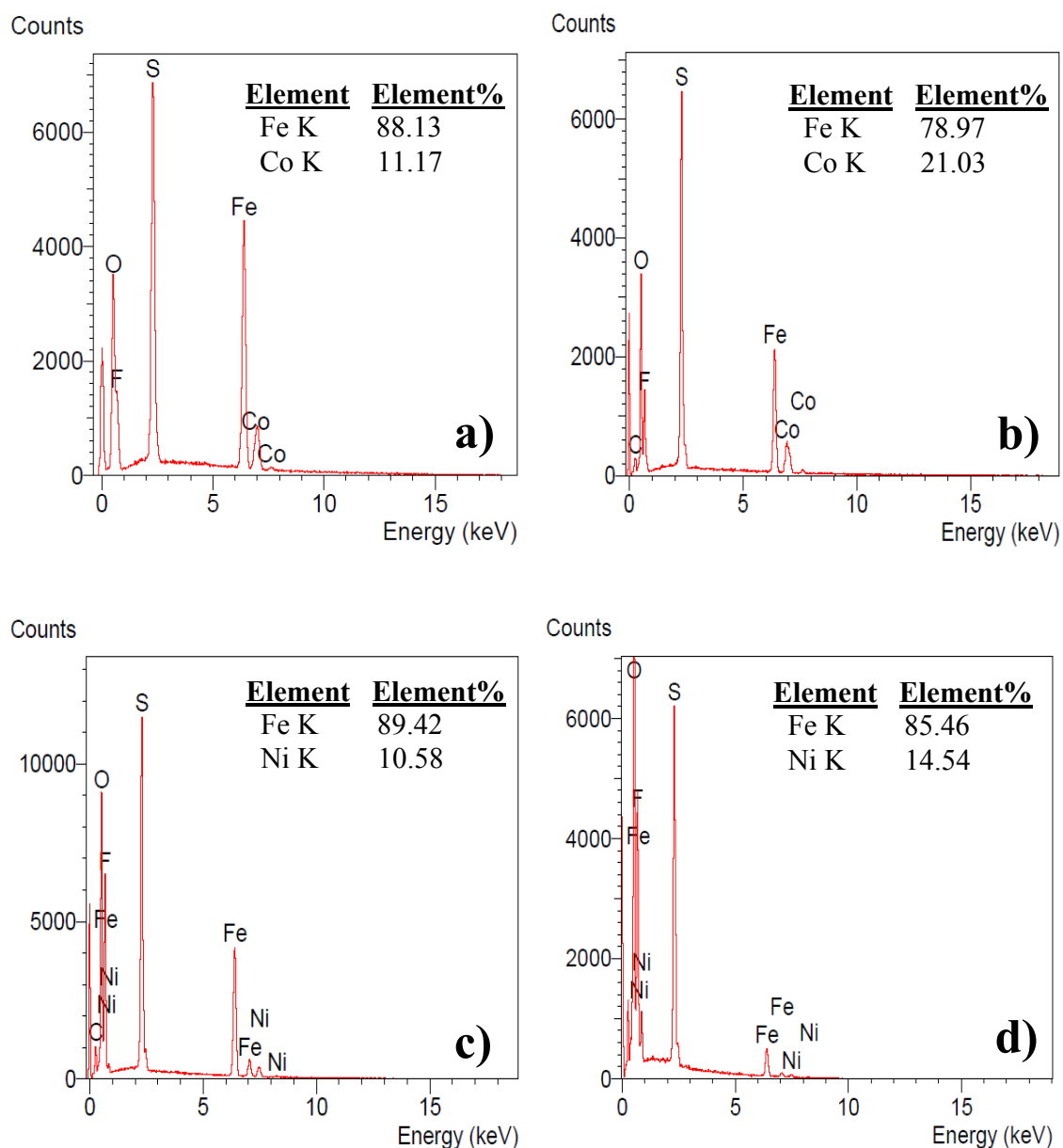


Fig. S3 Some representative EDAX elemental analysis done in conjunction with SEM study to confirm the stoichiometric composition and uniform distribution of individual 3d transition metals in $\text{Li(Fe}_{1-x}\text{M}_x\text{)SO}_4\text{F}$ products; (a) $\text{Li(Fe}_{0.90}\text{Co}_{0.10}\text{)SO}_4\text{F}$, (b)

Li(Fe_{0.80}Co_{0.20})SO₄F, (c) Li(Fe_{0.90}Ni_{0.10})SO₄F and (d) Li(Fe_{0.80}Ni_{0.20})SO₄F products. The elemental analysis values are enlisted in the inset of each figure.

