## Layer-dependent electrical and optoelectronic responses of ReSe<sub>2</sub> nanosheet transistors

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## **EXPERIMENTIAL SECTION**

**Mechanical Exfoliation of ReSe**<sub>2</sub>. Single- and few-layer (four-layer) ReSe<sub>2</sub> nanosheets were isolated from bulk ReSe<sub>2</sub> single crystals (Lamellae Co.) and then deposited onto the freshly cleaned Si substrates covered by a 300 nm thick SiO<sub>2</sub> layer using the scotch tape-based mechanical exfoliation method.

**Thermal Annealing.** The samples were heated to 400 °C in a 25 °C/min rate and the temperature was held at 400 °C for one hour in vacuum. After annealing, the furnace was cooled down to room temperature and the samples were taken out of the furnace.

**FET devices were fabricated as follows.** Metal electrodes were fabricated using electron-beam lithography (Vistec EBPG 5000plus ES) followed by electron beam evaporation (EB700-I) of 5-nm-thick Cr and 50-nm-thick gold. Electrical characterization for the transistors was measured using a semiconductor parameter analyzer and shielded probe station. Photo-response experiments were all performed with a CHI660D electrochemical workstation in a conventional three-electrode electrochemical cell.

**Raman Spectroscopy.** Analysis of the monolayer ReSe<sub>2</sub> nanosheet by Raman spectroscopy was carried out on a WITec CRM200 confocal Raman microscopy system with the excitation line of 532 nm and an air-cooling charge-coupled device (CCD) as the detector (WITec Instruments Corp, Germany).

**SEM image and EDX analysis.** SEM image was measured by NanoSEM430 from FEI Company. EDX was utilized for the estimation of the composition of ReSe<sub>2</sub> as shown in Fig. S4. The EDX was equipped on the SEM instrument.

Molecule adsorption calculations. First-principles spin-polarized calculations are carried out within density-functional theory (DFT) as implemented in the Vienna ab *initio* simulation package (VASP) <sup>1,2</sup>. The 4×4 supercell of ReSe<sub>2</sub> monolayer with single gas molecule adsorbed to it, is chosen as the studied model. The exchange-correction interaction is treated by the van der Waals density functional  $(vdW-DF)^{3}$  to describe the adsorbed system. The kinetic cutoff energy is set to be 450 eV and the Monkhorst-Pack grid  $^4$  of 5×5×1 is employed for the Brillouin zone integration. The vacuum layer is larger than 15 Å so that the interaction between two adjacent ReSe<sub>2</sub> monolayers can be eliminated. The geometric structure is fully relaxed by using conjugate gradient method until the Hellmann-Feynman force on each atom is smaller than 0.02 eV/Å. The charge transfer between ReSe<sub>2</sub> and gas molecule is obtained based on the Bader analysis <sup>5</sup>. The adsorption energy is defined as  $E_a = E_{\text{ReSe}_2 + \text{molecule}} - (E_{\text{ReSe}_2} + E_{\text{molecule}})$ , where  $E_{\text{ReSe}_2 + \text{molecule}}$ ,  $E_{\text{ReSe}_2}$  and  $E_{\text{molecule}}$  are the total energies of molecule adsorbed ReSe<sub>2</sub>, bare ReSe<sub>2</sub> and isolated molecule, respectively. density difference calculated The charge is by the formula  $\Delta \rho = \rho_{\text{ReSe}_2 + \text{molecule}} - (\rho_{\text{ReSe}_2} + \rho_{\text{molecule}})$ , where  $\rho_{\text{ReSe}_2 + \text{molecule}}$ ,  $\rho_{\text{ReSe}_2}$  and  $\rho_{\text{molecule}}$ are the charge densities of molecule adsorbed ReSe2, bare ReSe2 and isolated molecule, respectively.

Electronic structure and density of states (DOS) calculations. The calculations are performed using the projector augmented wave (PAW) method <sup>6</sup> with the generalized gradient approximation of Perdew-Burke-Ernzerhof (GGA-PBE) <sup>7</sup> exchange-correction functional including van der Waals corrections <sup>3</sup>, as implemented in the VASP code <sup>1</sup>. Energy cutoff for plane-wave expansion is set to 500 eV. A large vacuum layer more than 15 Å is adopted to avoid the interaction between adjacent images. Brillouin zone sampling is performed with Monkhorst-Pack (MP) <sup>4</sup> special k points meshes including  $\Gamma$  point. K-points grids of 5×5×5 and 5×5×1 are chosen for the optimization calculations of bulk and two dimensional (2D) systems, respectively. And much denser meshes (15×15×15 and 15×15×1 for bulk and 2D systems) are used for the DOS calculations. All the structures are fully relaxed using the conjugated gradient method until the Hellmann-Feynman force on each atom is less than 0.01 eV/Å.



Figure S1. The structure of ReSe<sub>2</sub>. a. Undistorted ReSe<sub>2</sub>, b. distorted ReSe<sub>2</sub>.

In order to verify the stability of the distorted structure of ReSe<sub>2</sub>, we have constructed an undistorted model of ReSe<sub>2</sub> with the honeycomb structure, as shown in Fig. S1, similar to other transition metal dichalcogenides. The total energy of the undistorted one is -17.395 eV per molecular, while that of the distorted one is -18.473 eV per molecular, which is more stable than the undistorted one by 1.078 eV per molecular.



Figure S2. AFM height image of few-layer ReSe<sub>2</sub>.

The thickness of few-layer ReSe<sub>2</sub> is about 2.7nm, corresponding to four layer.



Figure S3. Raman spectra of ReSe<sub>2</sub> nanosheets.



Figure S4. Single NH<sub>3</sub> molecule adsorbed on the ReSe<sub>2</sub> monolayer. Charge density difference plots for **a**, without defects and **b**, with defects for side view and **c**, with defects for top view on monolayer ReSe<sub>2</sub>. The isosurface value without defects is set to be  $5 \times 10^{-4}$  e/Å<sup>3</sup> and that with defects is  $1 \times 10^{-3}$  e/Å<sup>3</sup>. Yellow and blue distributions correspond to charge accumulation and depletion, respectively.

For NH<sub>3</sub> molecules, the adsorption energies are determined to be -203 meV, indicating the physisorption of NH<sub>3</sub> on the pristine ReSe<sub>2</sub> monolayer. A NH<sub>3</sub> molecule acts as a charge donor and donates 0.024 electrons to the ReSe<sub>2</sub> nanosheet, reducing hole carrier density in the p-type monolayer ReSe<sub>2</sub> (Fig. S3a). For system with Se vacancies, such as those generated by annealing, more electrons (0.049) are transferred from gas molecule to the ReSe<sub>2</sub> nanosheet, leading to more significant reduction of carrier density of ReSe<sub>2</sub> monolayer (Figs. S3b and 3c).



Element	Wt%	At%
SeL	45.07	65.93
ReM	54.93	34.07
Matrix	Correction	ZAF

Figure S5. EDX of ReSe<sub>2</sub> nanosheet.

	O <sub>2</sub>	Air	NH <sub>3</sub>
$I_{dark}(A)$	$2.1 \times 10^{-8}$	$9 \times 10^{-9}$	$7.7  imes 10^{-9}$
$I_{light}(A)$	$4 \times 10^{-7}$	$8 \times 10^{-8}$	$5 \times 10^{-8}$
$R_{\lambda}(AW^{-1})$	95	17.8	10.5
EQE(%)	18,645	3,484	2,061
On/Off ratio	20	9	6

Table S1 The related parameters measured in different gas environment.

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