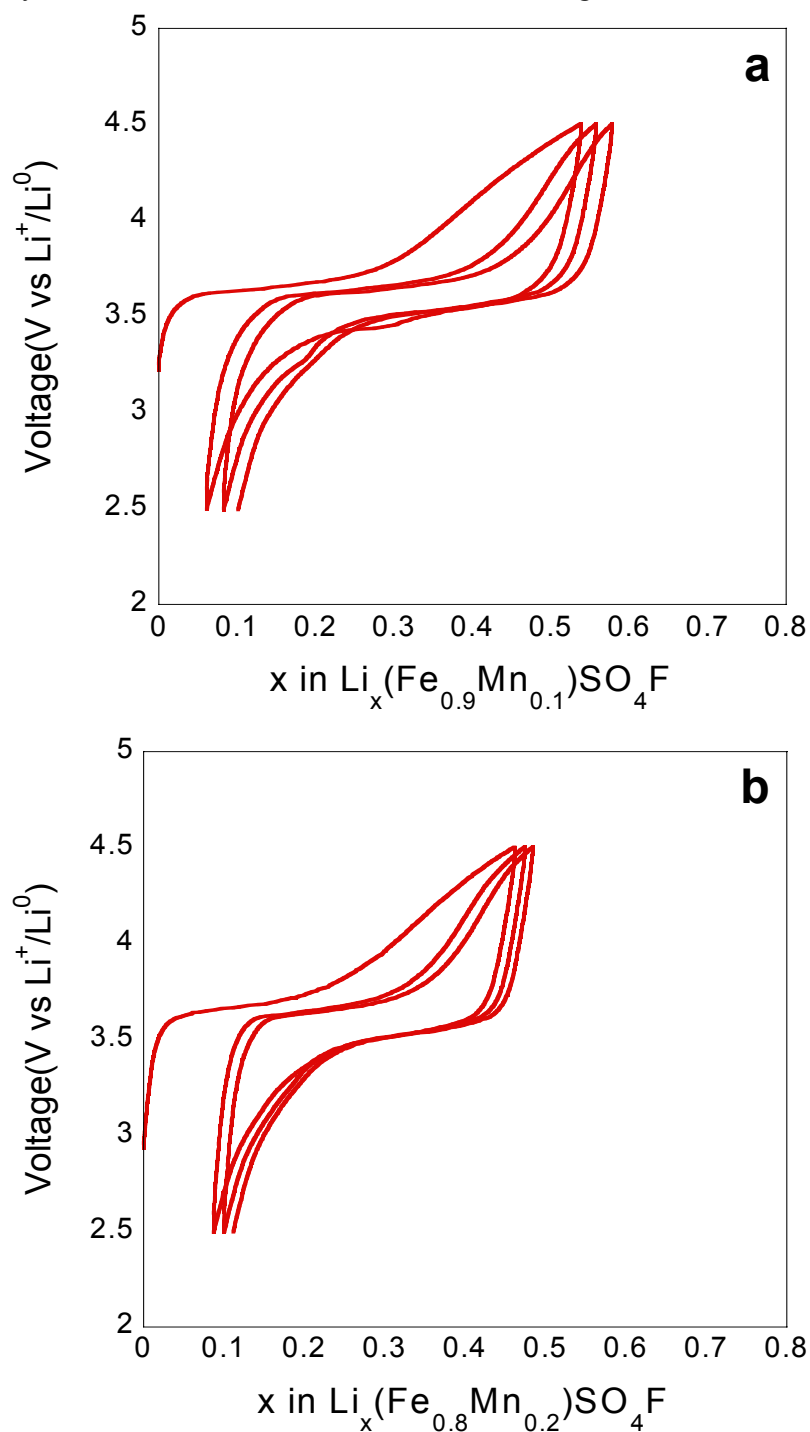


Fig. S4 Galvanostatic electrochemical cycling profiles of (a) Li(Fe_{0.9}Mn_{0.1})SO₄F, (b) Li(Fe_{0.8}Mn_{0.2})SO₄F compounds, cycled between 2.5 to 4.5 V at C/10 (1 Li in 10 hours) in Li-half cell swagelok setting. The electrodes were prepared by ball-milling

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Li(Fe_{1-x}Mn_x)SO₄F and carbon black (85:15 weight ratio) for 15 minutes. A 3.6 V plateau is observed in both cases with excellent cycling stability.